RESEARCH INVESTIGATION TO DETERMINE MECHANICAL PROPERTIES OF NICKEL AND COBALT-BASE ALLOYS FOR INCLUSION IN MILITARY HANDBOOK-5

VOLUME I

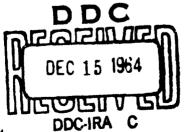
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Air Force Materials Laboratory
Air Force Systems Command
Unites States Air Force
Wright-Patterson Air Force Base, Ohio

Project No. 7381, Task No. 738103



Prepared under Contract No. AF 33(657)-8924 by Republic Aviation Corporation, Farmingdale, N.Y. Authors: A. Greene, H. Sieber, D. Wells, T. Wolfe

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### FOREWORD

This report was prepared by Republic Aviation Corporation, Production Engineering Division under USAF Contract No. AF33(657)-8924. The contract was initiated under Project No. 7381, "Materials! Application", Task No. 738103, "Materials! Information Development, Collection and Processing." The work was administered under the direction of the AF Materials Laboratory, Wright-Patterson Air Force Base, Ohio, by Mr. C. L. Harmsworth, Project Engineer.

This report covers work conducted from April 1961 to June 1964.

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### ABSTRACT

The purpose of this program was to develop design information on four nickel and cobalt base alloys for inclusion in Military Handbook-5. The alloys investigated were Rene 41, L-605, Inconel 702, and Incoloy 901.

The mechanical properties investigated were tensile, compression, shear, bearing, creep, stress-rupture, and fatigue. The general results obtained are presented in Section VII of this report and the data generated for Military Handbook-5 are presented in Section VIII.

The raw data generated in this program is presented in Volume II of this report.

This technical documentary report has been reviewed and is approved.

D. A. SHINN

Chief, Materials Information Branch Materials Applications Division Air Force Materials Laboratory

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SECTION I - INTRODUCTION

### SECTION 1 - INTRODUCTION

### 1.1 Purpose of the Program

The program objective was to obtain statistically sound 'A' and 'B' type data on mechanical properties of nickel and cobalt base alloys for inclusion in Military Handbook-5. Three different heats of each aircraft quality alloy material were evaluated. These alloys are:

a.	Rene <sup>†</sup> 41	AMS 5545 - plate, sheet and stri	P
		AMS 5512 - bars and forgings	
		AMS 5513 - bars and forgings	

- c. Inconel 702 AMS 5550 sheet
- d. Incoloy 701 AMS 5660A bars and forgings

The mechanical property tests performed were tension, compression, bearing, shear, stress-rupture, creep, and fatigue at temperatures ranging from ambient to 1880°F. The data generated was compared with data obtained by means of a literature search.

### 1.2 Background

Although there have been concentrated research and developmental programs on other more refractory alloys over the last few years, nickel and cobalt base alloys must still be considered paramount for present high temperature, load carrying applications. Most high strength refractory metals are still in the relatively early stages of development and indications from Aerospace Industry are that it probably will be several years before refractory metals and alloys with compatible coatings will be commercially available for general use as primary aerospace structures.

Dependable design criteria for nickel and cobalt base alloys must be considered inadequate in comparison with the more common steels and aluminum alloys. A survey of literature on the mechanical properties of the subject alloys has shown a significant quantity of data but very little standardization. An examination of these data, much of which is producer generated, showed wide latitude in such variables as (1) the stage of development of a particular alloy at the time of testing; (2) heat treat condition of material; (3) difference in nominal composition; (4) sampling procedures;

(5) sample preparation; (6) testing techniques; (7) testing environments; (8) correlation with gauges, sized for sheet, bar and plate; (9) data presentation, and (10) the number of tests used to fix a point on a curve.

The high degree of overall confidence demanded by the Aircraft and Aerospace Industries requires an unprejudiced look at the reliability of design allowable strengths.

### SECTION II - SCOPE

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### SECTION II - SCOPE

### 2.0 Scope Discussion

The material data contained in Section 2.1 herein provides the pertinent history, source, chemistry, condition, and form for all the materials used in the program. Each material is listed with the applicable specification, vendor's name, and material heat number. The heat treatment condition is briefly described and the mechanical properties as reported in the vendor's test report are also included.

The scope of the test materials program is briefly summarized below as to material, source, and number of vendor heats.

<u>Material</u>	Source	Number of Vendor Heats
Rene' 41	General Electric Company	14
L-605	Haynes Stellite	6
Inconel 702	Huntington Alloy Prod Div. of International Nickel Company	3
Incoloy 901	Allvac Metal Company	3

In addition to the data resulting from the test program, data has also been incorporated from Republic's Quality Control Laboratory and test reports of the following vendors:

Cannon-Muskegon Corporation

Universal Cyclops Steel Corporation

Latrobe Steel Corporation

Firth Stirling, Incorporated

International Nickel Company

Section 2.2 lists the mechanical properties that have been determined from each type of test performed.

Section 2.3 discusses briefly the test conditions and contains detailed tables denoting test conditions for each type of test.

Table 1

### 2.1 Test Materials

### 2.1.1 Material: Rene! 41

Specification: AMS 5545 - sheet and plate

Specification Minimum Properties

-p	<del>,</del>			
	Gäuge	Room Temp.	1400 F	
Tensile Strength, kai	.010	170	130	
_	.020	170	135	
	.040 & up	170	140	
Yield Strength, 0.2%				
offset	All	130	110	
Elongation, % in 2 in.	All	10	3	

Vendor: General Electric Company, Detroit, Michigan

Vendor	Heat	Number:	R-405	
RAC Id	mtif:	ications	Heat .	A

### Chemical Composition

Carbon	0.12	Cobalt	11.08
Sulfur	0.005	Titanium	3.18
Silicon	0.07	Aluminum	1.50
Manganese	0.02	Boron	0.0053
Chromium	18.68	Iron	0.30
Molybdenum	2.74	Nickel	Balance

Heat Treat Condition - As Received

1975°F water quench 1400°F for 16 hours and air cool

### Vendor Test Report

Form	Test	Ult. Tensile	0.2% Yield	% Elong.
	Temp_of	Strength-KSI	Strength-KSI	2 In.
0.010 In. Sheet	Room	203 <b>.</b> 3	165.4	17.0
	11:00	154 <b>.</b> 1	130.1	6.0
0.020 In. Sheet	Room 1400	-	-	-
0.040 In. Sheet	Room	204.9	159.6	18.0
	1400	164.3	138.3	17.0
0.080 In. Sheet	Room	189 <b>.</b> 7	156.7	14.0
	1400	16կ․կ	137.8	19.0
#0.375 In. Plate	Room	196.4	147.6	22.4
	11:00	155.3	123.7	9.7
1.00 In. Plate	Room	188 <b>.</b> 0	134.0	14.7
	1400	150 <b>.</b> 9	111.9	10.3

\*1975°F for 2 hours and water querch

1400°F for 16 hours and air cool

TABLE 1 (cont'd)

Chemical Composition RAC Identification: Heat B

### Vendor Heat Numbers

	<u>R-248</u>	R-329-22	R-274-8	R-279	R-4102	R-403
Carbon	0.11	0_06	0 <b>.0</b> 5	o.o6	0.10	0.10
Sulfur	0.006	0.007	0.006	0.006	0.007	0.006
Silicon	0.08	0.07	80.0	0.06	0.08	0.09
Manganese	0.07	0.03	0.05	0.04	0.03	0.04
Chromium	19.28	18.84	19.09	19.24	18.92	18.98
Molybdenum	9.60	9.75	9.83	9.71	9.83	9.89
Cobalt	19.93	11.04	10.85	10.98	10.93	11.06
Titanium	3.10	3.23	3.14	3.11	3.06	3.13
Aluminum	1.41	1.50	1.44	1.55	1.40	1.42
Boron	0.004	0.004	0.004	0.005	0.004	0.005
Iron	0.30	0.30	0.30	0.30	0.89	1.18
Nickel	Bal.	Bal.	Bal.	Pal.	Bal	Bal.

Form	Vendor	Test	Ult. Tensile	0.2% Yield	% Elong.
	Heat No.	Temp. OF	Strength-KSI	Strength-KSI	2 In.
0.010 In.	R-329-22	Room	199.6	136.9	18.0
Sheet		1400	141.2	135.4	4.0
0.020 In.	R-248	Room	1740°9	163 <b>.</b> 3	20.0
Sheet		1400	507°3	121 <b>.</b> 6	5.0
0.040 In.	R-274-8	Room	192.2	134.7	17.0
* Sheet		1400	156.8	127.9	4.0
0.080 In. Sheet	R-279-L	Room U400	202.8 158.4	159 <b>.</b> 3 113 <b>.</b> 4	9•0
**0.375 Ine	R-402	Room 1400	192.2 150.2	139 <b>.</b> 4 112 <b>.</b> 7	32.8 16.1
1.00 In.	R-403	Room	195.8	146.3	23.2
Plate		1400	152.8	121.0	11.6

<sup>\* 1975°</sup>F and water quench 1975°F for 30 mins. and air cool 1400°F for 16 hours and air cool

<sup>\*\* 1975°</sup>F for 2 hours and water quench 1400°F for 16 hours and air cool

Chemical Composition RAC Identification: Heat C

Alberta and the second fluid for the second second

Vendors	Heat	Numbers

	<u>r-264</u>	R-332	R-317	R-274	R-286	R-403	R-402
Carbon	0.08	0.07	0.06	0.06	0.07	0.10	0.10
Sulfur	0.005	0.007	0.007	0.006	0.006	0.006	0.007
Silicon	0.08	0.07	0.07	0.08	0.07	0.09	0.08
Manganese	0.05	0.03	0.01	0.05	0.04	0.04	0.03
Chromium	19.13	18.93	19.04	19.09	19.02	18.98	18.92
Molybdenum	9.82	9.62	9.70	9.83	9.89	9.89	9.83
Cobalt	11.06	11.06	11.06	10.85	11.03	11.06	10.93
Tit anium	3.13	3.18	3.14	3.14	3.16	3.13	3.06
Aluminum	1-44	1.54	1.46	1.44	1.47	1.42	1.40
Boron	0.004	0.004	0.0044	0.004	0.005	0.005	0.004
Iron	0.30	0.30	0.30	0.30	0.30	1.18	0.89
Nickel	Bal.	Bal.	Bal.	Bal.	Bal.	Bal.	Bal.

Form	Vendor Heat No.	Test Temp. F	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 2 In.
0.020 In. Sheet	R-317	Room 1400	201.9 146.9	136.3 120.0	23 <b>.</b> 5 3.0
* 0.040 In. Sheet	R-274	Room 1400	192 <b>.</b> 2 156 <b>.</b> 8	134.7 127.9	17.0 4.0
0.080 In. Sheet	R-286	Room 1400	203 <b>.</b> 3 160 <b>.</b> 8	179.4 136.3	21.0 8.0
0.375 In. Plate	R-403	Room 1400	195.0 148.7	138 <b>.</b> 3 116 <b>.</b> 1	32.1 13.5
1.00 In. Plate	R-402	Room 1400	188.0 148.0	132.8 110.8	19.7 9.3

<sup># 1975°</sup>F and water quench 1975°F for 30 mins. and air cool 1h00°F for 16 hours and air cool

### Table 1 (cont'd)

Material:

Renet 41

Specification:

AMS 5713 • bar and forgings

Specification Minimum Properties:

	Room Temp.	1400 F
Tensile Strength ksi	170	135
Yield Strength, 0.2% offset	130	105
Elongation, % in 4D	8	5

Vendor: General Electric Company, Detroit, Michigan

Vendor	Heat	Number:	R-42	0
RAC Ide	entif:	ication:	Heat	E

### Chemical Composition

Carbon	0.09	Cobalt	10.92
Sulfur	0.006	Titanium	3.09
Silicon	0.09	Aluminum	1.50
Manganese	0.03	Boron	0.006
Chromium	18.96	Iron	1.01
Molybdenum	10.02	Nickel	Balance

Heat Treat Condition - As Received

1975°F for 4 hours and water quench 1400°F for 16 hours and air cool

### Vendor Test Results

Form	Test	Ult. Tensile	0.2% Yield	% Elong.
	Temp. OF	Strength-KSI	Strength-KSI	2 In.
0.500 In. Bar	Room	203.0	155 <b>.2</b>	23.0
	1400	166.0	127 <b>.</b> 0	10.8
1.00 In. Bar	Room	196.8	149.8	21.1
	1400	162.5	122.3	12.2
# 1" x 3" Forging #	Room	204.0	15և ւհ	22 <b>.</b> 4
	1400	153.4	127 <b>.</b> 1	9 <b>.</b> 0

<sup>\* 1975°</sup>F for 2 hours and water quench 1400°F for 16 hours and air cool

### Stress Rupture

Form	Test Temp. OF	Stress-KSI	Life-Hours	% Elong.
0.500 In. Bar	1350	85.0	84	32.6
1.000 In. Bar	1350	85.0	103	25.0
1" x 3" Forging	1350	85 <b>.</b> 0	115	14.3

### TABLE 1 (cont'd)

		Number:	R-383	3
RAC Id	entif	ication:	Heat	r

### Chemical Composition

Cerbon	0.11	Cobalt	11.10
Sulfur	0.005	Titanium	3.09
Silicon	O <b>. O8</b>	Aluminum	1.45
Manganese	0.06	Boron	0.0032
Chromium	18.63	Iron	1.46
Molybdenum	9.57	Nickel	Balance

Heat Trest Condition - As Received 1975°F for 4 hours and water quench 1400°F for 16 hours and air cool

### Vendor Test Report

Form	Test	Ult. Tensile	0.2% Yield	% Blong.
	Temp. OF	Strength-KSI	Strength-KSI	2 In.
0.500 In. Bar	Room	199•5	146.8	22.5
	14 00	155•0	119.1	8.0
* 1.00 In. Bar	Room	190.0	141.5	20.7
	1400	155.8	117.5	10.5
1" x 3" Forging	Room	198.0	146.8	21.4
	11:00	149.0	119.4	8.7

\*.1975°for 2 hours and water quench 1400°F for 16 hours and air cool Stress Rupture

Form	Temp. OF	Stress-KSI	Life-Hours	% Elong.
0.500 In. Bar	1350	85.0	82	15.9
1.00 In. Bar	1350	85.0	64	5.1
1" x 3" Forging	1350	85.0	92	•

### TABLE 1 (cont'd)

		Number:	R-410	)
RAC Ide	ntif:	ication:	Heat	G

### Chemical Composition

Carbon	0.07	Cobalt	10.98
Sulfur	0 <b>.00</b> 6	Titanium	3.09
Silicon	0.07	Aluminum	1.49
Manganese	0.0H	Boron	0.00hh
Chromium	19.02	Iron	0.116
Molybdenum	9.94	Nickel	Balance

Heat Treat Condition - As Received

1975°F for 4 hours and water quench 1400°F for 16 hours and air cool

### Vendor Test Report

Form	Test	Ult. Tensile	0.2% Yield	% Elong.
	Temp. OF	Strength-KSI	Strength-KSI	2 In.
# 0.500 In. Bar	Room	196.0	135.5	26.2
	14 00	160.6	118.5	11.5
1.00 In. Bar	Room	194.0	11։14.5	21.5
	11100	163.0	122.0	12.9
* 1" x 3" Forging	Room	203 <b>.</b> 0	152.2	26.4
	1400	152 <b>.</b> 5	119.2	9.4

\* 1975°F for 2 hours and water quench 1400°F for 16 hours and air cool Stress Rupture

Form	Temp. F	Stress-KSI	Life-Hours	% Elong.
0.500 In. Bar	1350	85.0	106	10.7
1.00 In. Bar	1350	85.0	89	10.1
1" x 3" Forging	1350	85.0	67	6.2

2.1.2 Material;

**1.**≈605

Specification:

AMS 5537A - Sheet and Plate

Specification Minimum Properties:

Tensile Strength, ksi		130
Yield Strength 0.2% offse	t, ksi	55-80
Elongation. % in 2 in.	0.020	30
_	0.040	40
over	0.040	45

Vendor: Haynes Stellite Company, Kokomo, Indiana

Vendor	Heat Number:	L2-1787
RAC Ide	entification:	Heat A

### Chemical Composition

Chromium	19.91	Nickel	10.34
Tungsten	14.41	Manganese	1.52
Iron	1.97	Phosphorous	0.014
Carbon	0.11	Sulfur	0.010
Silicon	0.48	Cobalt	Balance

Heat Treat Condition - As Received 2250°F for two (2) hours and water quench

Form	Test Temp. OF	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 2 In.
0.020 In. Sheet	Room	136.95	71.2	45.0
0.040 In. Sheet	Room	136.25	65.25	54.0
0.080 In. Sheet	Room	139.2	66.45	58.0
0.375 In. Plate	Room	•	<b>÷</b>	-
1.00 In. Plate	Room	143.55	67.75	60.0

TABLE 2 (cont'd)

		TABLE 2 (con	t'd)	
Vendor Heat Number: RAC Identification:	L2-1782 Heat B			
Chemical Composition				
Chromium Tungsten Iron Carbon Silicon	19.91 14.79 2.13 0.10 0.60	M: Pl Si	ickel anganese nosphorus ulfur obalt	10.01 1.47 0.015 0.014 Balance
Heat Treat Condition	- As Reco	ived 2250°F	for 2 hours and	water quench
Vendor Test Report				
Form	Test Temp. OF	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	# Elong. 2 In.
0.020 In. Sheet	Room	137.75	72.3	45.0
0.040 In. Sheet	Room	142.8	67.25	55.0
0.080 In. Sheet	Room	140.3	65.25	58.0
0.375 In. Plato	Room	140.0	68.55	55.0
1.00 In. Plate	Room	144.45	67.85	60.0
Vendor Heat Number: RAC Identification:	12-1754 Heat C			
Chemical Composition	L			
Chromium Tungsten Iron Carbon Silicone	20.85 15.06 2.17 0.11 0.58		Nickel Manganese Phosphorus Sulfur Cobalt	9.91 1.555 0.11 0.18 Balance
Heat Treat Condition	- As Rece	eived 2250°F	for 2 hours and	water quench
Vendor Test Report				
Form	Test Temp. <sup>O</sup> F	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 2 In.
0.020 In. Sheet	Room	170.1	70.15	146.0
0.040 In. Sheet	Room	1կև.7	69 <b>.</b> L	52.0

142.95

140.750

142.5

Room

Room

Room

0.080 In. Sheet

0.375 In. Plate

1.00 In. Phate

69.15

69.0

67.75

57.0

55.0

62

### Table 2 (cont'd)

Material:

L=605

Specification:

AMS 5759B - Bar and Forgings

Specification Minimum Properties:

Tensile Strength, ksi 125 Yield Strength 0.2% offset, ksi 45 Elongation, % in 4D 30

Vendor: Haynes Stellite Company, Kokomo, Indiana

	Number:	12-1756
	ication:	Heat E

### Chemical Composition

Chromium	20.53	Nickel	9.84:
Tungsten	15.31	Manganese	1.49
Iron	1.97	Phosphorus	0.013
Carbon	0.11	Sulfur	0.013
Silicon	0.51	Cobalt	Balanca

Heat Treat Condition - As Received

2250°F for 2 hours and water quench

### Vender Test Report

Form	Test Temp. OF	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 2 In.
0.500 In. Bar	Room			
1.60 In. Bar	Room			
1" x 3" Forging	Room	142.1	70.45	58,0
Vendor Heat Number: RAC Identification:	12-1729 Heat F			
Chemical Composition	1			
Chromium Tungsten Iron Carbon Silicon	20.37 15.08 1.98 0.09 0.56	Nickel Hanger Phospl Sulfu Cobalt	nese norus	9.97 1.40 0.013 0.013 Eakance

Heat Condition: As Received

2250°F for 2 hours and water quench

Form	Test Temp. OF	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 2 In.
0.500 In. Bar	Room	11/4.35	69.8	40.0
1.00 In. Bar	Room	141.3	70.0	59.0
1" x 3" Forging	Room	141.5	71.85	61.0

### TAPLE 2 (cont'd)

Vendor Heat		L2-17	737
RAC Identif	ication:	Heat	G

### Chemical Composition

Chromium Tungsten Iron Carbon Silicon	19.95 11.79 1.91 0.10 0.62	Nicke]. Manganeso Phosphorus Sulfur Cobalt	10.05 1.37 0.012 0.019
-1310011	0.02	Cobalt	Balance

Heat Treat Condition - As-Received

 $2250^{\circ}\text{F}$  for 2 hours and water quench

Form	Test Temp F	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 1 In.
0.500 In. Bar	Coom			
1.00 In. Bar	Room			
1" x 3" Forging	Room	1.34.85	66.2	60

### TABLE 3

2.1.3 Material:

Inconel 702

Specification:

AMS 5550 - Sheet

Specification Minimum Properties:

Tensile Strength, ksi

Yield Strength, 0.2% offset, ksi

Elongation, % in 2 in.

0.020
17

0.040 25

Vendor: Huntington Alloy Products, Div. Internation Nickel, Huntington, W. Virginia

Vendor Heat Number: HT 580LD RAC Identification: Heat A

### Chemical Composition

Carbon	0.04	Copper	0.06
Manganese	0.07	Chromium	15.35
Iron	0.71	Aluminum	3.02
Sulfur	0.007	Titanium	0.53
Silicon	0.17	Nickel	Balanca

Heat Treat Condition - As Received

1975°F for 1/2 hour and air cool 1400°F for 5 hours and air cool

Vendor Test Report

Form	Test Temp. F	Ult. Tensile Strength-KSI	0.2% Yield Strength.KSI	% Elong. 2 In.
0.010	Room	139.0	80.5	28.0
0.020	Room	141.0	77.0	35.0
0.040	Room			

Vendor Heat Number: HT 5807D HAC Identification: Heat B

### Chemical Composition

Carbon	0.03	Copper	0.06
Manganese	0.06	Chromium	14.64
Iron	0.37	Aluminum	3.10
Sulfur	0,007	Titanium	0.51
Silicon	0,21	Nickel	Balance

Heat Treat Condition - As Received

Form	Test	Ult. Tensile	0.2% Yield	% Elong.
	Temp. F	Strength-KSI	Strength-KSI	2 In.
0.010	Room	136.0	78.0	29.0

### TABLE 3 (cont'd)

<u>RAC Identification:</u> HT 5806D

### Chemical Composition

Carbon	0.0h	Copper	0.07
Manganese	0.06	Chromium	Tr-80
Iron	0.37	Aluminum	2.99
Sulfur	0.007	Titanium	0.52
Silicon	0.18	Nickel	Balance

Heat Treat Condition - As Received

Form	Test Temp. CF	Ult. Tensile Strength-KSl	0.2% Yield Strength-KSI	% Elong. 2 In.
0.010	Room	134.0	79.0	26
0.040	Room	143.0	85 <b>.</b> 0	<b>3</b> 5

### TABLE L

2.1.4 Material:

Incoloy 901

Specification:

AMS 5660A - Bar and Forgings

Specification Minimum Properties:

Tensile Strength, ksi 150
Wield Strength, 0.2% offset, ksi 100
Elongation, % in 4D 12

Vendor: Allvac Metals Company, Monroe, North Carolina

Vendor Heat Number: 5036 RAC Identification: Heat E

### Chemical Composition

Carbon	0.066	Titanium	2.67
Sulfur	0.012	Aluminum	0.13
Manganese	0.05	Boron	0.013
Silicon	0.06	Copper	0.05
Chromium	12.41	Nickel	43.20
Molybdenum	ક.મ	Phosphorus	0.01
Cobalt	0.18	Iron	Balance

Heat Treat Condition - As Received

2000°F for 2 hours and water quench 1/150°F for 2 hours and air cool 1325°F for 21 hours and air cool

Hardness: Rc 35-38

Vendor Test Report

Form	Test Temp. °F	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 1.0 In.	Hardness Tested Rc
0.500 In. Bar	Room	182.5	116.0	19.8	34
1.00 In. Bar	Room	182•5	116.0	19.8	34
1" x 3" Forging	Room	169.9	111.6	19	32.5

### Stress Rupture

Test Specimen: 0.252 in comb. smooth and notched bar

Form	Test Temp. <sup>C</sup> F	Stress-KSI	Life Hours	% Elong1.0 In.
0.500 In. Bar	1200	80.0	ىل.190	28
1.00 In. Bar	1200	80.0	190•2	28.1
1" x 3" Forging	1200	80.0	128.4	20.0

TABLE 4 (cont'd)

Vendor Heat Number: 3164 RAC Identification: Heat F

### Chemical Composition

Carbon	0.061	Titanium	2.95
Sulfur	0.008	Aluminum	0.22
Manganese	0.06	Boron	0.014
Silicon	0.08	Copper	0.04
Chromium	12.26	Nickel	43.25
Molybdenum	6.20	Phosphorus	0.007
Cobalt	0.08	Iron	Balance

Heat Treat Condition - As Received

2000°F for 2 hours and water quench 1450°F for 2 hours and air cool 1325°F for 24 hours and air cool

Hardness: Rc 32-37.5

### Vendor Test Report

Form	Test Temp. CF	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 1.0 In.	Hardness Tested Rc
0.500 In.Ber	Room	185.0	120.3	14.0	36
1.00 In. Bar	Room	185.0	120.3	nr•0	36
1"x3" Forging	Room	183.1	127.և	18.4	37.5

Stress Rupture

Test Specimen: 0.252 comb. smooth and notched bar

Form	Test Temp.OF	Stress-KSI	Life Hours	% Elong. 1.0 In.
0.500 In. Bar	1200	80 <sub>•</sub> 0	158.9	16
1.00 In. Bar	1200	80.0	158.9	16
1 " x 3" Forging	1200	80.0	<b>L95•7</b>	5.1 Did not fail

### TABLE 4 (cont'd)

		Number:	3192	
RAC Ide	entif:	ication:	Heat	G

### Chemical Composition

Carbon	0.057	Titanium	2.60
Sulfur	0.009	Aluminum	0.23
Manganese	0.05	Boron	0.018
Silicon	0.03	Copper	0.05
Chromium	12.28	Nickel	42.68
Molybdenum	6.10	Phosphorus	0.007
Cobalt	0.12	Iron	Balance

Heat Treat Condition - As Received 2000°F for 2 hours and water quench 1450°F for 2 hours and air cool 1325°F for 21 hours and air cool

### Vendor Test Report

Form	Test Temp. CF	Ult. Tensile Strength-KSI	0.2% Yield Strength-KSI	% Elong. 1.0 In.	Hardness Tested-Rc
0.500 In. Bar	Room	167.2	109.3	22.0	32.5
1.00 In. Bar	Room	167.2	109.3	21.6	32.5
1" x 3" Forging	Room	164.5	103.5	21.0	30.5

### Stress Rupture

Test Specimen: 0.252 comb. smooth and notched bar

Form	Temp. or	Stress-KSI	Life Hours	% Elong. 1.0 In.
0.500 In. Ber	1200	80	102	26 <sub>*</sub> 0
1.00 In. Bar	1200	80	102	25.6
1" x 3" Forging	1200	80	108	19.0

### 2.2 Mechanical Properties Determined

The various mechanical properties determined from each type of test performed are as follows:

- Tensile Tests Sheet and Bar
  - a. Ultimate tensile strength
  - b. Tensile yield strength
  - c. Elongation
  - d. Modulus of elasticity
- 2. Compression Tests Sheet and Bar
  - a. Compression yield strength
  - b. Modulus of elasticity
- 3. Bearing Tests of Sheet e/D = 1.5 and e/D = 2
  - a. Ultimate bearing strength
  - b. Bearing yield strength
- 4. Shear Tests Sheet and Bar
  - a. Ultimate shear strength
- 5. Creep Sheet and Bar
  - a. Percent deformation versus time as a function of temperature and stress
- 6. Stress-Rupture Sheet and Bar
  - a. Time to fracture as a function of temperature and stress
- 7. Fatigue
  - a. Stress versus number of cycles to failure, as a function of temperature and stress-ratio

### 2.3 Test Conditions

Room and elevated temperature, tension, compression, bearing, and shear mechanical property data for both the longitudinal and transverse directions for each material in air atmosphere. The range of elevated temperature data was from 400°F to 1800°F, in increments of 200°F. Data on the effects of exposure in air at temperatures from 400°F to 1800°F for times of 10, 100, 500, and 1000 hours, were compiled from tests at exposure temperature and at room temperature after exposure.

Creep test data was obtained at temperatures 1000°F through 1800°F in air atmosphere. Sheet test specimens were in the transverse direction only. Creep deformations of 0.05, 0.1, 0.3, 0.5 and 1% were obtained.

Stress-rupture data was compiled at temperatures of 400°F through 1800°F, and for lives of 0.1, 1, 100, and 1000 hours.

Axial fatigue test data was obtained at room temperature, and at elevated temperatures from  $400^{\circ}$ F to  $1800^{\circ}$ F in increments of  $200^{\circ}$ F. The data includes lives of 100 cycles through  $10^{7}$  cycles, and at stress ratios of A = 1, 2.0, 0.98, 0.67, and 0.25 where:

# $A = \frac{Alternating Stress}{Mean Stress}$

All alloy materials under this contract were tested in the "as received" con ditions (described in Section II - Test Materials), except Rene' 41 AMS 5712. Finished test specimens were fabricated from the AMS 5712 alloy material, and subsequently aged at 1400°F for 16 hours and air cooled prior to testing. In all other instances, with the exception of exposure tests, there was no further thermal processing.

Test conditions for the complete program are shown in Tables 5 through 16.

1

TABLE 5
TENSION TESTS

Tests Per Alloy	100 100 100 00 00 00 00 00 00 00 00 00 0	17 30 10	3356	9 9 9 9
1800	*******	ww i	wii	ννξ
1600	1118111	w i i	N I I	wii
00 17 21	\$ \$ \$ \$ \$ \$ \$	ညီကက	W I I	ကက္ရရို
emperatu 1200		N I I	wii	rv 1 1
1000 1000	1 100 4 5 5 5 5	<i>ww</i> 1	~ 1 1	៷៷៰៓
of Tes	1110 111	1 1 1		w i i
Numb 009	5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	rv * *	~ I I	พพเ
001	604 104 104 104 106 109 109 109 109 109 109 109 109 109 109	1 1 1	ř I i	m i i
707	*******	អង <b>ម្ព</b>	ងងង្គ	288
Size and Form	.005 Strip .010 Sheet .020 Sheet .040 Sheet .080 Sheet .375 Plete	0.5 Bar 1.0 Bar 1x3 Forging	0.5 Bar 1.0 Bar 1x3 Forging	Incoloy 901 0.5 Bar (AMS 5660) 1.0 Bar 1x3 Forging
Material	Rene' b1 (AMS 5545) and 1—605 (AMS 5537)	Rene': 1,1 (AMS 5713) and L-605 (AMS 5759)	Rene' - 41 (APS 5712)	Incoloy 901 . (AMS 5660)

Tests are equally divided between three heats of material.

1910

Total Tension Tests

Half of these tests are in the transverse direction. All other tests are in longitudinal direction.

TABLE 6

COMPRESSION TESTS

Testa Per	Alloy	09 09 88 88	25 <b>%</b> 25	<i>አ</i> ፠୪	2,5 80 80	Total Compression Tests 1215
	1300	1941	ညီကက	IWI	៷៷ៜ	ssion Te
	7600	1 1 1	1 1 1	1 1 1	I W I	1 Compre
	200	10,4%	IWI	IWI	WW I	Tota
erature	1200	104 [ 1	1 1 1	1 1 1	I W T	
Per Tem	1000	104	ကက္ခရီ	IWI	៷៷៰៓្ន	
Number of Tests Per Temperature	800	ığıı	111	111	11/1	
Number	000	1011	IW I	ואו	11/1	
	9	1 1 1 1 1		i 1 i	IW I	
	lg lg	****** *******	15 30 40 80	ងង <b>ខ្</b> ្	<b>\$</b> 000	
	Size and Form	.020 Sheet .040 Sheet .080 Sheet .375 Plate	0.5 Bar 1.0 Bar 1 x 3 Forging	0.5 Bar 1.0 Bar 1 x 3 Forging	0.5 Ber 1.0 Ber 1 x 3 Forging	
	Material	Rene' 41 (AMS 5545) and L-605 (AMS 5537)	Rene' 41 (APS 5713) and 1~505 (APS 5759)	Rene' 41 (AMS 5712)	Incoloy 901 (Ars 5660)	

1 Tests are equally divided between three heats of material.

Half of these tests are in the transverse direction. All other tests are in the longitudinal direction.

TABLE 7

BEARING TESTS

	Total	350 33 33 33 33 33 33 33 33 33 33 33 33 33	33 33 33 33 33 33 33 33 33 33 33 33 33
	1800°F	๛ฐัพพพ <b>ฐ</b> ัพพ	๛฿ฺํ๛๛๛฿ฺํ๛๛
	150005	1 \$ 1 1 1 \$ 1 1	1 \$ 1 1 1 \$ 1 1
	11,000F	๛฿๎๛๛๛฿๎๛๛	๛ฺรักกกรุ้กก
<b>.</b> .	12000F	ιξίιιξιι	* 5   1   1 5   1
חומו חודי	1000CI	๛อุ้๛๛๛อุ้๛๛	๛฿ฺ้๛๛๛฿ฺ้๛๛
TO CO	800°F	1 2 1 1 1 2 1 1	י זָּלְיי י לָּיי
	6000F	18111811	ו זַּלָּוווּלָּזוּ
	1,000F	ığırığır	١٥٠١١٥
	70°F1	<u> </u>	2468868888888
	e/D	4444444 444444444444444444444444444444	44444466 774476666
	Thickness	920 940 375 920 940 375	020 040 375 020 040 080
	Material	Rene' !11 (AMS-5515 + Age)	L—605 (AMS—5537A)

1 Tests are divided between three (3) heats of material at 700F.

1028

Total Bearing Tests

\* Half of these tests are in the longitudinal direction, all others in transverse direction.

TABLE 8

# SHEAR TESTS

Material	Size and Form	707	8	Number 600	of Tests	Per Temp	erature 1200		1600	1800	Tests Per Alloy	
Rene' 41 (AMS 551,5) and 1~605 (AMS 5537)	.020 Sheet .010 Sheet .080 Sheet .375 Plate	****	11011	NNNN	intl	NNNN	IW I I	<i>ប</i> េសស	11/11	<i>አ</i> ምም	80 30 80 80	
Rane' 11 (AMS 5713) and L-605 (AMS 5759)	0.5 Bar 1.0 Bar 1 x 3 Forging	स् इ.स.कू	w t t	<u> የ</u>	WII.	יי ויי	<b>10 1 1</b>		wii	w i w	<i>X</i> 23 62	
Incoloy 901 (AMS 5660)	1 0.5 Bar 1.0 Bar 1 x 3 Forging	<b>8</b> 88	ww 1	NNN	I WW	www	ww i	$\kappa \kappa \kappa$	տտ 1	NNN	70 50 50	

Tests are equally divided between three heats of material.

0111

All Alloys - Grand Total

Half of these tests are in transverse direction. All other tests are in longitudinal direction.

TABLE 9

ELEVATED TEMPERATURE CREEP TESTS

		Num	ber of Te	Number of Tests Per Temperaturel	lemperatur	- J	Total Per
Material	Size and Form		81	2	8	140	Material
Rene' 1:1 (Avis 5545)	.005 Sheet	<b>4</b> 1	7 -	44	ন ।	44	
	.040 Sheet .080 Sheet	<b>ച</b> ା	<i>コ</i> ।	ゴニ	<b>≂</b> 1	기리	95
1-605 (ANS 5537)	.005 Sheet .020 Sheet .040 Sheet	a   a	7 - 7	বৰন	حا د	<b>ವ</b> ವನ	
	.080 Sheet		. 1	7	ā.	· <del></del>	26
Rene' 41 (AMS 5713)	0.5 Bar l x 3 Forging	<b>≂</b> 1	ا تہ	77	r*1	=#	33
1-605 (AMS 5759)	0.5 Bar l x 3 Forging	<b>4</b> :	<i>2</i> 1	77	, <b>7</b> ,7	77	32
Incoloy 901 (ANS 5660)	0.5 Bar l x 3 Forging	<b>ચ</b> !	77 -	*† 17*	77	r r*	32
				Total 1 Plu Grend	Total 1 Plus Data from SR Grend Total	om SR	208 56 <u>5</u> 26 <u>1</u>
							,

All data to be used for plotting of Larson-Miller curves from which the desired data will be replotted.

All S-R tests that are intended to last for 1000 hours (approx, 36 tests) will also record creep data. Approximately 20 other S-R tests will also record creep data.

Half of these tests are in the transverse direction. All other tests in longitudinal direction. \*

STRESS-TO-RUPTURE TESTS

Material	Size and Form	001	90	Total 800	Specime 1000	1200	Specimens Per Temperaturel 1000 1200 1400 1	1600	1800	Total Per Material
Rene' 41 (AMS 5545)	.005 Sheet .020 Sheet .010 Sheet	1 00 1	0 0 0 0 O	ର ର ଦ ର	<b>4484</b>	<b>ಸವ</b> ವನ	15* 15*	99779	6 12* 12*	278
1—605 (AKS 5537)	.020 Sheet .020 Sheet .040 Sheet	1 00 1	0 0 0 0 0	0 0 C 0	7787 7	17 77 7	12.8	8 77 8 9	6 12* 12*	278
Rene' 41 (ANS 5713)	0.5 Bar 1.0 Bar 1 x 3 Forging	1 1 1	0 I	۰ ۱ م	<i>a</i> 1	ا تـ	6 12*	<b>v</b> ) #	6 12*	*
1_605 (APS 5759)	0,5 Bar 1,0 Bar 1 x 3 Forging	1 1 1	0 I	∾ I	<b>ا</b> ت	<b>=</b> 1	66 12*	9 1	8 15 15 8	8
Incoloy-901 (AMS 5660)	0.5 Bar 1.0 Bar 1 x 3 Forging	<b>ω</b> ι ι	∞ ι	ထေး၊	ω ι	2ľ	₹ ₹	₹ .	75 15* 15*	164

30.0, 100, and 1000 hours. Those materials tested at 4 load levels at one temperature will adjust to fail in 10, 30, 100, and 1000 hours. The balance of testing will be for determining tested at h load levels at 400, 600, 800, and 10000F with 2 tests per load level at 8 load levels at 1200, 1400, 1600, and 18000F with 3 tests per load level. The eight load levels will be adjusted to ideally produce failure in the following times: 0.1, 0.3, 1.0, 3.0, 10.0, 30.0, 100, and 1000 hours. Those materials tested at 4 load levels at one temperature will The .(No sheet material of each sheet alloy and the 0.5 diameter Incoloy 901 her will be Larson-Killer curves.

852

Grand Total

Half of these tests will be in the transverse direction. All other tests in the longitudinal direction.

TABLE 11 - FATIOUE TESTS

MATERIAL	Gauge (Inches)	STRESS	20	700	LOAD IE	LOAD LEVELS PER 600 800	TEN PERA	TEV PERATURE OF 1000	(See Note 1	te 1)	1800	TOTALS
Rene' 41 AMS 5545 - Aged	0.040 Sheet	0.98 0.67 0.25	(21) 6 6	77	mm0	-3	<u>*</u> ~~	74	mm#	7	mma	361
Rene' 41 AMS 5545 - Aged	0.080 Sheet	0.67 0.25	99			3 (15)		3 (15)		3 (15)		150
Rene' 41 AMS 5713	1,000 Bar	29*0	3 (29)	(22)		7	<b>m</b> .	7		3 (21)		157
1-605 Ans 5537a	0.040 Sheet	0.98 0.67 0.25	(22) 6 6	N	нчн	8	448	~ ~	448	α	апн	202
1-605 Ams 5537a	0.080 Sheet	0.67 0.25	(27) (27)			44		ч ч		H H		814
1-605-A'S 5559A	1,000 Bar	29.0	9	8		8		8		8		70
Incoloy 901 A'S 5660A	1.000 Bar	2.000 0.98 0.67	οννο	4	๛๛๛๛	7	๛๛๛๖๎	*	๛๛๛ฃ๎	77	mmmm	067
Inconel 702 Ams 5550	0.040 Sheet	0.98 0.67 0.25	9119	~	244	8	212	0 0	o u o	2	0 1	225
NOTE 1: Five test	Five tests per load level, except numbers	vel, exce	pt numb		enclosed in	ď			<b>=</b>	IUIAL IESIS	a	1(3)

parenthesis, which indicate actual number of tests.

Indicates region where 1800 CPM and 3600 CPM will be investigated.

TABLE 12 EXPOSURE TESTS

	Ar. 4 4 . 9 1						Minimum Total Tests		
Test <sup>2</sup> Type	Material & Form	10	100	emperat 500	1000	Tests, R.T.		R.T.	Exp.T
Tension	005 Sheet (Ren# 41) (L-605)	1800 1600	1800 1600 1400	1800 1600 1400 1200	1800 1600 1400 1200	12 12 9 6	12 12 9 6	24 24 18 12	2կ 2ს 18 12
	Olio Sheet (Rene' 1,1) (L=605) &  Ost Bar (Incoloy 901)	1800 1600	1800 1600	1800 1600 1400	1800 1600 1400 1200 1000 800 600 400	12 12 6 3	12 12 6 3 3 3	36 36 18 9 9	36 36 18 9
	.080 Sheet		1800 1600	1800 1600	1800 1600 1400	9 9 3	9 9 3	18 18 6	18 18 6
Compression	e040 Sheet (Rend 41) (1-605)	1800	1800 1600	1800 1600	1800 1600 1400 1200	12 9 3 3	3 3 3	24 18 6 6	24 18 6 6
Shear	e040 Sheet (Rene 41) (L-605)	1800	1800 1600	1800 1600	1800 1600 1400 1200	12 9 3 3	12 9 3 3	24 18 6 6	24 18 6 6
	080 Sheet (Rene 41) (L-605)	1800	1800 1600	1800 1600	1800 1600 11:00 1200	12 9 3 3	12 9 3 3	24 18 6 6	6 6 6 6
Bearing (e/d=1.5)	1040 Sheet (Rens 41) (I-605)	1800	1800 1600	1800 1600	1800 1600 11:00 1200	12 9 3 3	12 9 3 3	24 18 6 6	24 18 6 6
	080 Sheet (Rene: 41) (L-605)	1800	1800 1600	1800 1600	1800 1600 1400 1200	12 9 3 3	12 9 3 3	24 18 6 6	24 24 24
		Tota Tota Tota	al Compa al Shear al Bear: IMUM TO:	ing Tes CAL TES	Tests ts	R.T. 255 54 108 108 525		255 54 108 108 525	mp a
			3	0					

# TABLE 12 (cont'd)

- Rene' 41 tested as AMS5545 Age L-605 tested as AMS5537A Incoloy 901 tested as AMS5660A

- All tests in the transverse direction

TABLE 13
STATIC PROPERTIES OF INCONEL 702

Ĭ.

Number of Tests Per Temperature SIZE & 700Fl 1200°F 1406°F LOOPF 600°F 800°F 1000°F 1600°F 1800°F SHAPE TENSION: 100 60m 10\* 10\* 10\* .005 Strip 10\* .020 Sheet 60× 10\* 10\* 10\* 10\* 100 .O40 Sheet 60**\*** 10# 10# 10# 10\* 10\* 10\* 10# 10# 1110 TOTAL TENSION 340 COMPRESSION: 60 .020 Sheet 60\* 10# 10# 10<del>\*</del> 10\* .OhO Sheet 60\* 10# 10# 10\* 10# 140 TOTAL COMPRESSION 200 SHEAR: .020 Sheet 60# 5 80 5 <u>-</u>5 .040 Sheet 60# 100 TOTAL SHEAR 180 BEARING: (e/D = 1.5).020 Sheet 39 140 30 3 10\* .OLO Sheet 60\* 10\* 10\* 10# 10\* 10% 10\* 10% (e/D = 2.0).020 Sheet 30 3 10# 39 .OLO Sheet 60# 10\* 10\* 10\* 10<del>\*</del> 10\* 140 TOTAL BEARING 358

All room temperature tests are equally divided between three (3) heats of material with the exception of the bearing tests on .020" sheet.

<sup>\*</sup> Half of tests in transverse direction.
All others in longitudinal direction.

TARLE 14

EXPOSURE TESTS ON INCONEL 7021

TYPE OF	SIZE &	TIM	E & TH	PERATUR	E <sup>2</sup>	SPECIMENS TO	be tested
TEST	SHAPE.	10	100	500	1000	R.T.	Exp. T.
TENSION:	•005 Strip	1800 1600	1800 1600 1400	1800 1600 1400 1200	1800 1600 1400 1200	12 12 9 6	12 <b>12</b> 9 6
	•040 Sheet	1800 1600	1800 1600	1800 1600 1400	1800 1600 1400 1200 1000 800 600 400	12 12 6 3 3 3 3	12 12 6 3 3 3 3
COMPRESSION:	●040 Sheet	1800	1800 1600	1800 1600	1800 1600 1400 1200	12 9 3 3	12 9 3 3
SHEAR:	.O4O Sheet	1800	1800 1600	1800 1600	1800 1600 1400 1200	12 9 3 3	12 9 3 3
BEARING: (e/D = 1.5)	•040 Sheet	1800	1800 1600	1800 1600	1800 1600 1400 1200	12 9 3 3	12 9 3 3
			TOTAL	FXPCSUF	Œ	165 R.T.	165 Exp. T.

<sup>1 -</sup> All tests in longitudinal direction.

TABLE 15

CREEP TESTS ON INCONEL 702<sup>1</sup>

SIZE & FORM	1000°F	1200 <b>°F</b>	11,000F	1600°F	1800 <b>°</b> F	TOTALS
.005 Sheet .020 Sheet .040 Sheet	<u>1</u> 4	<u>1</u>	7† 7†	4	7 7	20 8 20
				TOTAL CREE PLUS DATA GRAND TOTA	FROM S.R.	13 61

TABLE 16
STRESS-RUPTURE TESTS ON INCONEL 7021

SIZE &		<u>N</u>	umber o	f Tests	Per Temp	erature			
FORM	400°F	600°F	800°F	1000°F	1200°F	1600 <b>°</b> F	1600°F	1800°F	TOTALS
.005 Sheet .020 Sheet .040 Sheet	- - 8	2 2 8	2 2 8	14 14 8	կ հ 24	6 6 48*	6 6 24	6 6 48#	30 30 176
						TOTAL S	TRESS-RU	PTURE	236

The .040 sheet material will be tested at four load levels at 400, 600, 800 and 1000°F with two tests per load level and at eight load levels at 1200, 1400, 1600 and 1800°F with three tests per load level. The eight load levels will be adjusted to ideally produce failure in 0.1, 0.3, 1.0, 3.0, 10.0, 30.0, 100 and 1000-hours. The four load levels at 400 to 1000°F will be adjusted to fail in 10, 30, 100 and 1000-hours. The remainder of the tests will be used to determine Larson-Killer curves.

All others are tested longitudinally.

<sup>#</sup> Half of these tests will be in the transverse direction.

# SECTION III - TEST SPECIMENS

# SECTION 111 - TEST SPECIMENS

## 3.1 Specimen Identification Codes

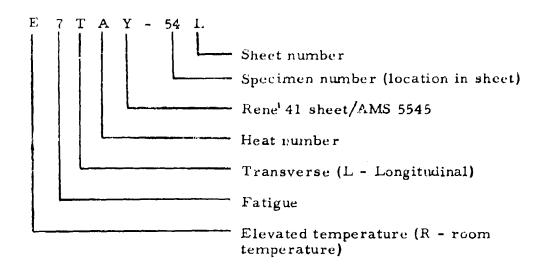
Two typical specimen identification numbers are £51AX-5G and R4TBYE-2H. In general, code numbers will have the above form with each letter or number having a specific meaning as indicated below. The only exception to this system, also described below, involves the two letters immediately following the L or T direction notation (in examples above, AX and BY).

- 3.1.1 The initial E or R indicates the testing temperature level intended for the specimen (E elevated, and R room).
- 3.1.2 The number following the E or R indicates the type of test; for sheet material, the numbers are as follows:
  - l Tension
  - 2 Stress to rupture
  - 3 . Creep
  - 4 Compression
  - 5 Shear
  - 6 Bearing with 1,5 edge distance
  - 67 Bearing with 2.0 edge distance
    - 7 " Fatigue
- 3.1.3 The letter following the type of test number indicates the specimen direction:
  - L Longitudinal
  - T Transverse
- 3.1.4 The two letters following the directional letter give the material used and the heat from which the specimen was taken. This information is given in one sequence for plate, bar, and forgings, and in the reverse sequence for sheet specimens.
- a. Plate, Bar, Forgings The first of the two letters gives the material, and the second shows the heat.
  - 1) Material Code

A	1.605	AMS 5537 (Sheet)
$\mathbf{B}$	L605	AMS 5759 (Bar)
C	Rene! 41	AMS 5545
D	Rene <sup>t</sup> 41	AMS 5512
$\mathbf{E}$	Rene! 41	AMS 5513
G	Incoloy 901	AMS 5560

- 2) Heat Code
  - S Heat A
  - V Heat B
  - W Heat C
  - X Heat E
  - Y Heat F
  - Z Heat G
- b. Sheet (through .080 inch) The first of the two letters gives the heat and the second indicates the material.
  - 1) Heat Code
    - A Heat A
    - B Heat B
    - C Heat C
  - 2) Material Code
    - X L605
    - Y Rene 41
    - Z Inconel 702
- 3.1.5 Exposure notation where exposure is required prior to testing, a letter E is included following the material-heat or heat-material designation (Example R6TAXE-3G or E7LASE-5H).
- 3.1.6 The final number (immediately following the hyphen) indicates the specimen location on the sheet from which the specimen was taken.
- 3.1.7 The final letter (G and H in the initial examples shown) indicates the sheet used within a given heat. When a heat contained more than one sheet, letter designations were arbitrarily assigned to differentiate one sheet from another.
- a. The absence of the sheet designation indicates the particular specimens involved was taken from a heat represented by one sheet (Example-R1LAX-2).
- 3.1.8 Miscellaneous notations where duplication of numbers under the standard coding system can occur due to similar specimens received from the same material (same specimen from 1 inch plate or 1 inch bar or same specimen from 1/2 inch bar or from forging), a further breakdown is made by including a notation in front of the standard code.
  - a. F indicates specimen was made from a forging (Ex. FR1LBX-2).
- b. Number and letter indicate size and type of stock (Ex. 1PE5LAX-2) 1P-1 inch plate; 3/8P-3/8 inch plates; 1/2B-1/2 inch bar; etc.
- c. Absence of an extra notation generally means that no duplication of codes can occur.

d. An example of the coding system is shown in the diagram below.



# 3.2 Specimen Sampling

A typical example of controlled randomization of test specimen sampling for sheet material is shown in Figure 1\*. The material is 0.040 inch Rene 41 sheet, 36 x 96 inches. Oversize specimen blanks were sheared in groups, with each group consisting of at least one specimen for each type of test, in order that the test results would reflect any material property variation within a sheet. Test specimens were taken from both transverse and longitudinal directions for all sheet, plate, and 1 x 3 inch forgings. The 0.500 and 1.00 inch diameter bar were sampled in the longitudinal direction only. Controlled sampling was carried out for each of the three representative heats, and for all gauges of each of four alloy materials.

# 3.3 Specimen Design

Specimen designs used are shown in Figures 2 through 11. All test specimens are full size, and conform where applicable to ASTM Standards and Aerospace Industries Association Report No. ARTC-13.

## 3.4 Specimen Preparation

In an effort to obtain uniformity in test results and to insure that the variations in test results would be due to material properties, the various types of test specimens were machined to close tolerances. The conditions for machining and grinding were chosen to provide a minimum of distortion and residual stress.

# \* Asterisk - Ref. Section V - Test Procedures

# SECTION IV - TEST EQUIPMENT

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## SECTION IV - TEST EQUIPMENT

# 4.1 Static Tests

## a. Loading Apparatus

The loading apparatus used in the performance of tension, compression, bearing, and shear tests, are as follows:

One - Instron Universal Testing Machine - 10,000 pounds capacity
One - Riehle Universal Testing Machine - 20,000 pounds capacity
Seven - Baldwin-Lima-Hamilton Universal Testing Machines of
50,000, 60,000, 100,000, 120,000 and 150,000 pounds
capacity.

All of the above mentioned machines are equipped with strain-rate pacers, and Baldwin-Lima-Hamilton Type MA-1 autographic recorders.

#### b. Extensometers

Baldwin-Lima-Hamilton Type T-1M and PSH-8MS nicroformer type extensometers were used in conjunction with Type MA-1 autographic recorders to measure load versus deformation.

#### c. Furnaces

Static test specimens (with the exception of 0.005 and 0.010 inch gauge tensile specimens) were heated by Marshall or Arcweld Type F-6 furnaces with a core 16 inches long and 3 inches in diameter. Splittype three-zone Marshall furnaces 16 inches long and 3 inches in diameter were used to accommodate the larger test fixtures and jigs required for compression and bearing testing. The 0.005 and 0.010 inch gauge sheet tensile specimens were heated by the specimen's electrical resistance. A Thermac Voltage Controller, with a Vernier controlled variable output was used in conjunction with Leeds-Northrup Potentiometers to control and record temperatures. Either Minneapolis-Honeywell Brown Electronik or Leads and Northrup dual range multipoint controller-recorders with 20 A.W.G. diameter chromel-alumel thermocouples, were used to monitor temperature for the furnace heated test specimens.

# 4.2 Creep and Stress-Rupture Tests

A simple beam loading type test frame having a maximum lever arm ratio of 20 to 1 was used for loading creep specimens. The electrical resistance furnaces were designed by the New England Material Laboratory,

with an 18 inch long alundum core wound with nichrome wire with the upper lower halves controlled. An inner 'shielding tube' was incorporated into the furnace for the purpose of smoothing out hot spots. The furnaces were constructed with nichrome wire wound window tubes also rheostat controlled, through which creep measurements were made. Three chromel-alumel thermocouples were wired to gauge section of test specimen, and specimen temperature was maintained by Minneapolis-Honeywell Brown Electronik controller-recorders.

Creep measurements were made optically by means of a platinum wire-in-tube arrangement mounted on the shoulders of the test specimen.

# 4.3 Fatigue Tests

A total of nine Universal fatigue testing machines were used; two each Baldwin SF-10-U, SF-11U, and 1V-4F machines were used for the 1800 CPM tests, and one Wiedemann-Baldwin SF-4 for the 3600 CPM tests. All machines were equipped with 5-1 multipliers and constant load maintainers.

Furnaces were either Marshall split type or heavy duty type MK-4010, electrical resistance type with 2200°F capability. Temperature recorder-controllers were Minneapolis-Honeywell Brown Electronik or Leads and Northrup.

# 4.4 Thermal Processing Equipment

The furnace equipment used for heat-treating and the elevated temperature exposure of test specimens in air is as follows:

One - Pereny Model EX-9-SP-103

One - Precision Model 31281

One - Blue M Model CHH-16

One - Lindberg Model B-6

Two - Temco Model 1525

Two - Dyna-Trol Model P93H

Temperature surveys were made of all furnaces prior to use. Test specimens were always located in the central portion of a furnace where the exposure temperature could be maintained at \$\frac{1}{2} \text{10}^{\text{O}} \text{F}\$.

# SECTION V - TEST PROCEDURES

Tables None

Figures 1 thru 32 (SPC.5.8, page 51)

#### SECTION V - TEST PROCEDURES

# 5.1 Tensile Tests

Tensile tests were performed on Instron, Riehle, and Baldwin-Lima-Hamilton Universal testing machines equipped with integral automatic strain pacers, see Figures 12 and 13. Each testing machine was calibrated periodically to insure loading accuracy within ± 0.2% on all scales, in compliance with applicable ASTM Standards and the Aircraft Industries Association Report ARTC-13 specifications for loading accuracy and axiality. Shackles and grips were designed to give load concentricity within 0.003 inches.

Specimens tested at elevated temperatures were generally heated by means of resistance wound (Arcweld and Marshall) furnaces of 2200°F capability and with a 16 inch long core which was 3 inches in diameter. Each furnace was equipped with a (Wheelco) pyrometer with 0 to 2400°F range capable of maintaining  $^{+}_{2}$ 5°F over a 2 inch gauge length, and a (Honeywell Brown "Electronik") dual range multipoint recorder with an accuracy of  $^{+}_{2}$ 10°F through 1800°F range. An exception was made in the case of the foil gauge specimens, which were heated by the electrical resistance method.

Specimen temperature measurement was by means of 20 A.W.G. diameter chromel-alumel thermocouples attached to the center of test specimen. All thermocouples were checked to a standard calibrated by the National Bureau of Standards. When potentiometers were used, they were calibrated daily to insure accuracy within the standard specifications.

Measurements for area determination were made with a micrometer suitable for measuring to † 0.0001 inch of the nominal dimension. The average of a minimum of five (5) readings spaced over the gauge length was used to determine the thickness of each test specimen.

Strain measurement was by means of an extensometer attached to the gauge length of a specimen. Since yield strength and the modulus of elasticity were determined for each specimen, a Baldwin-Lima-Hamilton Type T-1M or PSH-8MS Class B-1 extensometer was used in conformance with ASTM Standard E83-57T. The extensometers were modified with insulated knife edges for the testing of the electrical resistance heated foil gauge specimens, and with special care being taken that the extensometers did not impose bending moments or axial loads to the test specimens in excess of 1% of the failing load. Total elongation was measured by the use of gauge marks on specimen surface.

A typical tensile test began by mounting a clean test specimen with appropriate holders and grips into the loading apparatus as shown in Figure 14. The specimen was then instrumented with extensometers (and thermocouples when the test was being run at other than ambient temperature). In the case of an elevated temperature test, the specimen was brought to testing temperature as quickly as possible; and after a 30 minute soak, the specimen was loaded at a strain rate of 0.005 in./in./min. up to yield. After yield strength the strain rate was increased to 0.04 in./in./min. or a strain rate to induce failure in one minute.

# 5.2 Compression Tests

The loading machines shown in Figures 15 and 16 have been described in paragraph 5.1 (Tensile Tests). In addition, care was taken to assure proper alignment and parallelism of the machine cross-heads.

The heat source for all elevated temperature compression tests was provided by a three-zone clam shell (Marshall) furnace, of 2200°F capability. The three zones could be independently adjusted to give the desired temperature gradient along the furnace axis. A Leeds-Northrup Control Recorder was used to respond to the signals generated by the three chromel-alumel thermocouples, that were intimate with the test specimen surface, producing continuous load versus deformation curves.

A temperature calibrating procedure was performed at each test temperature to determine optimum furnace control settings for the temperature. The resulting temperature gradient was  $\pm$  3°F for all test temperatures. A bar compression test specimen is shown in Figure 18.

Compression test specimens fabricated from sheet materials (0.040 inch and 0.080 inch thickness) were tested in a specially constructed fixture, see Figure 17, which provided lateral support to prevent buckling prior to yielding. In a series of calibration tests, the magnitude of frictional force that was transferred from lubricated specimen to specimen support guides was determined to be negligible. The bar specimens did not require lateral supports in testing, but because of the higher loads required to cause the material in these shapes to yield, the bar test fixtures were designed to accommodate loads up to 75,000 pounds, see Figure 16. The selection of the fixture material was critical due to range and repeated exposure to elevated temperatures. Rene' 41 was chosen and it exhibited satisfactory properties throughout the test temperature range. Carbide inserts were used in the top and bottom subpress to resist deformation from concentrated compressive loads transmitted by test specimens.

Deformation measurements were made by means of a Baldwin-Lima-Hamilton Microformer Compressometer, calibrated and fitted with arm extensions to remove it from the elevated temperature environment of the furnace. This compressometer was connected to an autographic recorder to give load-deformation curves for each test.

A typical compression test began by recording the dimensions of a clean test specimen and then placing it into the test fixture. For sheet specimens, the specimen lateral support guides were positioned. Care was taken to assure axiality and alignment before application of load. The compressometer was then attached and the furnace closed. When the test temperature was stabilized, load was applied at 0.005 in./in./min. until yield is observed on a load deformation curve being autographically plotted.

### 5.3 Bearing Tests

The descriptions given of loading apparatus and heat source for the tension and compression tests are applicable for bearing tests. Bearing tests differ in respect to the type of fixtures employed, method and load application, and the manner in which the mechanical properties are determined. Clevis fixtures were used of various slot widths to accommodate each particular sheet specimen thickness. The application of

the load through clevis fixture to test specimen was by means of 0.250 inch diameter pin through reamed hole of specimen as shown in Figures 18, 19, and 20.

Measurement of bearing deformation was by means of a microformer extensometer with one set of arms attached to the loading clevis, and the other set attached through point contacts to the specimen edge on a line tangent to the loaded side and on the horizontal center line of bearing hole. The displacement detected by the extensometer was equivalent to the vertical deformation of the 0.250 inch diameter hole. The change in displacement sensed by the extensometer was fed into an autographic recorder which produced the load deformation plots.

The clevis fixtures were fabricated from Rene' 41 alloy material. The most severely stressed fixture component was the 0.250 inch diameter bearing pin, whose service life varied with test temperatures. Bearing pins of several materials were employed in the test program; the following table indicates the most effective material selection for each test temperature.

Dearmy Lin March	Pin Material	P	aring	Bea
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	Room	400	<u>600°</u>	800°	1000°	1200°	<u>1400°</u>	1600°	1800°
Vascojet M-A	x	x	X	x	x				
Vascomax 300	x	x	X	X	x				
Rene' 41					X	X	X		
Haynes 713C								X	X

A bearing test specimen was prepared for tests by cleaning; hole dimensions and thickness measurements were noted. The specimen was placed in the clevis fixtures and secured in place with two pins. The specimen was then instrumented with thermocouple and extensometer, and then inserted into a clam-shell furnace. When both fixture and test specimen were stabilized at test temperature, the specimen was loaded at 0.02 in/in./min. and a load-deformation curve was obtained. After yield, the strain rate was increased to produce failure in less than a minute.

### 5.4 Shear Tests

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Single shear tests were performed on sheet materials room and elevated temperatures, utilizing a heat source and axial loading apparatus described previously in paragraph 5.1. A clevis-pin fixture was used to attach the test specimens to self-aligning loading rods, as shown in Figure 21. Since the only value of interest was ultimate load, no extensometers were used.

Examination of test specimens after initial shear tests revealed the mode of failure to be other than true shear. The failures resulted from twisting of the test section perpendicular to the thickness dimension followed by tearing parallel to the slot, the latter action being accompanied by a lateral shift in the width direction, see Figure 22a.

The specimen configuration for sheet shear was chosen for this contract originally is recommended by ARTC-13 and elsewhere, and widely used throughout the industry. Ideally this specimen design should give a straight line shear failure

between the two 1/16 inch reamed holes which were 0.19 inches apart.

Test equipment was re-checked for alignment and axiality of load application, and found satisfactory. Additional test specimens were fabricated to incorporate changes in slot angles, i.e., 25°, 60° and 90°, as well as changes in shear height ratios. In other specimens, the 1/16 inch diameter holes were varied in position, some were placed on the center line, while others were made tangent to the center line.

When jigs were constructed to restrain the test section rotation, so that loading would be axial, the specimens failed by coining nuggets.

The problem of shear path instability was resolved satisfactorily by adjusting the distance between reamed holes. The distance between reamed holes for 0.020 inch thick sheet material was reduced to 0.070 inch and a subsequent w/h ratio of 9, and for the 0.040 and 0.080 inch thick sheet, the distance was made 0.100 inch for a w/h ratio of 6. (In this regard the thickness to shear path ratio seemed more critical than the w/h ratio). Consequently, the design of all single shear sheet specimens, for the four alloys tested, was changed in respect to hole distances.

The shear test fixture, see Figures 23 and 24, for double shear pin tests (pin specimens taken from bar and forgings) consisted of two clevis connected with a 1/4 inch center plate. A 1/2 inch diameter bolt was used for one clevis pin; the shear test specimen was the other clevis pin. Replaceable carbide inserts with 1/4 inch inside diameters were fitted to hold the specimen in the center plate and bottom clevis. These inserts prevented deformation of the fixture in the vicinity of load application to pin shear specimen. The inserts were replaced as soon as they became damaged or worn.

Both sheet and pin shear tests were performed with strict adherence to ASTM Standards specifications for heating and soaking time, as well as specimen temperature control. The shear path distance, and specimen thickness were measured for all sheet specimens; (pin diameter in the case of pin specimens) prior to test. Upon application of loads, sheet shear specimens failed along shear path, and pin specimens by the formation of two new planes perpendicular to pin specimen axis.

# 5.5 Creep Tests

Creep tests were performed using the simple beam loading type apparatus, as shown in Figure 25, having a maximum lever arm ratio of 20 to 1. Depending on the loading requirement, a 10 to 1 ratio or direct loading pan was used. To minimize variability of results in elevated temperature creep, care was taken to insure that misalignment between load application and longitudinal axis of test specimen was less than 1% of the working range. Eccentricity of loading is critical for creep tests, especially when small deformations (less than 1%) with time must be measured. Reproducibility of results also depends on rigid adherence to temperature measurement standards (as outlined by ASTM Standards), as well as the dependability of equipment and/or thermocouples to monitor these temperatures over periods of 1000 hours.

The electrical resistance furnaces used were specially constructed with an 18 inch long alundum tube wound with nichrome wire, with the upper and lower halves of main winding rheostat controlled. A pyrex "window" tube, covered at each end through which the creep measurements were made, contained rheostat controlled

windings, so that exceptionally fast temperature adjustments were possible. In addition, the furnace was designed with an inner "shielding tube" for the purpose of reducing hot spots. Indicated temperature deviations from nominal could be maintained at a maximum  $\pm 2^{\circ}F$  for each test duration. A diagram of furnace is shown in Figure 26.

Three chromel-alumel thermocouples were attached to test specimen gauge section, and test temperature was maintained by Minneapolis-Honeywell "Electronik" on/off controllers. Furnace control was through a centrally located thermocouple. All temperatures could be simultaneously printed on the same recorder chart when a temperature check was necessary. A minimum of three creep deformation readings were made daily during the first fifty hours of a test. More frequent readings, as many as ten per hour, were made when the nature of the creep curve warranted it. Instrumented test specimens are shown in Figures 27 and 28.

Extension measurements were made optically, using notched platinum wire-intube extensometers mounted on the shoulders of the test specimens. Two sets of the platinum wire-in-tube arrangement were mounted on each specimen, so that corresponding readings on two sides could be averaged to compensate for any unavoidable load eccentricity. The filar eyepiece inicroscopes used read directly to 0.00004 inches.

Prior to loading of test specimens, the furnaces were always brought to the desired temperature. The correct specimen temperature and gradient were achieved, and the specimen stabilized for 1/2 hour. Initial loadings were done in small increments well within the elastic range of test specimen previously determined. With the addition of each small increment, an elongation measurement was made. Readings of total elongation on loading could be made (after incremental loadings) so that the elastic contribution could be determined and subtracted.

## 5.6 Stress Rupture Tests

For the stress-rupture testing, the procedures and apparatus were generally the same as outlined for creep testing, except that the platinum wire-in-tube extensometers were not used. In this phase of the testing, we were concerned with time-to-rupture under a constant tensile load at a constant temperature.

Upon fracture of a specimen a timing device, actuated at the beginning of the test, would stop automatically to give test time duration. Another switch actuated by specimen fracture would stop the furnace power.

## 5.7 Fatigue Tests

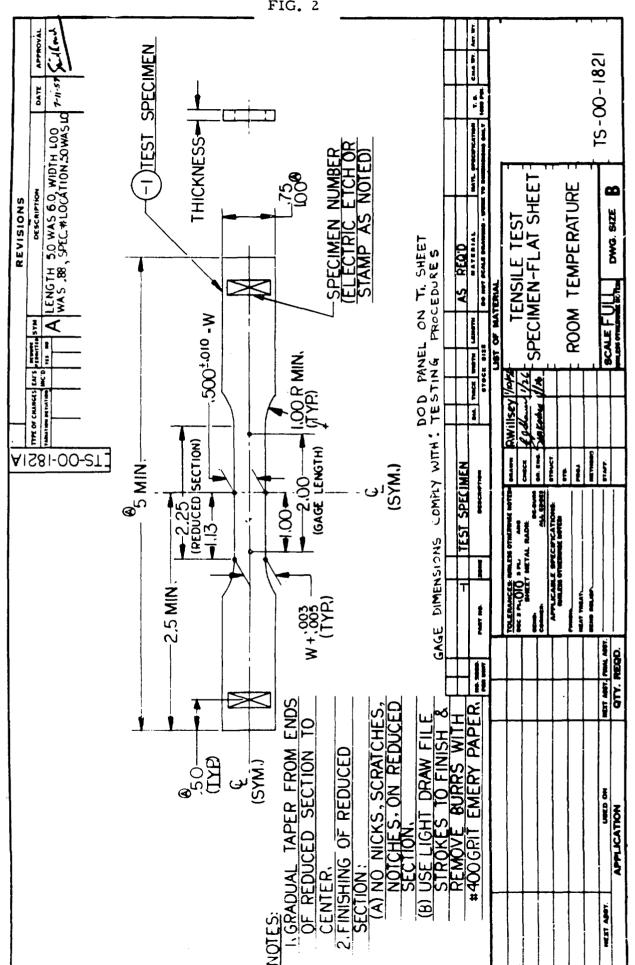
Several types of fatigue equipment were used to perform the axial tension-tension fatigue tests under this contract (see Figures 29 and 30). Selection of a test machine was on the basis of required stress-level, cycling rate, and stress-ratio. All 1800 CPM cycling speed fatigue tests were performed on Baldwin SF-10-U, SF-1-U, and IV-4F type machines. The machines were equipped with 5 to 1 multipliers. A Wiedemann-Baldwin SF-4 was used for testing in the 3600 CPM range. These machines were equipped with automatic pre-load maintainers. All dynamic systems of fatigue testing machines were checked to insure conformance to cycling speeds. Dynamic loading systems were checked and calibrated periodically.

Electrical resistance heating furnaces with 2200°F capability were used, as shown in Figure 31. Chromel-alumel thermocouples were calibrated at each test temperature against standards traccable to the National Bureau of Surviards. Separate thermocouples were used for controlling and monitoring. On initial tests at each temperature, dummy specimens were instrumented with 4 thermocouples at each temperature. By utilizing Brown "Electronik", and/or Leeds-Northrup indicating controlling recorders, temperatures were controlled within ± 5°F to 1800°F. Care was taken to insure alignment of test specimens, as shown in Figure 32.

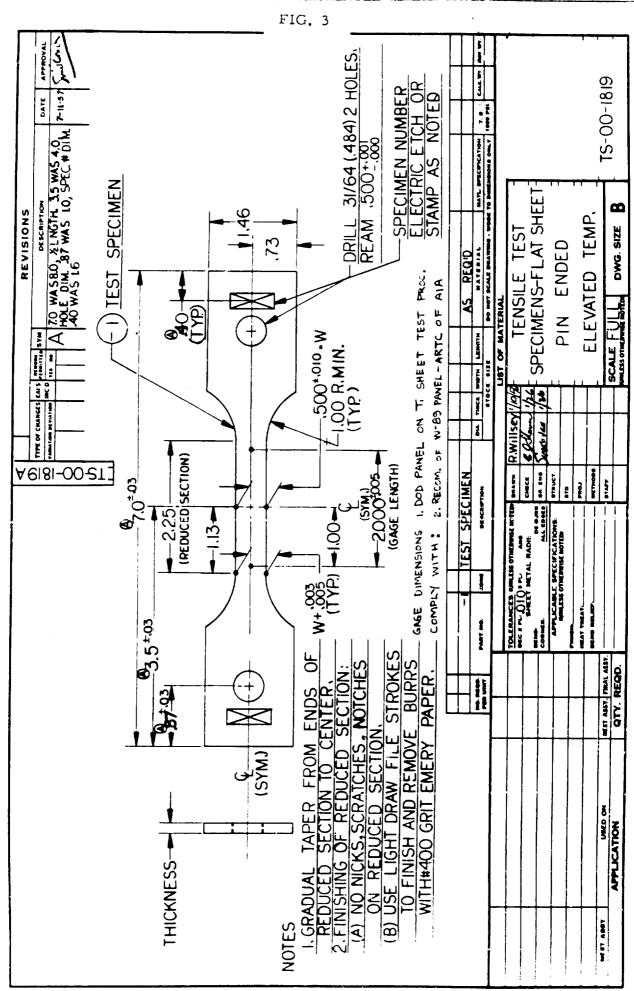
# SECTION V - TEST PROCEDURES

5.8 Figures 1 through 32

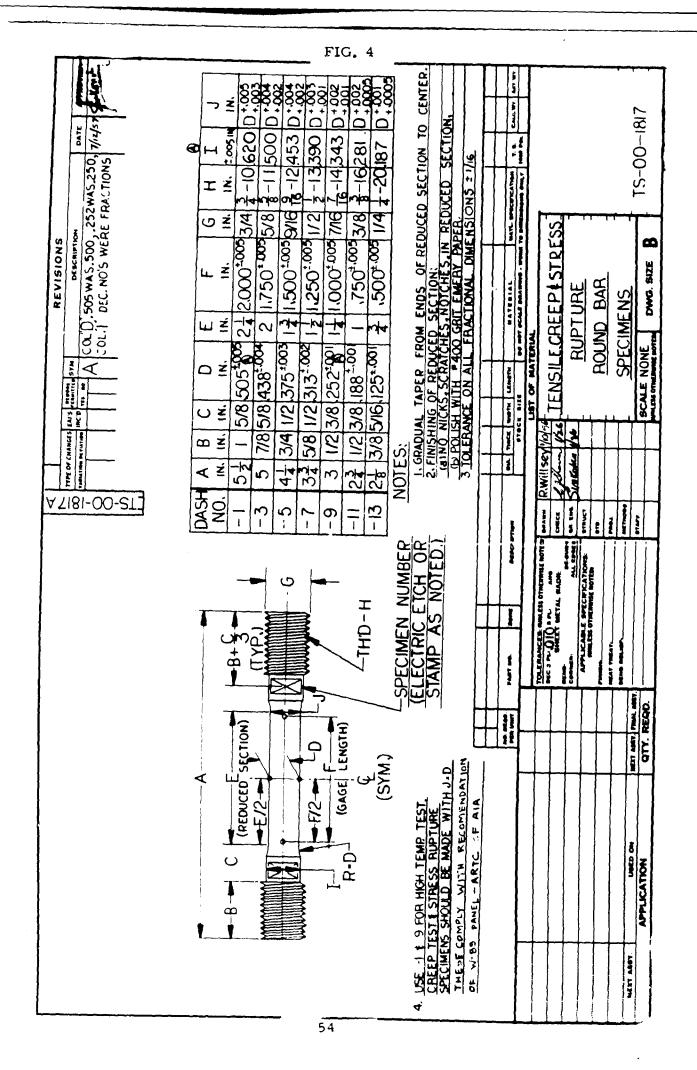
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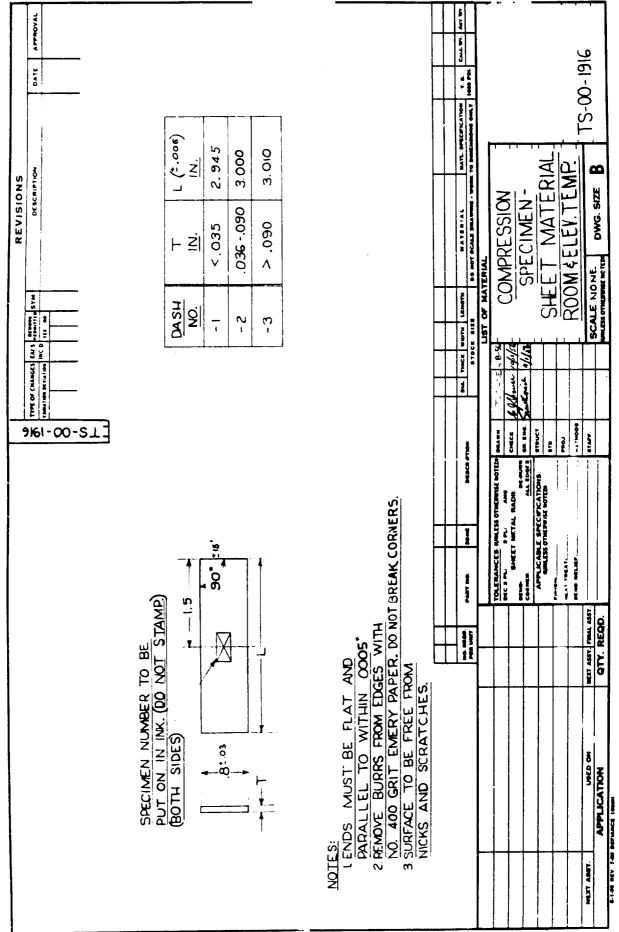


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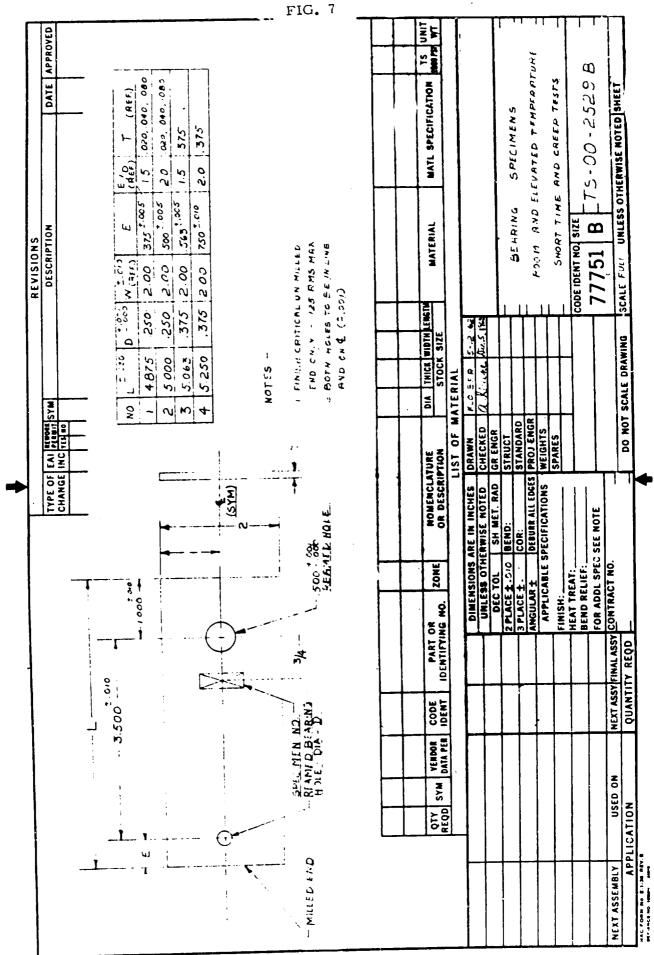


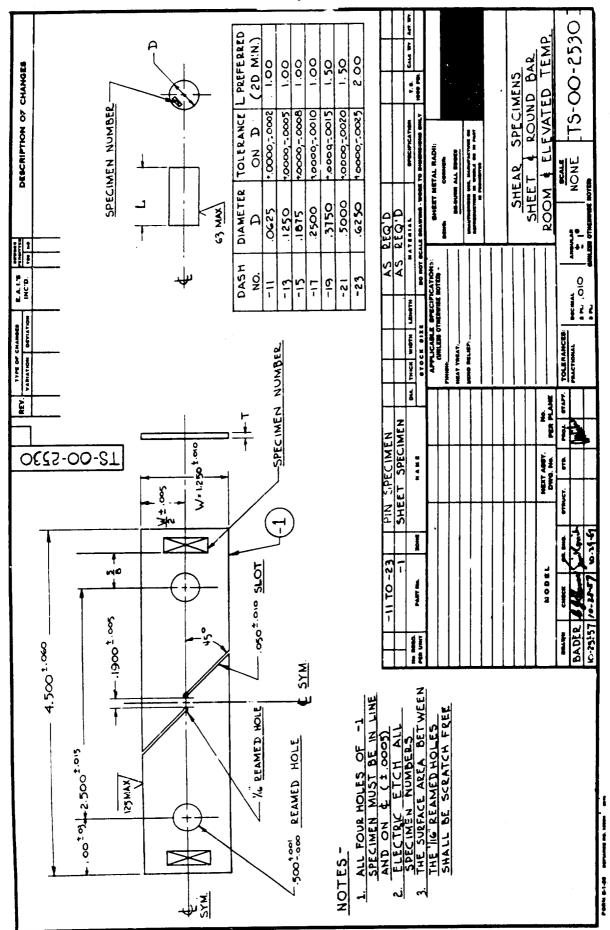




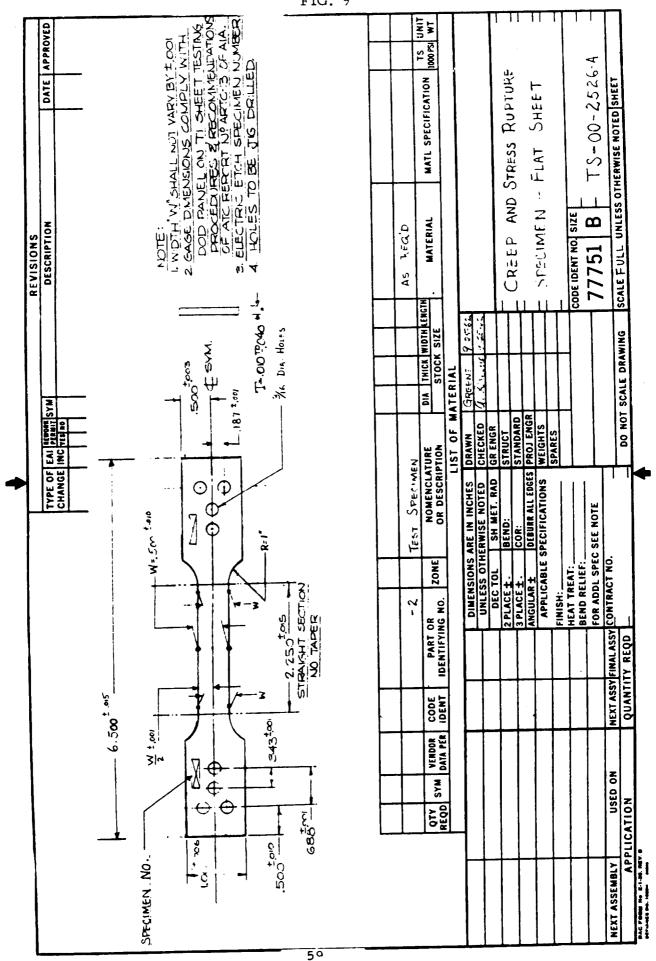
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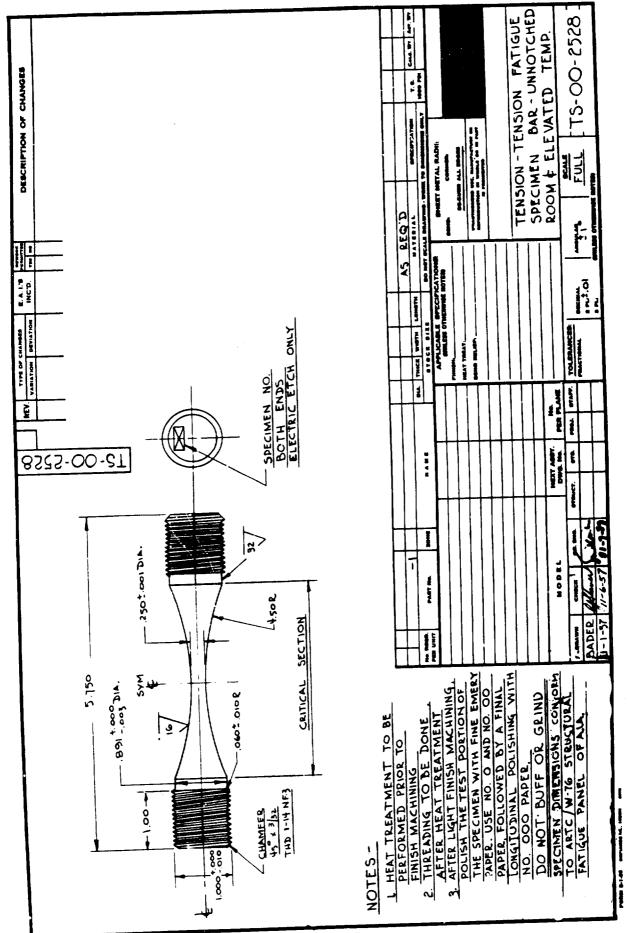
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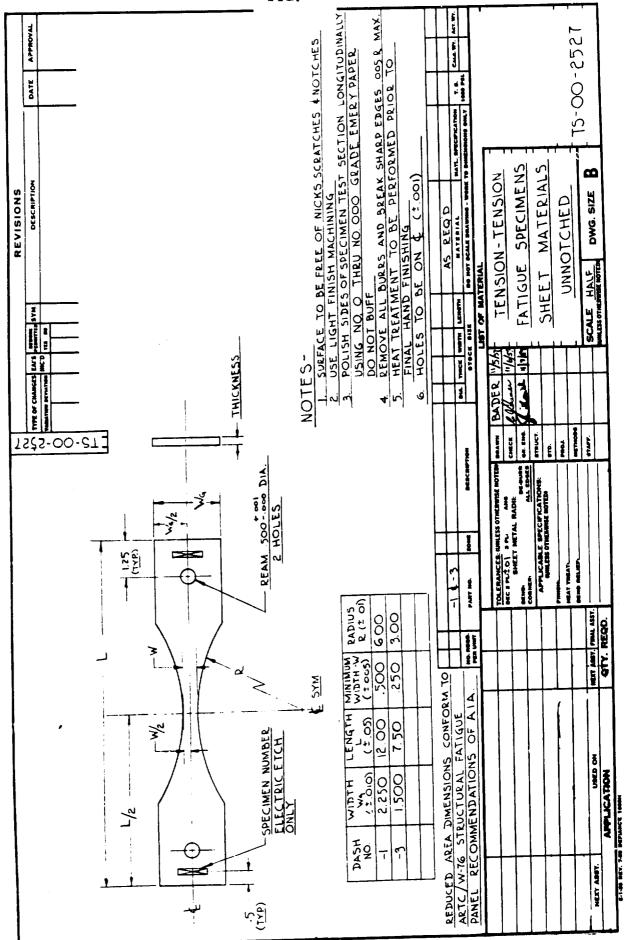




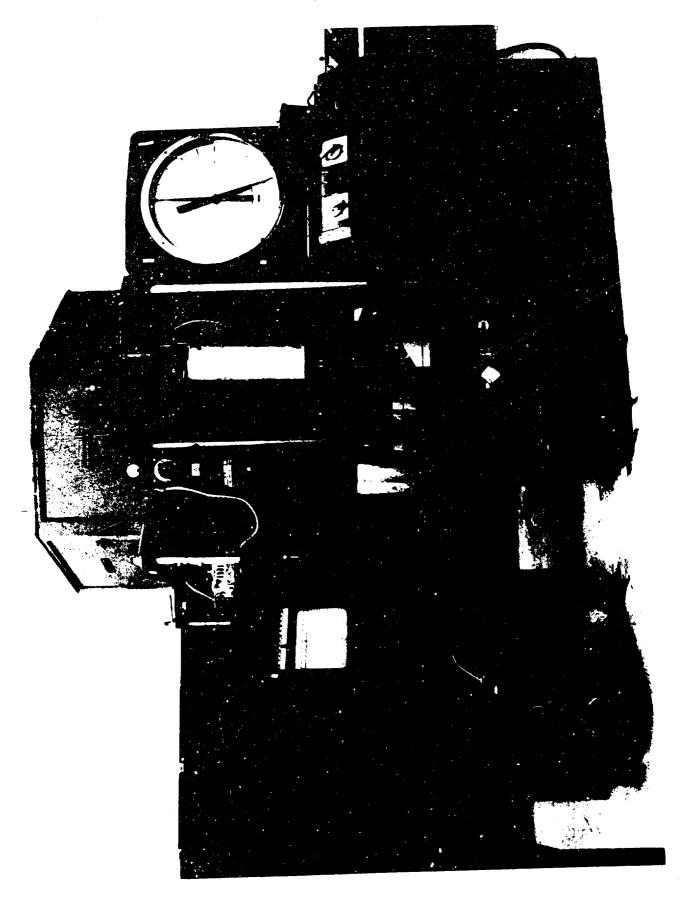
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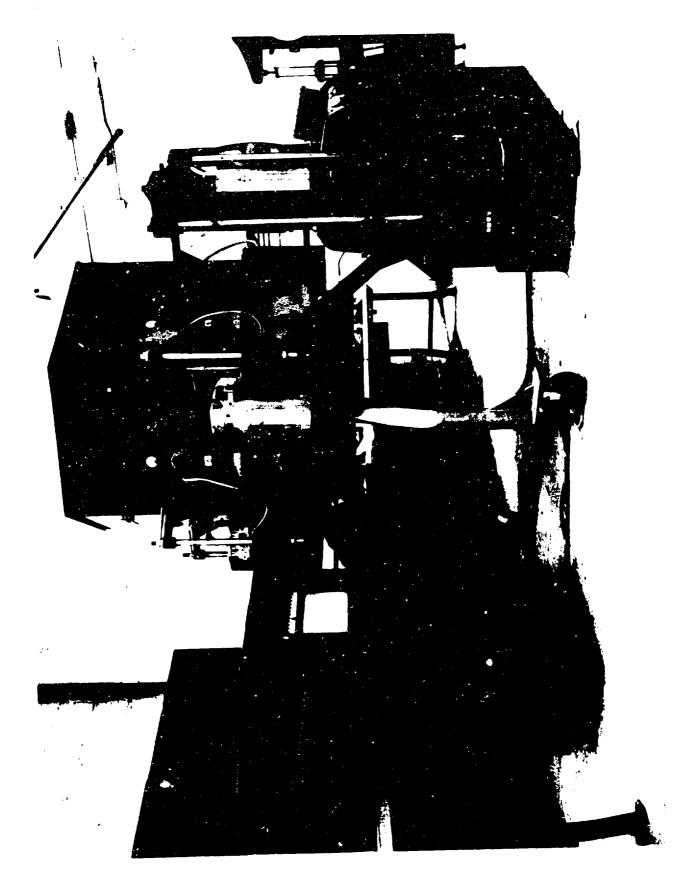


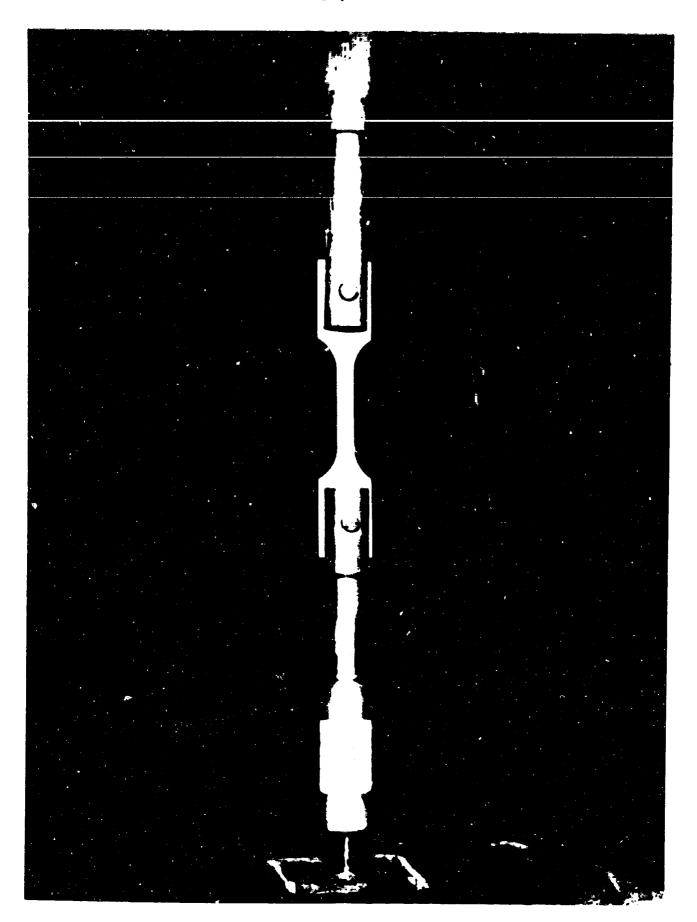


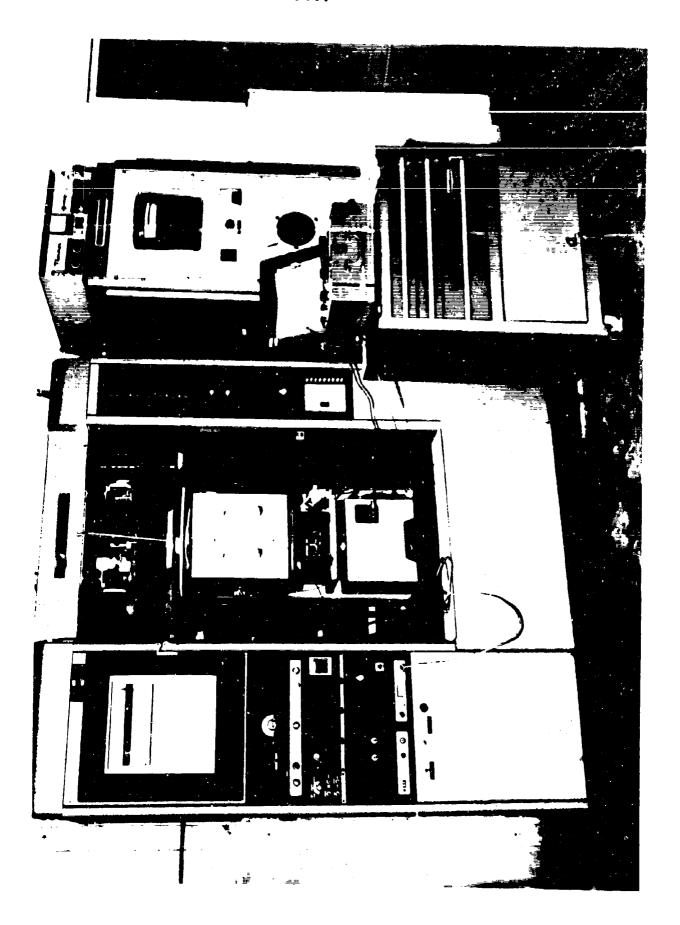


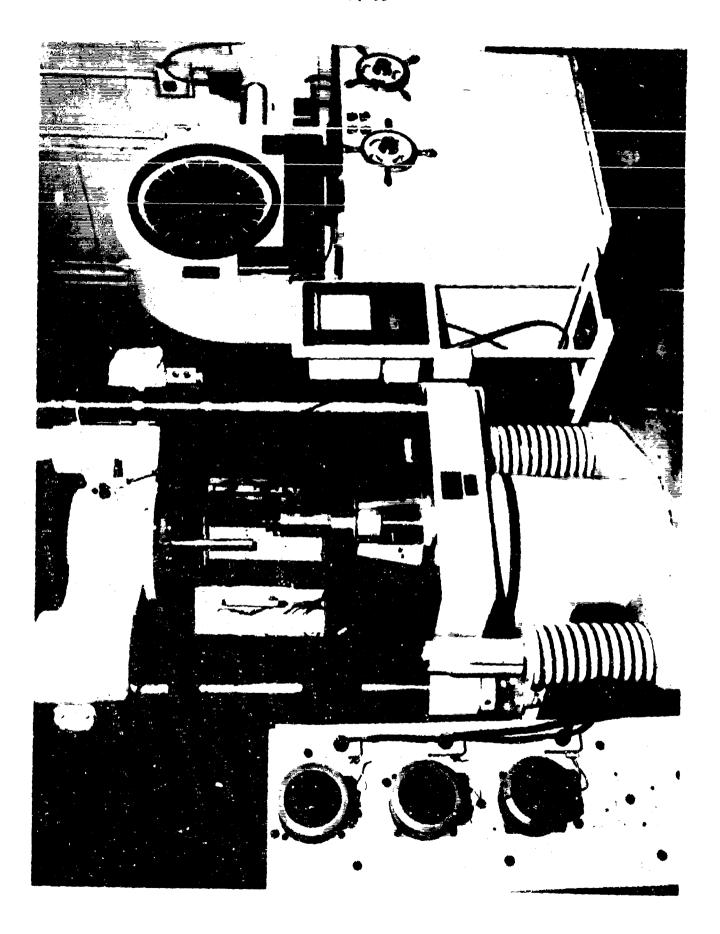
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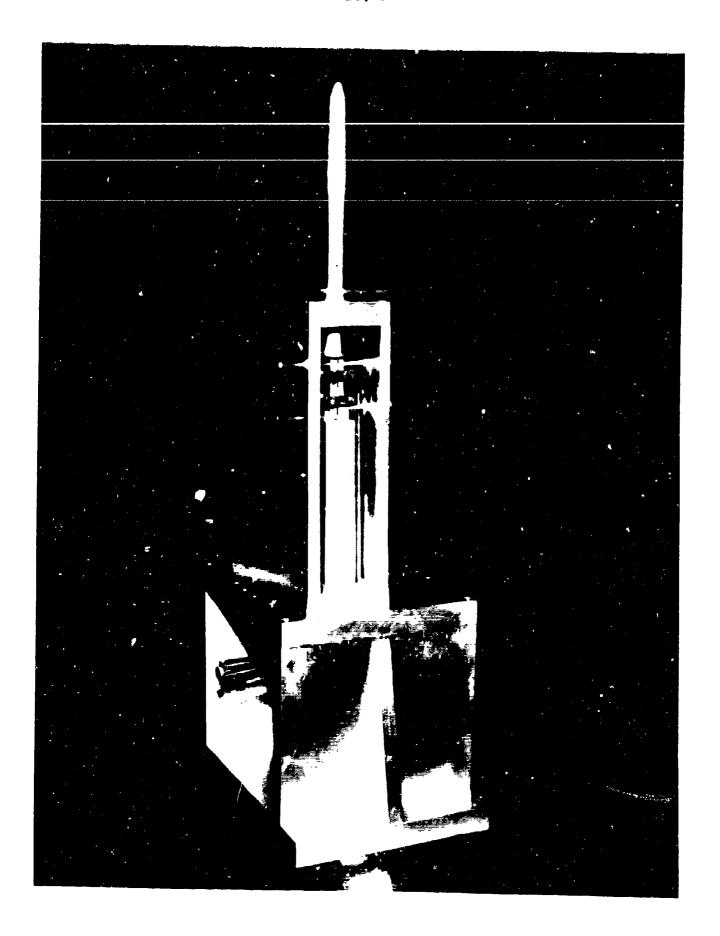


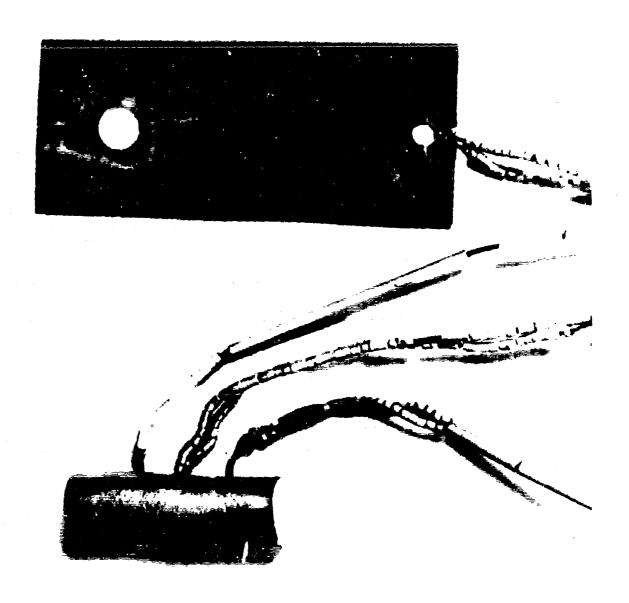


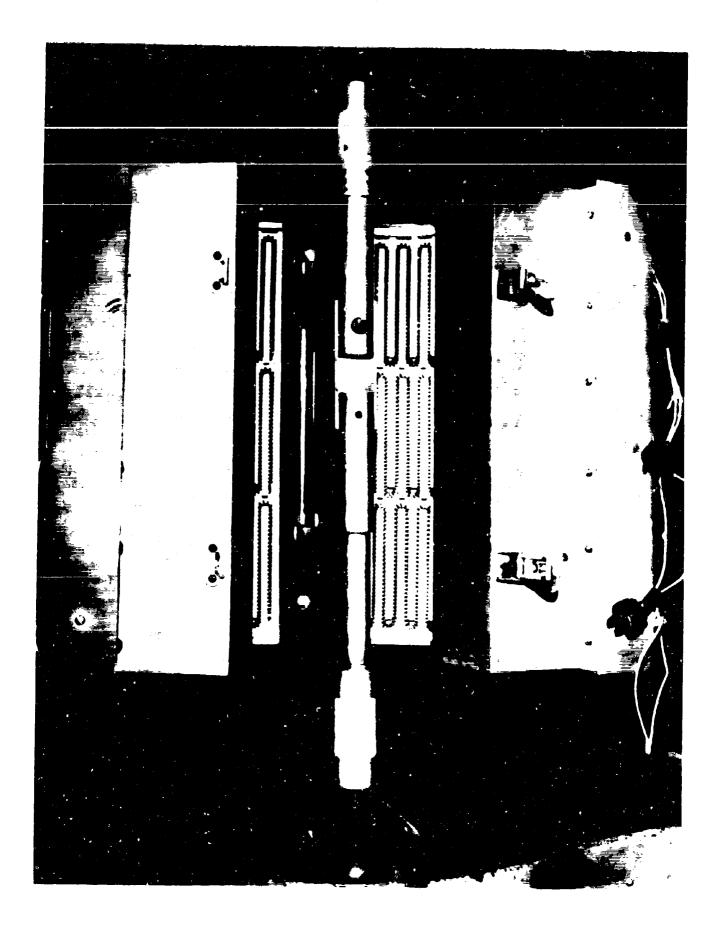


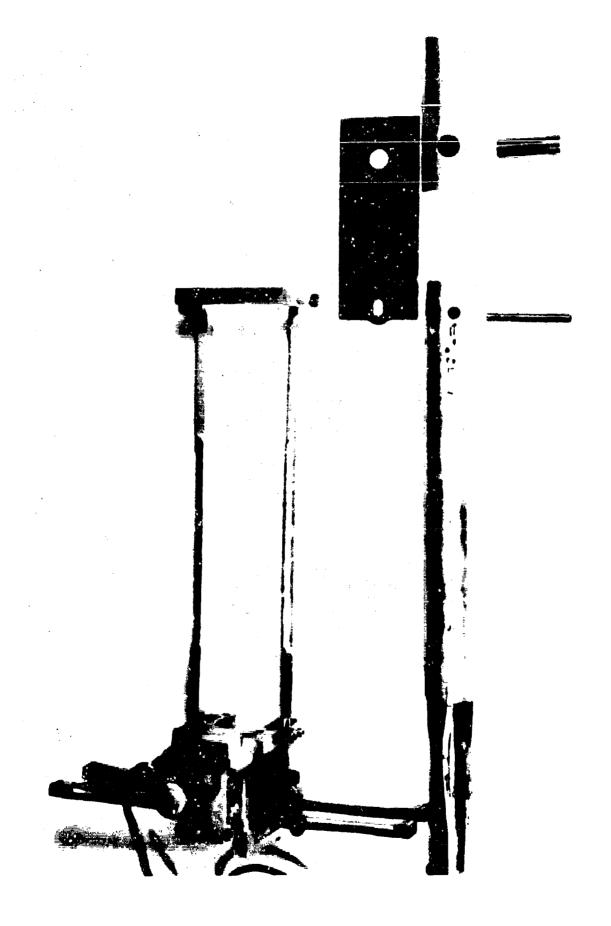


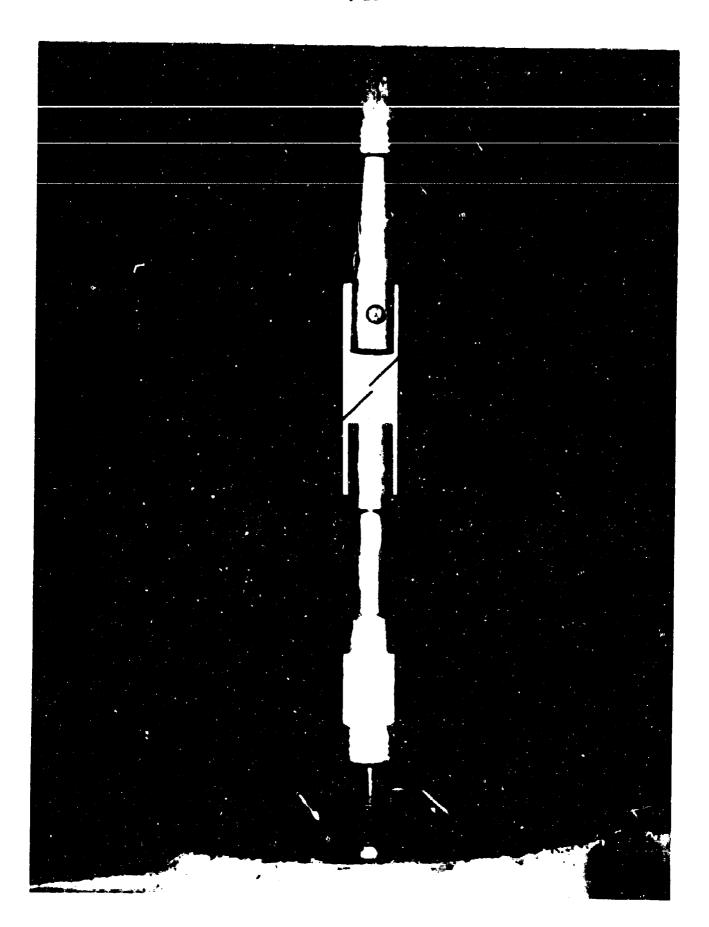


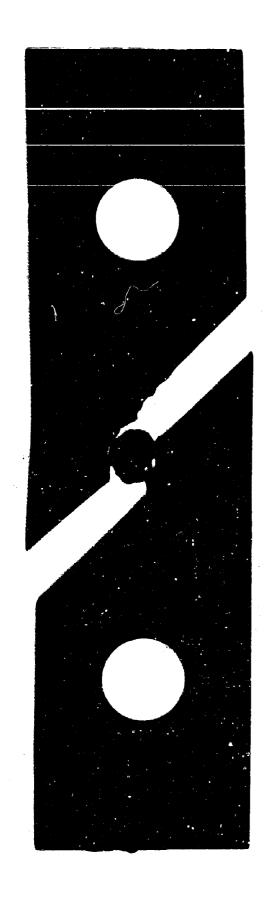


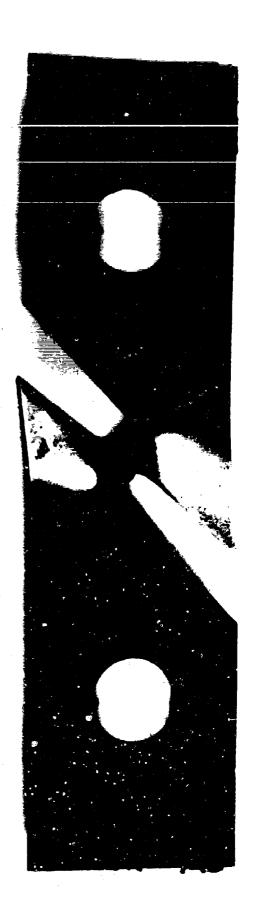


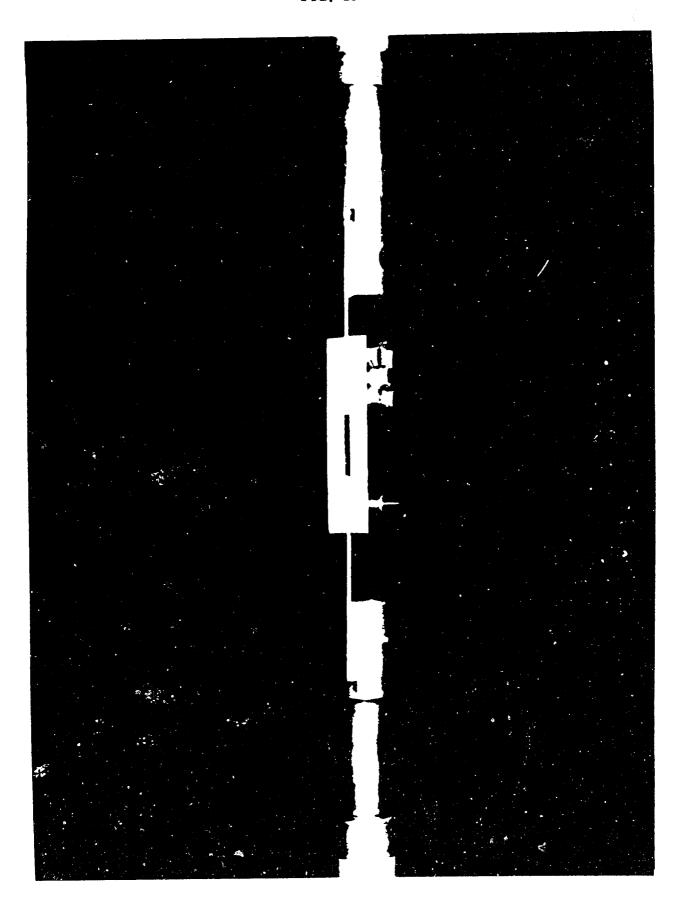


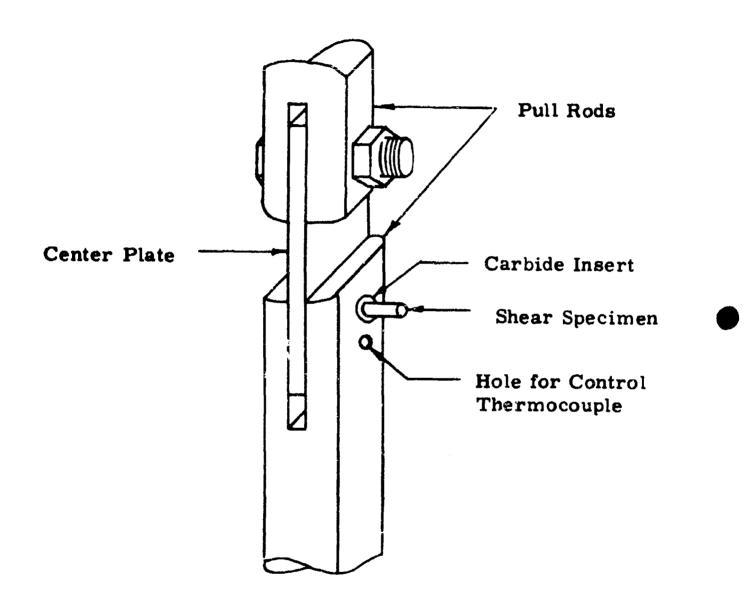


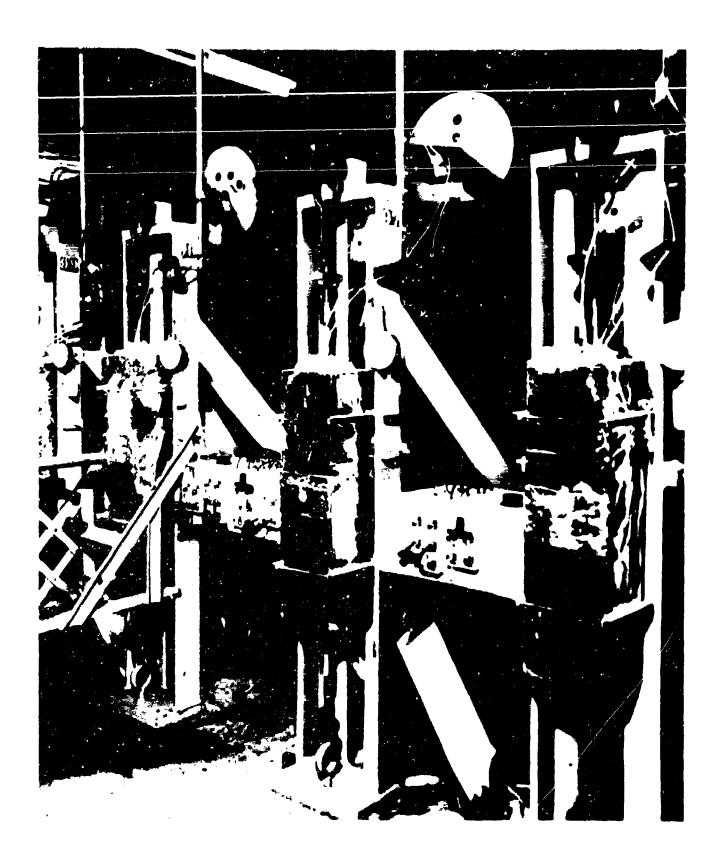


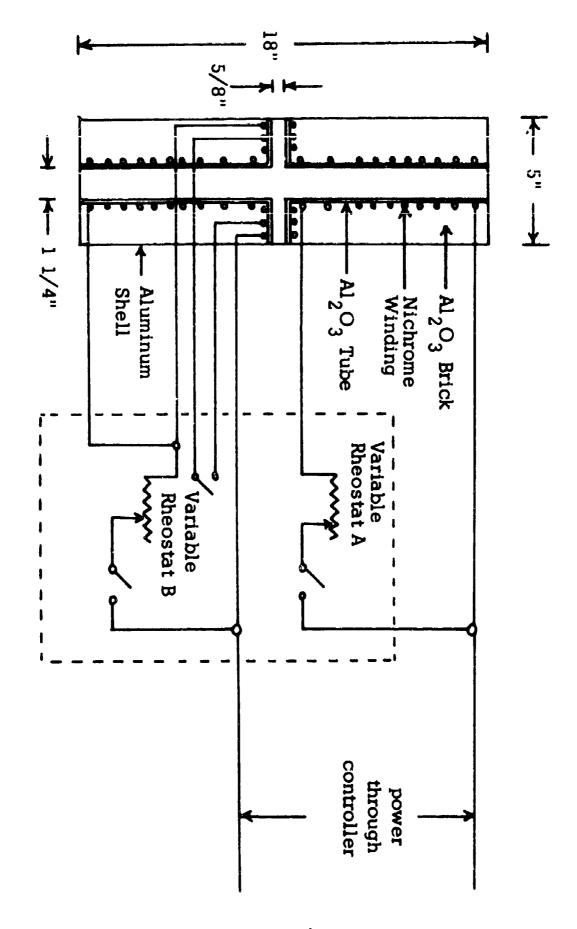


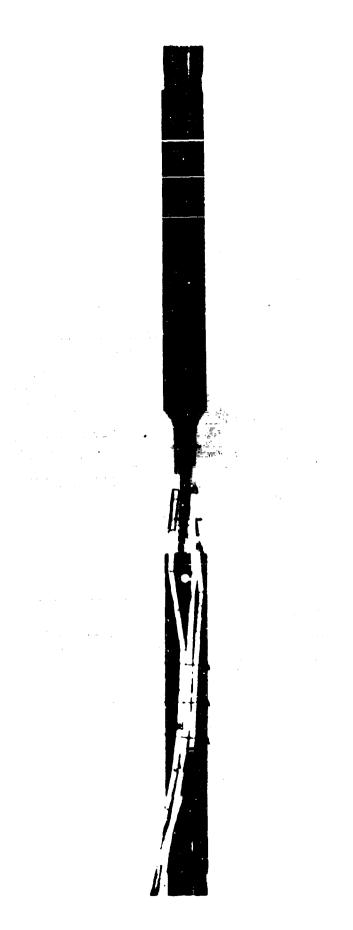




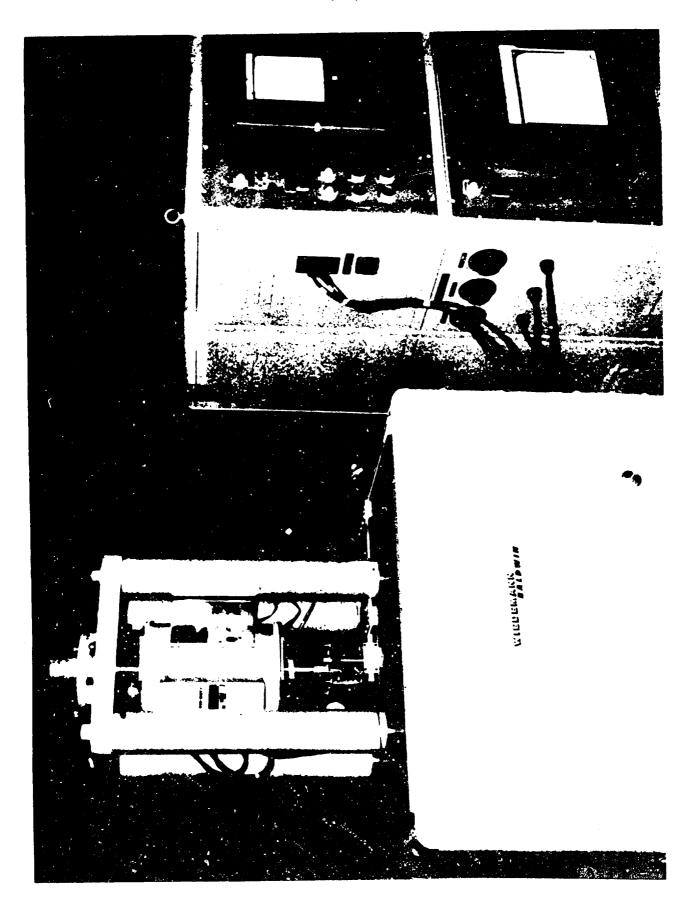


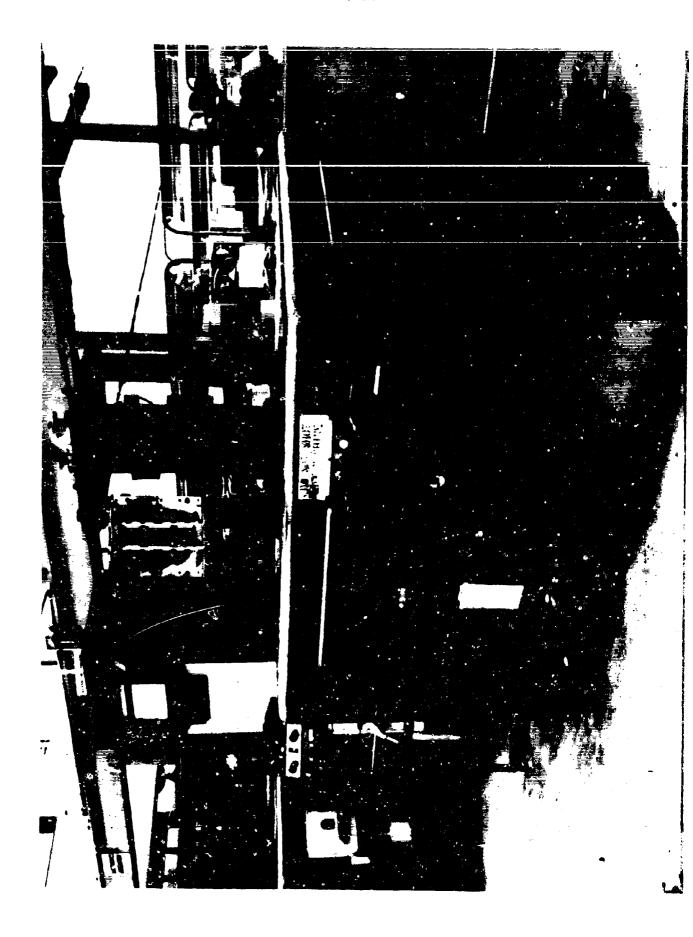


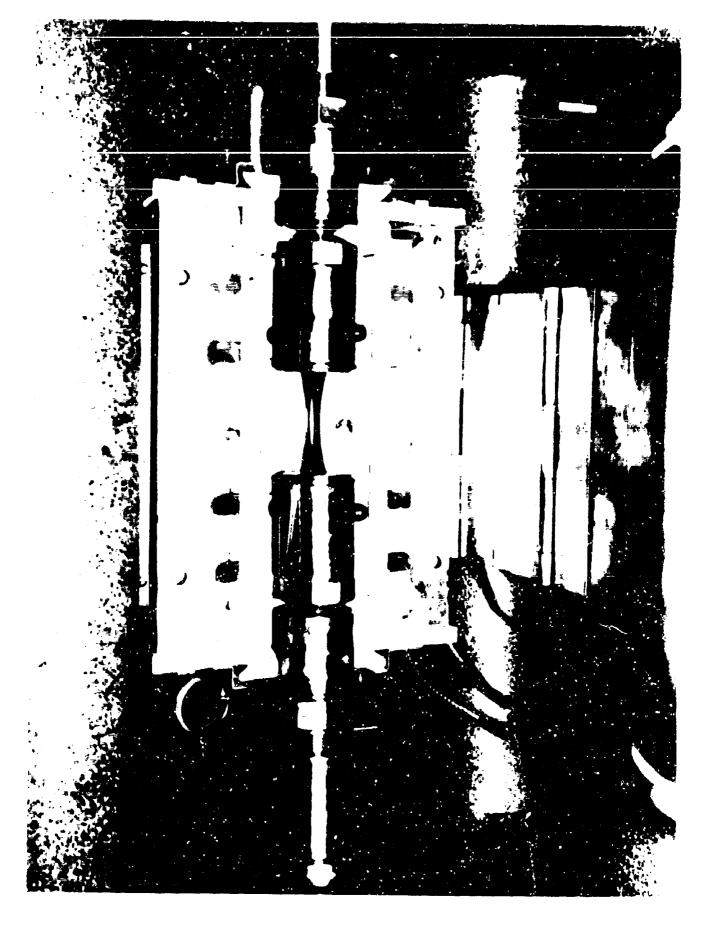


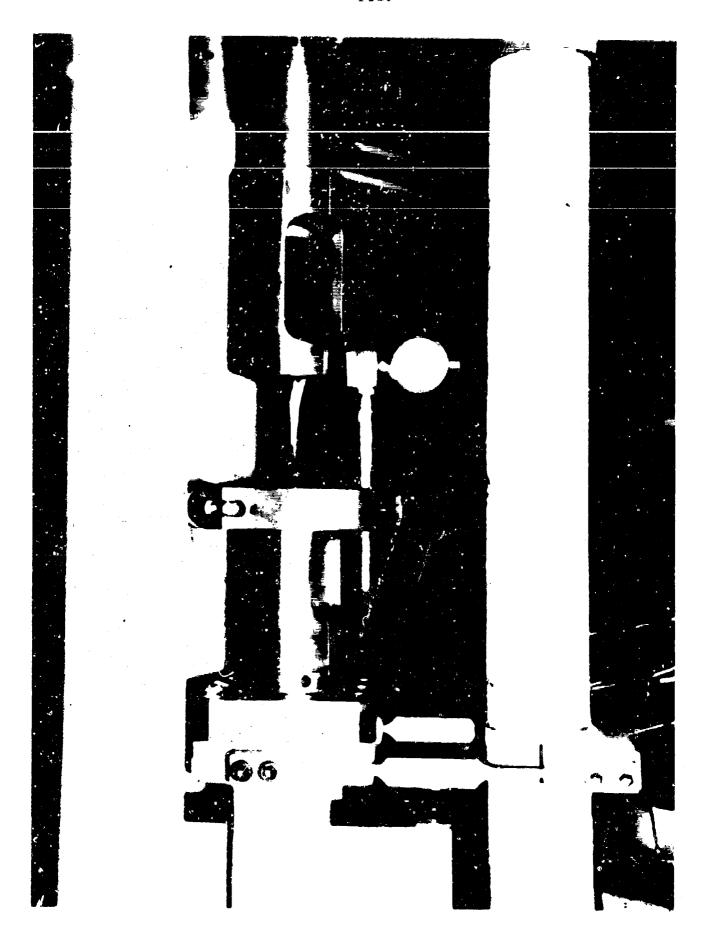












# SECTION VI - SUMMARY OF TEST RESULTS

#### SECTION VI - SUMMARY OF TEST RESULTS

#### 6.1 Methods of Statistical Analysis

This analysis was accomplished in accordance with PROFOSED REVISIONS OF, OR ADDITION TO, CHAPTER 1 - MIL-HDBK-5(Revised March 1961) - ATTACHMENT 59-29(23), Battelle Memorial Institute.

The mechanical properties presented herein are identified by a letter (i.e., A or B) to indicate the basis upon which they were established. An 'A' value is the property above which 99 per cent of the population is expected to fall with a confidence of 95 per cent. A 'B' value is the property above which 90 per cent of the population is expected to fall with a confidence of 95 per cent.

There are two methods of obtaining these values and they are:

Directly Calculated Values - the directly calculated 'A' а. values are obtained as follows.

$$^{\dagger}A^{\dagger} \text{ value} = \bar{x} - KS_{x} \tag{1}$$

where 
$$\bar{x} = \frac{\sum x}{n}$$
, (2)

where 
$$\bar{x} = \frac{\sum x}{n}$$
, (2)  
 $S_{x} = \sqrt{\frac{\sum (x - \bar{x})^{2}}{(n-1)}}$ 

where  $\bar{x}$  is the average value of individual measurements,  $S_{x}$  the standard deviation of individual measurements, n the number of individual measurements and K, the one-sided tolerance factor for normal distribution and specified probability, confidence, and population (i.e., for 'A',  $K = K_{,99}, _{,95}, _{,9}$ 

The 'B' values are calculated as follows:

'B1 value = 
$$\bar{x} - KS_{x}$$
 (1a)  
where  $K = K_{.90}$ , .95, n.

The values of K were obtained from the table 'One-sided Tolerance Factors for the Normal Distribution and a Confidence & of .951, in Tables of Normal Probability Functions, N.B.S., Applied Mathematics Serics 23, (1953).

An additional requirement is that the population, n, must consist of at least 100 points from a minimum of ten different heats of material. Because of the paucity of available data, this requirement usually can be satisfied for room temperature tensile ultimate and yield only.

b. Derived Values - these values are established through their relationship to directly calculated 'A' or 'B' values of  $F_{tu}$  or  $F_{ty}$  as obtained in the foregoing section. This method consists of pairing individual ultimate strength values (i.e.,  $F_{tu}$ ,  $F_{su}$ ,  $F_{bru}$ ) with individual tensile ultimate strength values, or individual yield strength values (i.e.,  $F_{ty}$ ,  $F_{bry}$ ) with individual tensile yield strength values, determining the mean ratio of these pairs with a probability of 95 per cent and multiplying the directly calculated 'A' or 'B' values of  $F_{tu}$  or  $F_{ty}$  by this factor. Derived values are therefore equal to:

$$(-t_{.05}S_{r})$$
  $F_{tu}$  (A or B) (4)

or 
$$(\bar{r} - t_{.05}S_{\bar{r}}) F_{ty} (A \text{ or } B)$$
 (5)

where 
$$\bar{r} = \frac{\sum r}{n}$$
 (6)

$$S_{\overline{r}} = \sqrt{\frac{\sum (r - \overline{r})^2}{n (n - 1)}}$$
 (7)

and t.05 is the two-sided tolerance factor for the 't' distribution, a probability of 95 per cent and the population, n, involved. The values of t.05 were obtained from a table in Statistical Methods for Research Workers by R. A. Fisher. The derived values of the mechanical properties have the same validity (A or B) as the values of  $F_{tu}$  or  $F_{ty}$  used in equations (4) and (5). Ten pairs of measurements (n = 10) are the minimum for establishing a derived allowable.

Statistical methods were used to establish the design allowables presented in this report. The testing performed for this program did not include specimens from a sufficient number of different heats to permit direct calculation of any of the values. Additional data was obtained from material vendors and other sources, such as inspection acceptance data, to make up this deficit. The extra data consisted of results from many heats in different forms; in many cases one specimen from each form was reported. Using the data available from all sources with each specimen as a point in the population would bias the results in favor of the heats containing a large number of data points, such as the heats tested in this program. To prevent this, a slight change was made in the method of directly calculating the allowables. The change consisted of using the average value of all the specimens in a heat-form combination as a single point in the population. The population then becomes the number of heat-form combinations instead of the number of individual specimens. The tolerance factor, K, was still chosen based upon the number of specimens. This procedure is justified by the

results obtained under this contract which show that, except for the foil gauges, there is a relatively small difference in properties from sheet to sheet, plate to plate, bar to bar, or forging to forging within a heat.

The values of all the mechanical properties of Rene' 41 (plate, bar, and forgings), Inconel 702 (sheet), and the shear ultimate of L-605 and Incoley 901 (bar) are not statistically sound 'A' and 'B' values, and therefore are reported as tentative in the tables of MIL-HDBK-5 data.

Tables 17 to 22 indicate the populations and other pertinent information used in the calculation of the allowables.

# 6.1.1 LIST OF STATISTICAL SYMBOLS

Á	'A' basis for mechanical property values
В	'B' basis for mechanical property values
К	one-sided tolerance factor for the normal distribution and the specified probability, confidence and population
n	the number of individual measurements or paired measurements - population
r	ratio of two paired measurements
<del>-</del>	the average ratio of paired measurements
S-r	standard error of paired measurement ratios
S <sub>x</sub>	standard deviation of individual measurements
t	two-sided tolerance factor for the 't' distribution and the specified probability and population
x	value of an individual measurement
x	the average value of individual measurements
Σ	the summation of

Populations Used to Calculate A and B Values

•							Number of		
Material Form		Values	,	Heat-Porm				Different	
& Property Rene'41 Foil	Direction	Obtained	Method	Combinations	Paire	Heats	Vendors	Thicknesses	Specimens
Ftu	u	45	Derived	•	55	ო		2	55
	H	¥ ¥	Direct	94	•	45	6	e	306
Pty	ų	45	Derived	1	55	m		2	55
•	H	A & B	Direct	97	•	45	ю	e	306
Sheet									
Ftu	u	4	Derived		98	m	-	٣	86
	H	A & B	Direct	506		<b>50</b> ¢	ς,	7	1089
Fty	1	拍 5 V	Derived		85	e	1	m	85
•	H	A A	Direct	506		<b>50</b> %	2	7	1089
Fcy	ы	At & B*	Derived	í	79	ო	pomel	<sub>,</sub> м	79
	H			•	87	m	-	m	87
10 m	ħ			•	9/	ю	-	m	76
2	H			•	83	m	-	m	83
Fbru(@/D=1.5)				ì	8	m (	<b>-</b>	<b>-</b>	30
	<b>.</b>			•	78	m	-	m	78
Fbry (e/D=1.5)				•	8	m (		<b>~</b> (	8
	H			ı	9/	m	~	m	76
Pbru (e/D=2.0)				•	30	٣	7	m	90
	H			1	<b>8</b> 3	m	-	m	<b>8</b> 9
Pbry (e/D=2.0)	) r			•	8	٣	-	-	30
).		At 6 Bt	Derived	ı	82	m	<b>-</b>	m	82

TABLE 18

Populations Used to Calculate A and B Values

							Number of		
Material Form		Values		Heat-Porm				Different	
& Property Rene'41	Direction	Obtained	Hethod	Combinations	Paire	Heate	Vendors	Thicknesses	Specimens
Plate, Ber									
Ftu	ı	Al & Bl	Direct	15	ı	•		5	145
Fty	1	Al & Bl	Direct	15	ı	9	1	5	145
Fcy	1	A <sup>2</sup> & B <sup>2</sup>	Derived	•	105	•		4	105
Plate & Forging Ptu	H	A <sup>2</sup> & B <sup>2</sup>	Derived	t	<b>78</b>	ý	1	m	48
Pty	Ħ			•	82	9	1	n	82
Fcy	H			•	53	٠	<b>,-4</b>	2	53
Par	႕ 뉴				33	99		88	0 7 7
Pbru(e/D-1.5)	5) T			•	20	М	-	1	20
Fbry (e/D=1.5)	5) T			•	20	٣		1	20
Fbru(e/D=2.0)	0) T				25	٣	-	-	25
Fbry (e/D-2.0)	D T	A <sup>2</sup> & B <sup>2</sup>		•	25	ю		-	25
Bar Fau	H	A3 & B3	Derived		8	7	<b>~</b>	7	30

Populations Used to Calculate A and B Values

						No.	Number of		
		Values		Heat-Porm				Different	
ırty	Direction	Obtained	Method	Combinations	Paire	Heats	Vendors	Thicknesses	Specimens
Sheet	H	A* 6. B*	Derived	•	69	m	<b>,</b>	(r	9
;	н		Direct	211		80	-	ı m	370
n M	-1	A* 6 B*	Derived	•	69	ო	-	m	649
î	H	<b>Q 3</b>	Direct	211		80		) m	370
FC.	H	A* G B*	Derived	•	69	m	<b>,-4</b>	m	69
•	H			•	69	ო	-	m	69
Fou	<b>-1</b>			•	27	m	1	7	27
	H			•	ဇ္ဇ	m	~	7	30
Fbru (e/D=1.5)	<b>1</b>			•	20	m	<b>~</b>	1	20
				•	3	m	-	m	8
Fbry(e/D=1.5)	H (			ı	19	m (	<b>~</b> ,	<b></b>	19
	H				<b>3</b>	m	<b>-</b>	m	3
Pbru (e/D-2.0)	٠ ١			•	91	m	<b>,-4</b> -	<b></b> (	16
	•				<b>ò</b>	า	4	า	ò
Pbry (e/D-2.0)	1			•	16	e	7	-	16
•		4.5	Derived	•	29	m	-	m	67
Plate, ler &									
Ftu	-1	A & B	Direct	181	•	92	-	11	305
Fty	u	A & B	Direct	181	•	92	-	11	305
Fcy	u	4 6 B	Derived	•	69	•		m	69

TABLE 20

Populations Used to Calculate A and B Values

								Number of		
	Material Porm		Values		Heat-Form				Different	
. ,	& Property L-605	Direction Obtained	Obtained	Method	Combinations	Pairs	Heats	Vendors	Thicknesses	Specimens
	Plate &									
	Ftu	H	A & B	Direct	87	•	78	-	10	174
	Fty	H	A & B	Direct	87	ı	78	<b>~</b>	10	174
	Fcy	卢႕	At 6 Bt	Derived		39 45	99	<b></b>	77	39
	Fau	H	A* 6. B*		•	45	9		2	45
9	Pbru(e/D=1.5)	F	A* 6 B*		•	12	м	1		12
1	Pbry(e/D-1.5)	H	A* & B*		•	12	m	1	1	12
	Ber	Ħ	A4 & B4	Derived	ı	30	m		7	30

TABLE 21

Populations Used to Calculate A and B Values

								Number of		
£	Material Form		Values		Heat-Porm				Different	
1177	& Property Incomel 702	Direction	Obtained	Hethod	Combinetions	Paire	Heats	Vendors	Thicknesses	Specimens
	Pru	,,	A <sup>2</sup> & B <sup>2</sup>	Derived	ı	57	ю	<b></b> 1	7	27
		H	Al & Bl	Direct	29	ı	19	1	<b>6</b> 0	82
	<b>7</b>	ы	A2 & B2	Derived	•	57	ю	-4	7	57
	,	H	Al & Bl	Direct	29		19	7	80	82
	Fcy	ы	A2 & B2	Derived	•	24	m	-	2	<b>አ</b>
	•	H			ı	99	ო		7	26
	200	H			•	84	m	-	2	84
		H			•	27	m	-	7	27
0.	Pbru(e/D-1.5)	H			1	53	ო	1	1	29
,	1	H			•	27	m	-	7	57
	Pbry(e/D=1.5)	ы			•	29	e	-	-	53
		H			•	23	e	7	8	57
	Pbru(e/D-2.0)				•	29	<b>6</b>	<b></b> 1	8	29
		H			•	27	m	<del></del>	7	57
	Fbry(e/D-2.0)		•		1	29	m	1	-	29
		H	A <sup>2</sup> & B <sup>2</sup>	Derived	•	27	က	1	7	57

TABLE 22

				Populations L	Populations Used to Calculate A and B Values	A And B		,		
	Material Form		Values		Heat-Form			Number of	Different	
	& Property Incoloy 901	Direction	91	Nethod	Combinations	Pairs	Heats	Vendors	Thicknesses	Specimens
	Bar 6 Forging									
	Ptu	H	A & B	Direct	97	1	39	4	٣	166
	Fty	H	A & B	Direct	97	ı	39	4	m	166
	Fcy	ы	A* & B*	Derived	•	88	E	-	٣	88
	Forging Ftu	<b>£</b> 4			•	30	m	<del></del> 4	1	30
Ć	Fty	H			•	30	e	~		30
93	Fcy	Ħ			ı	8	m	<b>,</b> i	<b>~</b> 4	30
	n e Ne	보	At & Bt			<b>3</b> 31	m m			21 23
	Ber	Ħ	A4 & B4	Derived	•	59	m	1	7	\$9

NOTES:

A, B Directly calculated values (as described in text).

A\*, B\* Derived values (as described in text).

 $A^1$ ,  $B^1$ 

Value obtained using same procedure as directly calculated values except that the the population was not large enough. These values are not statistically sound A data did not cover the minimum number of different heats, (i.e., ten), or else and B values.

Derived values based on A<sup>1</sup>, B<sup>1</sup> values of F<sub>tu</sub> and F<sub>ty</sub>. These values are not statistically sound A and B values.

 $A^2$ ,  $B^2$ 

transverse direction was paired with tensile ultimate data from the longitudinal Values obtained in manner similar to derived values except that data from the direction. These values are not statistically sound A and B values.

transverse direction was apired with tensile ultimate data from the longitudinal Values obtained in manner similar to derived values except that data from the direction. These values are not statisitically sound A and B values.

A<sup>3</sup>, B<sup>3</sup>

# 6.2 Data Presentation

# 6.2.! Effect of Temperature on Strength (1/2 Hour Exposure) -

These curves are presented as 'Per Cent Strength at Room Temperature vs Test Temperature'. The procedure used to obtain these curves is as follows.

- a. Plot the range of values (i.e., minimum to maximum) for the property at each temperature. Note: The elevated temperature tests were run on specimens from one heat only, and therefore the room temperature range is plotted for this heat only.
  - b. Indicate the average value at each temperature.
- c. Indicate the value 5 per cent above the minimum value at each temperature.
- d. Draw the curve passing through the average or 5 per cent above minimum value whichever is lowest at each temperature.
- e. Obtain the curve value at each temperature as a percentage of the curve value at room temperature.
  - f. Plot the per cent values and fit the curve.

To obtain a smooth curve in step (d), engineering judgement was used and the curves do not necessarily pass through the stated values at each temperature.

6.2.2 Effect of Exposure at Elevated Temperature on the Elevated Temperature Strength -

These curves are presented as 'Per Cent Strength of Room Temperature vs Temperature'; they show the effect of exposure time at temperature on the elevated temperature properties. Each of these figures contains a number of curves - one for each exposure time. The curves are drawn for each exposure time using the same technique as described in the foregoing section, and are plotted on the same graph for easy comparison.

6.2.3 Effect of Exposure at Elevated Temperature on Room Temperature Properties -

These curves are presented in 'Per Cent Strength at Room Temperature vs Exposure Temperature'; they show the effect on the properties of specimens tested at room temperature after exposure to elevated temperature. These curves are drawn using the same technique described in the preceding sections.

### 6.2.4 Stress-Strain Carves

The method used to obtain these curves is as follows:

- a. A smooth, weil defined curve typical of those obtained during testing was selected and repictted on regular graph paper.
- b. Percentages were taken of the .2 per cent yield strength and the plastic strains required to obtain these values were noted.
- c. The modulus to be used was selected by comparison between data generated in this program and data in published literature.
- d. The 'A' value of the yield strength was used as the room temperature value in these curves; the percentage of the 'A' value to be used at elevated temperature was obtained from the appropriate figure.
- e. The straight line portion of the curve was drawn using the selected modulus and then the remaining portion of the curve was plotted using the plastic strains noted in step (b) and the percentage of the appropriate yield strength.

## 6.3 Room Temperature Tensile Strength Distribution

This investigation did not contemplate conducting an analysis of variance to determine the contribution of such factors as testing, form, vendor, etc. to the overall variance of a given alloy. Such an analysis normally expressed in terms of root mean square deviations of individual observations from their mean would be beyond the scope of this program. However, in order to indicate the manner in which some of these factors may be influencing strength variations, the range of values (the lowest and highest values) are plotted against some of the more significant variables (i.e., heat, form, etc.). The presentation incorporates all data including testing accomplished under this contract as well as data available from other sources.

Only room temperature values are plotted since these data are least likely to be influenced by testing techniques and there seems to be a fairly good correlation between this parameter and other properties such as shear strength, bearing strength, etc. In addition, the elevated temperature strength in a uniform material will normally be proportional to the room temperature strength at least out to some critical temperature range.

This method of presentation has its limitations and in some instances, may lead to some confusion particularly when one considers that a plot of the range of values found within a heat will incorporate (in some instances) the range found from sheet to sheet within the heat as well as the range within a sheet. It does, nevertheless, give a good indication as to which factors are the predominating ones. If, for example, the range of values for a given vendor and gauge thickness is 30,000 psi from heat to heat, 10,000 psi from sheet to sheet within a heat and 5,000 psi with a sheet, it is safe to assume that if the number of observations is adequate in all categories that the greatest variation in strength results from the heat composition.

#### 6.3.1 Material Rene! 41

The range of room temperature tensile properties found within a single sheet, bar or forging are shown in Figure 33. Specimens were taken so as to survey the form in a random manner and only those values are plotted where a sufficient number of specimens were tested to make the data meaningful. The letter designation applies to the heat involved and repetitive letters indicate the time heat. All heats were produced by the General Electric Company.

Figure 33 indicates very little difference, if any, between longitudinal and transverse properties except for the significantly lower transverse strength in 1 x 3 inch forgings. The spread in values is relatively small, particularly in sheet gauges .020 to .080 inches. The AMS specification requirements were satisfied as to room temperature yield strength for all forms. The same is true of the ultimate strength except for one  $1 \times 3$  inch forging from heat F.

The range of room temperature tensile properties found from sheet to sheet within a single heat are presented in Figures 34 thru 27 broken down by Producer. Most of these data were collected by the Quality Control Laboratory of Republic Aviation Corporation as part of incoming inspection of production material over the last two years and as such are considered representative of the current quality of the alloy in sheet form. The number of sheets per heat is indicated for each gauge thickness. In the majority of cases, only one specimen was tested for each sheet and this in the direction transverse to the grain flow.

The maximum spread in values in Vendor A material, Figure 34 occurred in the 13 sheets of .050 inch material; however, the direction of scatter is favorable in that it is toward higher tensile properties (i.e., the lowest points are 140,000 psi yield, 190,000 psi ultimate). All points satisfy AMS specification requirements.

Only .013 and .025 sheets were purchased from Vendor B. The .013 inch material exhibits generally lower properties than the .025 inch material, however, this may be attributed to the difficulty in testing thin gauge material. In any event, all tests satisfy the AMS requirements except for minor deficiencies in the tensile yield (lowest value obtained 127,000 psi). Vendor C material (.020, .025, and .032 inch sheet) shows remarkedly little scatter. All values satisfy AMS yield and ultimate strength requirements. Ranges for V ndors B and C are shown in Figures 35 and 36 respectively.

Vendor D material (.020, .025, .032 and .050 inch) exhibits a normal scatter pattern except where the spread is large (i.e., the 21 sheats

from the .025 inch material), the values extend significantly below the AMS minimum requirements. Some of the actual values obtained which are cypical of this scatter when it occurs are shown in Table 23. The remarkable feature of this data is the consistency within a given sheet, as well as the consistency from sheet to sheet among the rejects. Rejects are generally tested with two additional specimens from the same sheet so that repetitive sheet numbers in the table represent the same sheet. Figure 37 shows range of values.

Figure 38 shows the variation from heat to heat, analyzed by vendor and gauge, and includes all data points available within a given heat. The number of heats per gauge are shown in parenthesis. All values from material produced by Vendors A, B, and C were above the specification minimums with the exception of the .013 inch material from Vendor B. In this latter case, approximately 3 sheets out of 233 did not satisfy the 130,000 psi minimum yield strength (the lowest value obtained being 123,000 psi). Vendor D material as anticipated shows the lowest values for .025, .032 and .050 inch material; the .020 and .070 inch sheet from the same vendor does not exhibit this deficiency. While the material which did not meet the strength requirements of the applicable AMS specification was rejected by Quality Control, the values obtained were used in the calculation for 'A' and 'B' properties.

It is interesting to note the erroneous conclusions that can result if a full picture of material variation is not known when a calculation is made for 'A' and 'B' values for MIL HDBK-5. For example, there were approximately 90 data points from 8 heats available for establishing room temperature tensile strength in the longitudinal direction. Since this population is very close to the minimum requirements of Reference 114, the 'A' and 'B' values directly calculated would give the following:

		A	В
$\mathbf{F}_{tu}$	Long.	188.2	193.5
$\mathbf{F}_{\mathbf{ty}}$	Long	139.9	145.5

The 'A' and 'B' values directly calculated for the transverse direction are as follows:

		A	B
$\mathbf{F}_{\mathbf{tu}}$	Trans.	177.7	185.8
F <sub>ty</sub>	Trans.	123.6	134.0

Figure 38 however, shows that there is very little difference between longitudinal and transverse properties. The reason for the discrepancy is that all of the data for the longitudinal analysis came from 2 of 3 vendors with demonstrated

superior tensile strength while the transverse data includes the low values from Vendor D (longitudinal properties were not available from Vendor D). A substantial upgrading in minimums would have resulted if these later values were eliminated from the transverse 'A' and 'B' analysis. For this reason, the longitudinal 'A' and 'B' values reported were derived from the transverse values as being more representative of the actual strengths.

While this presentation only considers room temperature tensile data, the trend established persists at least up to 1400°F. Typical values for room temperature versus 1400°F are presented in Tables 23 for Vendor D material.

#### 6.3.2 Material L-605

The range of room temperature tensile properties found within a single sheet, plate, bar or forging are shown in Figure 39. The presentation for this figure as well as Figures 40, 41 and 42 follow the same format as Rene! 41.

Referring to Figure 39, the degree of scatter is negligible except for the foil gauges (i.e., .005 and .010 inches). The longitudinal properties are somewhat higher than the transverse in sheet thicknesses and considerably higher in foil. With the exception of the foil gauges, form does not seem to exert any significant influences on strength.

The range of room temperature tensile properties from sheet to sheet within a heat are presented in Figure 40. Once again the foil gauges exhibits the greatest spread in values.

The range of values from heat to heat within a given thickness is shown in Figure 41 for bar, plate and forgings in the longitudinal direction, and this data was collected by the Haynes Stellite Company as part of their quality control inspection. The range of properties for the heats evaluated under this contract are indicated by an 'x' on the graphs.

In general, the uniformity of the alloy is excellent considering the number of heats represented, and all values satisfy the AMS specification requirements.

### 6.3.3 Material Incomel 702

The range of room temperature tensile strength found within a single sheet is shown in Figure 43. All material was purchased from the International Nickel Company. The foil was reduced from .020 to .005 inches by the Hamilton Watch Company.

The .005 inch material is considerably lower in strength than the .020 and .040 inch materials. The .040 inch material is very uniform in strength and does not show any difference in tensile strength between the longitudinal and transverse directions. The .020 inch sheet in this respect is very erratic and somewhat heat dependent. The longitudinal strength is lower in heat C. higher in heat B, and the same in heat A with respect to the transverse direction. The longitudinal and transverse strength are the same for .005 inch foil. The largest spread in values occurs in the .020 inch sheet; however, the tendency is toward higher strength particularly in the longitudinal direction, and the degree appears to be a function of the heat involved. The B heat for example, exhibits the largest spread in both the .020 and .005 inch thicknesses.

The range within a heat and from heat to heat is shown in Figures 44a and b respectively. Again, this data is limited but the spread within a category does not appear to be excessive.

#### 6.3.4 Material Incoloy 901

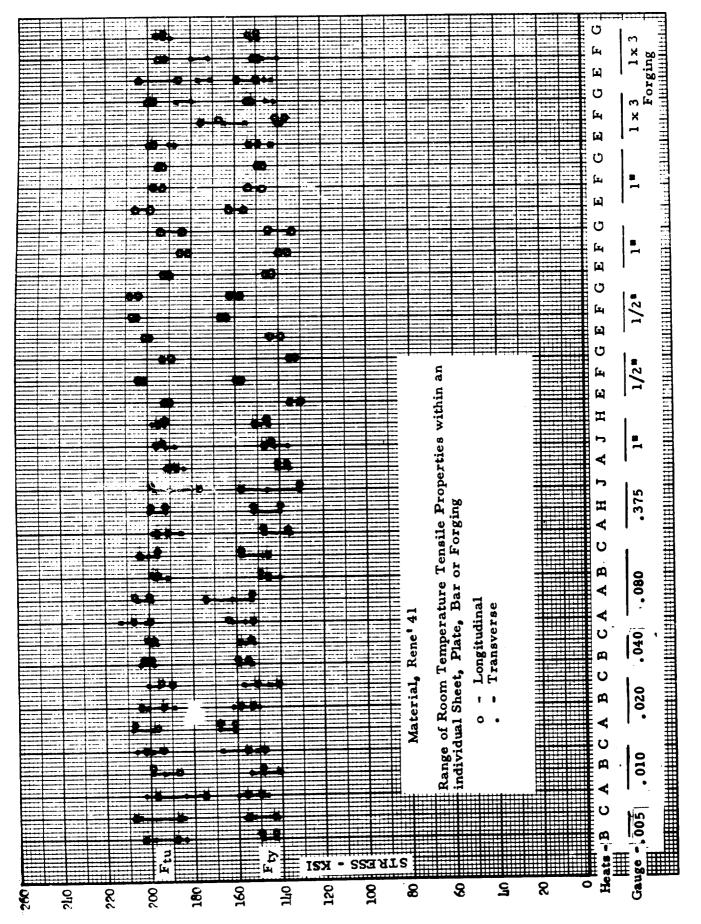
The range of values from room temperature tension and compression yield strengths found within a single bar or forging are shown in Figure 45. The spread appears to depend on the heat analyzed. Heat E exhibits the maximum range in all 3 forms tested (e.g., .5 inch, linch bar, and 1 x 3 inch forging). In general, the strength within a heat appears to be independent of section size. (Heat F is a good illustration.)

Figure 46 presents the range of tensile properties compared on a heat to heat basis for a given section size. The number of heat per gauge is shown in parenthesis. Letter designations are used to represent different vendors which are not the same as those analyzed under Rene! 41. The heat range from Vendor A material includes the variation within a single bar or forging whereas the others do not. This is responsible for the greater range exhibited by Vendor A even though a lesser number of heats were evaluated.

The limited data available for this analysis would suggest that the maximum variation in properties can occur within a given bar or forging, and that the spread is a heat characteristic, the variation from heat to heat does not appear to be excessive.

#### 6.3.4.1 Compression

The range of room temperature compression yield strength within a given heat for Rene' 41, L-605 and Inconel 702 are shown in Figures 47 - 49.



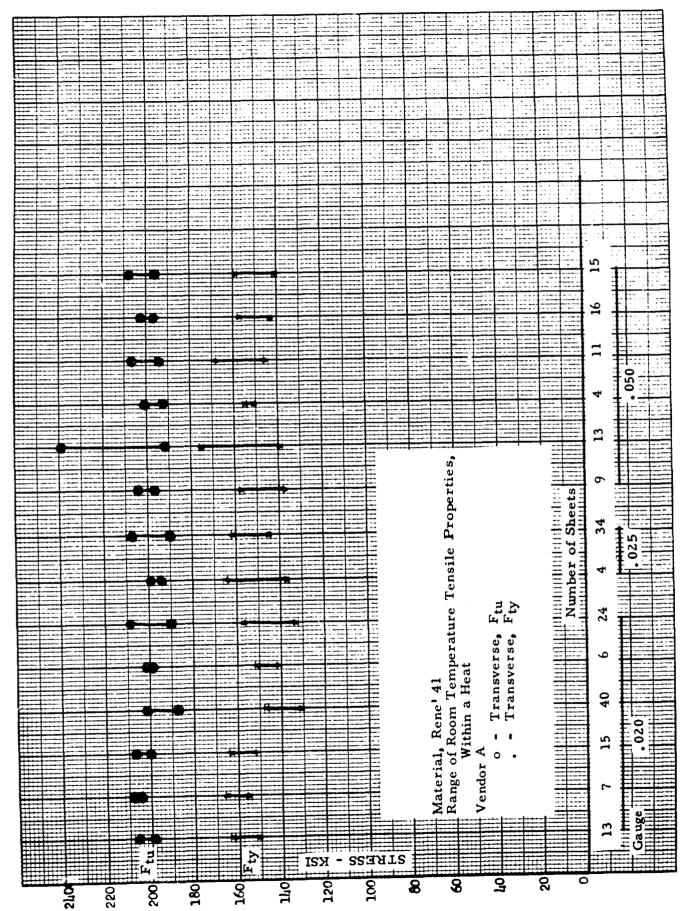
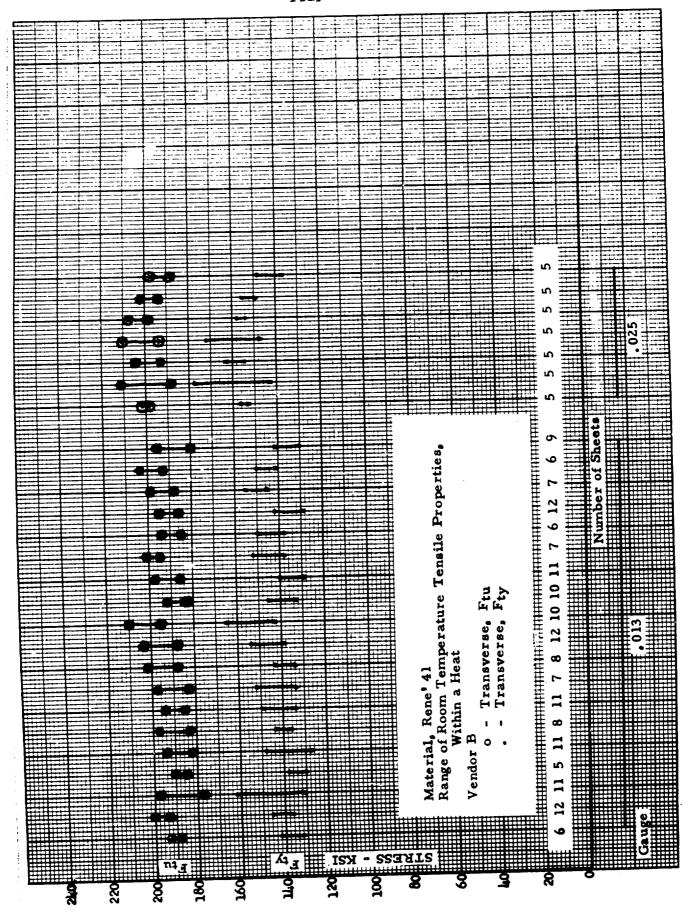
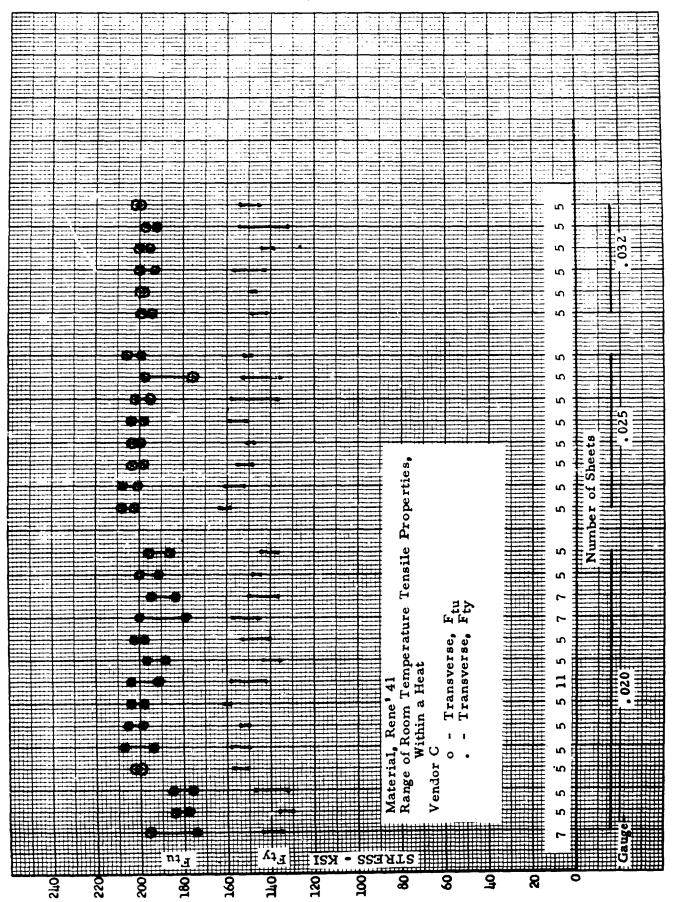
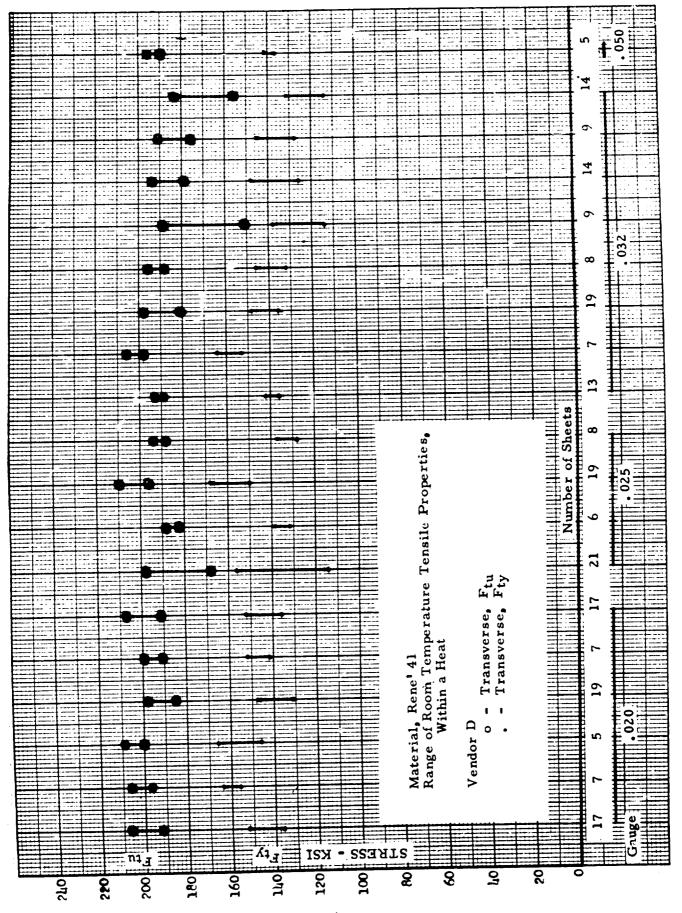
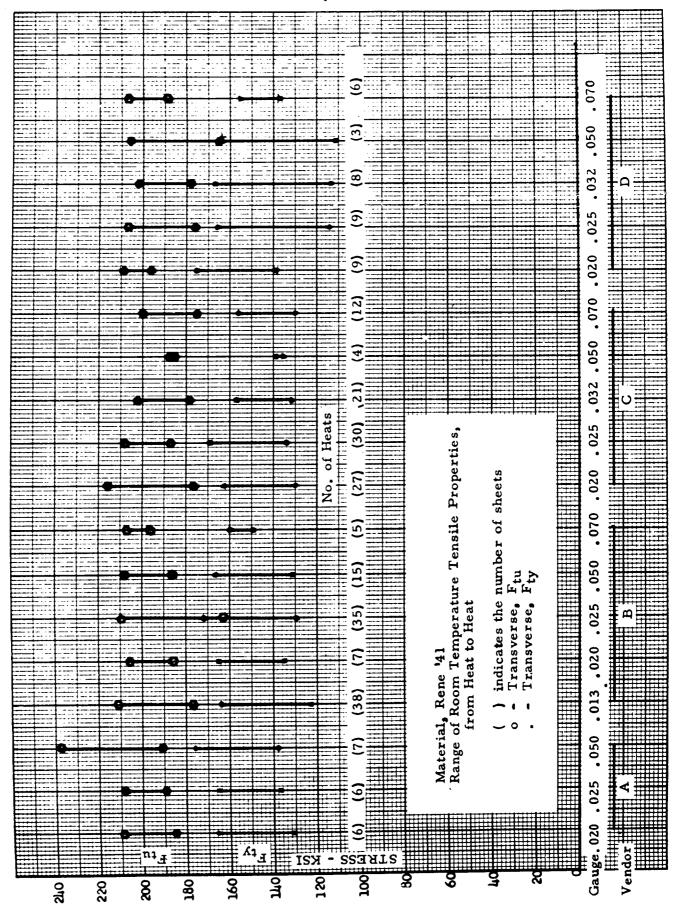


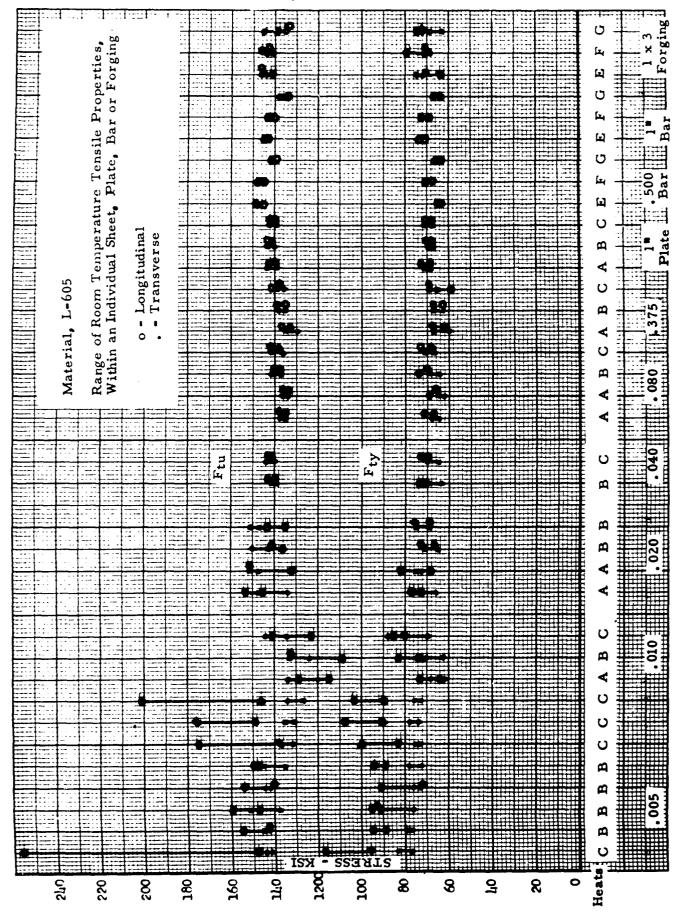
FIG. 35

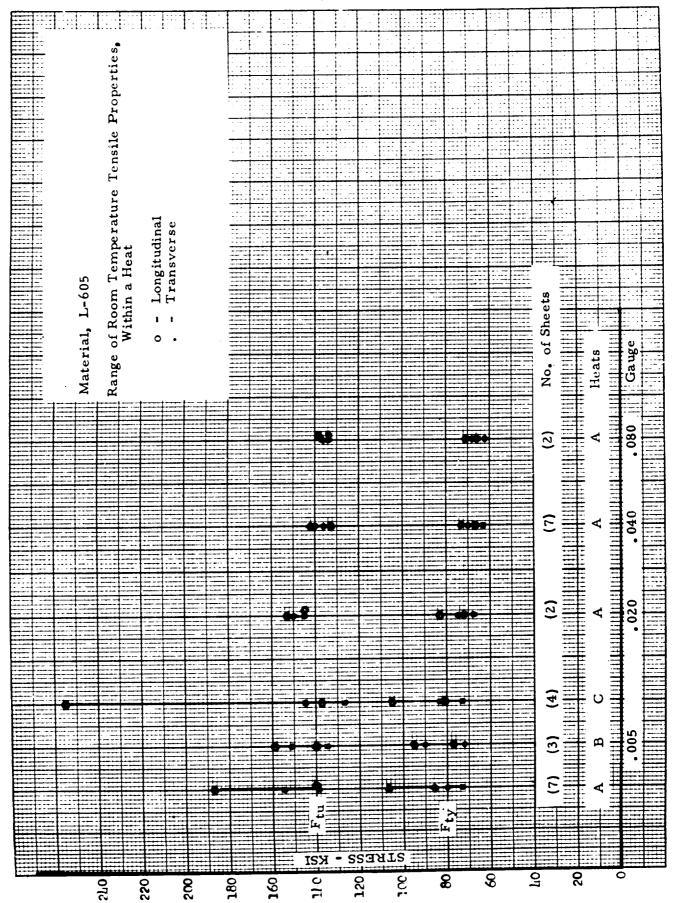


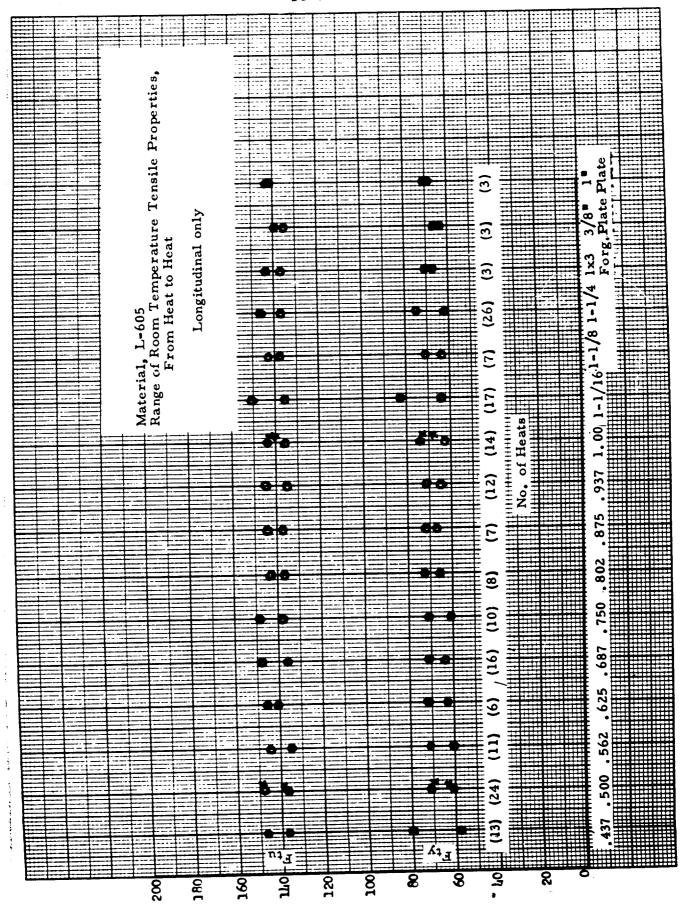












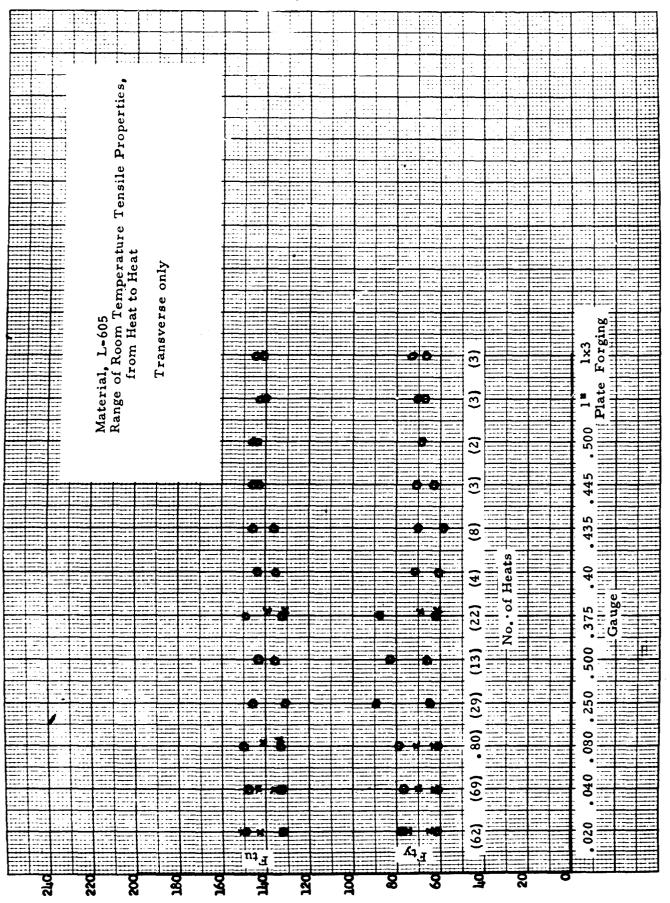
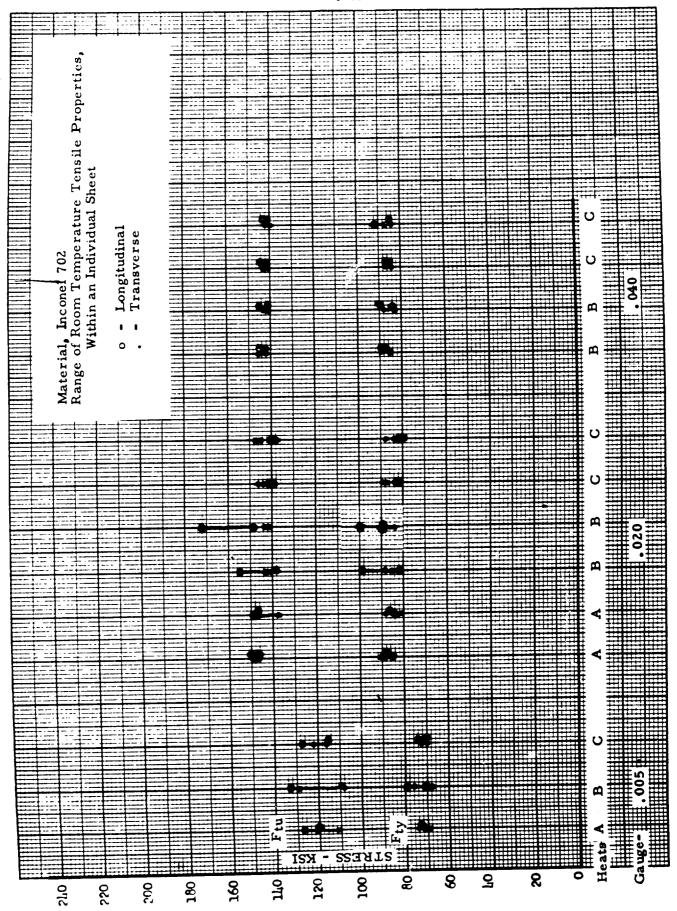
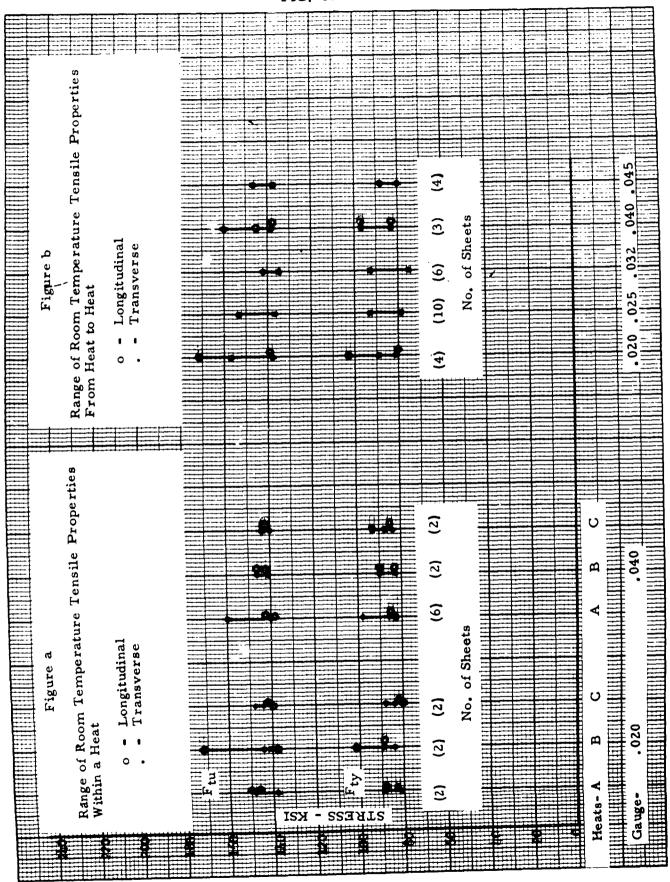
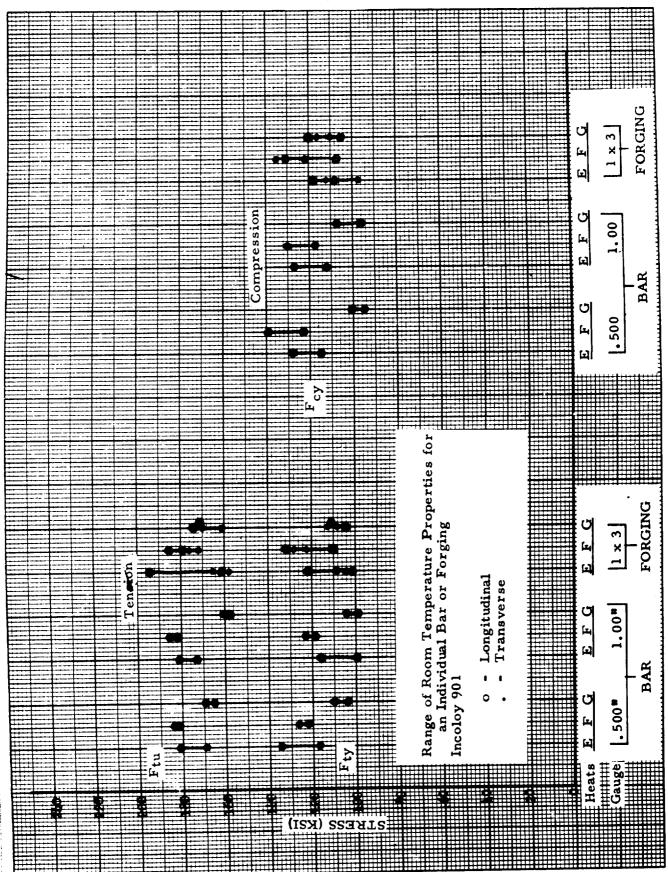
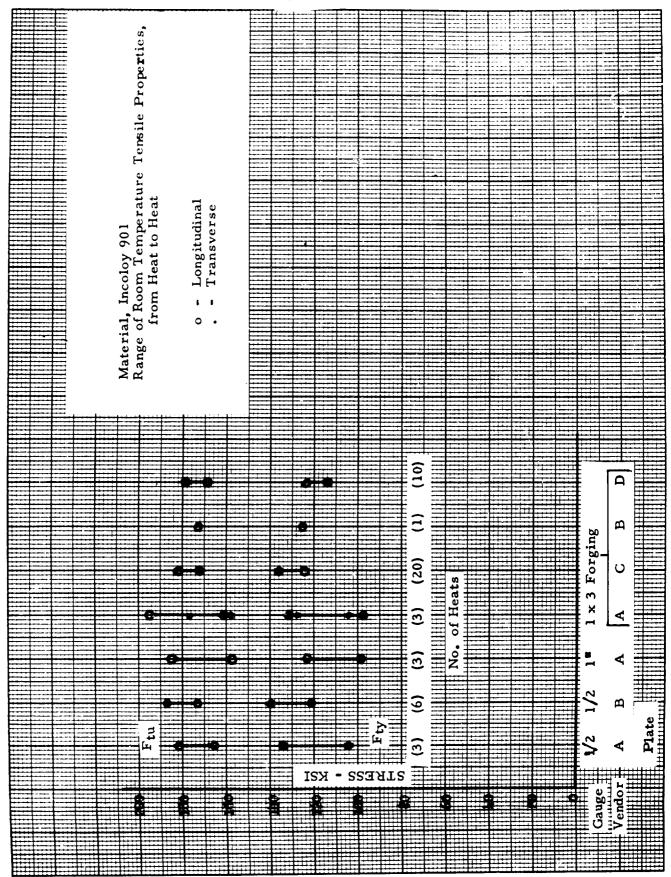


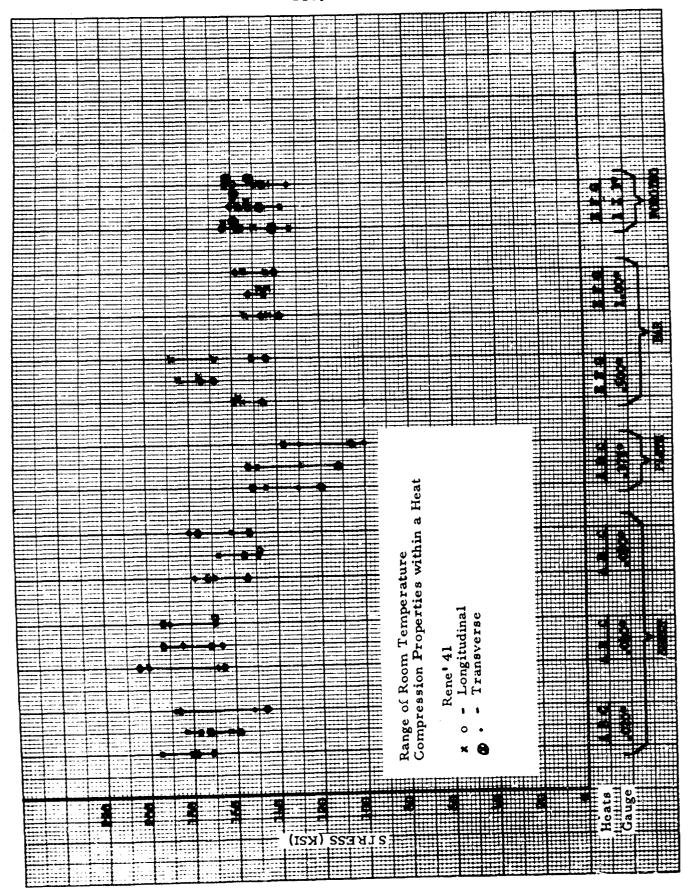
FIG. 43

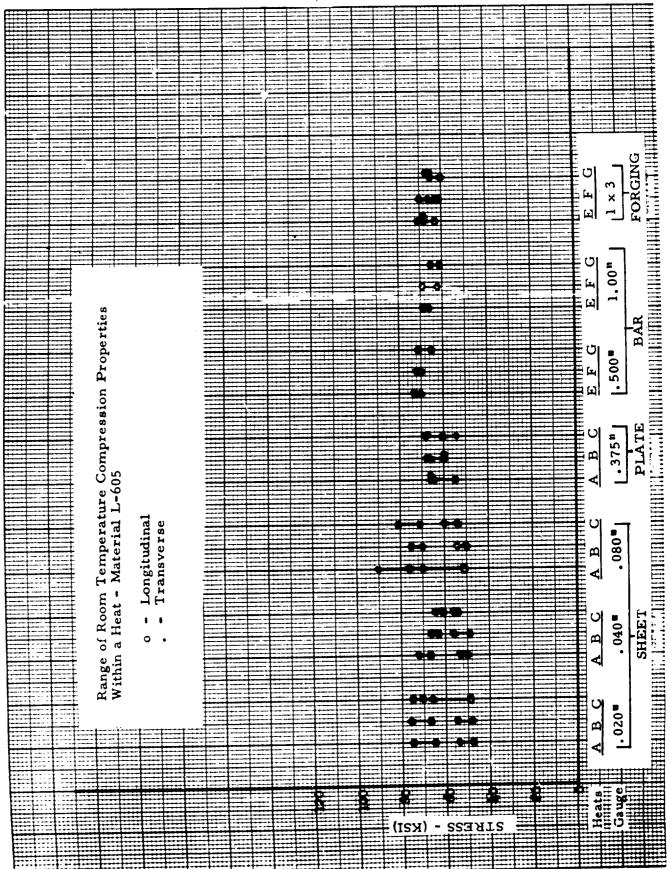


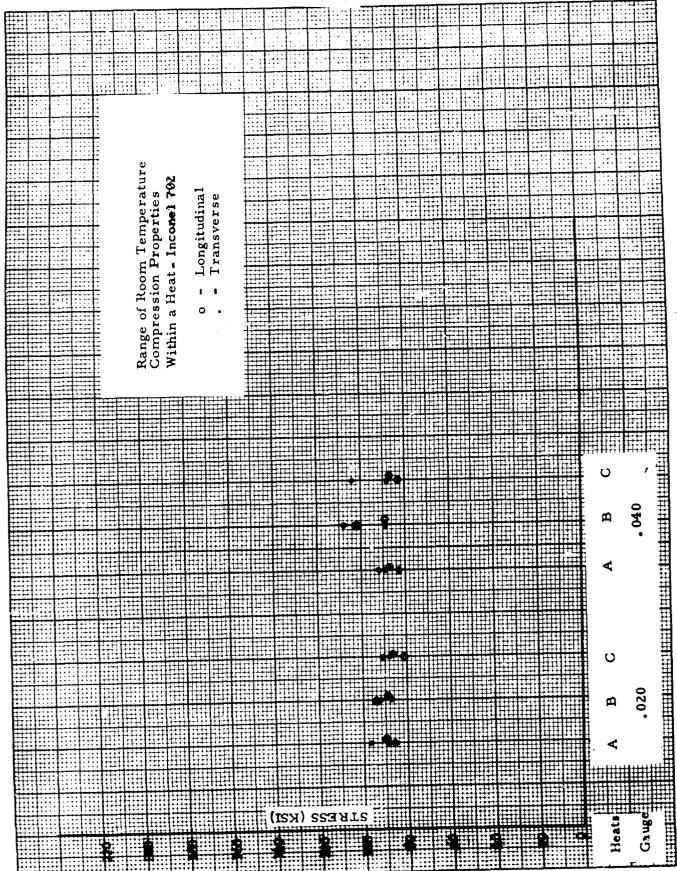












# SECTION VI - SUMMARY OF TEST RESULTS

# 6.3.5 RANGE OF ROOM TEMPERATURE STRENGTH AND ELONGATIONS

TABLE NOs. 23 and 24

FIGURE NOs. 33 thru 49

TABLE 23

TENSILE STRENGTH AND ELONGATION OF SHEET MATERIAL RENE\*41

VENDOR D MATERIAL

		Room Te	<b>moerat</b> ura	025" Gauge	1400°F		
	Sheet #	Ultimate	Yield	Elongation	Ultimate	Yield	Elongation
		KSI	KSI	Percent	KSI	KSI	Percent
Heat X	1	190.6	156.9	24.9	174.2	-	12.5
	2	189.4	139.8	25.0	176.3	-	13.0
	2 3	184.0	136.2	24.5	163.8	-	14.0
	4	188.8	140.6	24.5	174.3	-	14
	5	190.2	139.8	23.0	177.0	-	14
	6	189.1	140.7	24.0	170.7	-	12.5
	7	194.6	143.9	24.0	168.9	-	13.5
	8	192.0	139.3	24.0	169.0	•	12.5
	9	189.9	137.4	26.0	171.4	-	14.5
	10	191.9	144.1	20.0	184.1	-	12.0
	11	195.5	149.8	23.0	174.0	-	13.0
	12	195.1	149.6	23.0	178.7	-	13.0
	13	198.0	156.0	22.0	192.2	-	10.5
	14	194.9	150.8	23.5	179.6	-	12.0
	15	193.7	145.1	23.5	185.1	-	12.0
	16	195.7	152.5	23.0	186.4	-	11.0
	17	195.2	151.0	22.5	168.8	-	12.0
	18	178.8	119.1	26.0	156.3	-	15.0
	18	183.0	121.4	24.0	-	-	-
	18	174.8	118.4	17.5	-	_	•
	19	178.3	125.3	27.0	159.8	•	14
	19	172.5	115.3	20.5	-	•	-
	19	174.1	118.7	23.5	•	_	_
	20	171.3	114.7	24.0	147.7	-	13.5
	20	169.1	114.5	20.0	•	_	-
	20	175.4	115.6	25.0	-	_	•
	21	177.6	117.6	22.5	161.0	-	14.5
	21	182.0	123.7	24.0	-	-	•
	21	189.9	140.7	23.5	-	-	-
AMS 55	545 (min.	) 170.0	130.0	10.0	135.0	-	3.0

Repetitive numbers indicate retests from same sheet. All tests on this page come from the same Heat.

TABLE 23 (cont'd.)

TENSILE STRENGTH AND ELONGATION OF SHEET MATERIAL RENE 41

		Room Ten	Room Temperature 032" Gauge			1400°F		
	Sheet	Ultimate	Yield	Elongation	Ultimate	Yield	Elongation	
	<del></del>	KSI	KSI	Percent	KSI	KSI	Percent	
Heat	Y 1	182	124.9	27.0	149	108.6	5.5	
	2	189.1	137.8	23.5	146.7	121.1	11.5	
	3	179.2	124.7	24.5	149.3	109.1	11.5	
	4	172.7	122.1	19.5	143.7	107.9	11.5	
	5	182.1	126.7	27.5	159.6	117.3	13.0	
	6	175.6	119.9	27.5	158.7	112.7	14	
	7	173.8	114.2	26.0	138	105.8	9	
	7	178.8	121.6	18.0	-	-	-	
	7	168.9	125.2	11.5	-	_	-	
	8	176.8	124.4	21.5	141	104.5	9.5	
	9	162.0	118.4	14.5	138	106.0	8	
	9	162.0	118.4	14.5	•	-	•	
	9	151.0	119.9	9.0	-	-	-	
Heat Z	z 1	184	148.6	25.5	170.3	124.7	11.5	
		193.5	148.2	22.5	157.3	121.2	11.0	
	2 3	192.3	144.3	23.5	170	127	13	
	4	194.6	140.6	23.5	147.2	121.5	8	
	5	185.9	131.6	26.5	148.7	111.0	14	
	6	188.5	142.5	22.0	160.5	121.3	16.5	
	7	189.4	139.4	26.0	171.8	121.5	15	
	8	186.9	132.4	26.0	166.0	122.0	15	
	9	190.2	142.3	24.0	172.9	128.7	14	
	10	188.3	132.7	24.5	173	119.9	16.5	
	11	183.6	132.1	25.0	160.7	115.4	14	
	12	192.7	139.3	26.0	181.4	129.1	14	
	13	186.3	131.0	26.0	160.8	-	13	
	14	180.2	126.2	27.5	168.9	125	15.5	
	14	185.0	119.7	25.0	-	-	-	
	14	178.9	124.7	21.0	•	-	-	
	AMS 5545	170.0	130.0	10.0	140.0	110.0	3.0	

Retests from same sheet.

Sheet groups 1-9 - Same Heat Sheet groups 1-14 - Different Heat than 1-9

# 6.4 Michanical Property Discussion, Material Renet 41

# Sheet

Effect of Temperature on Strength (Short Time). Essentially there is no difference between the longitudinal and transverse properties except for the compressive yield strength where the transverse is higher to  $1400^{\circ}$ F at which point the curves join and remain the same to  $1800^{\circ}$ F. The compression yield is higher than the tension yield to  $1400^{\circ}$ F. There is only slight loss of strength for all properties up to  $1200-1400^{\circ}$ F after which degradation becomes rapid. All the curves are nearly parallel to  $1200^{\circ}$ F.

Exposure Effects - Exposure at elevated temperature has no degrading effect on the room temperature properties with the exception of bearing strength below 1400°F; above this temperature deterioration is rapid even for the 10 hour curve. The bearing ultimate and yield show a secondary hardening effect and deterioration for long time exposure (500-1000 hours) starts at 1200°F.

Length of time at temperature has no effect on the elevated temperature properties below 1400°F; in comparison to the 1/2 hour condition; above this temperature, long time exposure does cause some additional degradation but it is not excessive.

### Foil

Effect of Temperature on Strength (Short Time) - The curve for the ultimate tensile strength is parallel to that for sheet but consistently lower; the tensile yield strength is lower than for sheet to 1200°F where the curves join and only exhibit minor differences to 1800°F.

Exposure Effects - There is no effect of exposure at elevated temperatures on the room or elevated temperature strength below 1400°F; above this temperature deterioration is extremely rapid. When a time curve ends abruptly, it indicates that higher temperatures resulted in specimens which could not be tested due to severe degradation. Foil exposed was .005 inch.

# Bar, Plate and Forging

Effect of Temperature on Strength (Short Time). The sheet properties are generally higher to 1400°F where they tend to converge. The bearing and shear properties are almost identical with sheet. There is little loss of strength to 1400°F after which deterioration becomes rapid.

### 6.5 Material, L.609

#### L-605

The yield to ultimate strength ratio of this alloy in the solution treated condition is in the order of .45-.50. The alloy has very good ductility coupled with; high strain hardening rate. For this reason if the full ductility of which the alloy is capable is not achieved the full ultimate strength capability will not be developed. In the testing of the sheet and foil gauges this can create a real problem. Although the specimen is reduced by some .003 to .005 inches to force failure at the center of the reduced section the tremendous stretching that takes place over the whole gauge length causes the failure to be somewhat random over the entire test section. In addition the extensometer which bites into the material can be the deciding factor in inducing failure. All of these factors will tend to reduce ductility and ultimate tensile strength. The strain rate after yield may also influence the magnitude of the ultimate strength. The tests in this program were conducted in accordance with ARTC-12 which recommends .005 inch per inch per minute through the .2 per cent yield after which the rate is adjusted to induce failure in one minute. Since the percentage elongations in this alloy were as high as 60 per cent the strain rate to failure is abnormally high in comparison with most conventional materials.

These factors render the ultimate strength values somewhat questionable in many instances. This would not normally be an important factor in design since the low yield strength would be the determining factor in any analysis. If the data is being used to establish trends however, these factors should be considered.

#### Sheet

Effect of Temperature on Strength (Short Time) - There is no difference in the longitudinal and transverse properties throughout the temperature range investigated, except for the compressive yield strength (Fcy). The transverse compressive yield strength is higher than the longitudinal at room temperature, the curves tend to converge join at approximately 1400°F and are the same out to 1800°F.

The compressive yield (Fcy) and tensile yield (Fty) strengths are almost identical. The tensile and compressive yield strengths (Fty and Fcy) show very good retention of properties after an initial drop, (between room temperature and 800°F) out to 1800°F.

The tensile and shear ultimate strengths (Ftu and Fsu) are approximately parallel to about  $1300^{\circ}$ F, after which they converge meeting at  $1800^{\circ}$ F. The bearing ultimate strengths (Fbru) for  $e/D \approx 1.5$  and  $e/D \approx 2.0$  are approximately parallel to each other and to the tensile ultimate (Ftu). The bearing yield strengths (Fbry) for  $e/D \approx 1.5$  and  $e/D \approx 2.0$  are approximately parallel to each other and to the tensile yield Fty) up to  $1400^{\circ}$ F after which the bearing strength tends to fall off at a higher rate.

Exposure Effects - Exposure at elevated temperatures shows no significant loss in room temperature strength except for the ultimate tensile (Ftu) which exhibits a slight decrease for long exposures (500-1000 hours) in the 1600-1800°F range. This loss in tensile strength is attributed to a decrease in ductility. All other properties show a secondary hardening effect.

The curves illustrating the effect of exposure time on the elevated temperature strengths show all curves higher than the 1/2 hour curve except the 1000 hour curve which shows some slight degradation for temperatures in excess of 1400°F.

# Plate, Bar and Forging

Effect of Temperature on Strength (Short Time) - There is no difference between longitudinal and transverse properties. The ultimate tensile strength is almost the same as for sheet to  $1000^{\circ}$ F then becomes higher. The tensile yield strength is slightly lower for bar, plate and forging, then for sheet over the whole temperature range to  $1700^{\circ}$ F where they join. The compressive yield strength is almost identical to the tensile yield strength. The shear ultimate is lower than that of sheet until approximately  $1150^{\circ}$ F where they cross, then is higher to  $1800^{\circ}$ F where they join.

The bearing ultimate stress with e/D=2.0 starts out lower than for sheet but crosses the sheet curve at  $900^{\circ}F$  and is higher to  $1800^{\circ}F$ . The bearing yield strength with e/D=2.0 is lower than for sheet over the entire temperature range. The bearing ultimate strength with e/D=1.5 is lower than sheet over the entire temperature range while the yield strength with e/D=1.5 starts higher than for sheet but crosses at  $550^{\circ}F$  and remains lower.

#### Foil

The longitudinal and transverse tensile yield strengths (Fty) of .010 inch foil are identical.

The transverse tensile yield strength (Fty) of the .005 inch foil is slightly lower than the longitudinal over the whole temperature range investigated. The .005 inch foil has higher tensile ultimate (Ftu) and yield (Fty) strengths than the .010 inch foil.

The tensile yield strength (Fty) of sheet lies between the .005 inch foil and .010 inch foil over the whole temperature range.

Exposure Effects - All exposure data were obtained on .005 inch foil. There is no effect of exposure at elevated temperatures or the room temperature strengths below 900°F; above this temperature deterioration is rapid even after 10 hours.

exposure for the ultimate tensile strength (Ftu), while the yield strength (Fty) shows a secondary hardening effect at 1200°F and deterioration starting at 1400°F for 10 = 500 hours and at 1200°F for 1000 hours. Specimens exposed for 1000 hours at 1800°F could not be tested because of the warpage and general material degradation due to oxidation.

The effect of time at temperature on the elevated temperature tensile ultimate shows deterioration starting as low as 1100°F for long time exposures. The tensile yield curves in this category show secondary hardening at 1200°F and the start of degradation at about 1400°F for the longer exposure times.

- 6.6 Mechanical Property Discussion, Materials Inconel 702 and Incoloy 901
- 6.6.1 Material Incomel 702

# Sheet

Effect of Temperature on Strength (Short Time) - The longitudinal and transverse properties are the same throughout the temperature range investigated. The difference between the ultimate tensile (Ftu) and the tensile yield (Fty) strengths is large from room temperature to 800°F. Above 800°F, the ultimate strength (Ftu) starts deteriorating rapidly while the yield strength (Fty) starts to rise slightly at 1000°F, peaks at 1200°F and then falls off rapidly. From 1400 to 1800°F, the tensile ultimate (Ftu) and tensile yield (Fty) strengths are almost identical (the ultimate, Ftu, being slightly higher). The same effect (i.e., overlapping of ultimate and yield) is displayed in the bearing strength (Fbru and Fbry) for both edge distances (i.e., e/D = 1.5 and e/D = 2.0). The bearing yield strengths (Fbry) do not show the secondary hardening effect (i.e., rise of curve at 1000°F) seen in the tensile yield (Fty).

The shear ultimate strength (Fsu) runs out almost parallel to the ultimate tensile strength (Ftu) until 1200°F, then declines less rapidly and the curve crossing at approximately 1550°F.

The compressive yield strength (Fcy) is equal to the tensile yield strength from room temperature to 400°F, then starts to rise, peaks at 1000°F and then deteriorates rapidly meeting the tensile yield (Fty) at 1300°F and remaining equal to it out to 1800°F.

The curves (Fbru) are approximately parallel to the ultimate tensile (Ftu) curve. The shape of the bearing yield strength (Fbry) curves are almost the same as the tensile yield curve (Fty); they do not however display the secondary hardening effect.

Exposure Effects - For exposure times up to 1000 hours there is no degradation of room temperature strength properties for exposure temperatures below 1000°F. The curves for all the properties indicate a secondary hardening effect in the temperature range 800-1000°F. In the 1200-1800°F range the room temperature tensile ultimate and yield (Ftu and Fty) and the compressive yield (Fcy) strengths show an initial decline in properties which occurs within a ten hour period and subsequent exposure does not result in additional deterioration; in some cases, the longer exposure results in higher room temperature strengths. The room temperature bearing ultimate (Fbru) and yield (Fbry) and the shear ultimate (Fsu) strength curves do show increasing degradation with increasing exposure time in the 1400-1800°F.

All the curves indicate that the greatest degradation occurs at approximately  $1600^{\circ}$ F, after which the curves tend to flatten out.

There is very little effect of exposure time on the elevated temperature strengths of this material. In all cases the 1/2 hour and 1000 hour curves are almost identical. Significant differences only appear at temperatures above 1200°F and always disappear by 1800°F.

# Foil

Effect of Temperature on Strengths (Short Time) - There is little degradation of the ultimate tensile strength (Ftu) below 1000°F; above this point decay is rapid. The decline of the tensile yield strength (Fty) is slow below 1000°F, the rate of decay increases above this temperature but not as rapidly as the ultimate strength. The ultimate and yield strengths (Ftu and Fty) of the foil are lower than the sheet until approximately 1600°F where they tend to converge.

Exposure Effects. There is no effect of exposure at elevated temperature on the room temperature ultimate tensile (Ftu) or tensile yield (Fty) strengths below 1200°F. Above this temperature the rate of degradation increases with increasing exposure time. The foil does not display the secondary hardening effect seen in the sheet. Length of exposure at elevated temperatures has little effect on the elevated temperature tensile yield strength (Fty); the 1/2 hour and 1000 hour curves being almost identical with a maximum difference of approximately 16 per cent at 1400°F. Length of time at temperature has a greater effect on the elevated temperature ultimate tensile strength (Ftu); rapid degradation begins at a lower temperature and the maximum difference is approximately 30 per cent at 1300°F. This latter condition is accompanied by a loss in ductility probably due to oxidation effects.

# 6.6.2 Material Incoloy 901

Effect of Temperature on Strength (Short Time) - The longitudinal tensile ultimate strength (Ftu) is slightly higher than the transverse up to 1400°F where they converge. This is also true in the case of the tensile yield (Fty), compressive yield (Fcy) and shear ultimate (Fsu), except that the compressive yields do not join until a slightly higher temperature (1500°F). The retention of strength of this alloy is good, showing only slight losses in strength up to 1000-1200°F range.

Exposure Effects • Exposure for periods of time up to 100 hours have no effect on the room temperature ultimate tensile strength for temperatures below 1400°F; above this temperature, deterioration is rapid. For longer exposures (i.e., 500-1000 hours) deterioration begins at 1000°F and progresses rapidly. Exposure effects on the room temperature tensile yield strength are approximately the same as for the ultimate strength except that deterioration begins at a somewhat lower temperature (1200°F) for 10-100 hour exposure. Above 1600°F there is little effect of length of exposure on these properties. Only .005 inch foil was tested.

Exposure for periods up to 100 hours have no effect on the elevated temperature properties at temperatures below 1200°F; above this temperature the deterioration in comparison to the 1/2 hour curve reaches a maximum at 1400°F to 1500°F after which the 1/2 hour and 100 hour curves start to converge. For exposures of 500-1000 hours, the deterioration of elevated temperature strengths starts at 1000°F with the maximum degradation in comparison with the 1/2 hour curve occurring at 1400°F for the ultimate and 1200°F for the yield. Above 1600°F there is little effect due to length of time of exposure.

# 6.7 Creep

Creep tests were conducted for the materials - Rene '41, L-605, Inconel 702, and Incoloy 901. The tests were performed at temperatures of 1200, 1400, 1600, and 1800 °F for time periods up to 1000 hours. The various forms for each alloy are shown on the respective graphs in Section VII. The graphs plotted indicate the stress versus the time in hours for a family of percentages of plastic deformations ranging from 0.05 up to 1.0 per cent. The curves normally shown as a single family have been separated in many instances due to overlapping of test points and data scatter in order to more clearly observe the difference (if any) due to section thickness or material form. Where test points are not shown for the drawn curves, the test results had been obtained with little scatter.

#### 6.7.1 Rene! 41

The total plastic deformation curves at 1200°F (Figures 81, 88, 89) are closely grouped and diverge for the longer time periods for all forms except the 0.005 inch gauge foil. This latter data is shown as plotted points only due to data scatter. At 1400°F, (Figures 90 - 62), the curves are uniformly diverging as the time periods increase with the thinner sections dropping more rapidly. The data for the foil gauge is shown to have more scatter at the lower percentage deformations. At 1600°F, (Figures 93 to 95, 99, 100) the curves are more widely separated due to section thickness and longer time periods indicating an accumulative effect of timetemperature oxidation. The tendency for curve reversal at the longer time periods is noted since the slope again tends to approach that stress rupture curve. Similiar trends are noted at 1800 F (Figures 96 through 100) with the exception of the forgings curve. The basic tensule strength of the forging material had considerable variation within the forging as well as lower strength than the 0.5 inch diameter bar. However, as timetemperature oxidation effects increased, the forging material maintained a higher percentage of its initial strength, thereby crossing over the curves for the thinner gauge material as shown on the respective graphs.

#### 6.7.2 L-605

The resulting curves for the total plastic deformations for the material L-605 are shown in Figures 188 to 198 for the various temperatures. At 1200°F, separate curves are shown for the 0.005 inch gauge foil and the 0.040 inch gauge sheet. The data resulted in fairly consistent curves with the curves for foil falling slightly below the curves for 0.040 inch sheet.

The plotted results shown at 1400°F are relatively closely grouped for the various forms with consistent results shown for the .005 inch gauge foil except at the 0.05 per cent deformation for which no curve was drawn for the foil gauge due to scatter of test results. At the 0.05 per cent deformation and the 0.1 per cent deformation the time periods range from up to 20 hours and 50 hours respectively for the sheet material and up to 650 hours for forgings.

At 1600°F a single series of composite curves for all forms are plotted at the higher deformation (1.0 and 0.5 per cent), and are noted to be parallel to the stress-rupture curve for the 0.5 inch bar. At the lower deformations, the curve for foil is consistent but separated from the curve shown for all other forms.

The curves shown for data at 1800°F are near parallel to stressrupture curves up to 100 hours then widely diverge as a function of section thickness with foil data being limited to 100 hours in most cases.

#### 6.7.3 Inconel 702

The plastic deformation curve for Inconel 702 as shown in Figure 272 for sheet material only. At the 1200°F temperature, curves are drawn for the 0.040 inch gauge sheet for all deformations with a few points shown for the .005 inch gauge foil at the lower deformations. The 1400°F temperature curves develop slight reversals of curvature after relatively short time periods before continuing on parallel to the stress-rupture curve. Data for the 1.0 per cent deformation curve was not recorded since rupture occurred near the 1 per cent deformation point. The material evidently approached third stage creep deforming rapidly, shortly after the previous data had been recorded.

At 1600°F and 1800°F the curves are drawn as straight lines with the .05, 0.1 and 0.3 per cent deformation curves having a single break in slope occurring at time periods less than 100 hours. For each per cent deformation, a single curve is plotted as a composite of the three sheet gauges.

# 6.7.4 Incoloy 901

The test results for total plastic deformation of Incoloy 901 are plotted and shown in Figure 314 to Figure 319. At the 1200 F temperature, only data for 0.500 inch bar is plotted for all deformations and is noted to be straight lines, closely spaced and parallel to the stress—rupture curve up to 1000 hours. Data for both bar and forgings are shown at 1400 F temperature, the curves having a gentle decreasing slope at the center sections before dropping more rapidly at longer time periods. The curves are consistent for the whole family of deformations. At 1600 F temperature, however the data for 0.500 inch bar is almost straight and parallel to the stress-rupture curve for the 0.3, 0.5 and 1.0 per cent deformations, while a significant drop off occurs at approximately 50 hours before leveling off at the longer time periods over 100 hours.

At the 1800°F temperature, data for 0.500 inch bar and forging material is very consistent and composite curves are drawn for each of the respective deformations and are very similiar in pattern. The time periods range from 200 hours for the 0.05 per cent deformation curve to beyond 1000 hours for the 1.0 per cent deformation curve.

# 6.8 Stress Rupture

The stress rupture evaluation accumulate data for all alloys over the 1200 to 1800 F range in 200 F increments. Individual families of curves are given for each material form or gauge, and in some cases, for each grain direction.

The form or gauge of each alloy that had been subjected to maximum testing was used for computation of a best fit Larson Miller constant and master rupture curve. The remaining forms or gauges of a specific alloy were then examined for conformance with the parametric plot for the 'standard'.

Material forms selected as base data are as follows:

Rome' 41 0.040 inch sheet, transverse

L-605 0.040 inch sheet, transverse

Inconel 702 0.040 inch sheet, transverse

Incoloy 901 0.5 inch bar

Where a significant departure was observed for a different form or gauge, an additional curve was incorporated on the master rupture plot. This base data was first qualified with available published information. No major discrepancies were observed between the data generated in this investigation and average values obtained in a literature search.

Unfortunately, very little foil rupture data was available for supplementary information. As a consequence, foil properties were not included in master rupture plots. The behavior of this gauge was always lower in rupture life and sometimes erratic in behavior.

#### 6.8.1 Rene' 41

Stress rupture data obtained for Rene! 41 proved extremely uniform within a given form. Stress to rupture curves presented in Figures 103 through 107 indicate identical performance for all sheet gauges (with the exception of foil) regardless of grain orientation. The same is true of bar and forged materials, Figure 108 through 110. However, substantially higher rupture lives were exhibited by this class of material beyond 1400 F. A progressive increase in life is observed at 1600 and 1800 F over sheet with increasing time and temperature. This is assumed to be a result of oxidation since the short time elevated temperature strengths for both classes of material are essentially equal and the plots for each at a given temperature are markedly divergent rather than parallel.

The master rupture chart, Figure 102 for sheet and bar illustrates this by convergence of the two curves with decreasing values of parameter.

Foil testing indicated greatly reduced performance compared to sheet. Curves have been plotted only to 100 hours life in consideration of the relatively small quantity of data generated and the severe oxidation observed in thermal exposure specimens for times beyond this amount. See Figure 101.

#### 6.8.2 L-605

The L-605 stress rupture plots over the 1200 to 1800  $^{\circ}$ F range are provided in Figure 199 through 208. No major variation was found to exist within a given material form.

Bar and forging properties are generally superior to sheet material particularly at the longer times and higher temperatures where loss of area through oxidation is most pronounced. Oxidation effects, however, are not as great as was observed when comparing Rene' 41 sheet and bar, and only become appreciable at 1800°F for extended times.

A Larsen-Miller parametric constant of 19 was employed to produce the master rupture chart of Figure 102. Foil (.005 inch) was not included in the master plot because of limited data available. Examination of the stress to rupture curves for this gauge reveal generally lower performance than heavier sheet material. Some difficulty was experienced at 1200°F resulting in questionable data points higher in life than for sheet. Stress rupture curves for foil have only been plotted to lives of 100-300 hours, the level of reasonable confidence.

#### 6.8.3 Inconel 702

Stress rupture plots for Inconel 702 are given in Figure 274 through 278.

Larsen-Miller representation of data obtained for 0.040 inch sheet (transverse) resulted in a constant of 25 being most suitable. The balance of material, with the exception of 0.005 inch foil, exhibited good conformance with the basic plot.

Foil gauge testing indicated severe divergence throughout the 1200 to 1400 F range. Data for 1600 and 1800 F approximates that generated for heavier gauge sheet.

### 6.8,4 Incoloy 901

Stress rupture testing of Incoloy 901 evolved the curves shown in Figure 320 through 323. Rupture lives for 0.5 and 1.0 inch bar, and 1 x 3 inch forged bar were found to be equivalent.

The Larsen-Miller curve developed for the 0.5 inch bar resulted in a constant of 29 as being most suitable.

# 6.9 Axial Fatigue Data

Axial fatigue data was accumulated for all materials evaluated in this program. Table 11 provides an outline of the testing format for each alloy form.

All data has been presented as individual S/N plots for a specific form, stress ratio and temperature. A survey of available literature produced nothing in the way of supporting data to confirm or extend these fatigue diagrams.

In attempting to satisfy the low cycle fatigue requirements (10<sup>2</sup> ~ 10<sup>4</sup> cycles) of the investigation, a large number of tests were performed at maximum stresses well above the 0.2 per cent yield stress of the material at temperature. As a consequence, severe plastic deformation and attendant work hardening were encountered. The resultant strengthening and the effects of temporary preload changes occurring with specimen deformation are indeterminate in nature; resulting in data of questionable validity. This situation was ultimately corrected by limiting the maximum stress to values lower than yield strength. Unfortunately the change was made rather late in the program when a significant percentage of tests had already been completed.

The 0.2 per cent yield strength is indicated on each S/N plot by a dashed line. This yield strength represents the rated or design curve value generated from tensile tests performed during the course of this investigation.

The stress ratio 'A' being equal to the alternating stress/mean stress by reading the graphs for maximum stresses; the other pertinent stresses are determined as follows:

Mean stress = 
$$\frac{\text{Maximum stress}}{1+A}$$

Alternating stress = Maximum stress x 
$$\frac{A}{1+A}$$

Minimum stress = Maximum stress x 
$$\frac{1-A}{1+A}$$

### 6.9.1 Rene! 41

An axial fatigue evaluation was conducted on Rene! 41 in two sheet gauges, 0.040 and 0.080 inches, and 1.0 inch diameter bar. Stress ratios from A = 0.25 to A = 0.98 were employed up to 1800 F.

#### 6.9.1.1 0.040 Inch Sheet Transverse

# a. Stress Ratio A=0.25 (Figures 111 to 120)

The largest number of tests at room temperature were conducted above the rated (design curve value) 0.2 per cent yield strength. However, a valid endurance limit below yield was obtained at 135 ksi. Subsequent testing from 600 to 1200°F considered only a single load level under yield strength at temperature. All tests produced run-outs beyond 10° cycles at a stress level equal to the 10° endurance limit at room temperature.

A substantial reduction in fatigue life was found at 1400°F, coincidental with the fall-off in static strength and ductility.

No detrimental effects were apparent with increased cycling rate at 1000, 1200, and 1400 °F.

# b. Stress Ratio A=0.67 (Figures 121 to 129)

A format identical to that of A = 0.25 was followed for this stress ratio in that only part of the ambient temperature testing was performed below the rated 0.2 per cent yield stress.

Heavy scatter was encountered within 1200 to 1400 °F minimum ductility range. Beyond 10° cycles tests from 400 to 1200 °F produced fatigue lives equaling or exceeding room temperature performance.

#### c. Stress Ratio A=0.98 (Figure 130 to 134)

A room temperature endurance limit (10<sup>7</sup> cycles) was obtained at 80 ksi. Data taken at 600 and 1000 F tended to approximate the room temperature curve over the entire life range. At 1000 F wide scatter was evident but resulted in an endurance limit of the same order as room temperature. Maximum scatter was again produced at 1400 F.

#### 6.9.1.2 0.080 Inch Sheet Transverse

#### a. Stress Ratio A=0.25 (Figure 135 to 138)

The room temperature S/N curve effectively duplicates that of 0.040 inch sheet with an endurance limit of approximately 140 ksi. Data obtained for elevated temperatures were ambiguous due to the use of maximum stresses well above the rate 0.2 per cent yield strength.

# b. Stress Ratio A=0.67 (Figure 139 to 142)

Room temperature S/N data produced for 0.080 inch gauge material provides a reasonable approximation to the lighter gauge material previously evaluated. Elevated temperature S/N data were evalved on a composite of test results run above and below the rated 0.2 per cent yield strength.

#### 6.9.1.3 One Inch Bar

# Stress Ratio A=0.67 (Figure 143 to 147)

Testing of one-inch diameter bar at A=0.67 was conducted for the most part at maximum stresses above the 0.2 per cent yield strss. Some useful data was obtained in the vicinity of 10 cycles for temperatures of 800, 1200, and 1600 °F.

### b. Stress Ratio A= 🗢 (Figure 148)

Only ambient temperature data to 10<sup>6</sup> cycles was produced for this ratio. When attempting tests at 1000 and 1600 °F, extreme scatter was encountered at all load levels. Further evaluation at elevated temperatures was not attempted.

# 6.9.2 L-605

An axial fatigue program was conducted on L-605 in 0.040 and 0.080 inch sheet (transverse specimen orientation) and on 1.0 inch diameter bar.

A large percentage of testing was performed at maximum stresses above the 0.2 per cent yield strength. The extremely low proportional limit of this alloy in the solution treated condition made load selection to obtain low cycle data difficult. The high ductility and notch toughness of this material permits a  $10^{\circ}$  cycle run-out at ambient temperature with stresses slightly below the 0.2 per cent yield strength, for example, .040 gauge sheet at room temperature, the stress ratio (alt. stress/mean stress), A = 0.98.

#### 6.9.2.1 0.040 Inch Sheet (Transverse)

# a. Stress Ratio A=0.25 (Figure 209 to 217)

Ambient temperature S/N data was taken entirely above the material yield strength. Failures were induced over the 10 to 10 range. Elevated temperature requirements at 600 - 1200 °F to establish a 10 cycle run-out were satisfied with maximum stresses just below the rated (design curve value) yield strength. At 1400 °F, the maximum stress had to be reduced an additional amount below yield to obtain consistent run-out at 10 cycles.

Increased cycling speed (1800 versus 3600 cpm) at 1000, 1200, and 1400 °F produced no reduction in fatigue life.

#### b. Stress Ratio A=0.67 (Figure 220 to 228)

Room temperature S/N data for this stress ratio was obtained with a minimum of tests because of incorrect loads applied throught part of the testing. All tests represent maximum stresses above the 0.2 per cent yield strength.

Elevated temperature fatigue performance was obtained from 400 to 1800 °F in 200 °F increments. Maximum stresses for all tests were held below the yield strength. Stress to cause run-out at 10 °C.

cycles (minimum) were determined for all temperatures except 1800°F.

c. Stress Ratio A=0.98 (Figure 233 to 237)

The bulk of specimens at ambient temperature were again run above the rated (design curve value) yield strength. Some data was obtaine below yield establishing a 10 cycle run-out. Tests performed at 600, 1000, 1400, and 1800 F yielded valid 10 cycle lives.

#### 6.9.2.2 0.080 Inch Sheet Transverse

a. Stress Ratio A=0.25 (Figure 218, 219)

All tests were performed above the rated 0.2 per cent yield strength. Therefore, a comparison with 0.040 inch sheet material was not possible.

b. Stress Ratio A=0.67 (Figure 229, 230)

Both ambient and elevated temperature tests with the exception of 1600°F were conducted above the rated 0.2 per cent yield strength. Elevated temperature testing employed only a single load level for a given temperature, primarily to establish correlation with 0.040 inch gauge material. Actually, due to the excessive maximum stresses applied, a check with .040 inch material was only possible at 1600°F, where 0.080 inch sheet demonstrated higher fatigue life.

# 6.9.2.3 One Inch Bar A=.67 (Figures 231, 232)

a. All tests for ambient and elevated temperature were run above the rated 0.2 per cent yield strength. S/N data at seven stress levels was obtained at ambient temperature and at 2-4 stress levels at elevated temperatures.

#### 6.9.3 Inconel 702

Inconel 702 was examined at 3 stress ratios (A=0.25, 0.67, and 0.98) over the room temperature to 1800°F range. Transverse specimens in 0.040 inch sheet were used exclusively.

a. Stress Ratio A=0, 25 (Figure 279 to 287)

An ambient temperature endurance limit (10<sup>7</sup> cycles) was obtained at 85 ksi. S/N data was produced only for this temperature. The 85 ksi value represents the 0.2 per cent yield strength, hence all data on this plot were obtained after some degree of plastic deformation of the specimen.

Elevated temperature properties were checked at a single load level in most instances. This load was intended to be a practical maximum held just below the 0.2 per cent yield strength. At temperatures of 600, 1000, 1200 and 1800 F, no failures were encountered at 10 cycles. These tests were subsequently discontinued.

At 1400°F an equivalent value of stress produced failure at 10° cycles, possibly a result of minimum ductility inherent to the alloy at this temperature. A corresponding reduced life was observed in creep testing where deformations greater than 0.9 per cent could not be obtained before rupture. The majority of creep specimens failed in the 0.3-0.5 per cent deformation range.

The effect of cycling rate, 1800 versus 3600 CPM, was included in the 702 testing although not originally scheduled. Testing at 3600 CPM was conducted at 1000, 1200, and 1400°F. Comparative data at 1000° and 1200° are inconclusive since the stress levels selected for the higher cycling rate were excessively low, necessitating discontinuance of tests without failure between 10° and 10° cycles. At 1400°F where failure occurred at each of 2 stress levels, the 3600 CPM rate seemed to exert only a minor reduction in fatigue life.

### b. Stress Ratio A=0.67 (Figures 288 to 296)

The room temperature S/N data was taken both above and below 0.2 per cent yield strength. A 10 cycle endurance limit was obtained at 6 ksi.

Elevated temperature tests were performed from 400 to 1800 °F in increments of 200°. Depending on a specific temperature, data was produced for maximum stresses both higher and lower than rated 0.2 per cent yield strength at temperature.

### c. Stress Ratio A=0.98 (Figures 297 to 301)

As in testing at A=0.67, loads equaling or exceeding yield strength at temperature were employed for a large portion of the testing. A room temperature endurance limit (10 cycles) of 58 ksi was indicated.

#### 6.9.4 Incoloy 901

Incoloy 901 was evaluated in the form of 1.0 inch diameter bar stock of appropriate intervals between ambient temperature and 1800°F. Stress ratios of A=0.67, 0.98, 2.0 and were examined.

The majority of data generated for the 901 alloy was not influenced by work hardening effects incurred by applying maximum stresses above yield strength.

### a. Stress Ratio A=0.67 (Figures 324 to 334)

Considerable overlap is evident in testing within the 400-1200 °F range. Data for these temperatures reveal slightly superior performance over the room temperature plot, understandable in view of the constant

yield strength and increasing ductility up to 1200°F.

The first obvious reduction in fatigue life was observed at 1400°F at the low cycle end of the plot. At 10° cycles, the maximum stress was equivalent to that or room temperature. Tesis at 1200 and 1400°F with increased cycling rate (3600 CPM) resulted in insignificant changes in fatigue life.

### b. Stress Ratio A=0.98 (Figures 335 to 339)

The family of curves developed at A=0.98 closely resemble those of A=0.67. Data overlap occurred to 1000°F with the first substantial drop in life taking place at the low cycle end of the 1400°F curve. At 10° cycles all data from ambient to 1400°E tends to merge to what is apparently a common endurance limit at 10° cycles.

### c. Stress Ratio A=2.0 (Figures 340 to 344)

A minimum of data overlap was found in testing at A=2.0. S/N curves plotted for ambient, 600, 1000, 1400 and 1800 F show fairly clear ceparation, particularly at low and intermediate cycle ranges. At 10 cycles, the data again tended to converge. A clear picture of performance was not obtained since tests were not conducted beyond 10 cycles.

#### d. Stress Ratio A= (Figures 345 to 349)

The S/N curves developed for A=  $\infty$  fell into the same temperature relationship as found for A=2.0 possessing good low cycle separation and convergence at 10 cycles. A room temperature endurance limit was defined at 47 ksi.

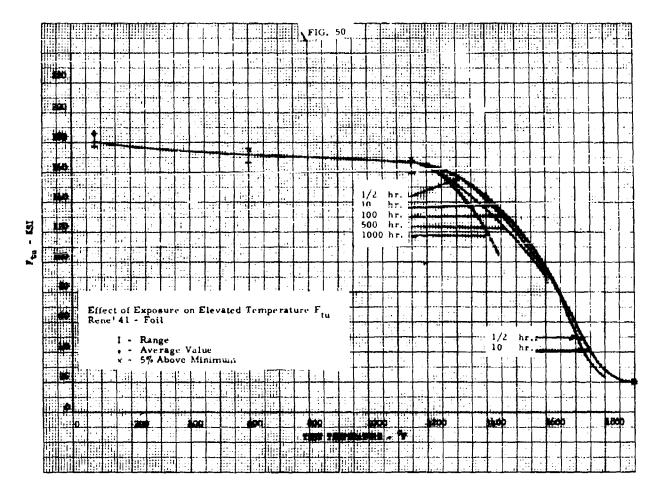
# SECTION VII - TEST RESULTS, TABLES AND GRAPHS

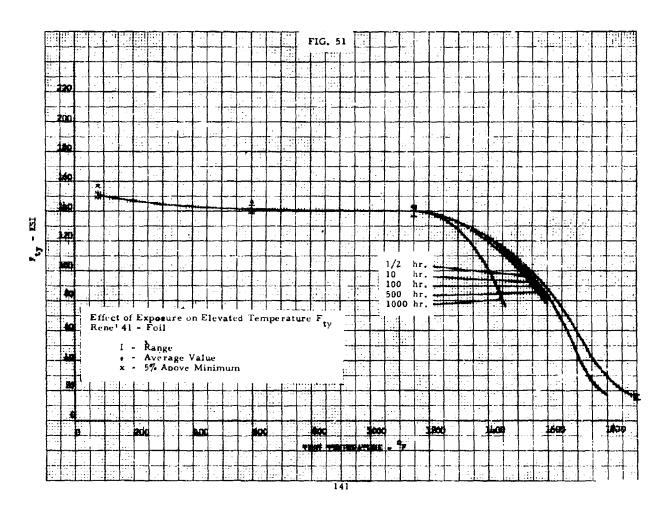
# SECTION 7.1 MATERIAL, RENE: 41

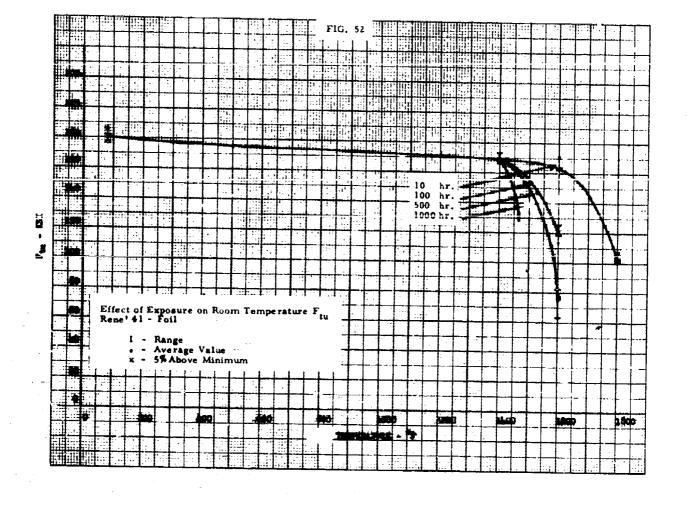
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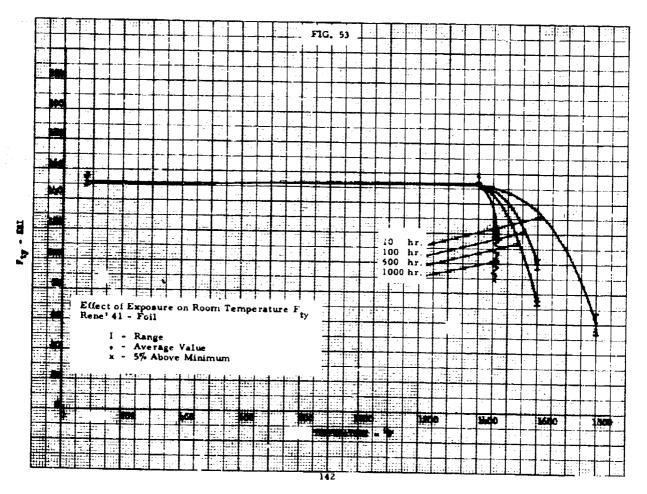
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SECTION 7.1.1 TENSION

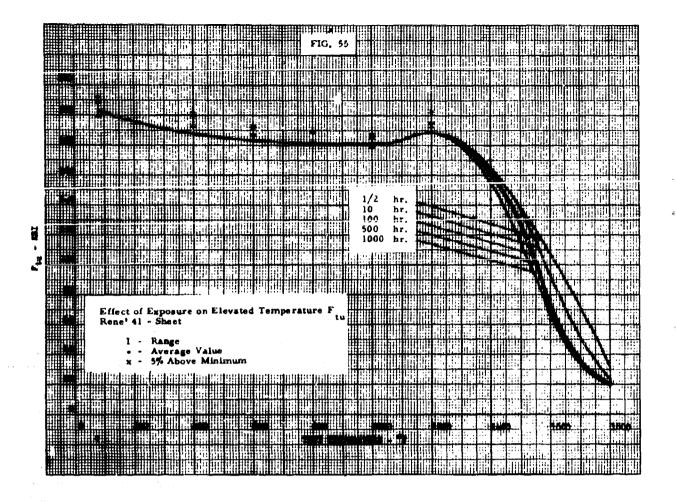


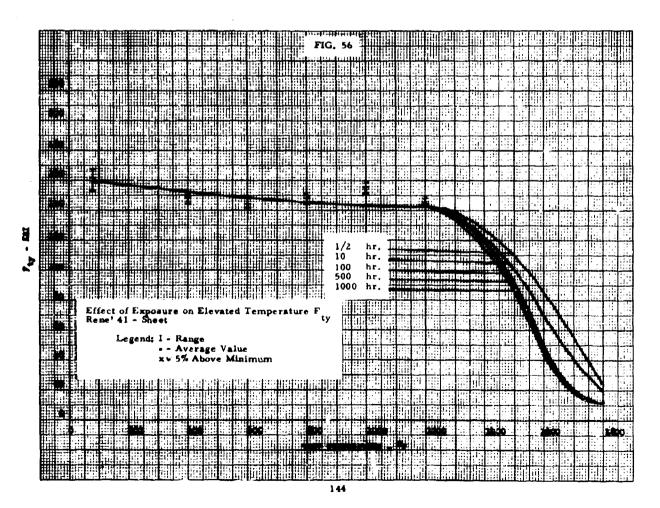


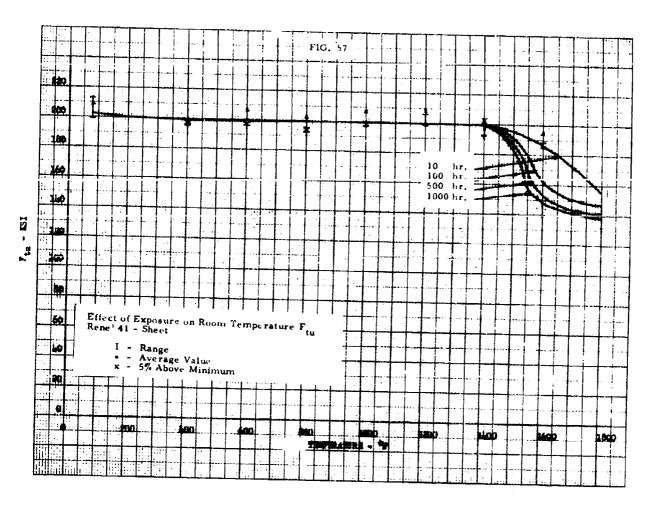


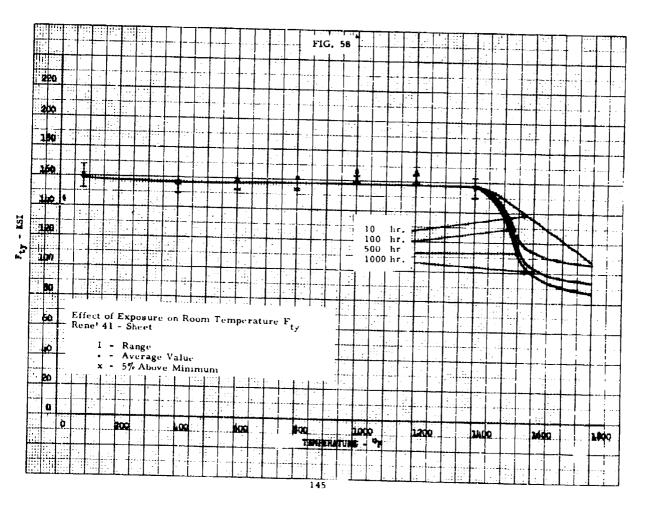


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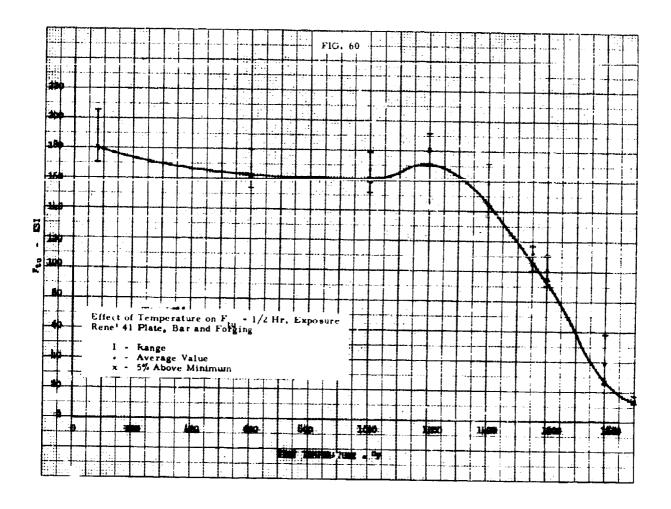


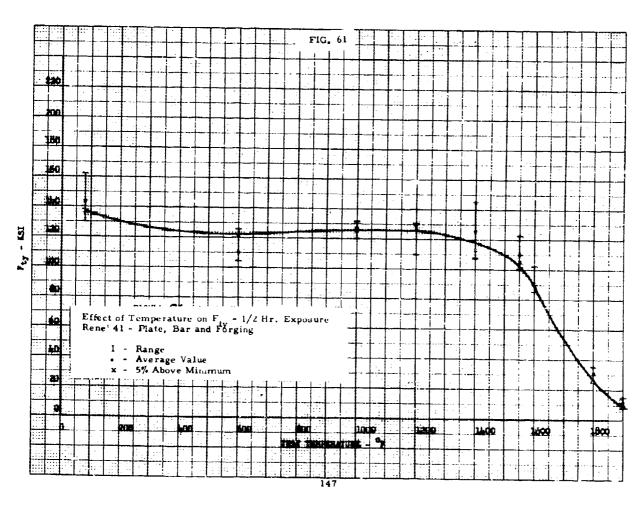


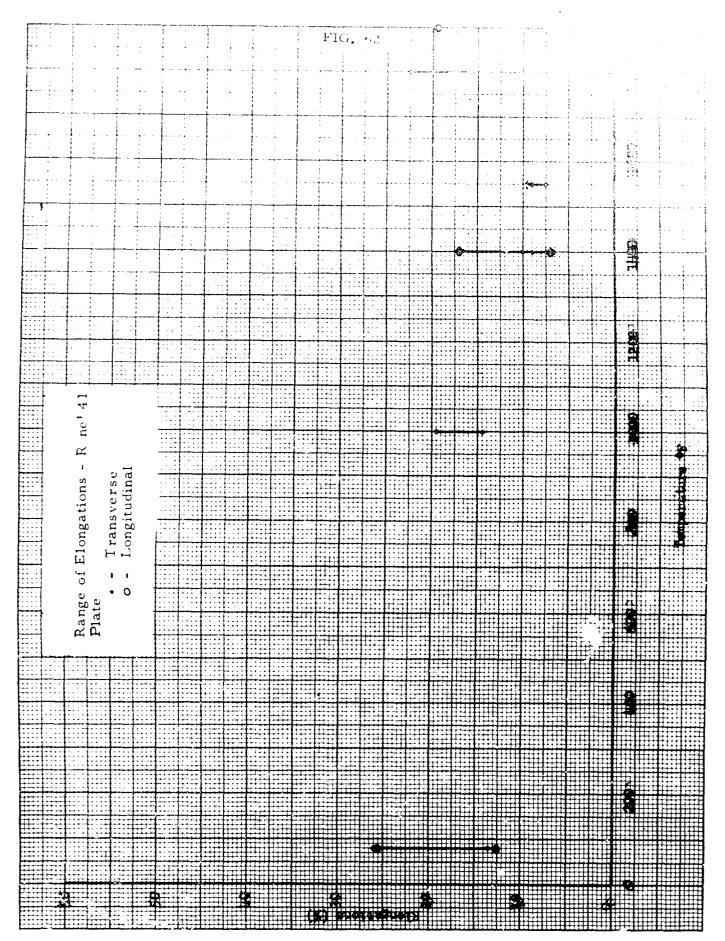




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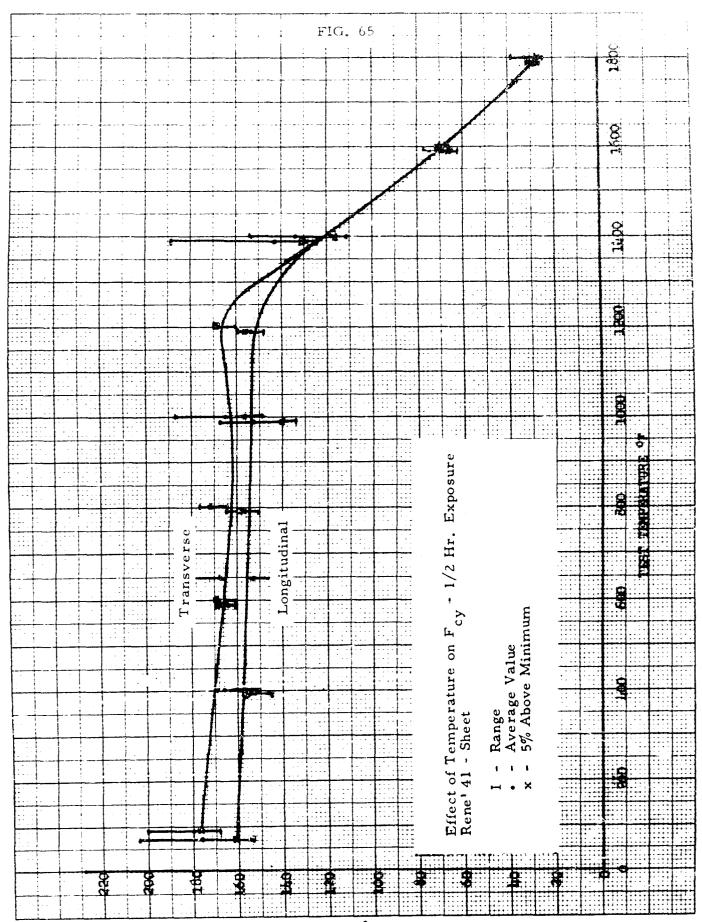




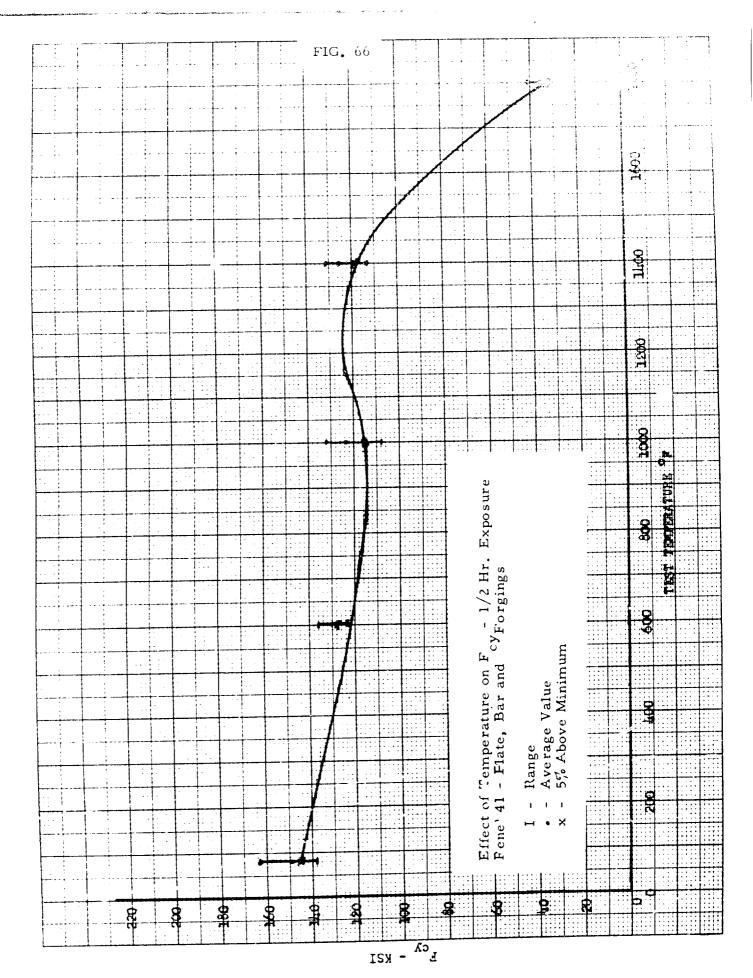
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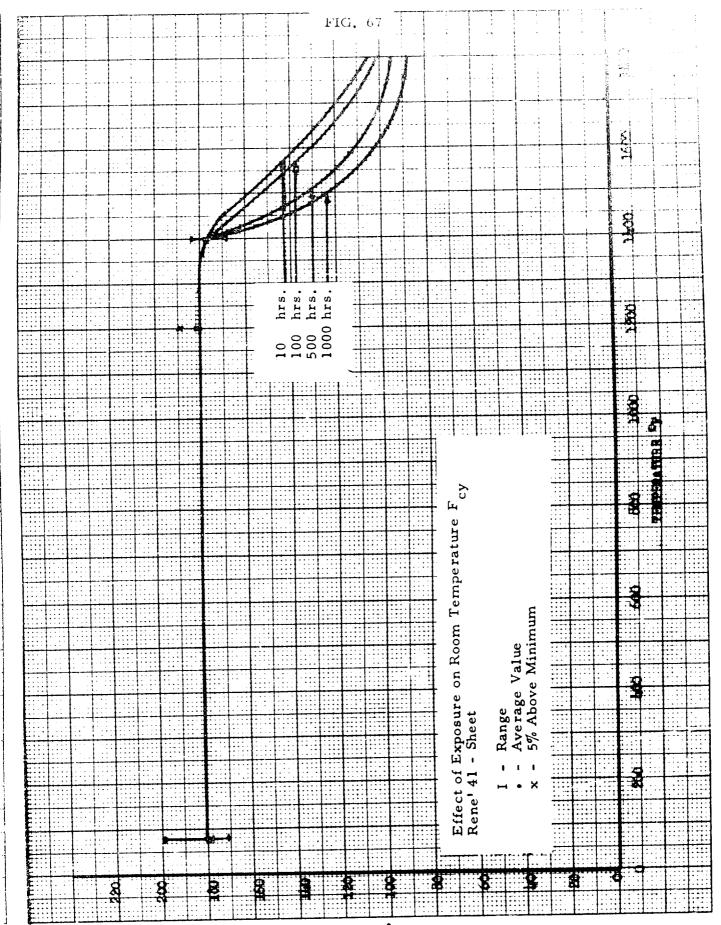
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### SECTION 7.1.2 COMPRESSION

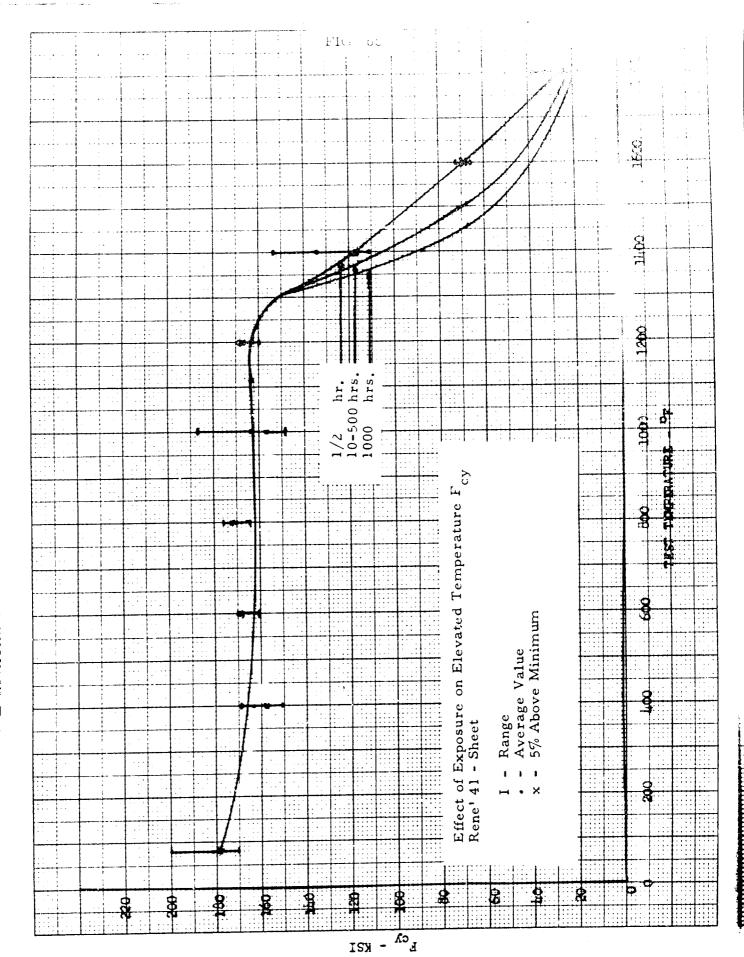


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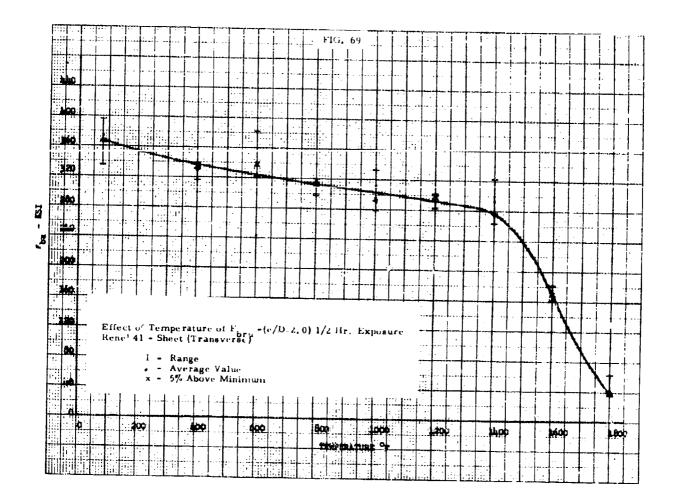


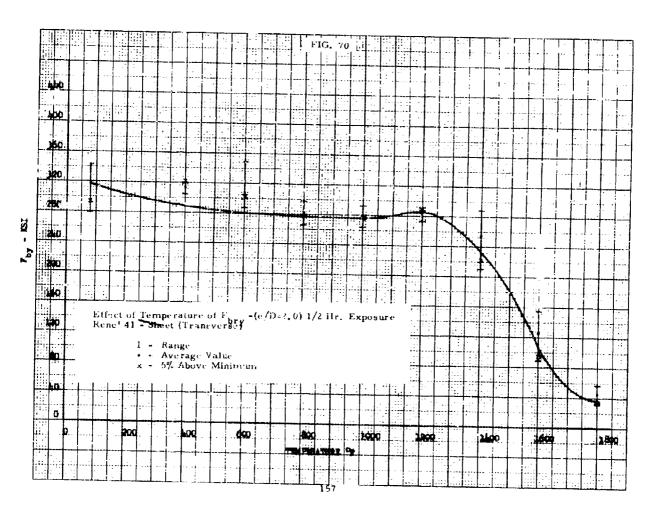


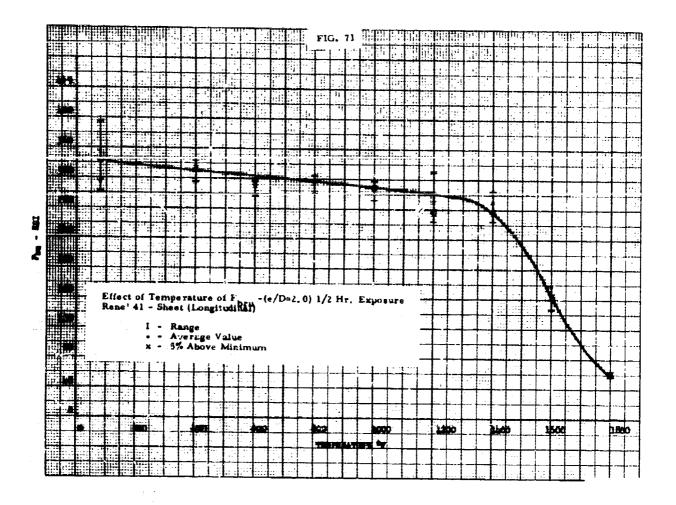
LCA - KSI

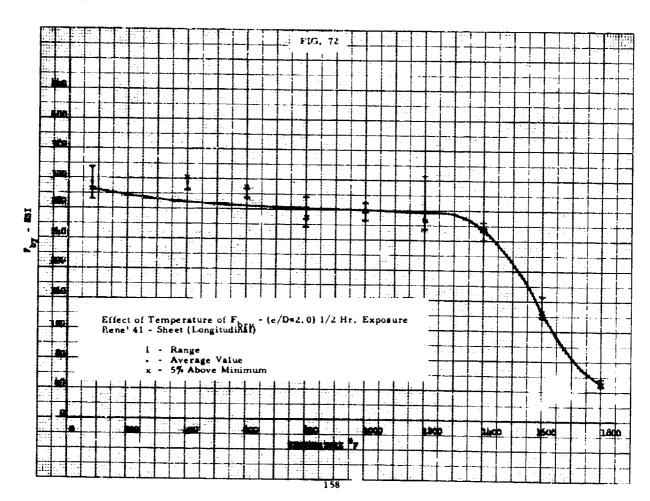


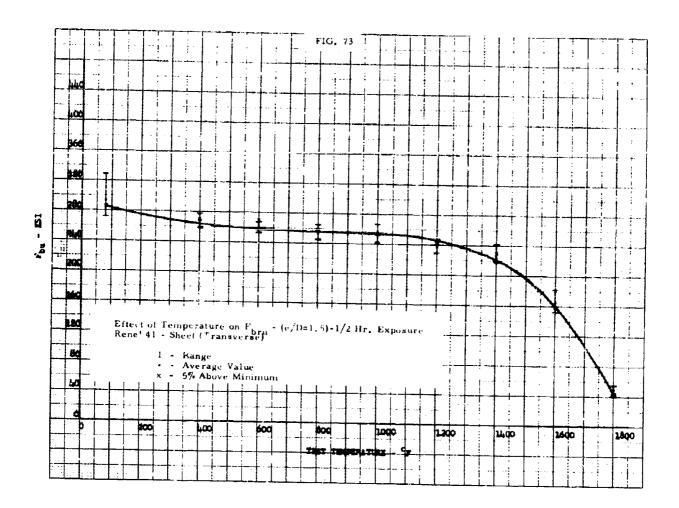
SECTION 7.1.3 BEARING

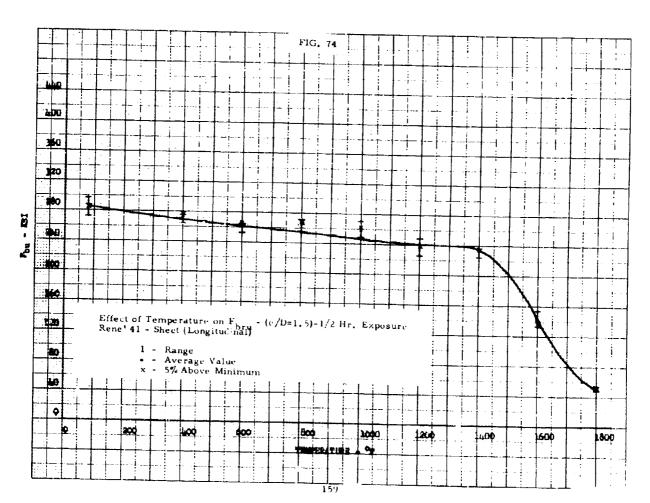


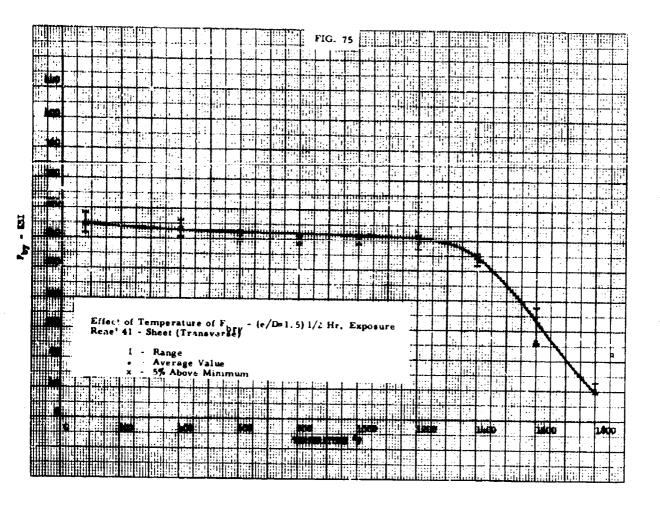


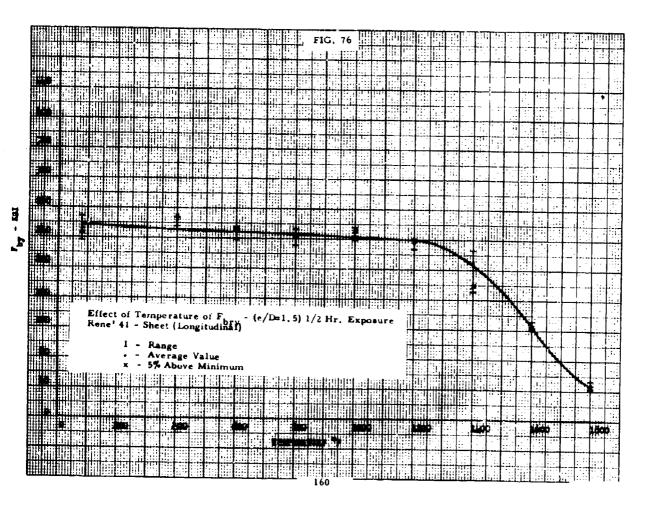


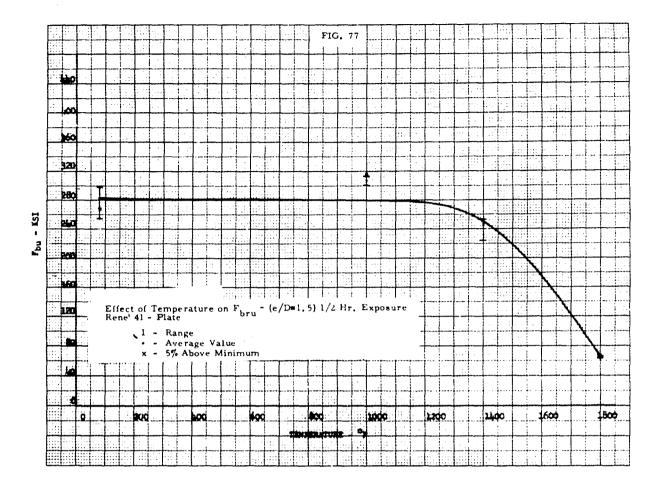


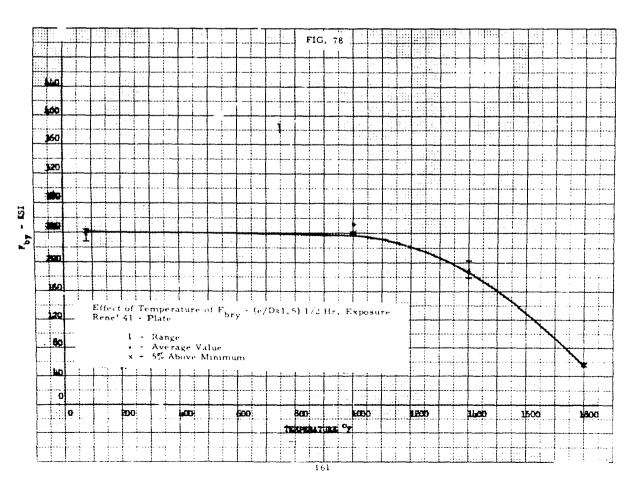


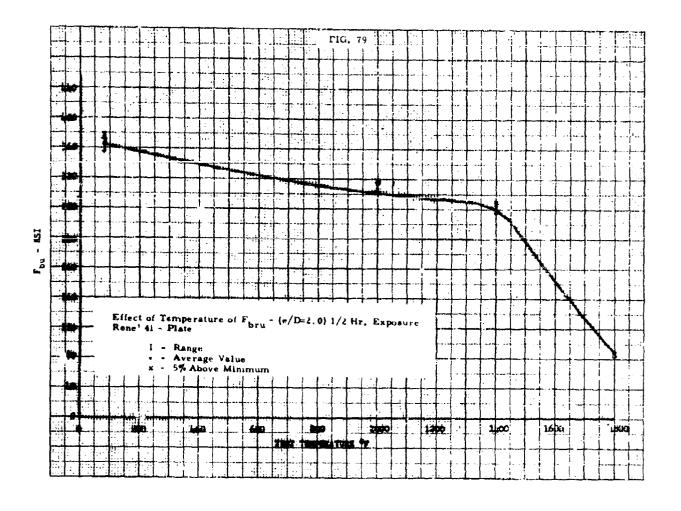


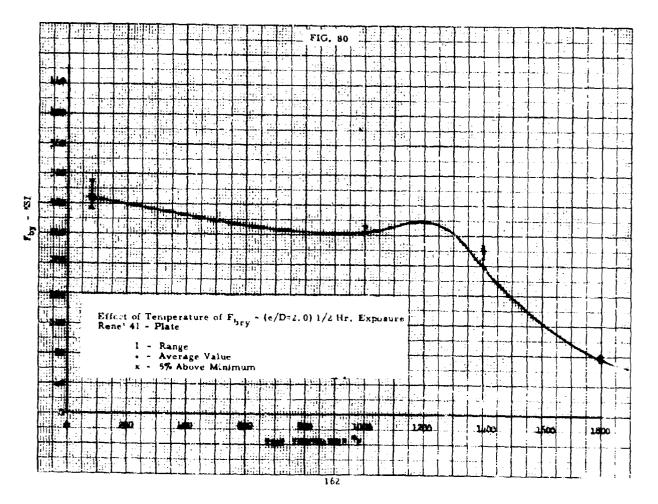


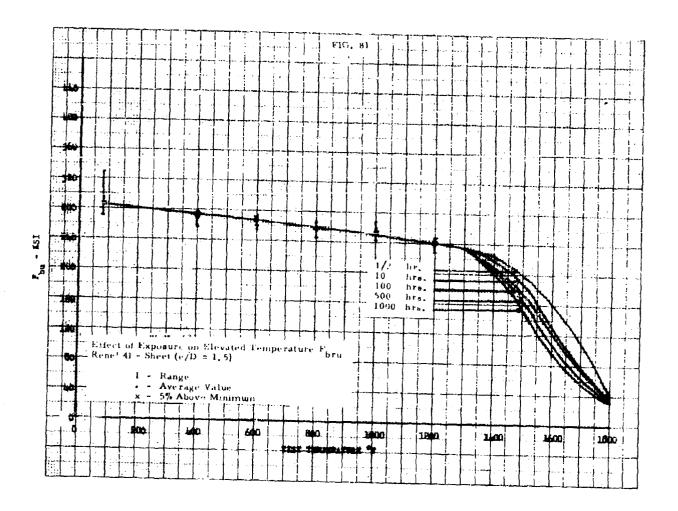


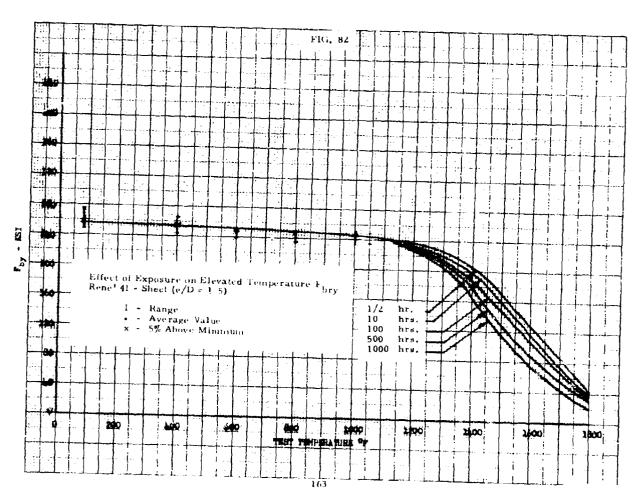


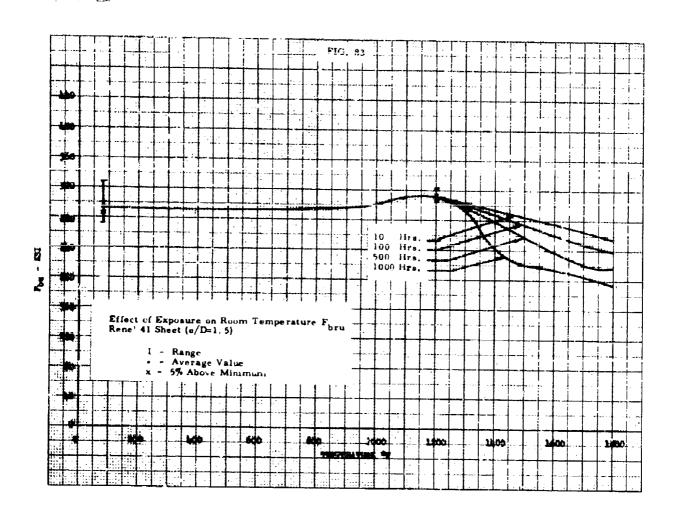


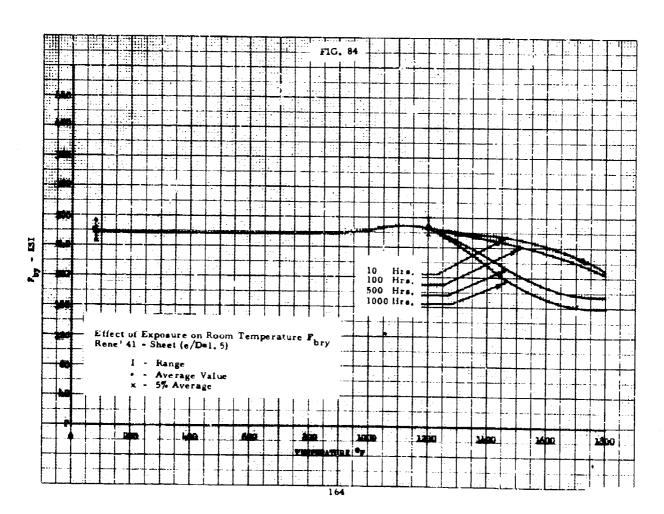




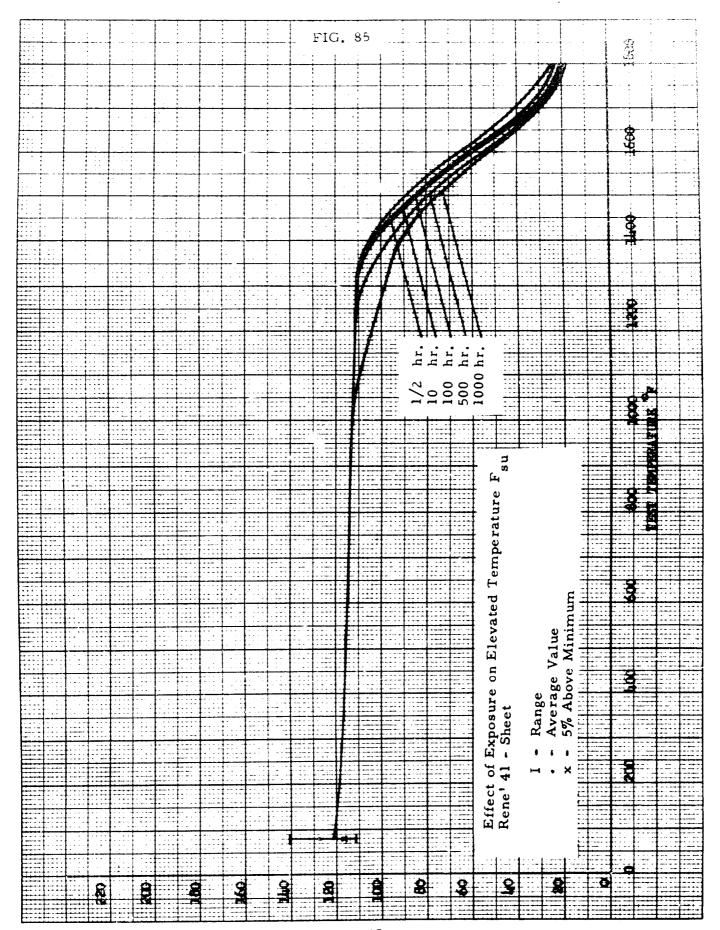




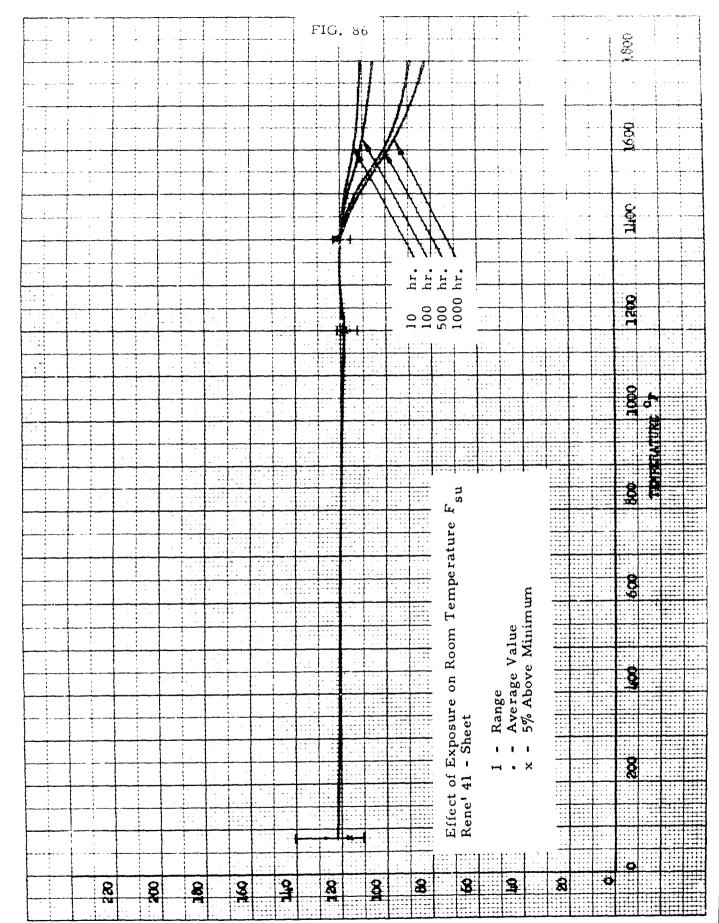




SECTION 7.1.4 SHEAR



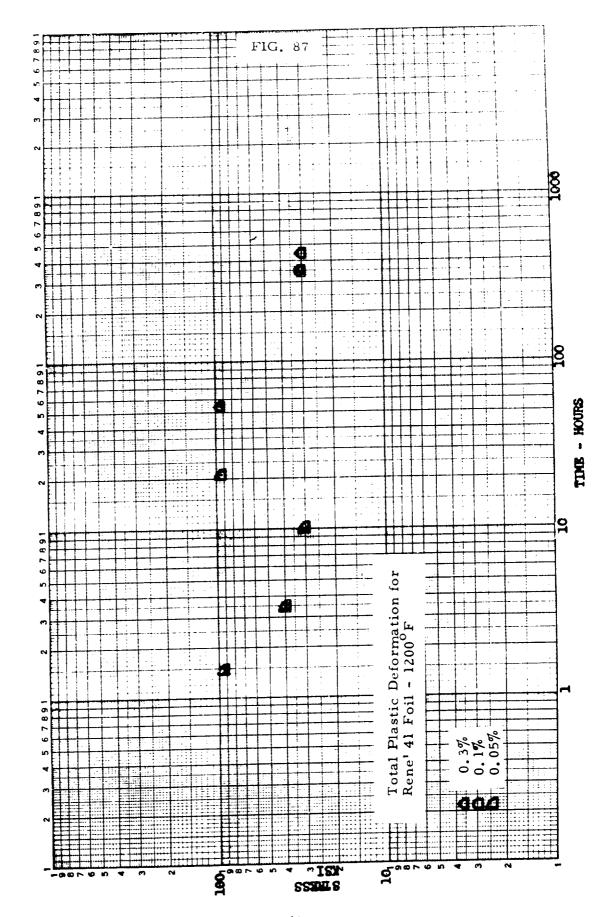
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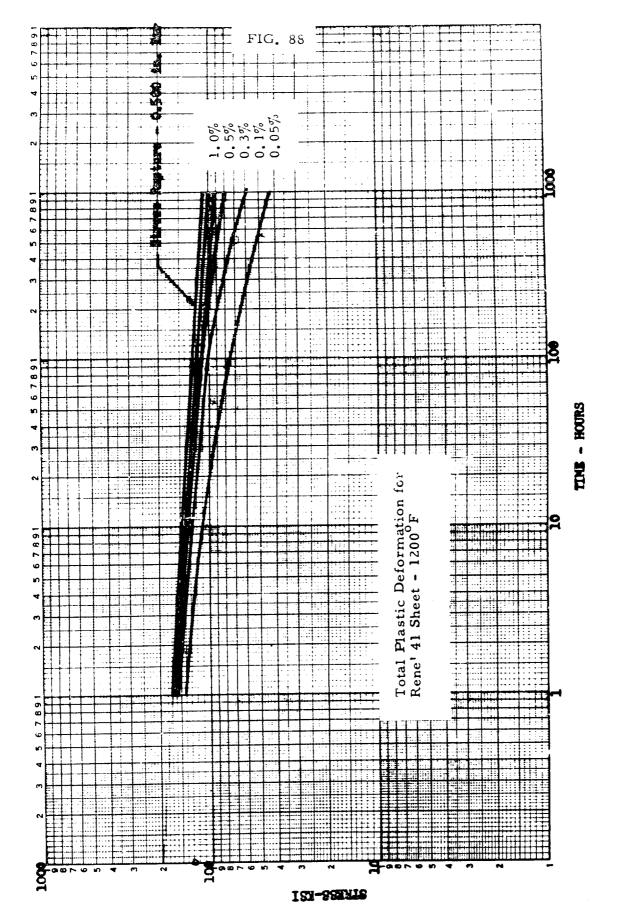


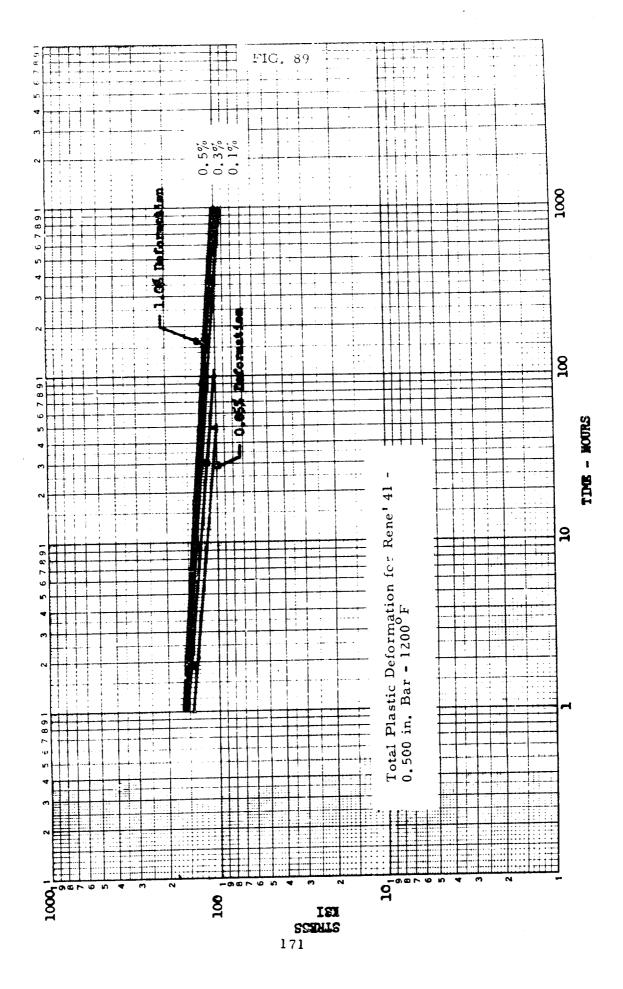
Len - KSI

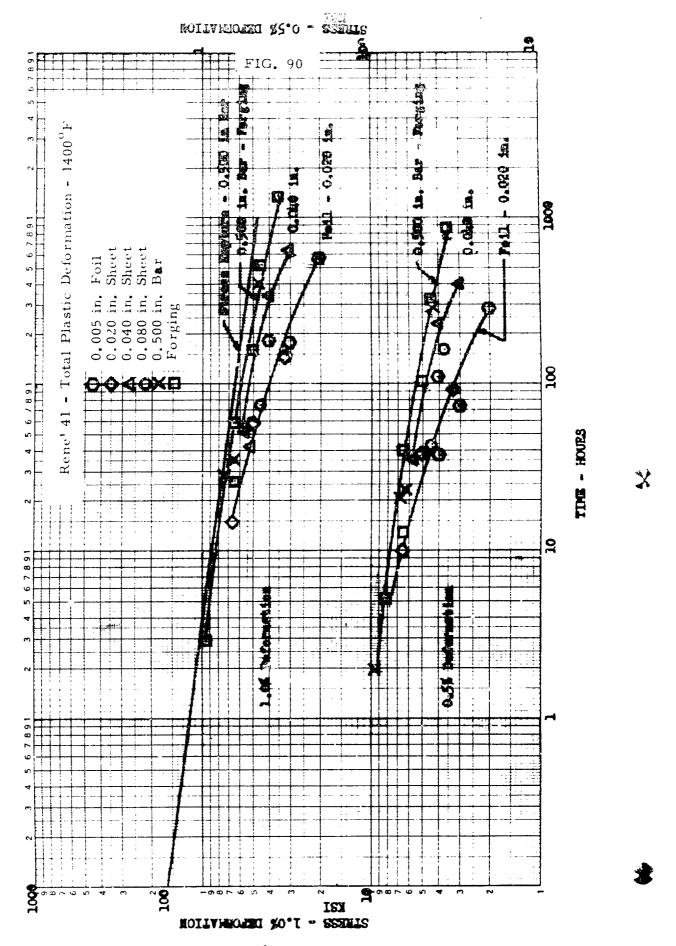
**SECTION** 7.1.5

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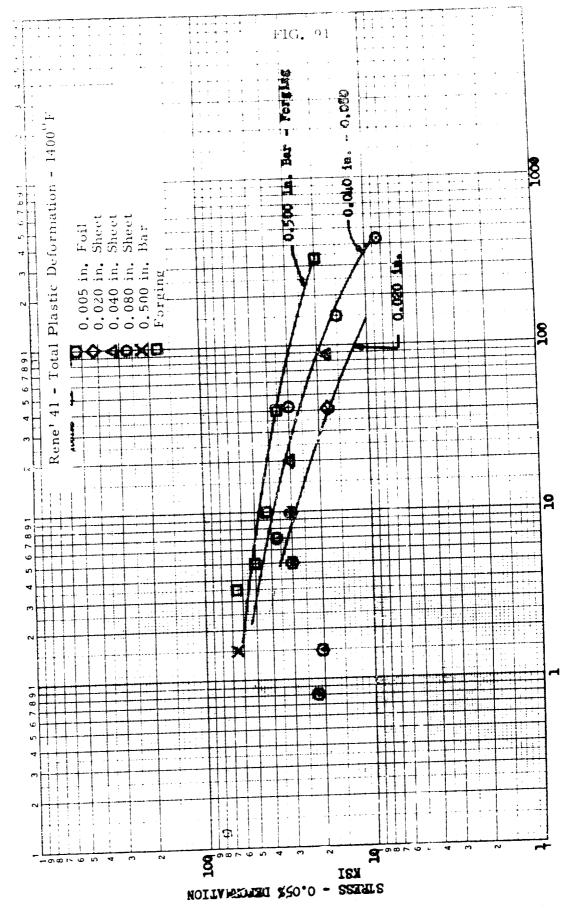


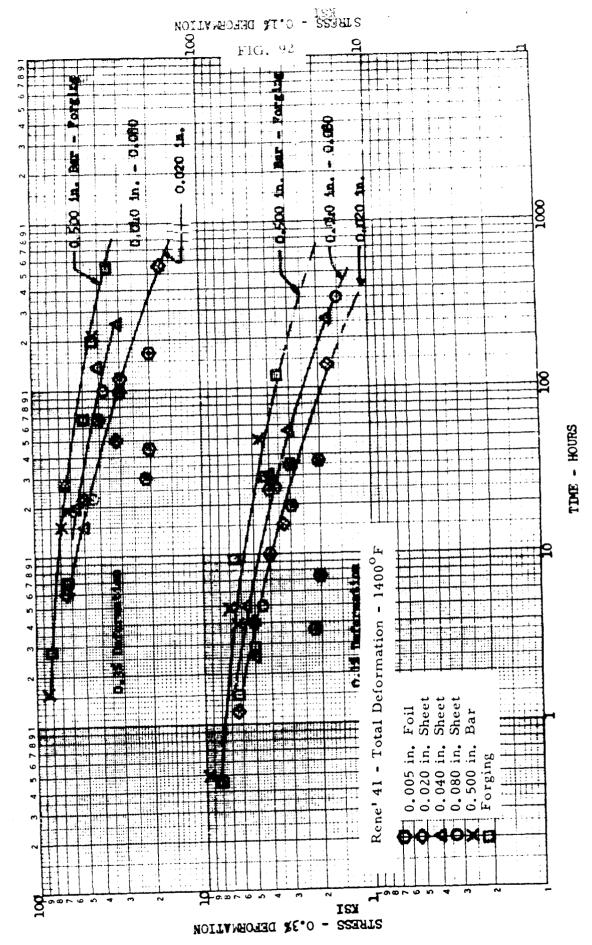


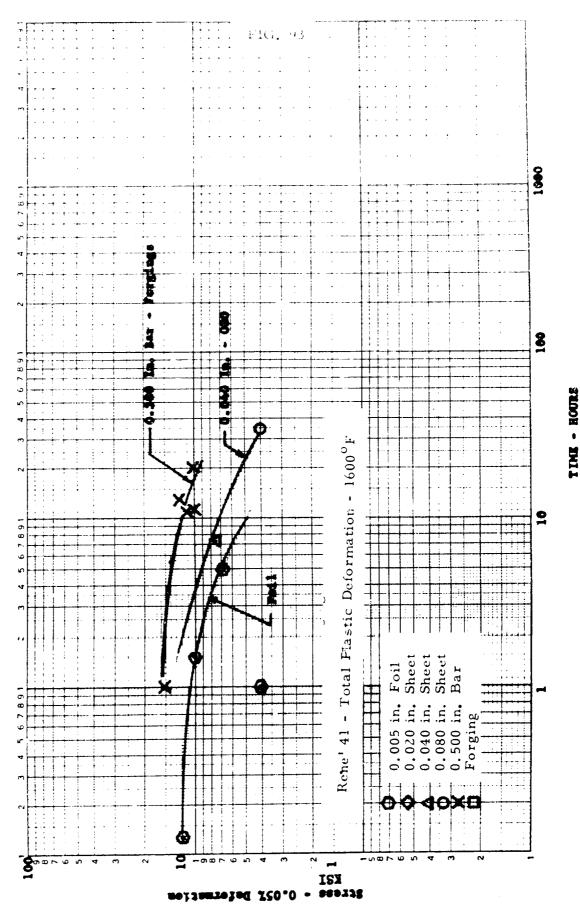


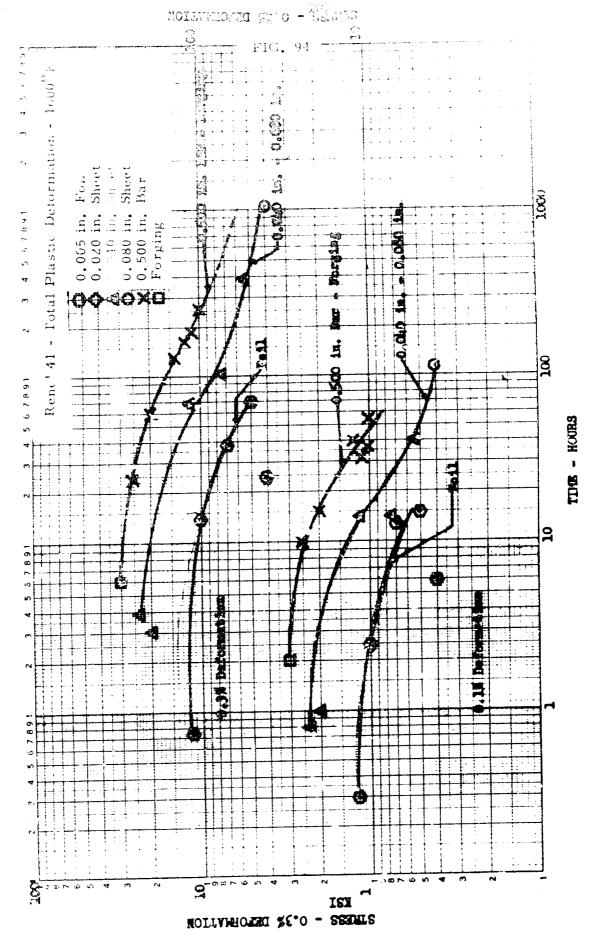


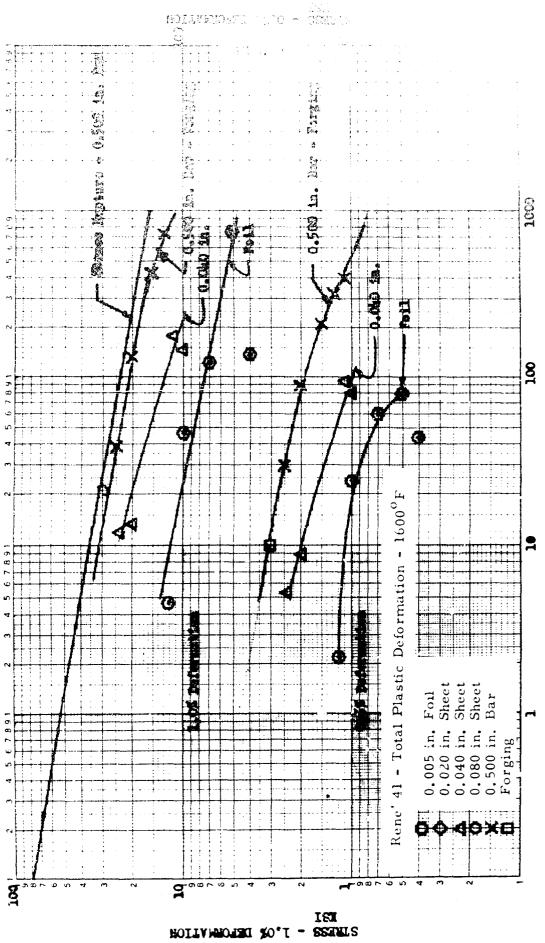






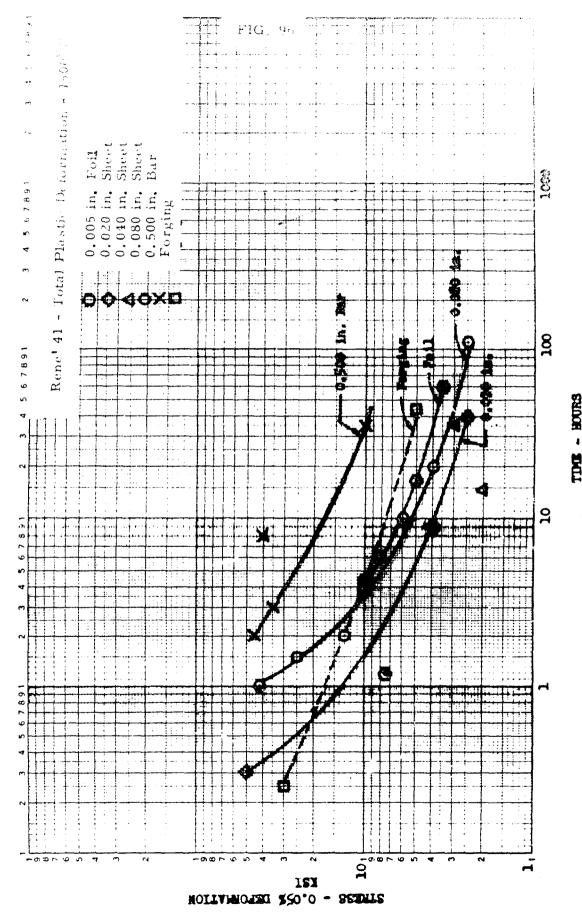


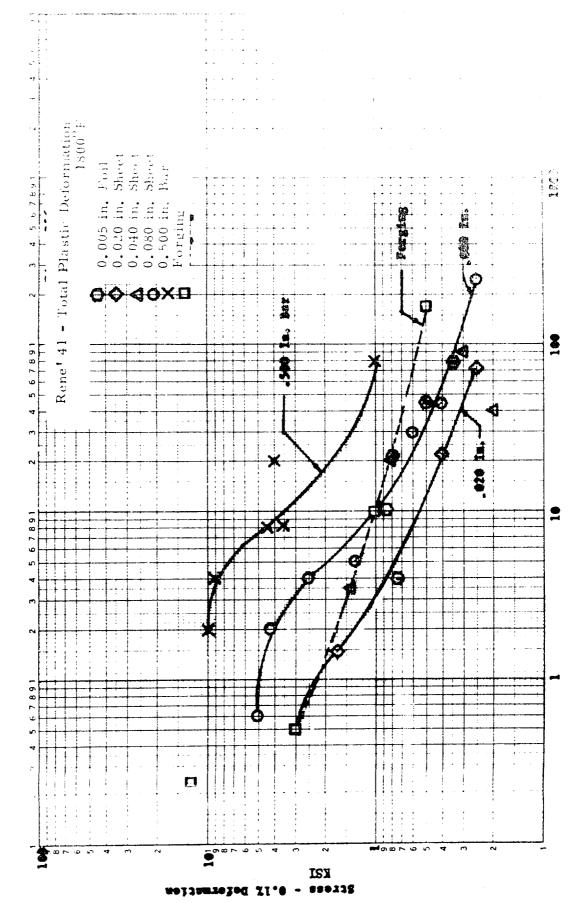


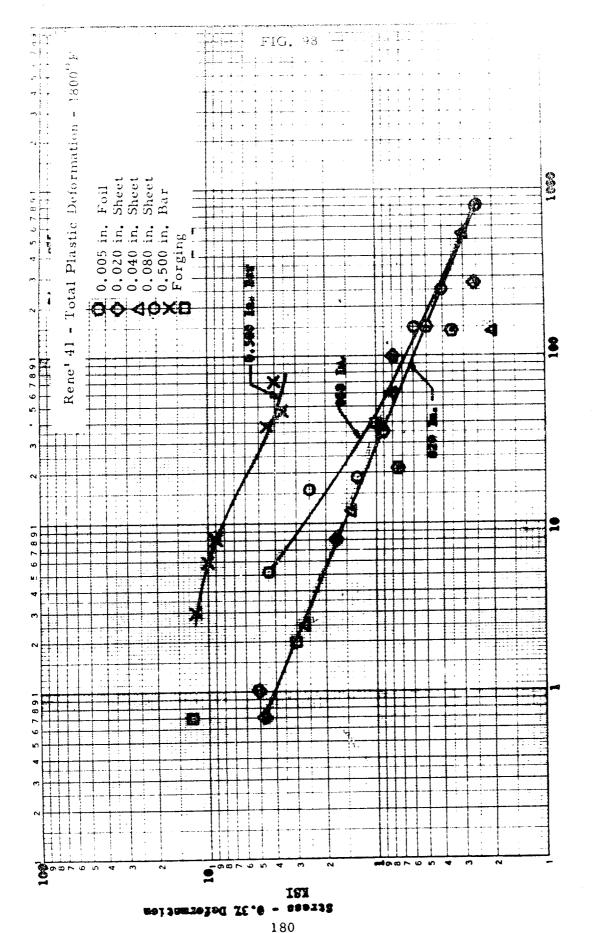


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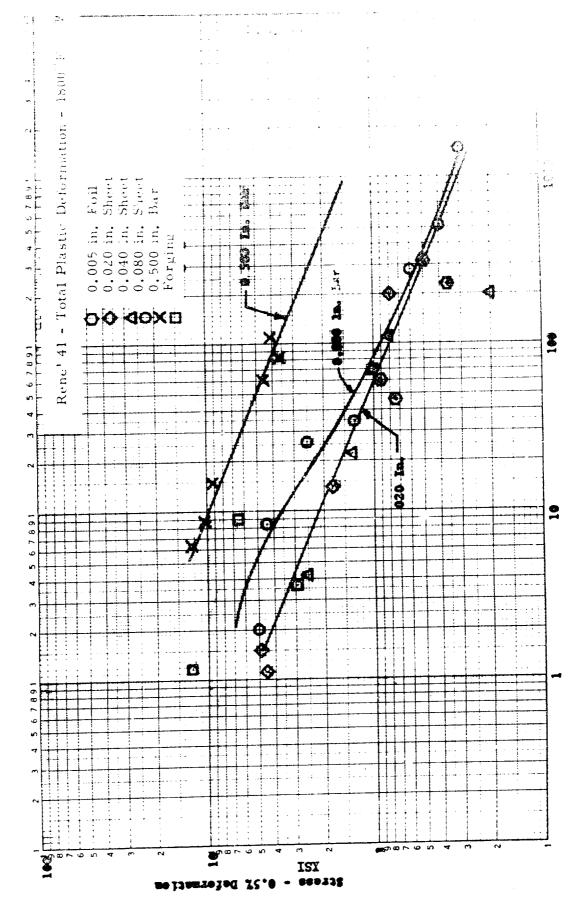
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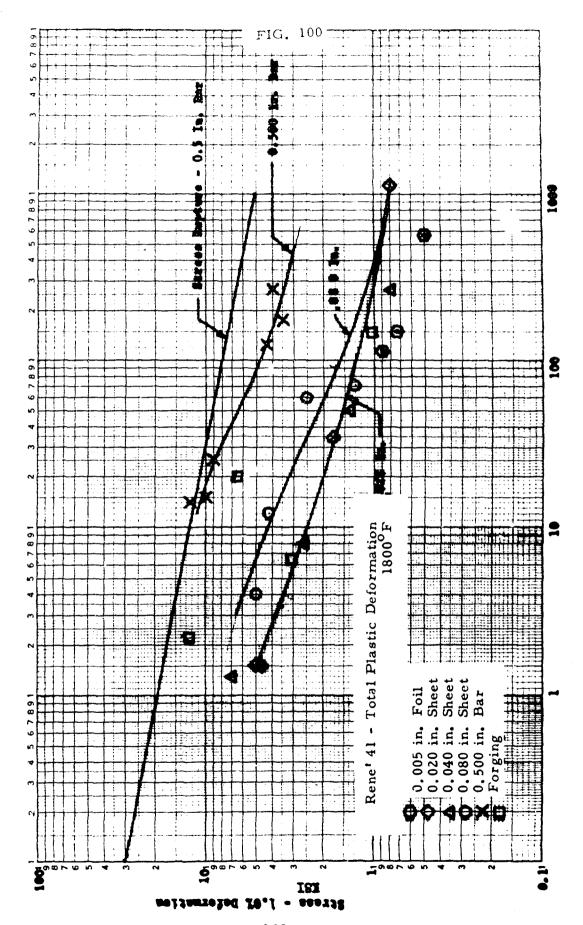




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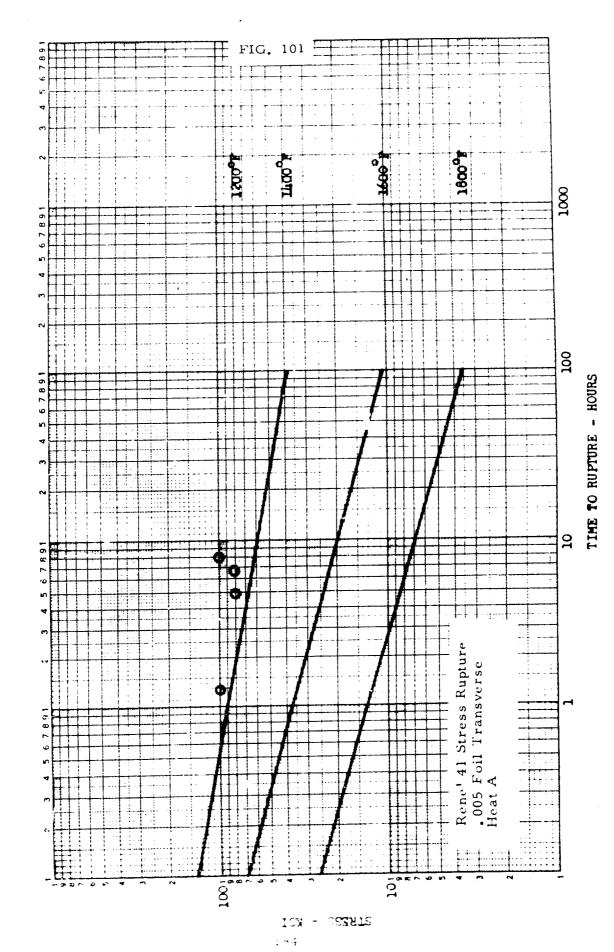


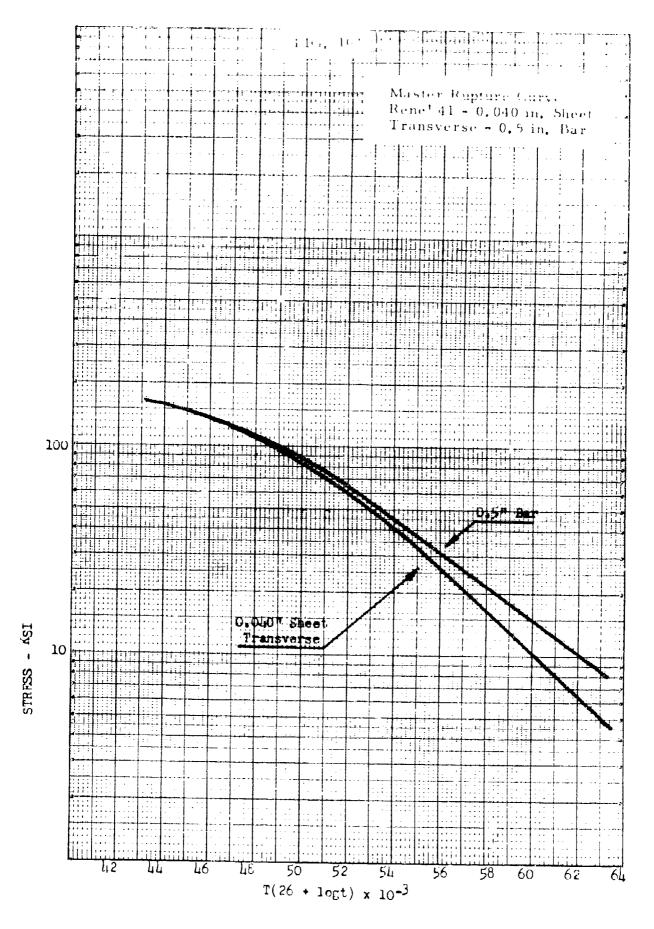


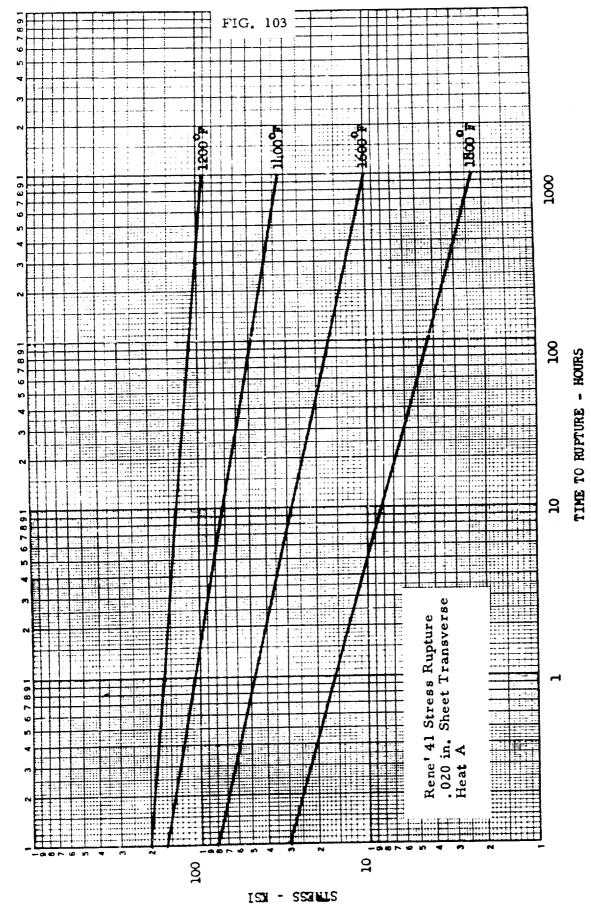


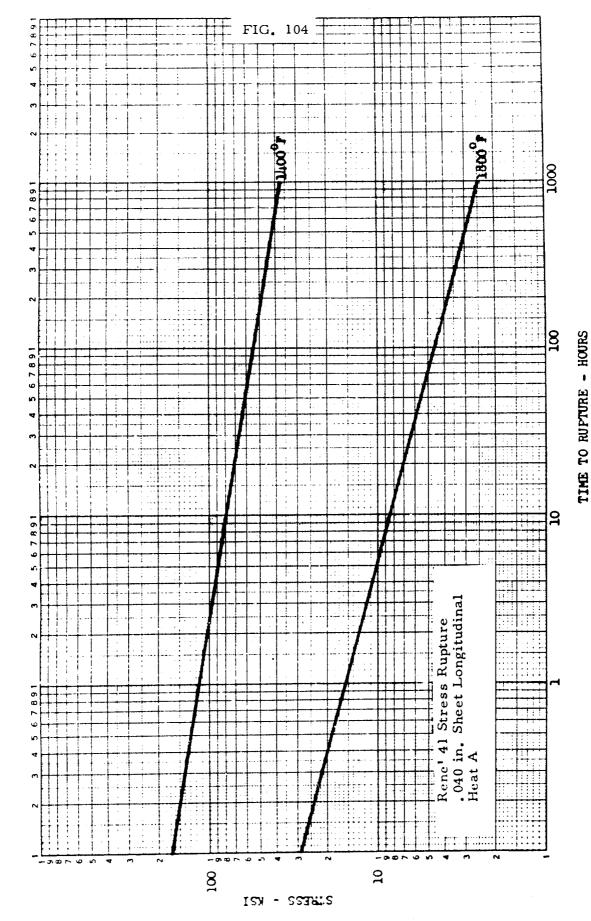
## SECTION VII

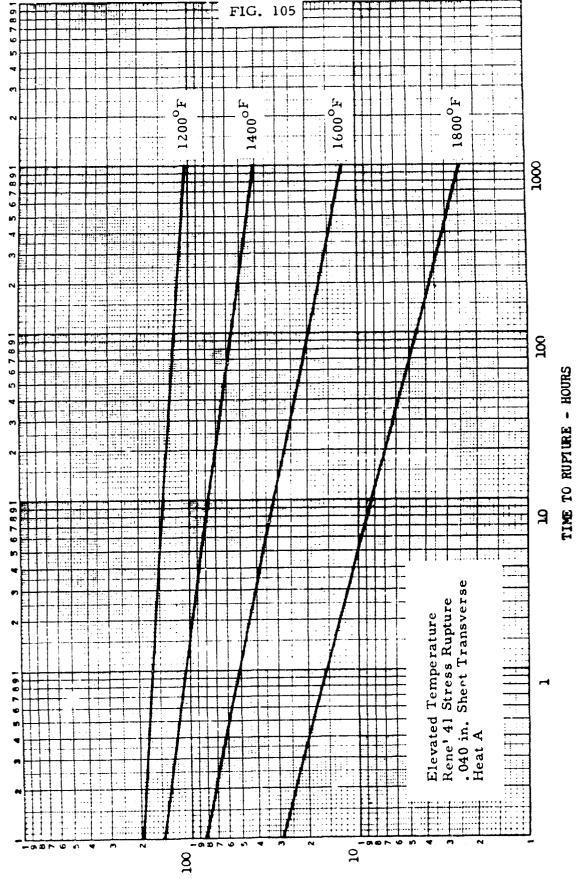
## SECTION 7.1.6 STRESS RUPTURE



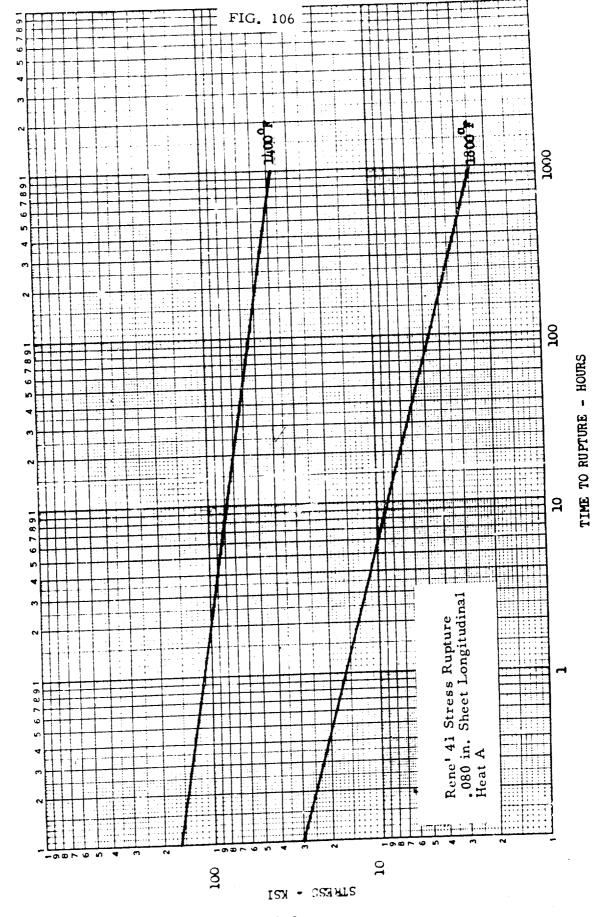


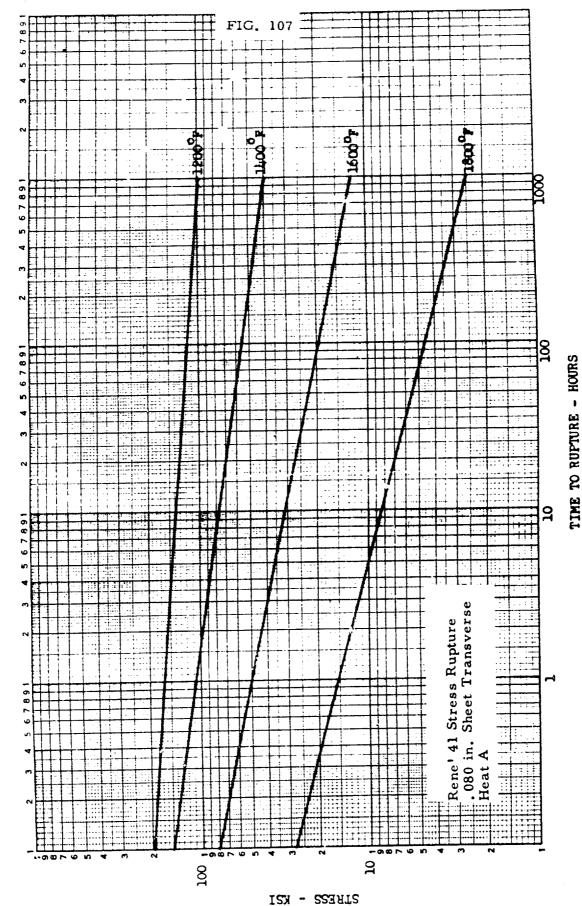






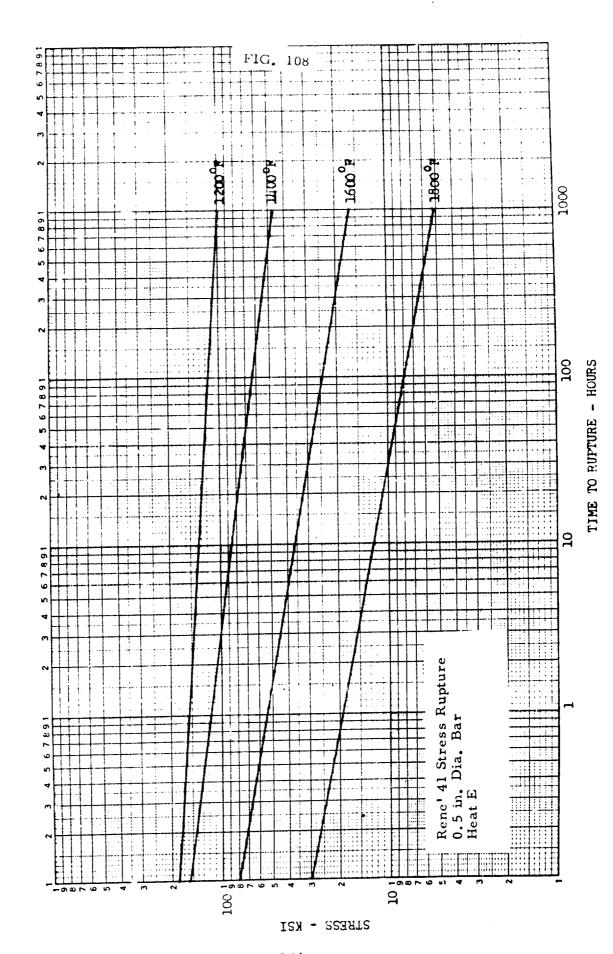
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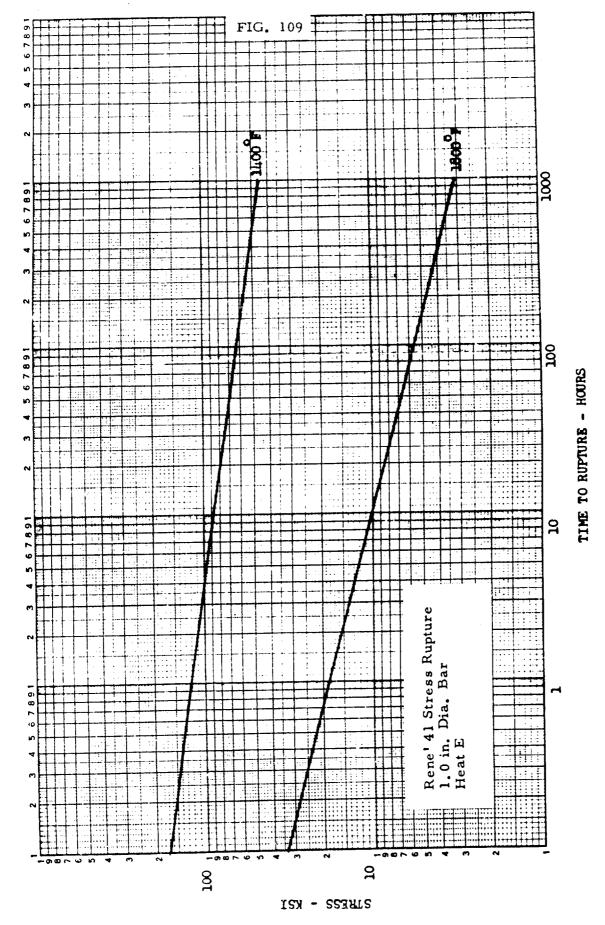


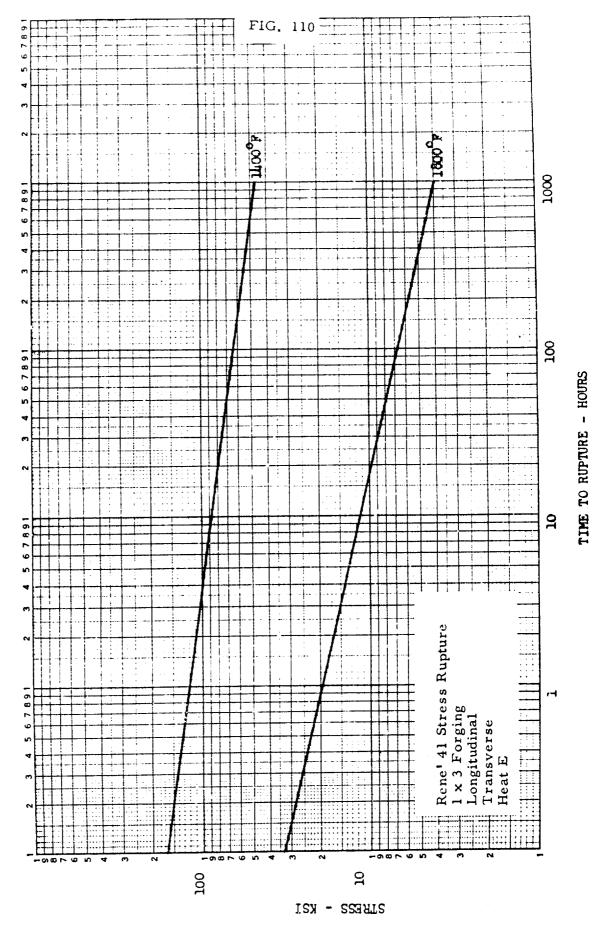


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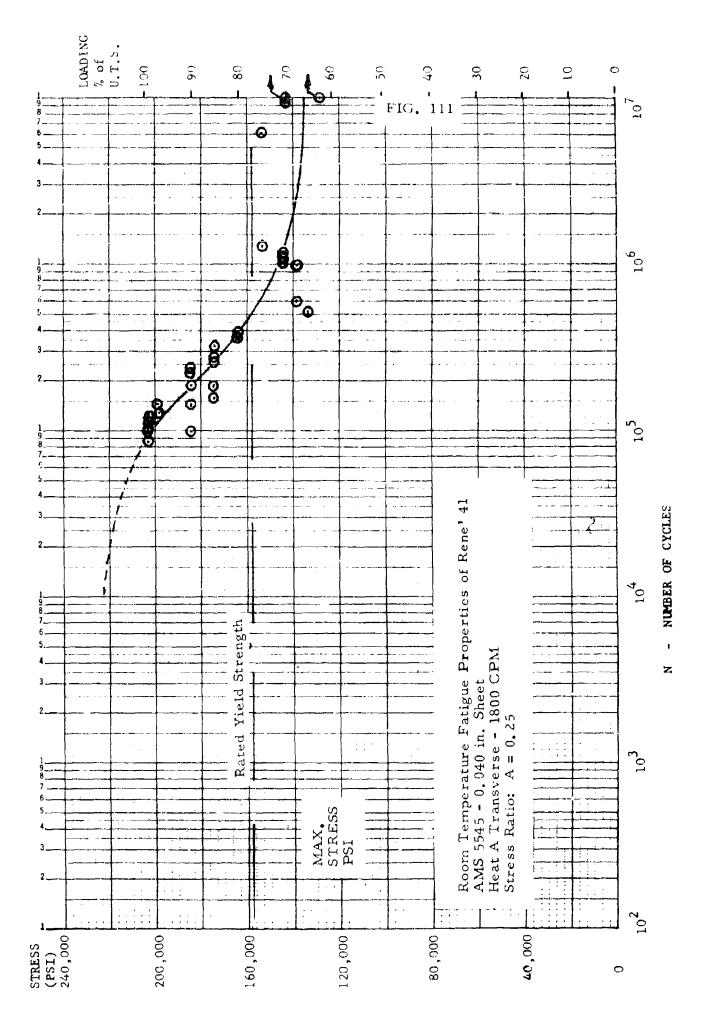


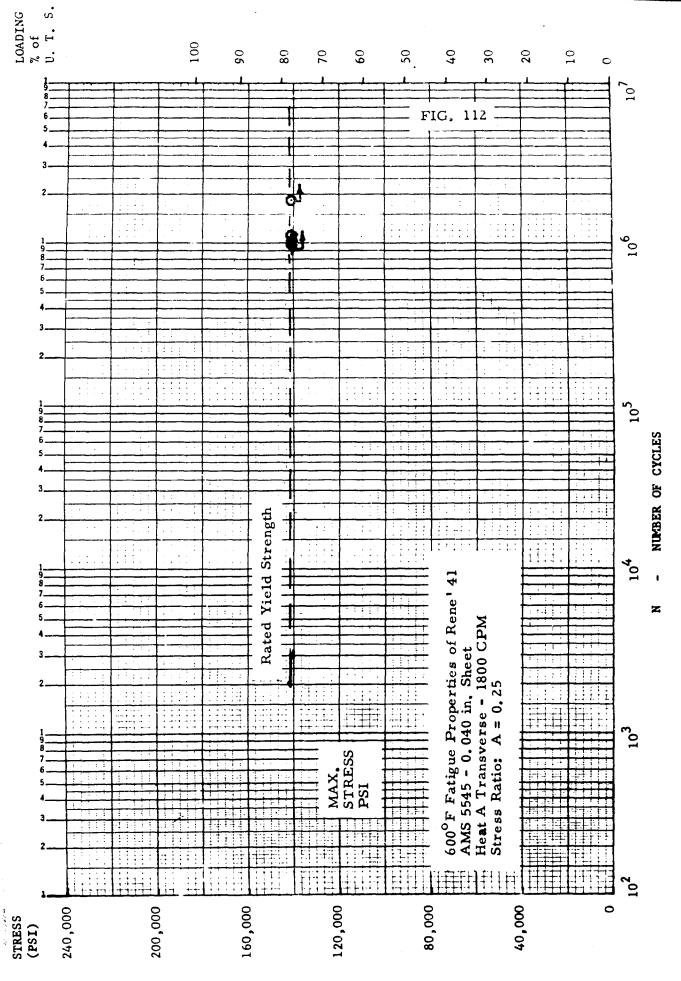


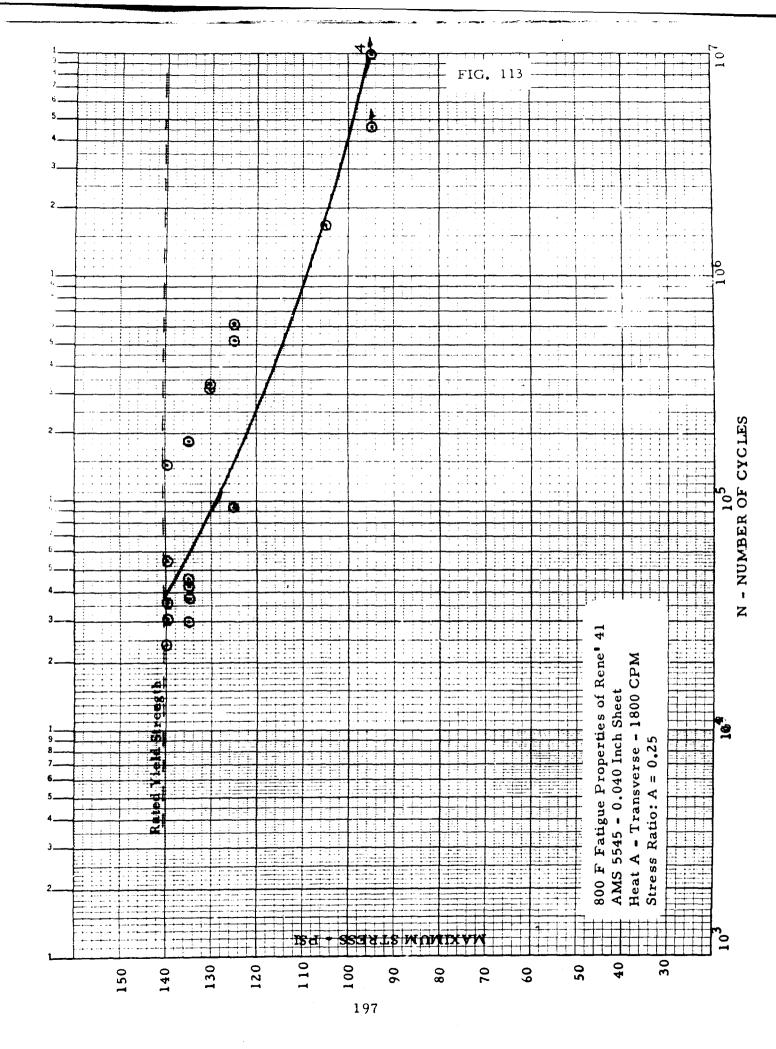


## SECTION VII

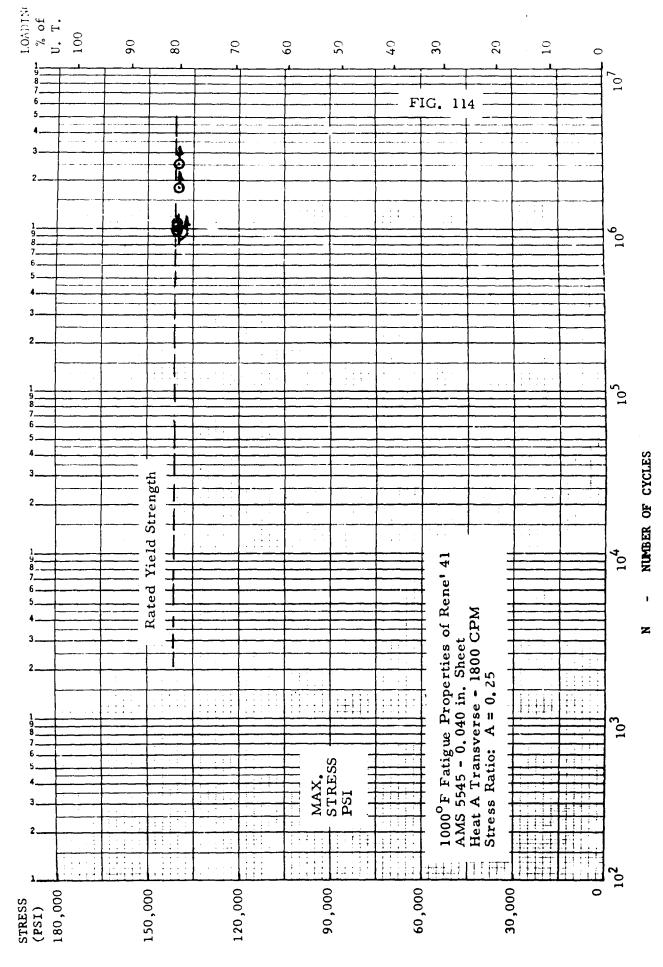
SECTION 7.1.7 FATIGUE

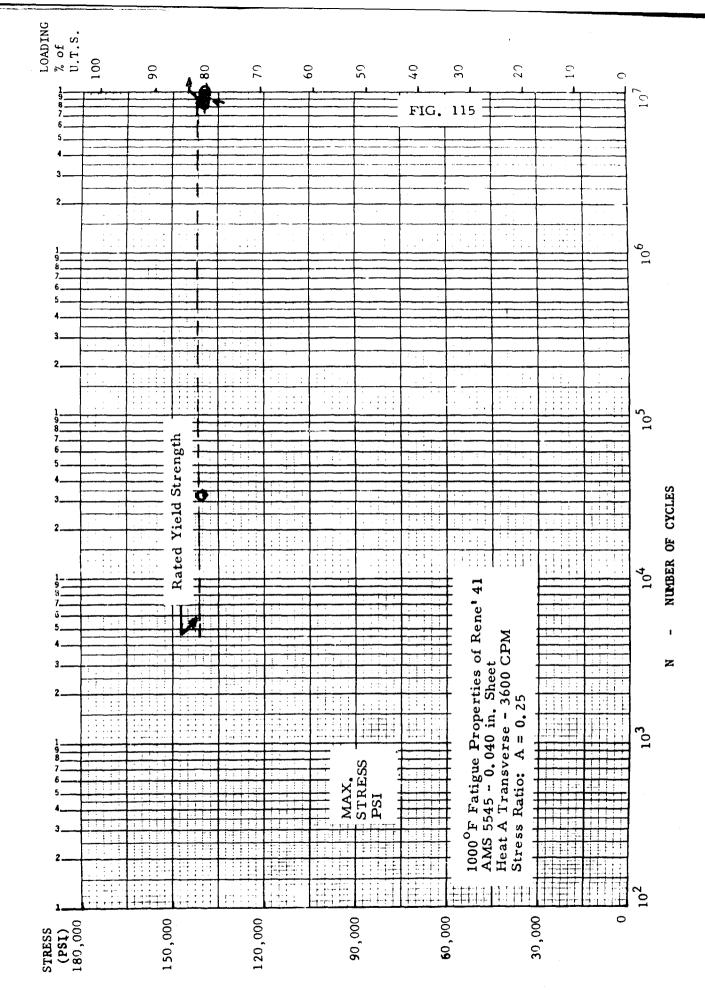


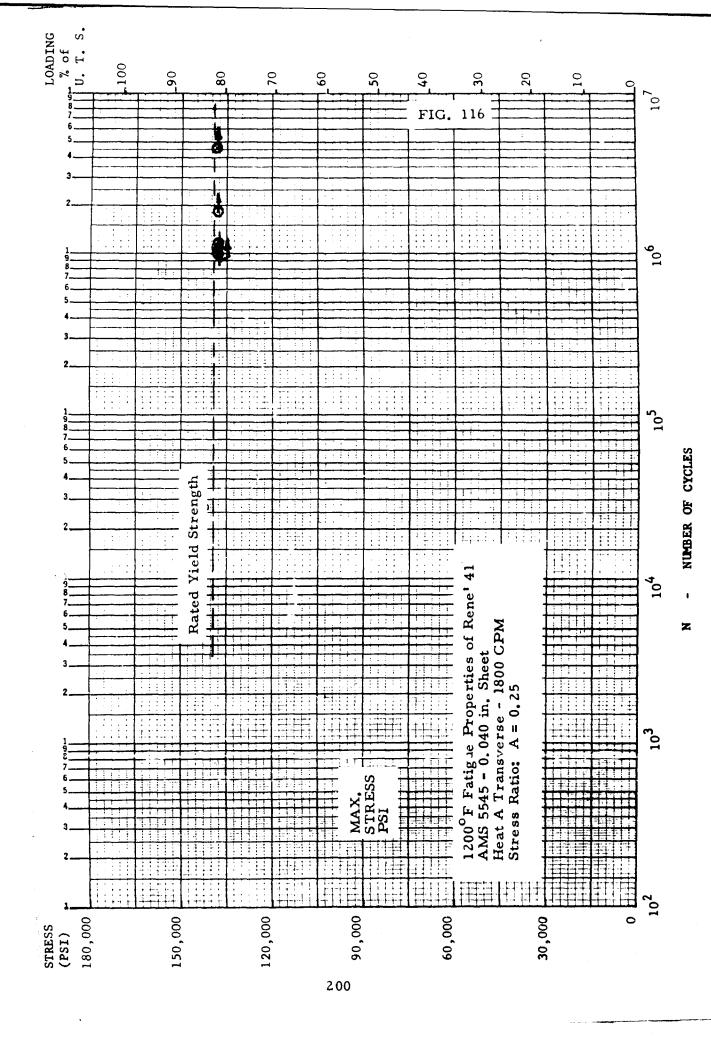


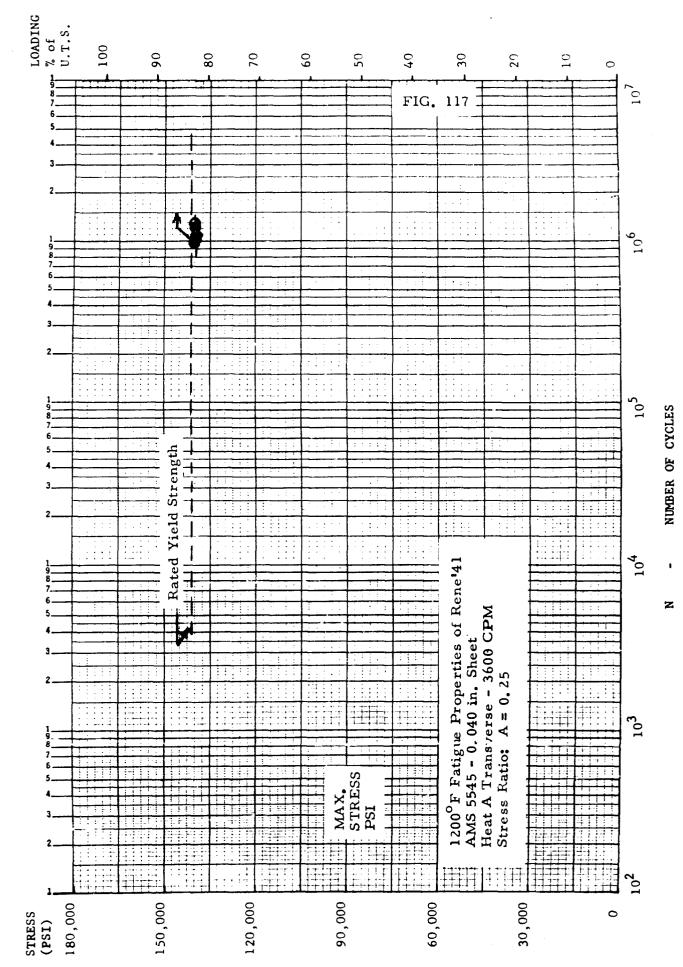


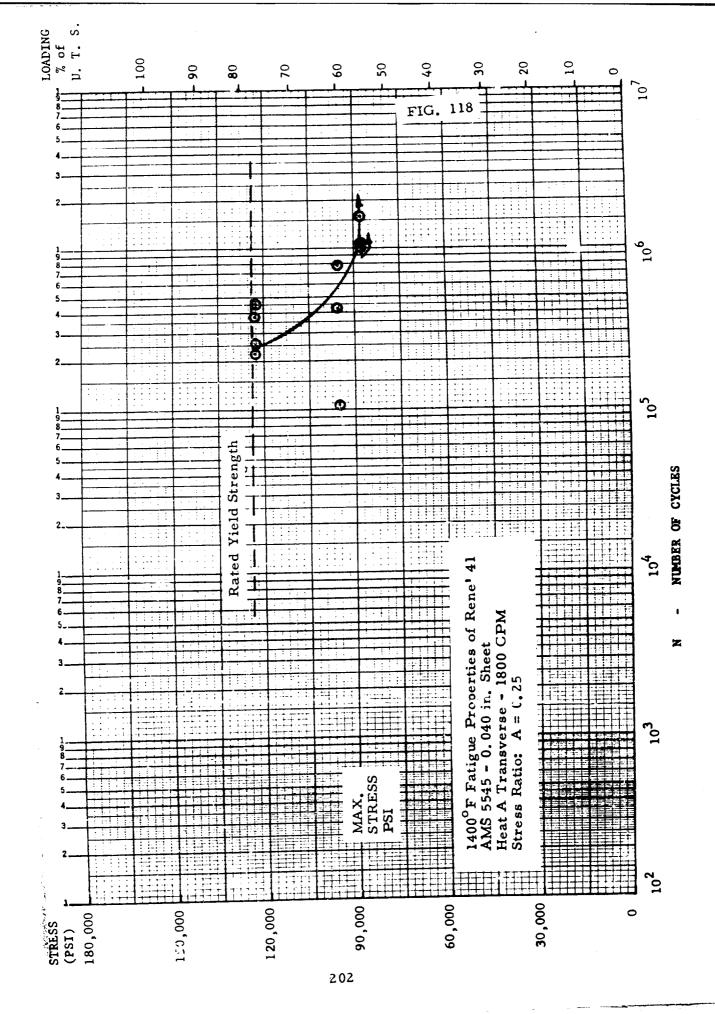
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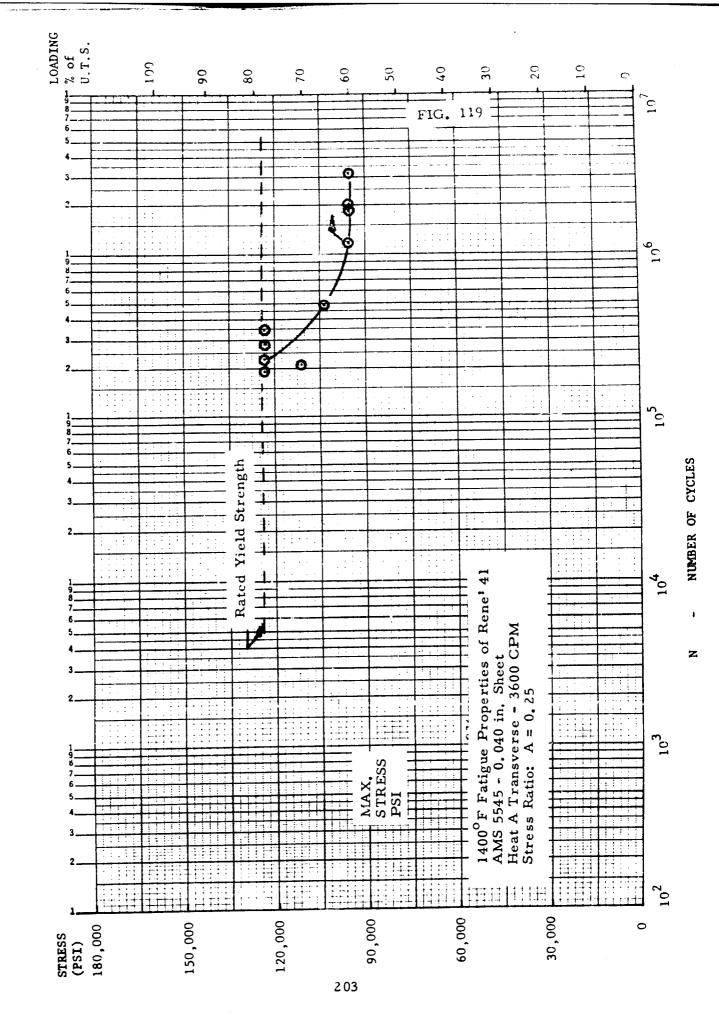




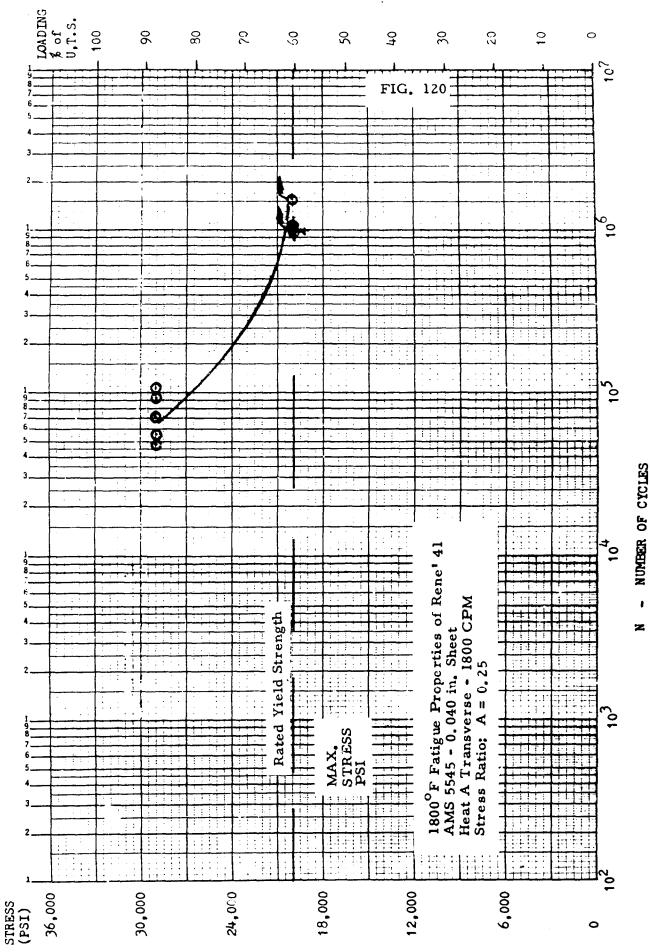




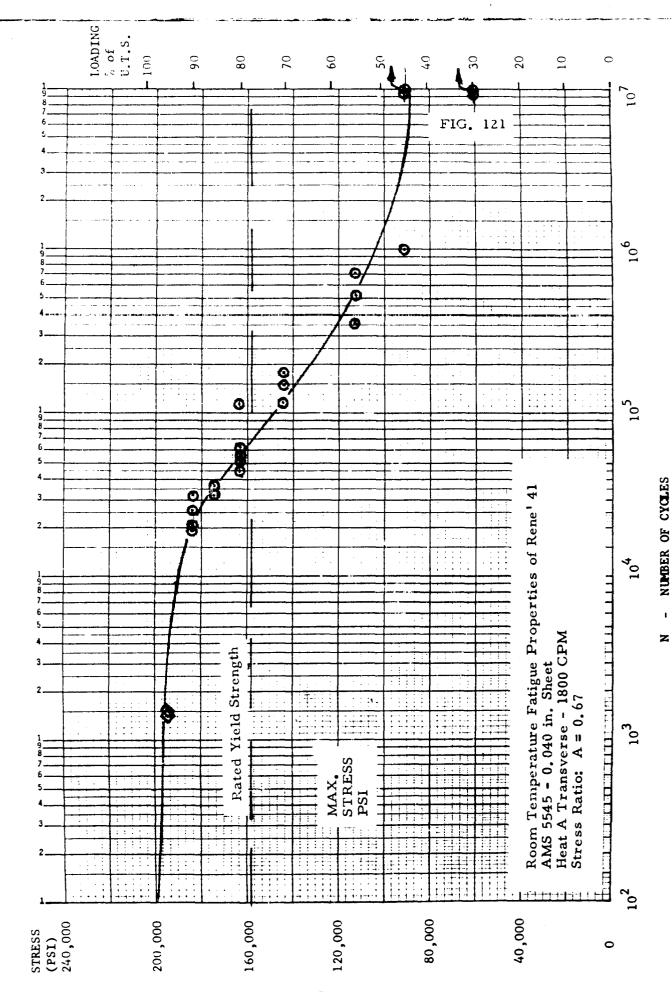




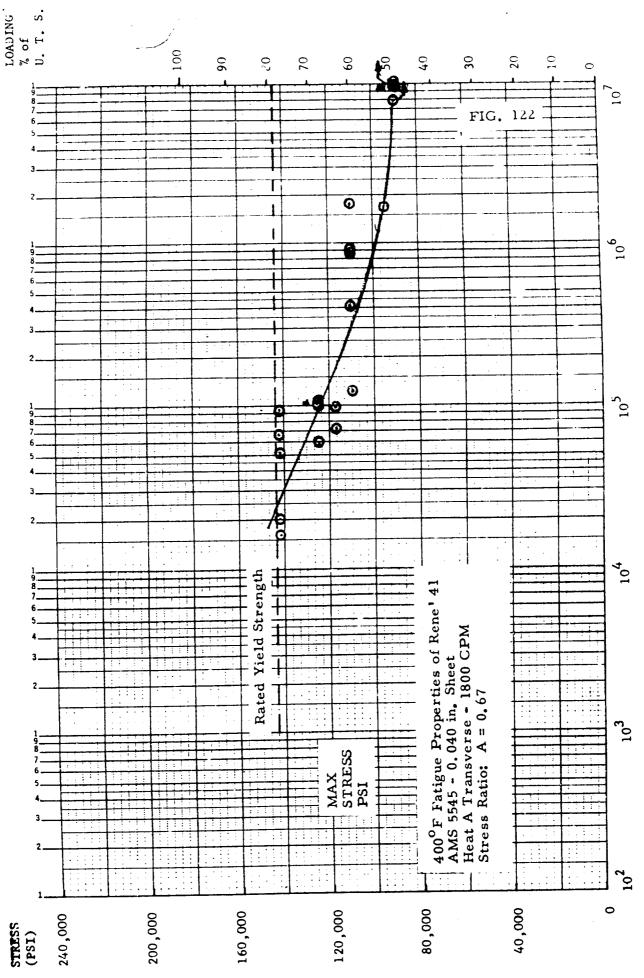
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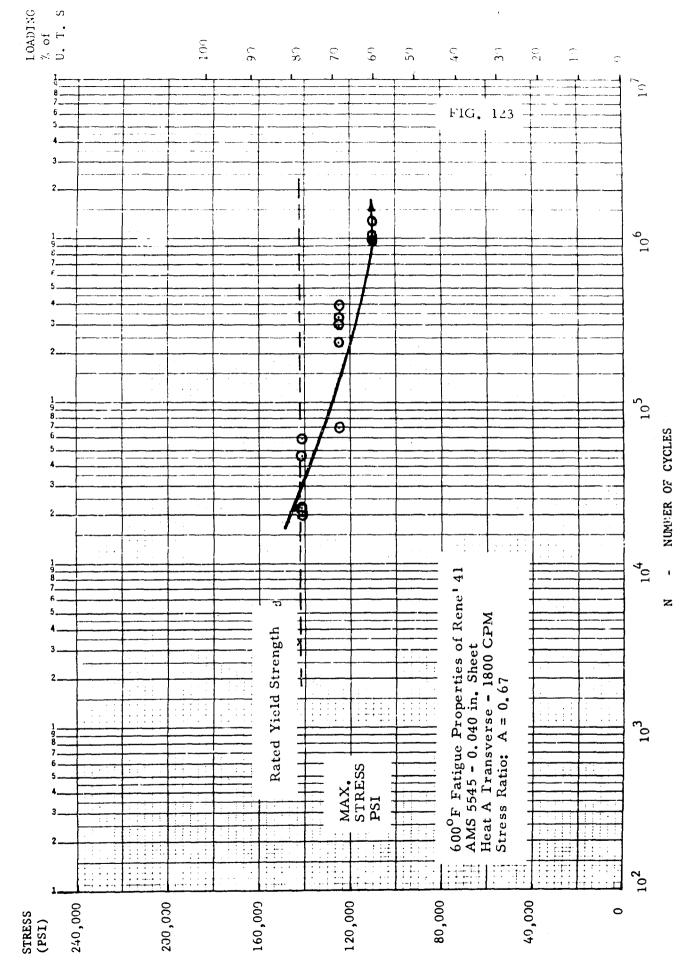


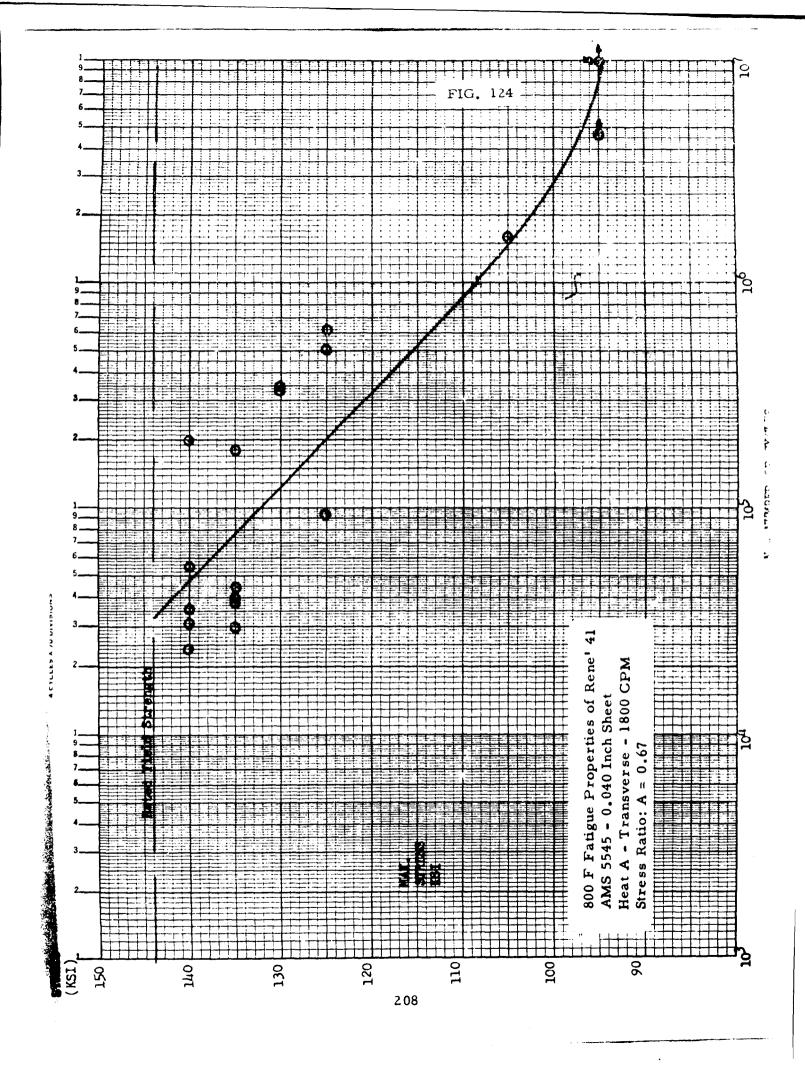
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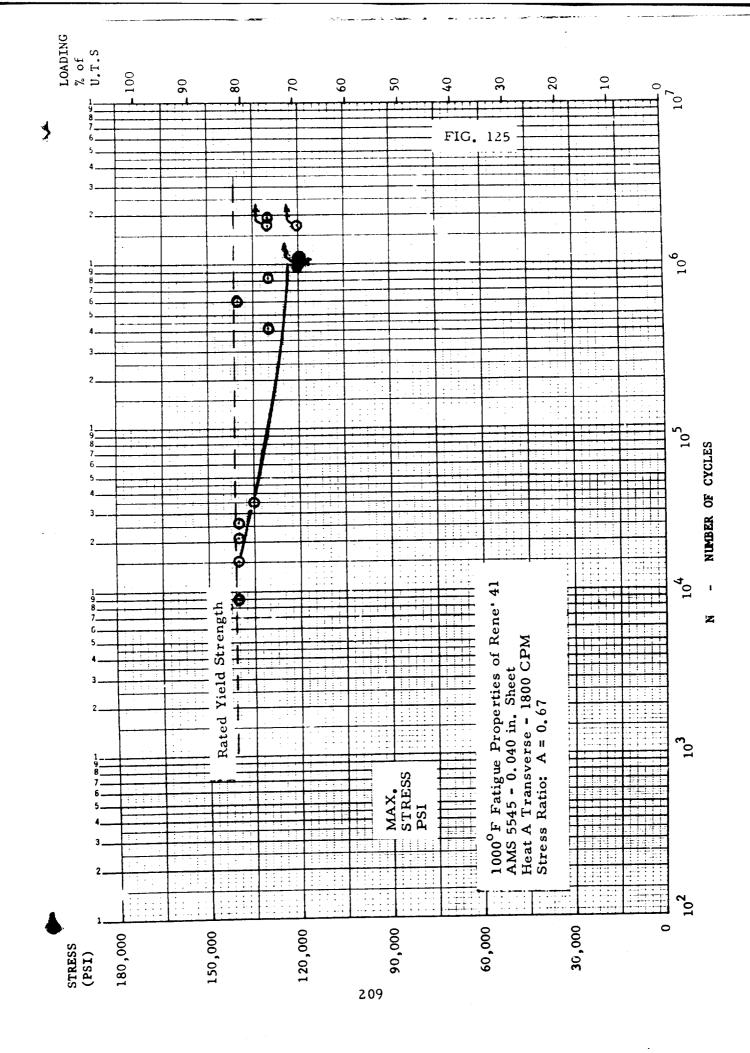


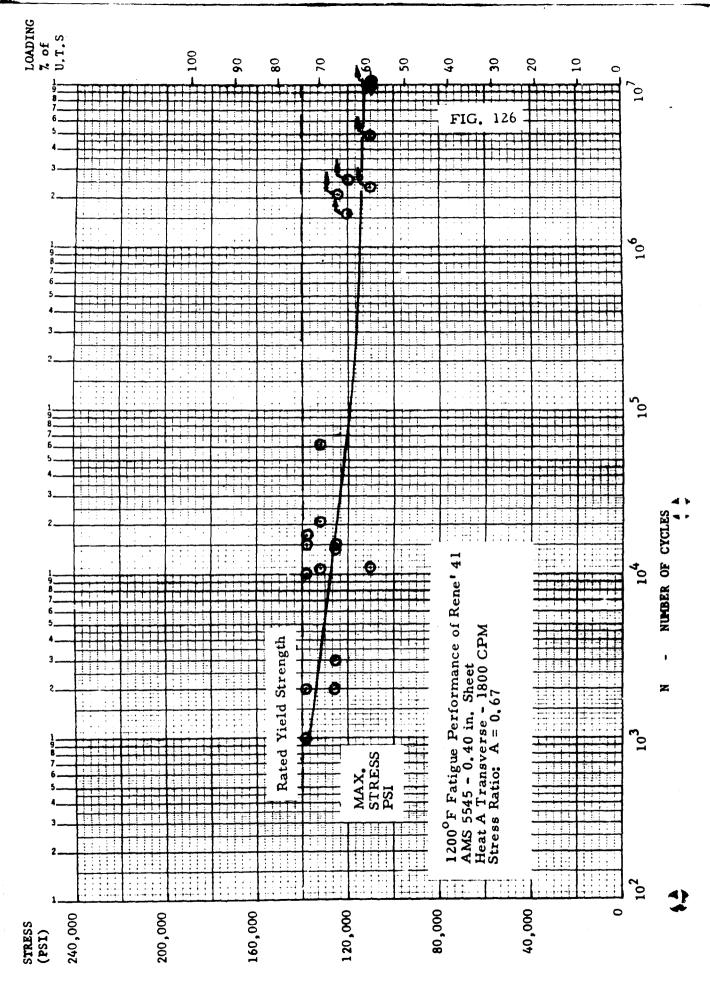
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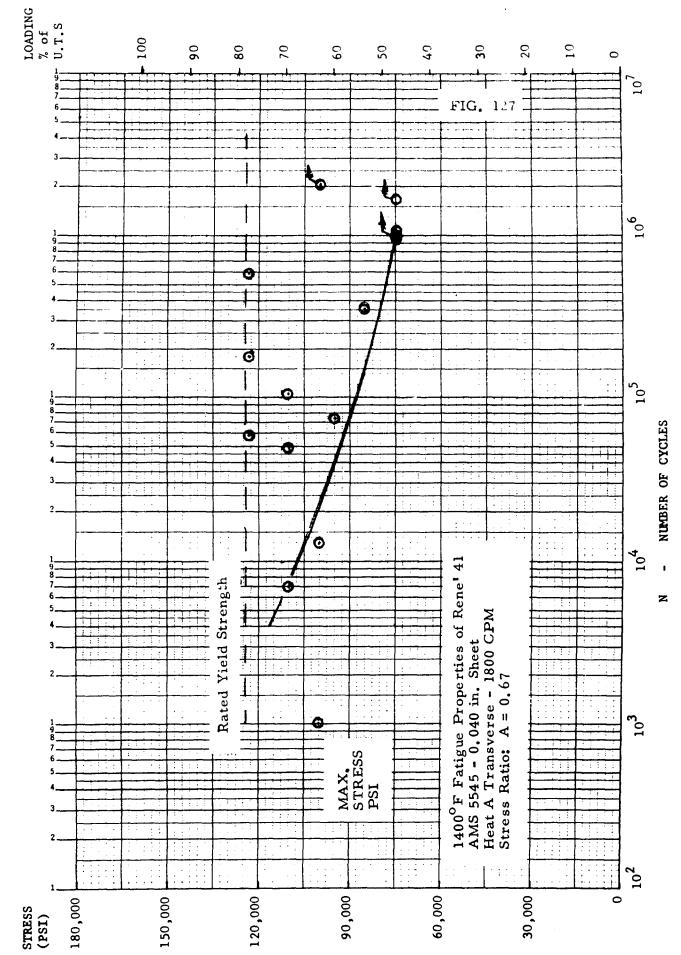
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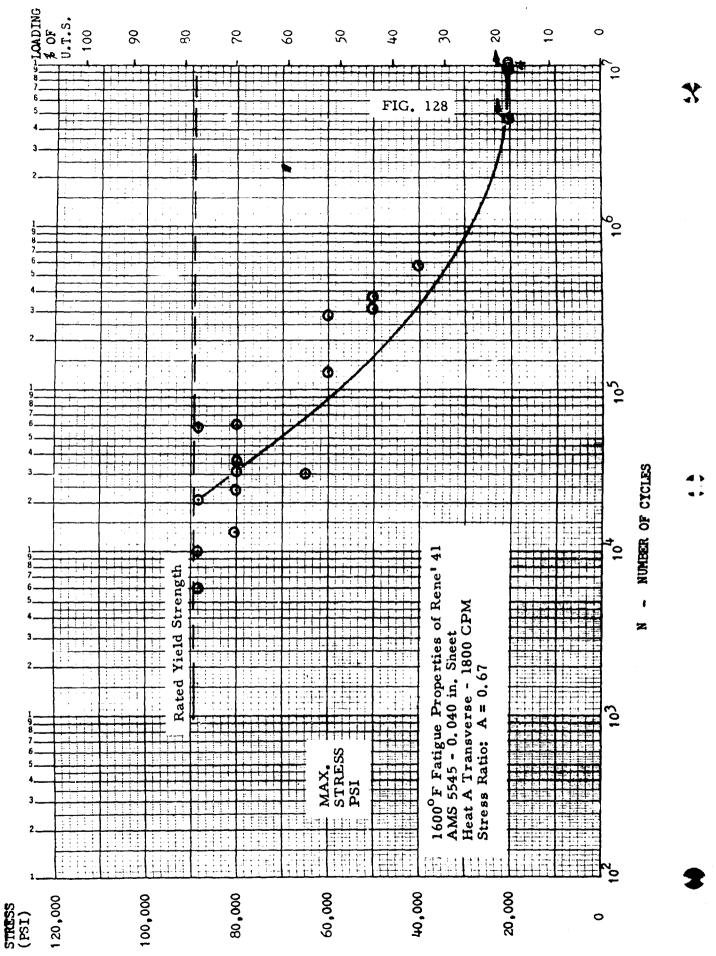


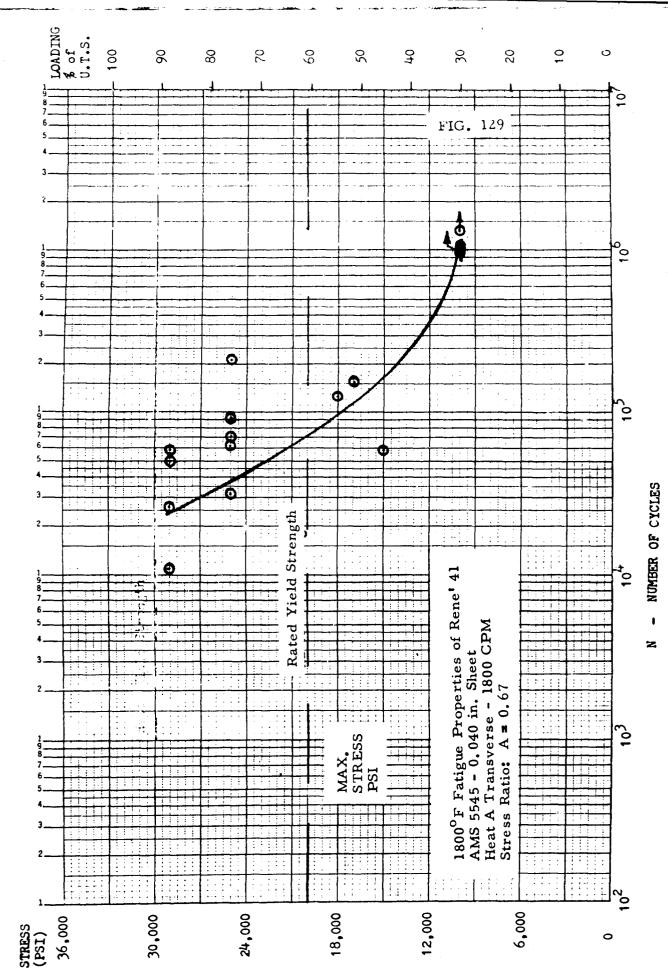


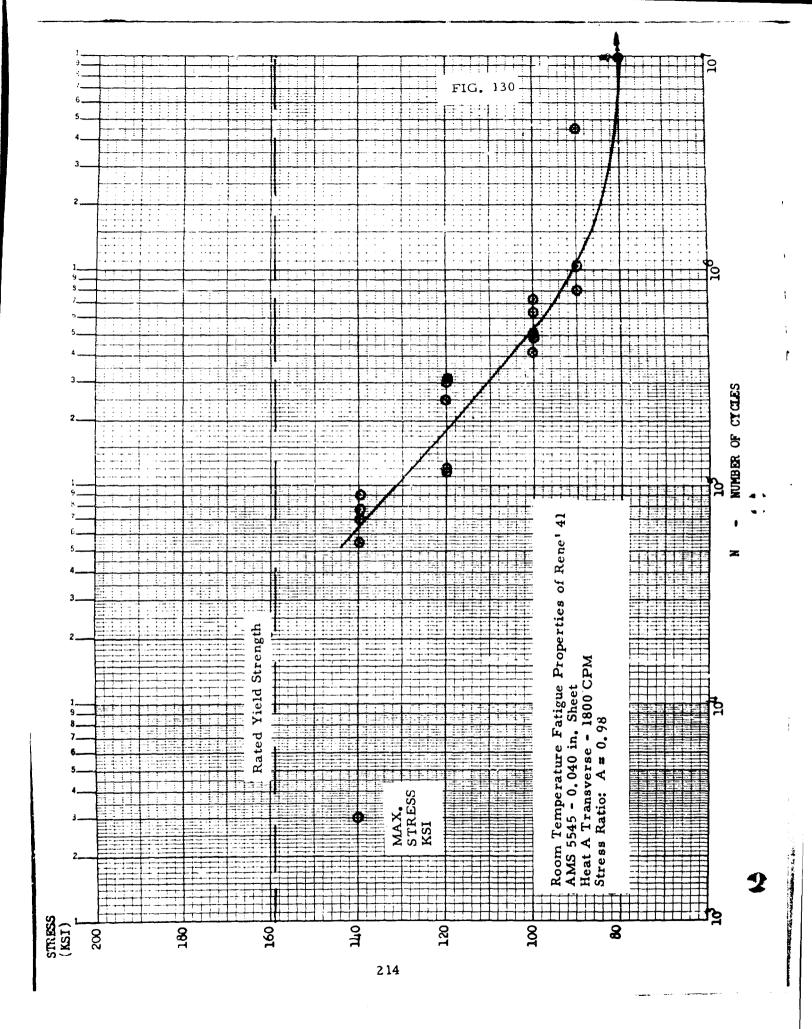


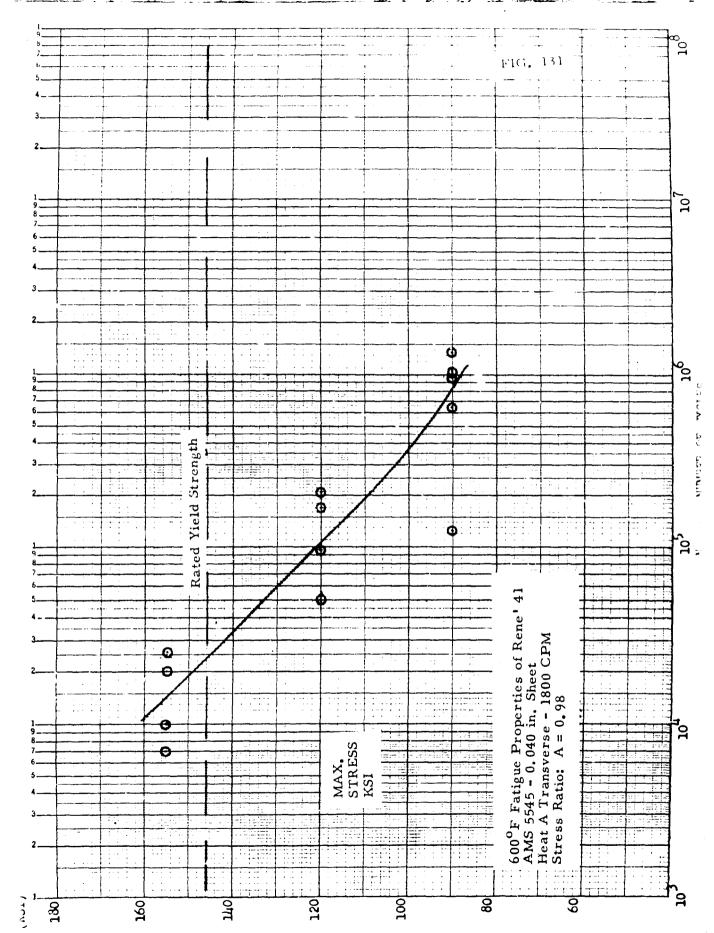


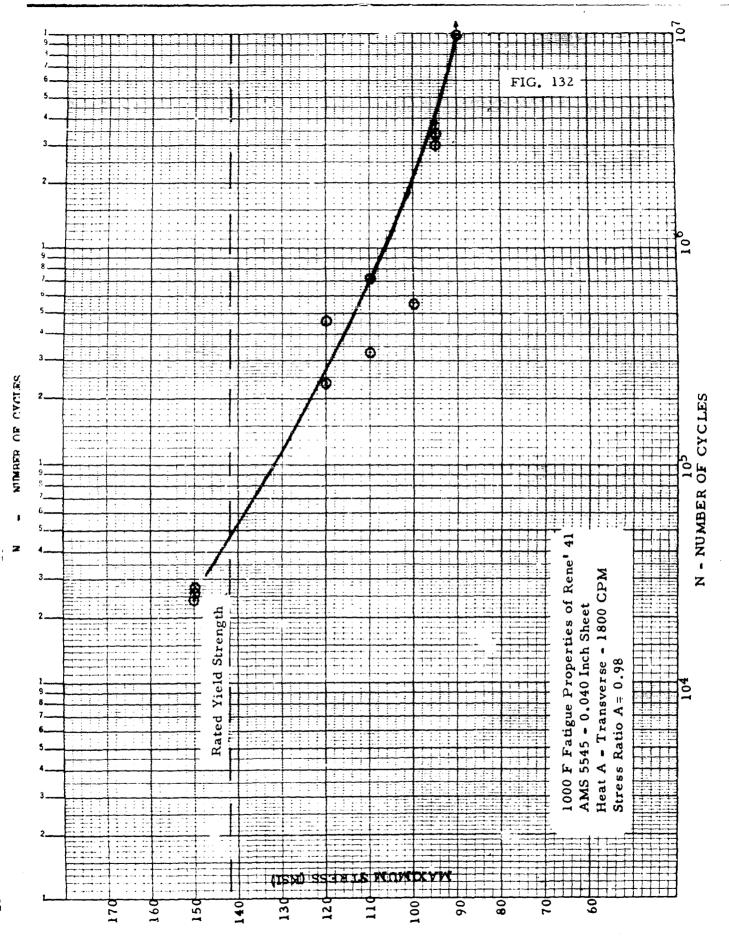


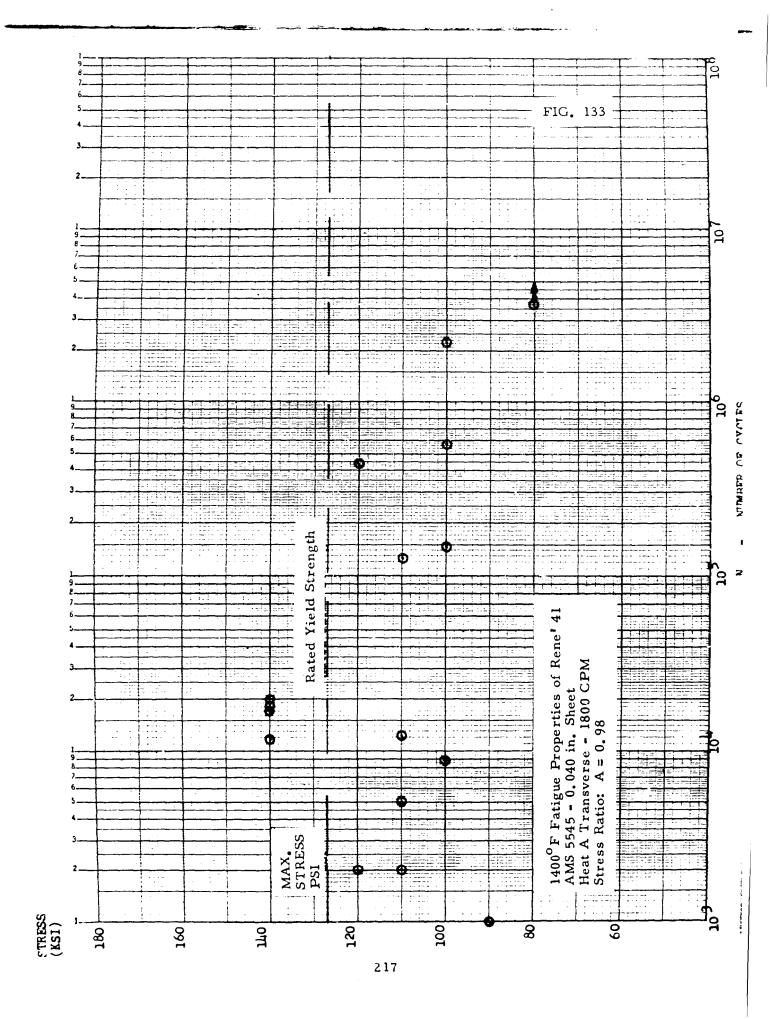


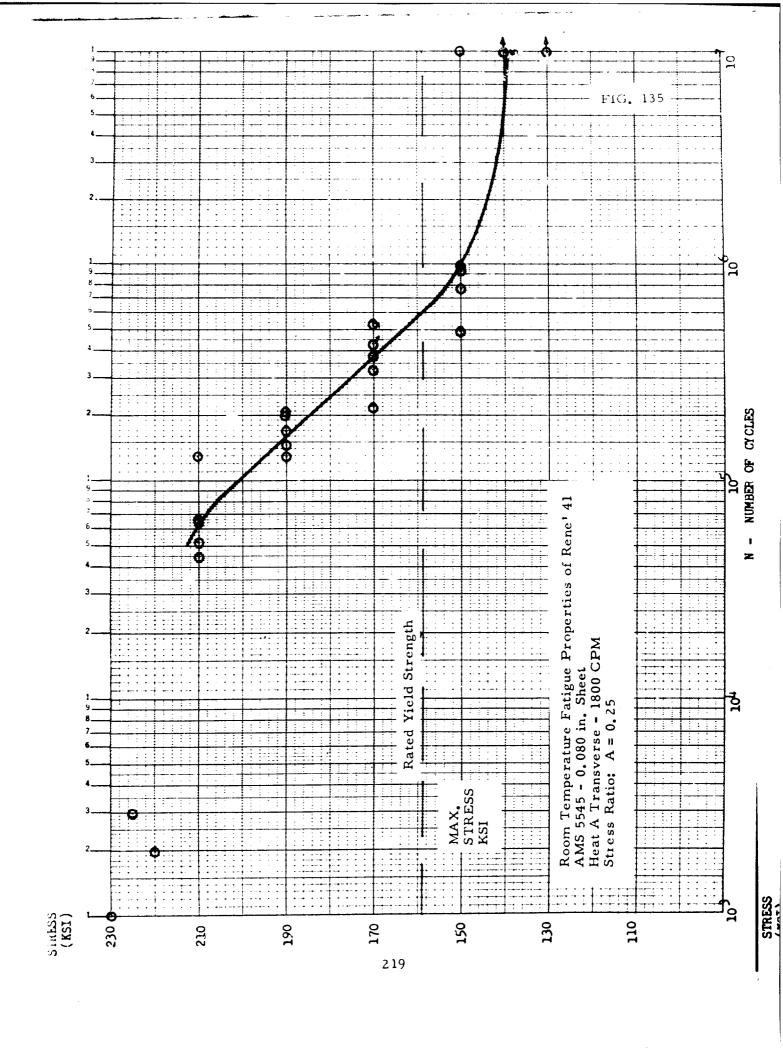


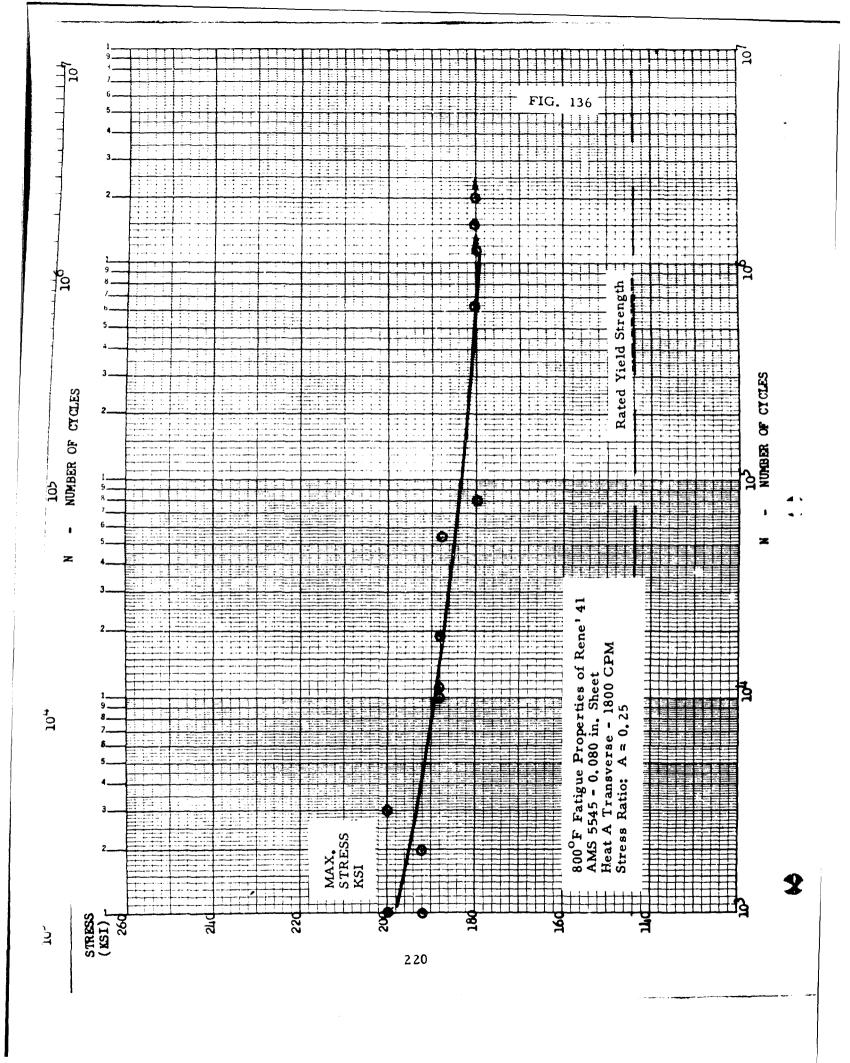


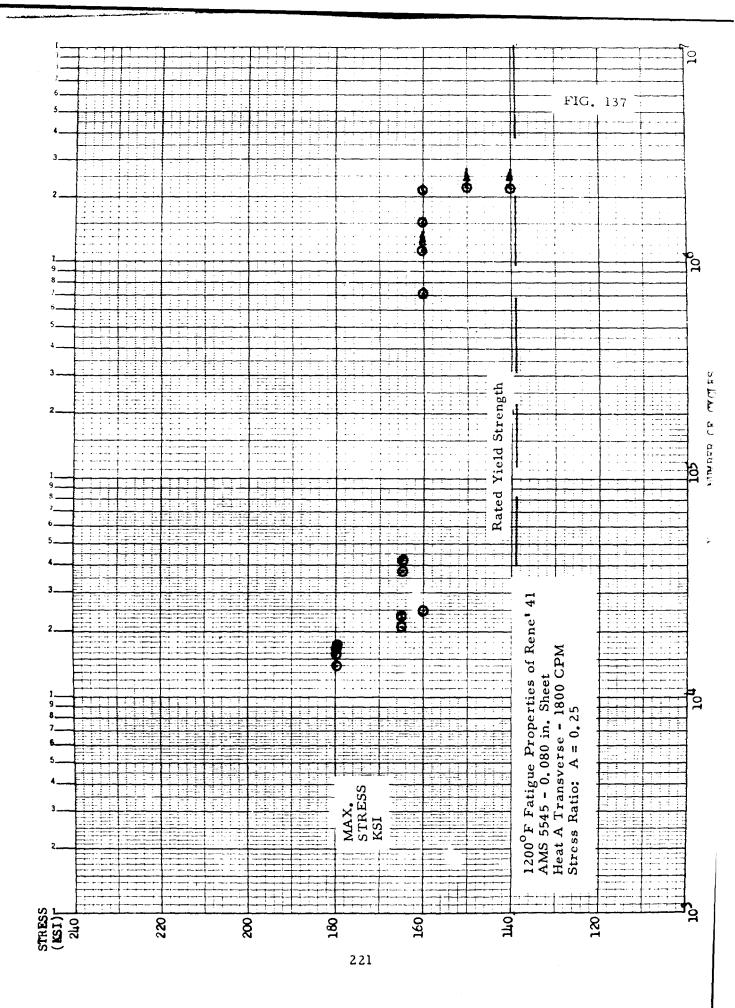


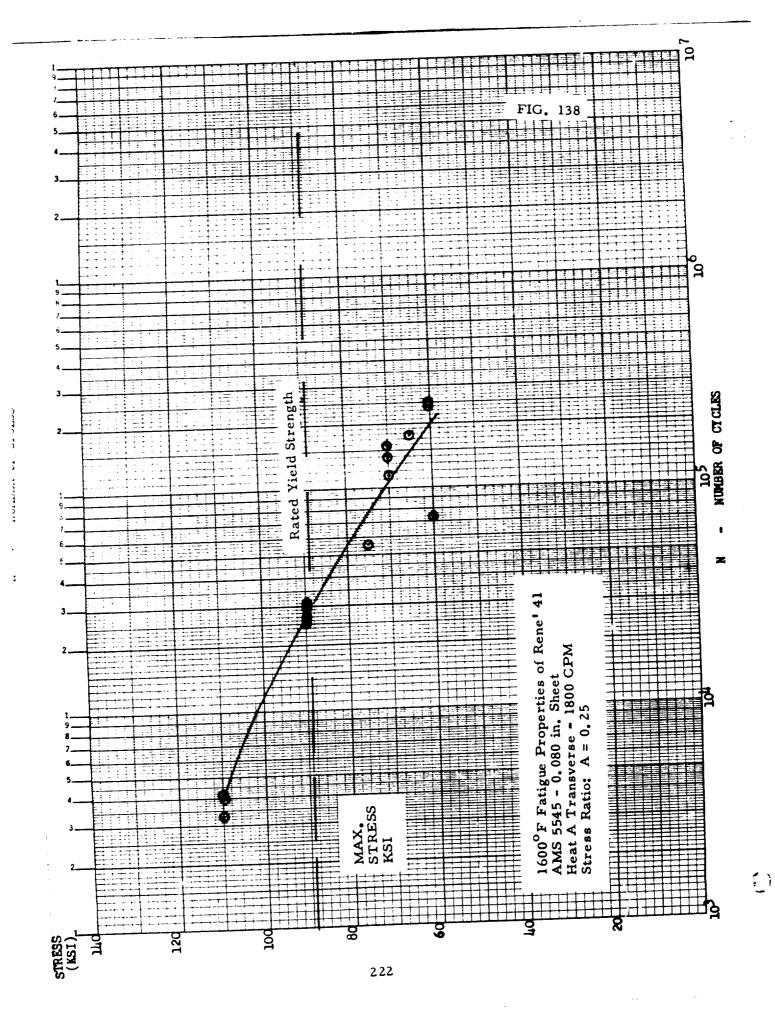


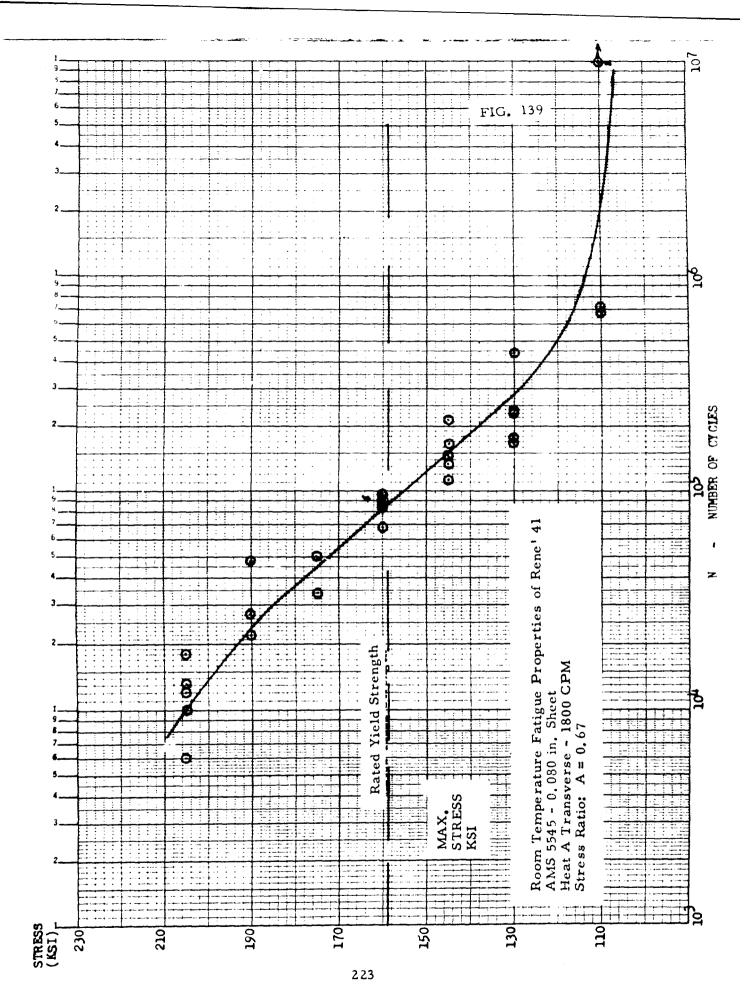


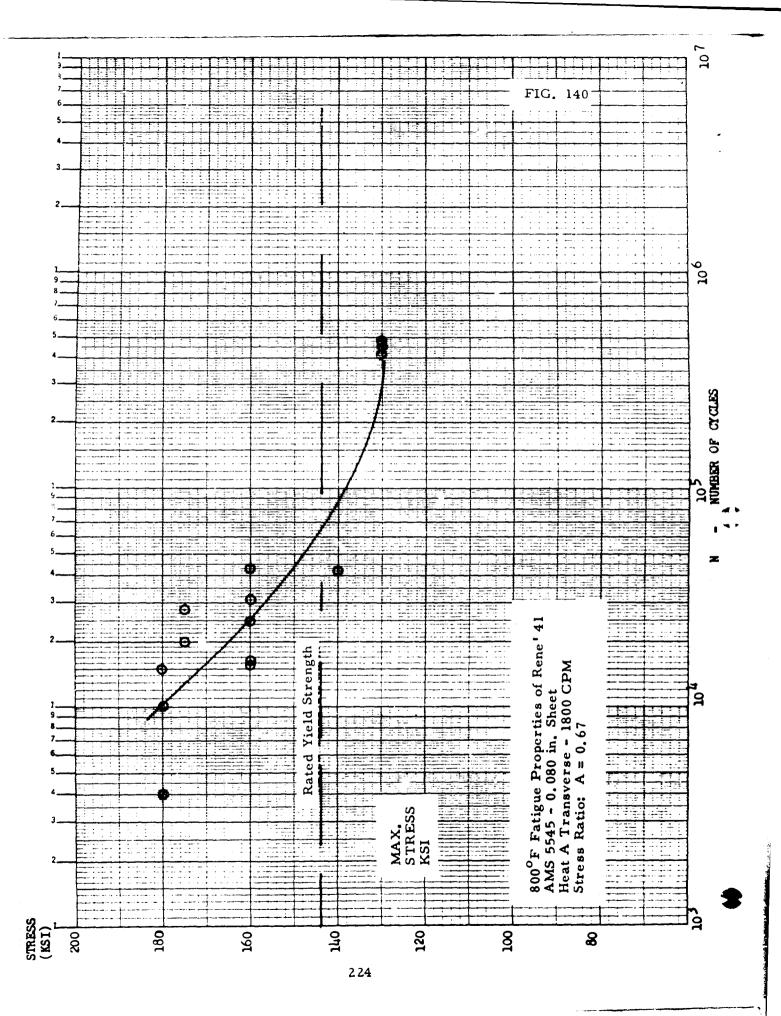


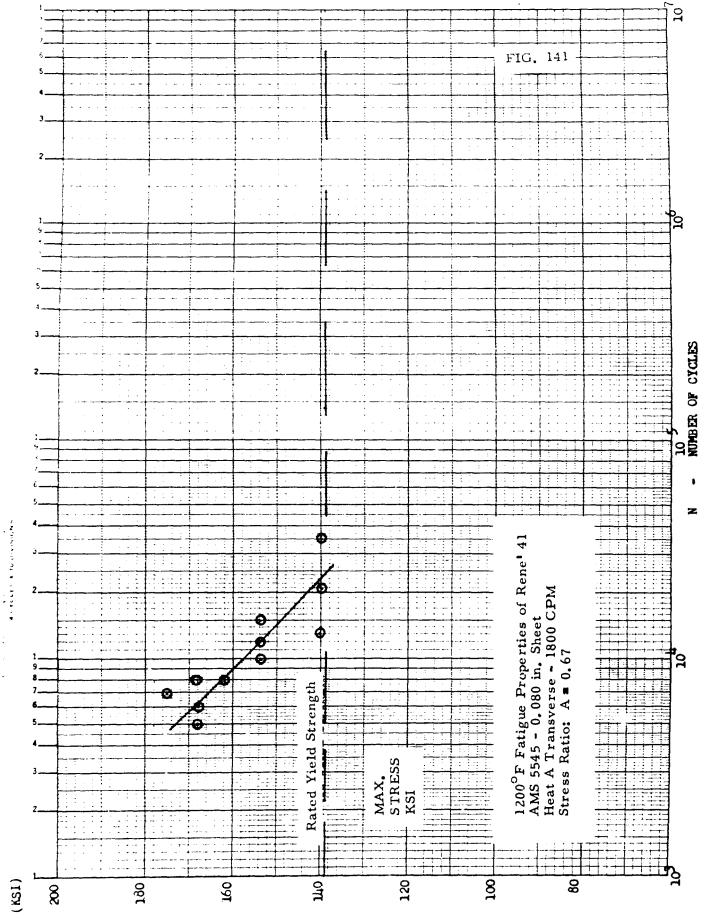


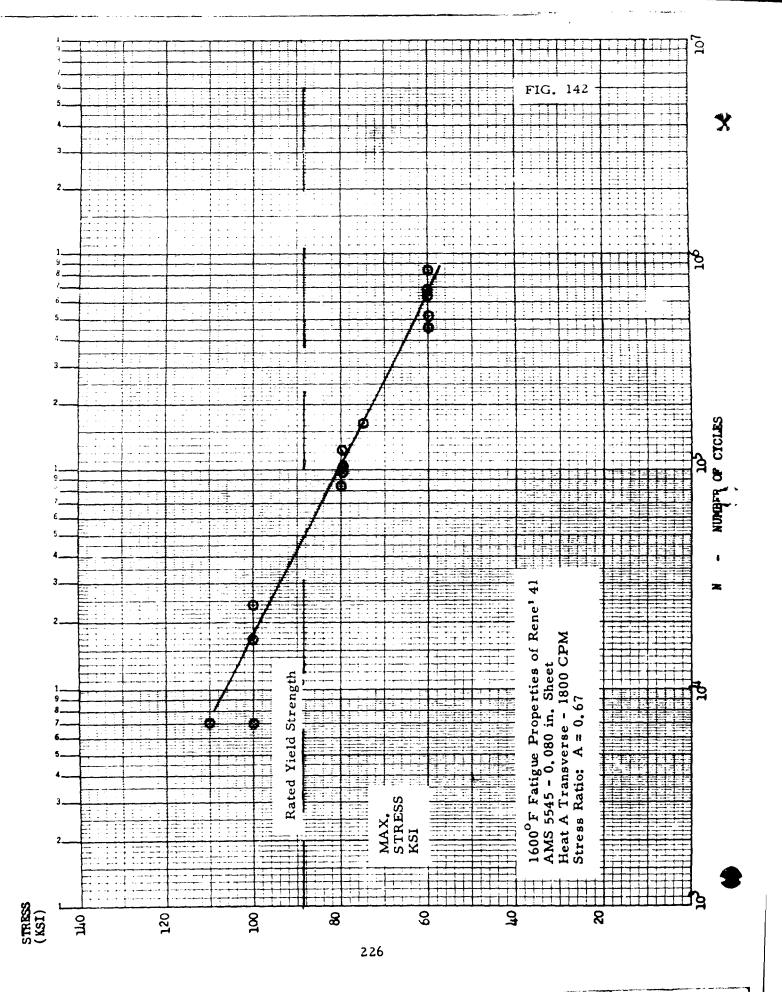




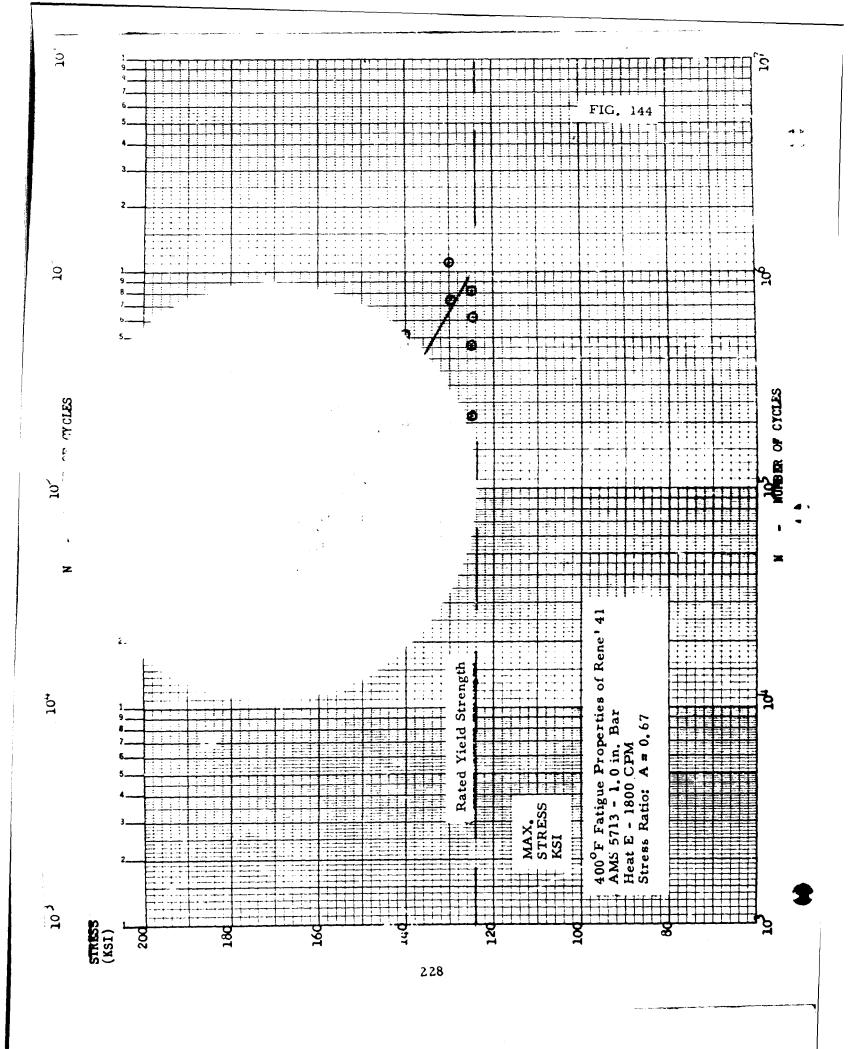


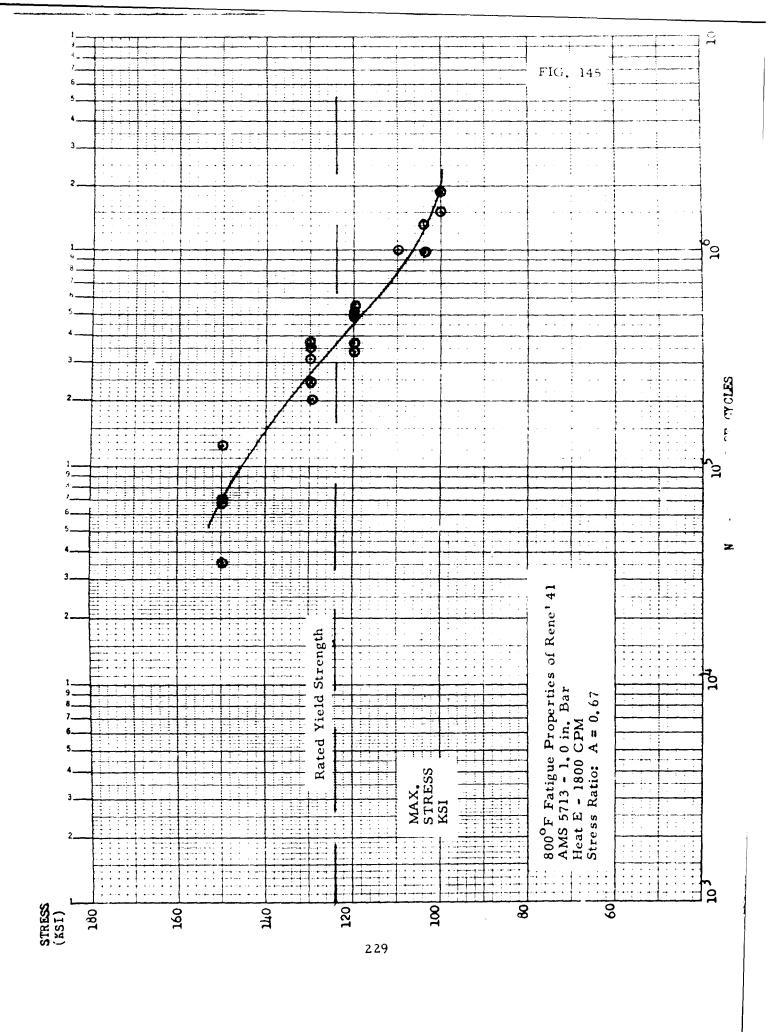


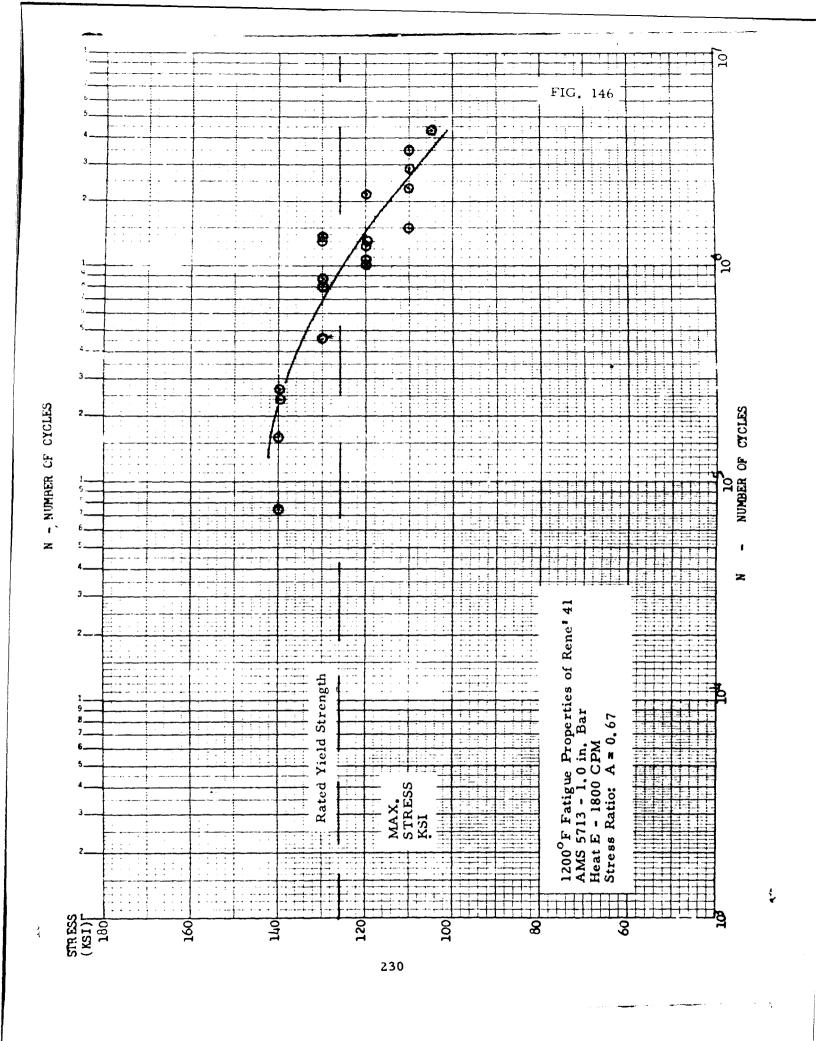


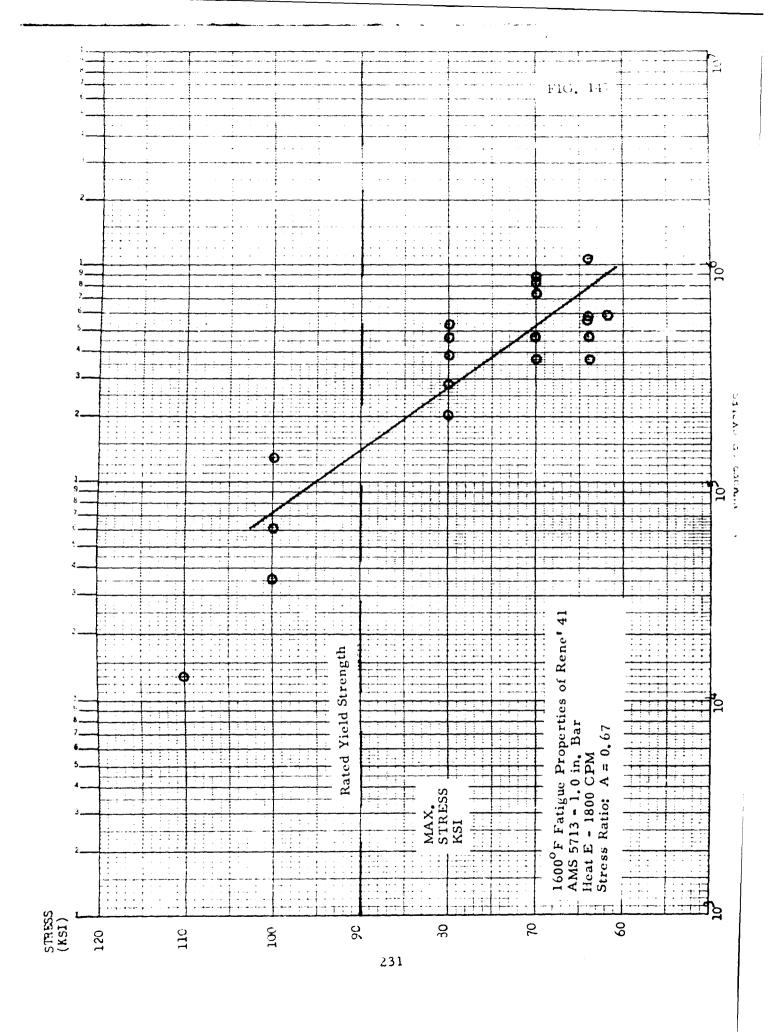


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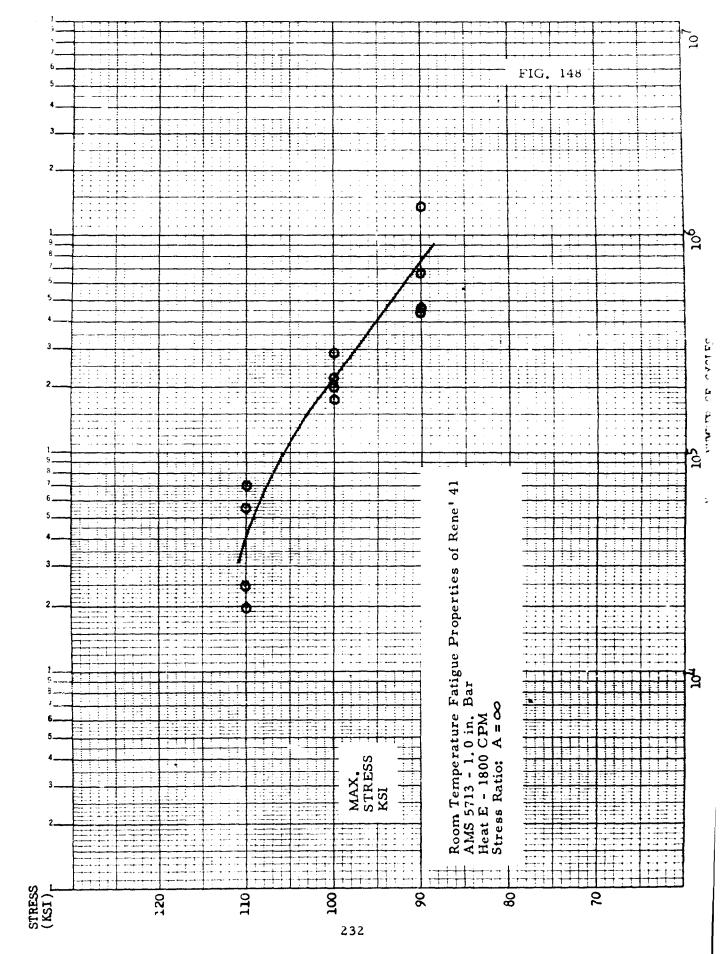




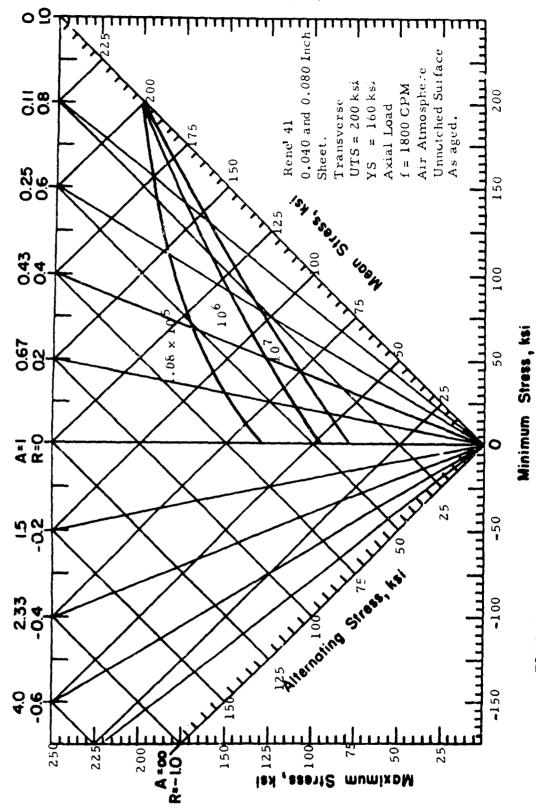




TYPICAL CONSTANT LIFE DIAGRAM FOR FATIGUE BEHAVIOR OF RENE' 41 SHEET MATERIAL AT ROOM TEMPERATURE FIGURE

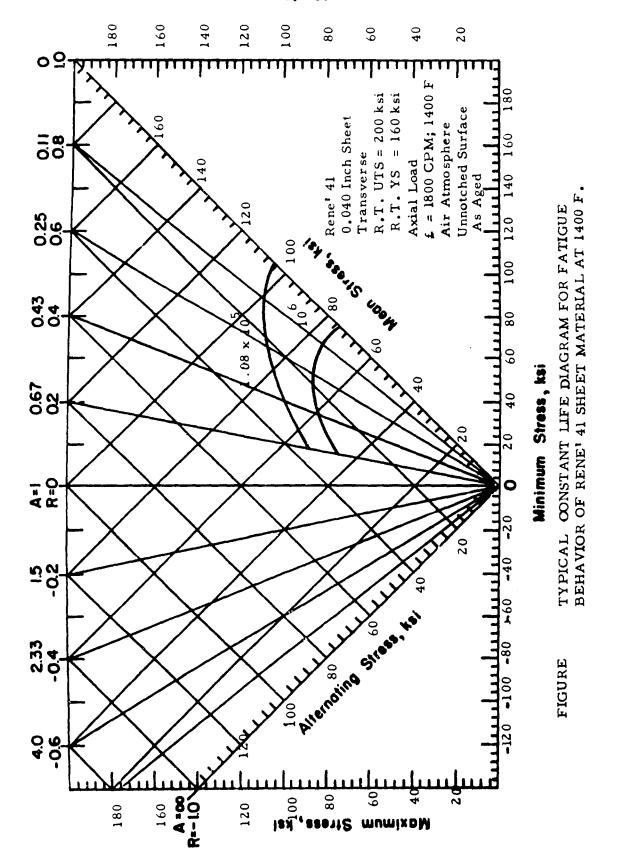






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FIGURE



SECTION VII - TEST RESULTS, TABLES AND GRAPHS

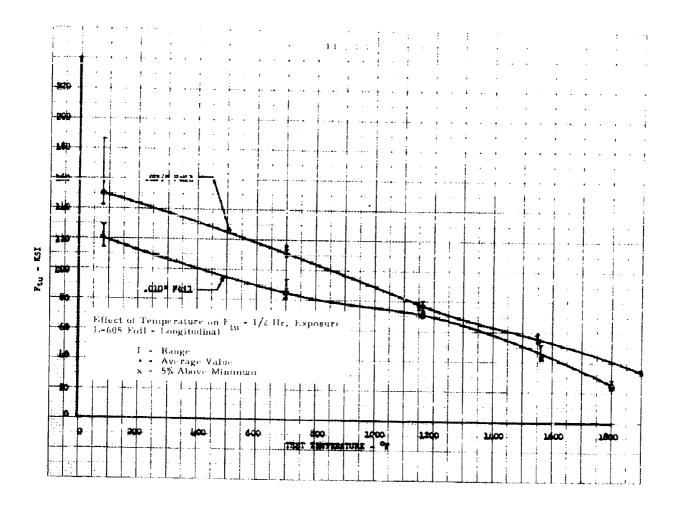
## SECTION 7.2 MATERIAL, L-605

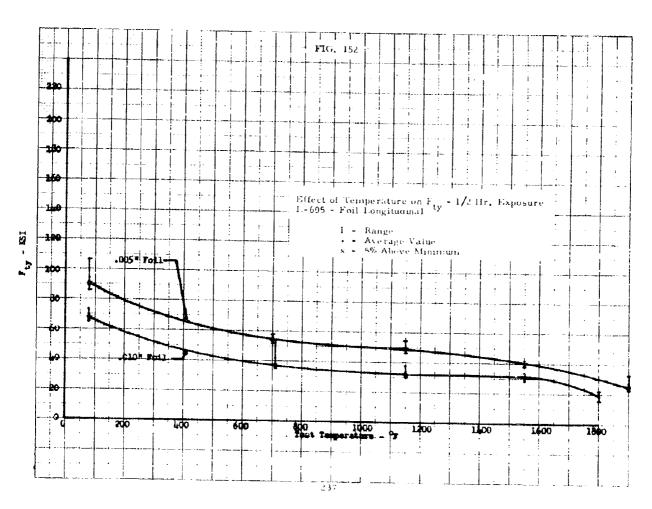
## TABLE OF CONTENTS

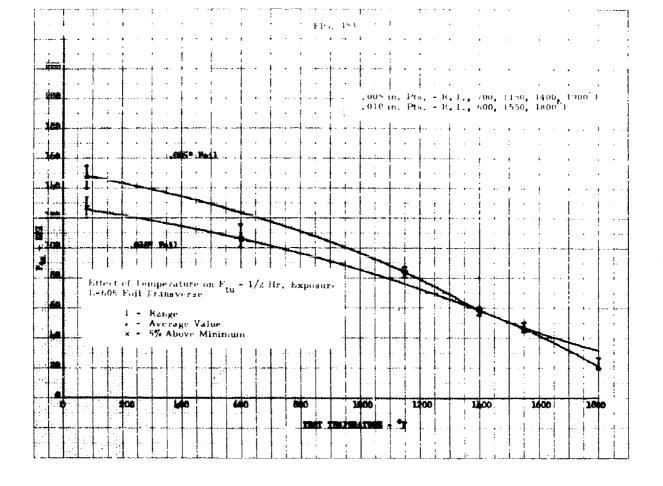
		FIGURE NUMBERS	PAGE NO.
7.2.1	TENSION	151 - 169	236 - 252
7.2.2	COMPRESSION	170 - 175	253 - 258
7.2.3	BEARING	176 - 185	259 <b>-</b> 266
7.2.4	SHEAR	186 - 187	267 - 269
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7.2,6	STRESS RUPTURE	199 - 208	282 - 292
7.2.7	FATIGUE	209 - 237	293 - 316
7.2.8	TYPICAL CONSTANT	238	317

## SECTION VII

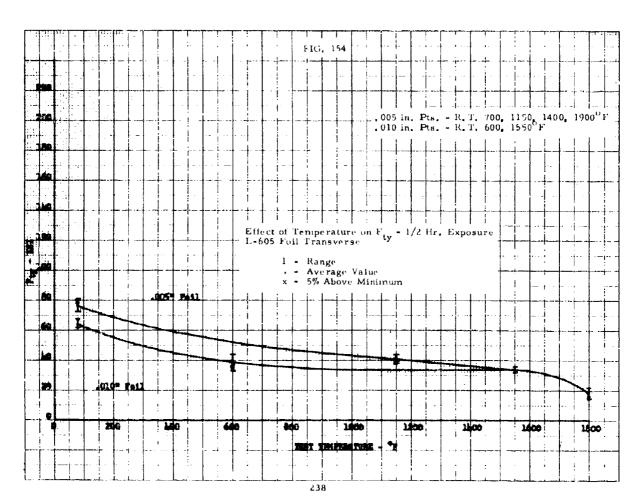
SECTION 7.2.1 TENSION

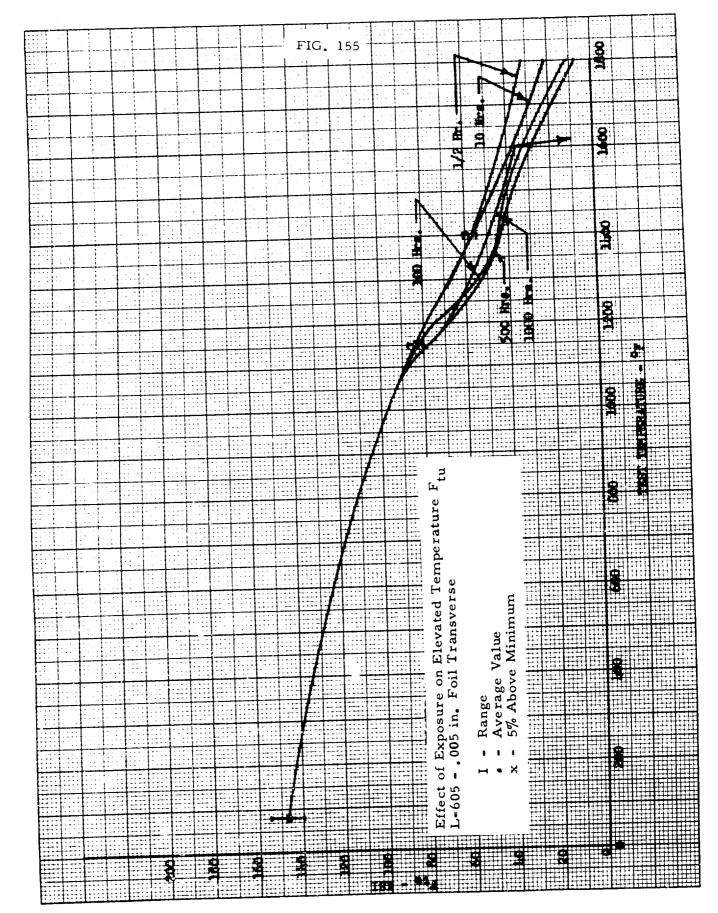




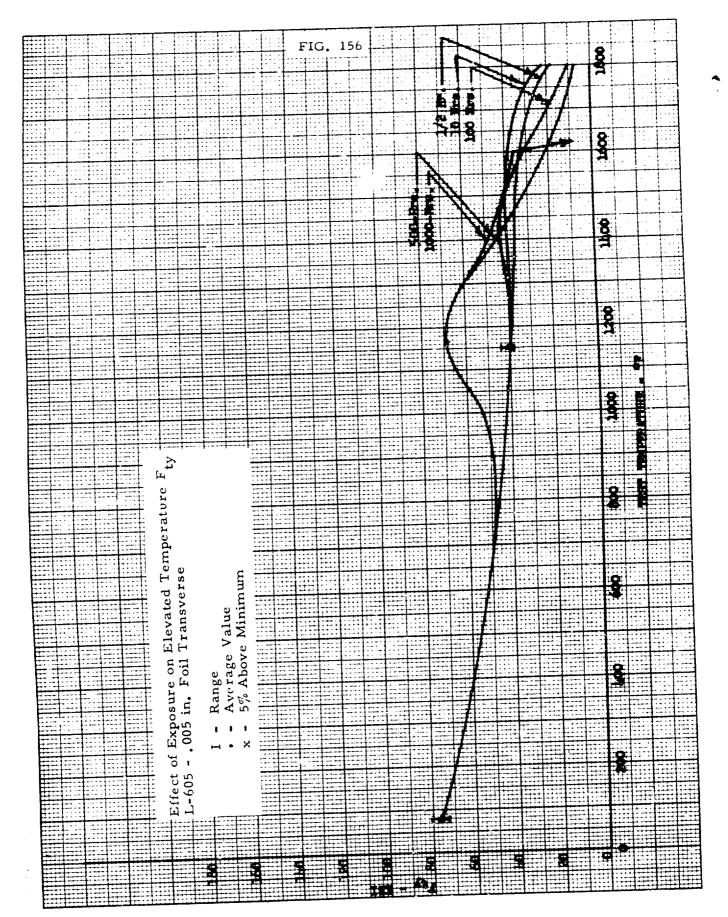


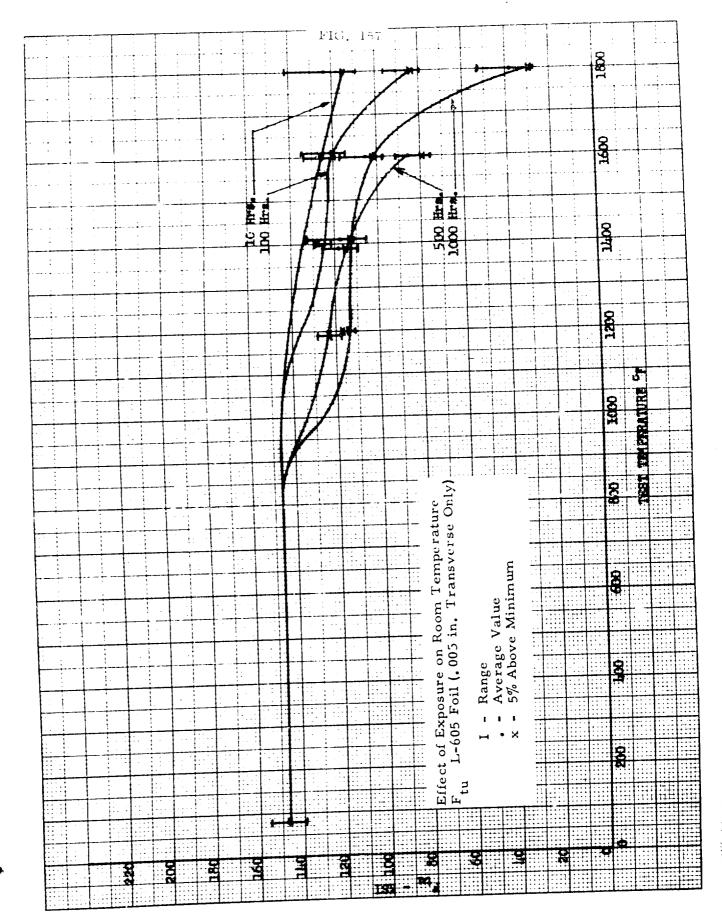
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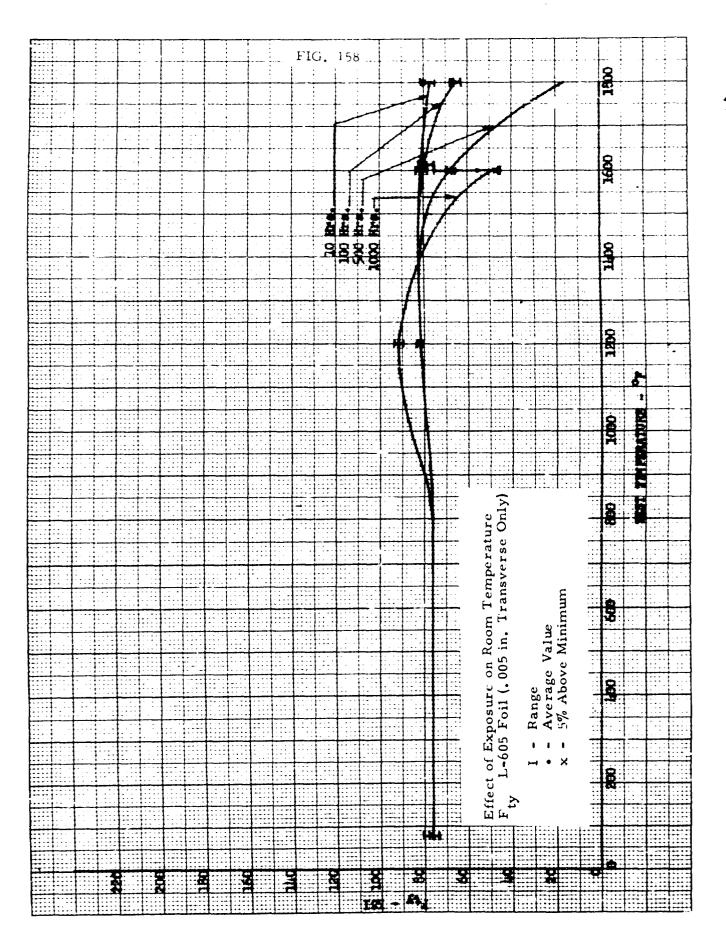


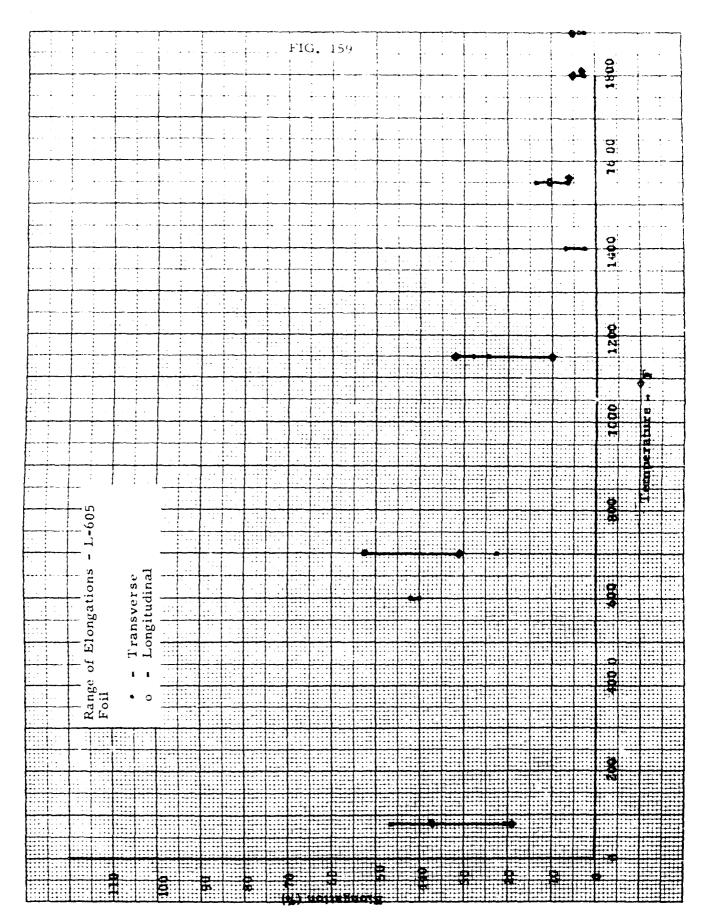


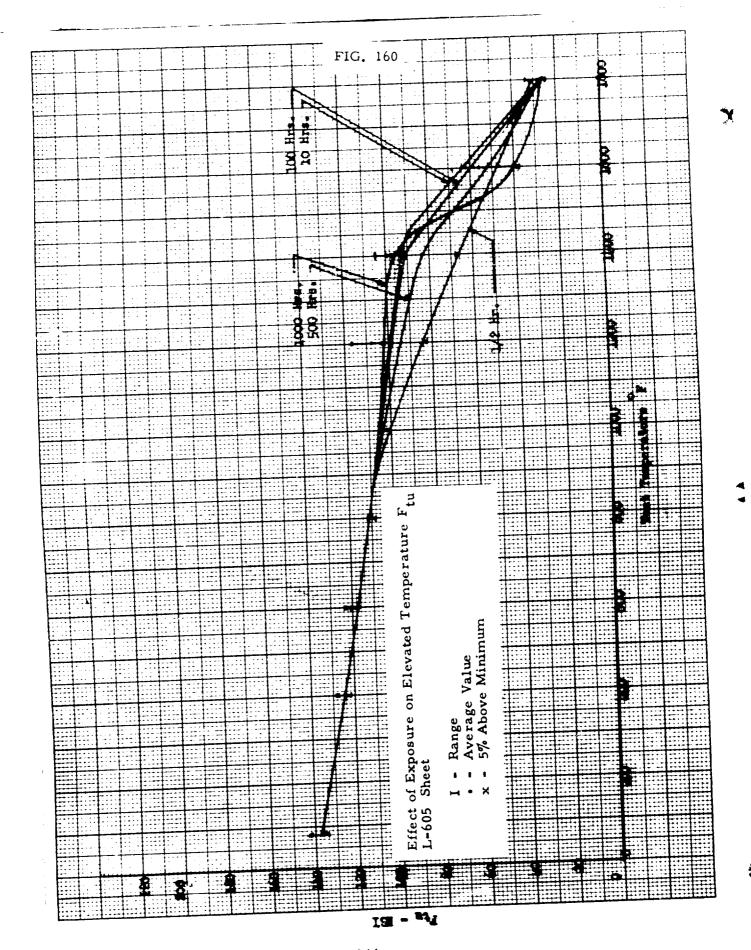
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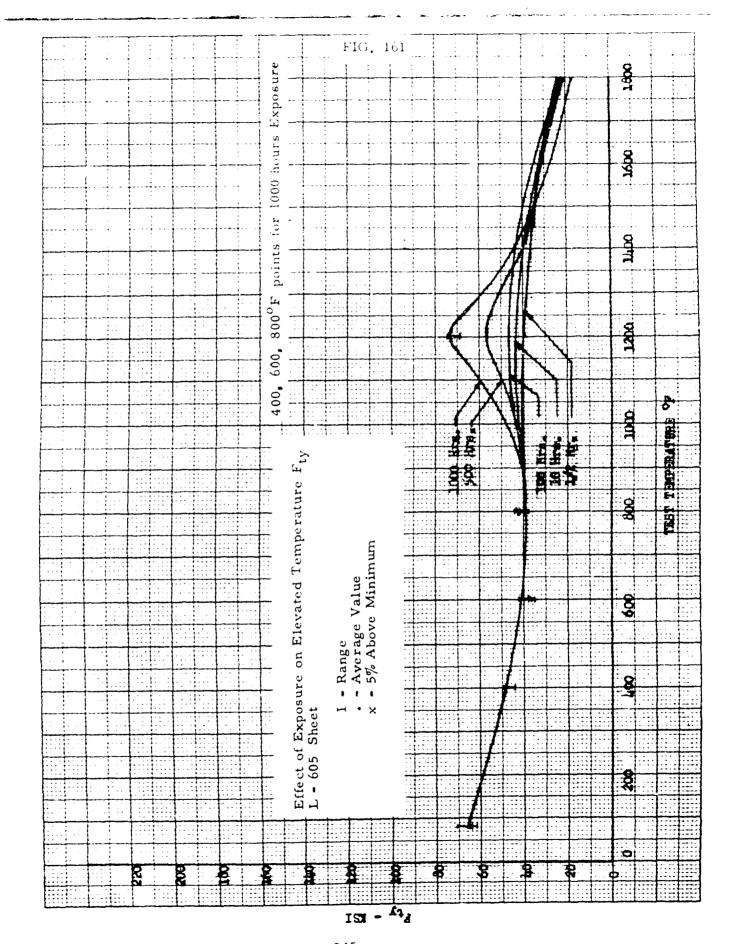


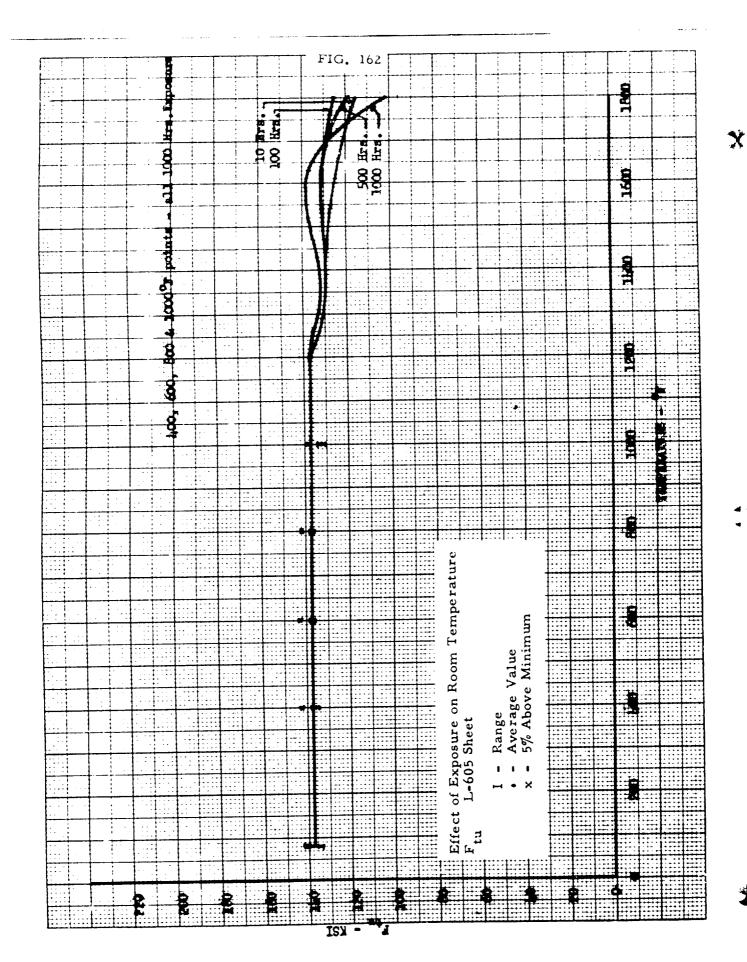


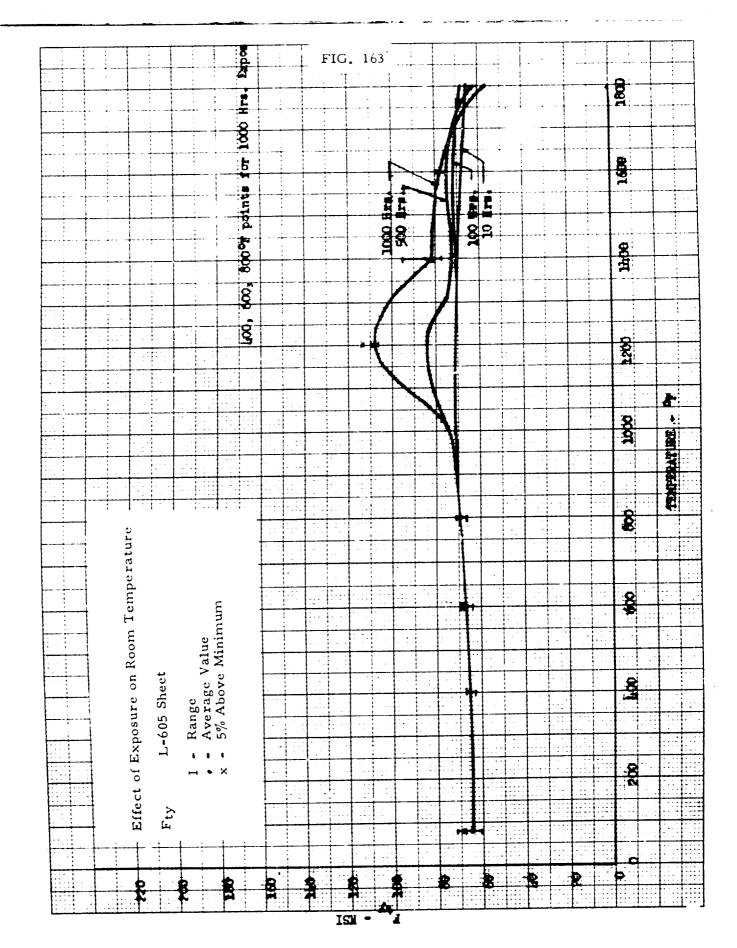


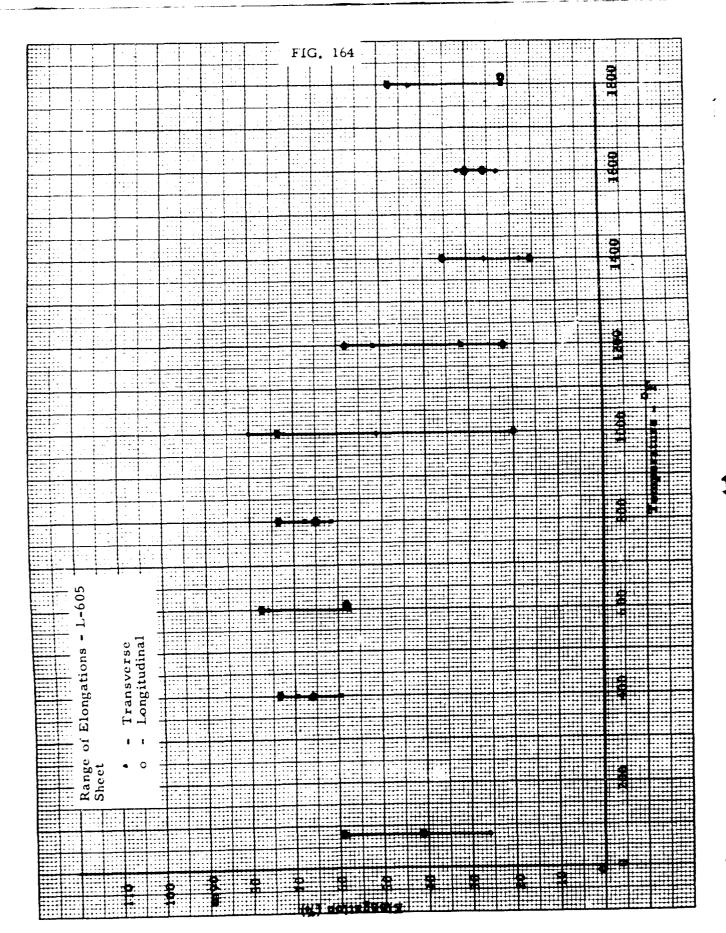


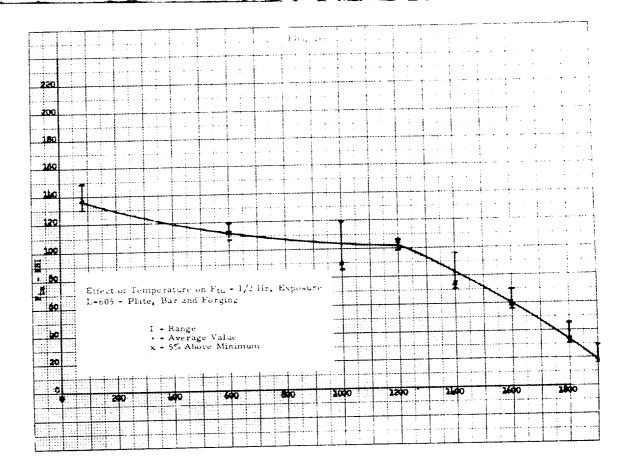


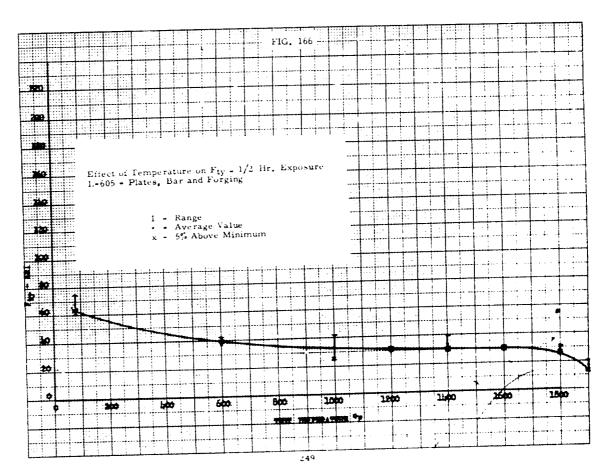


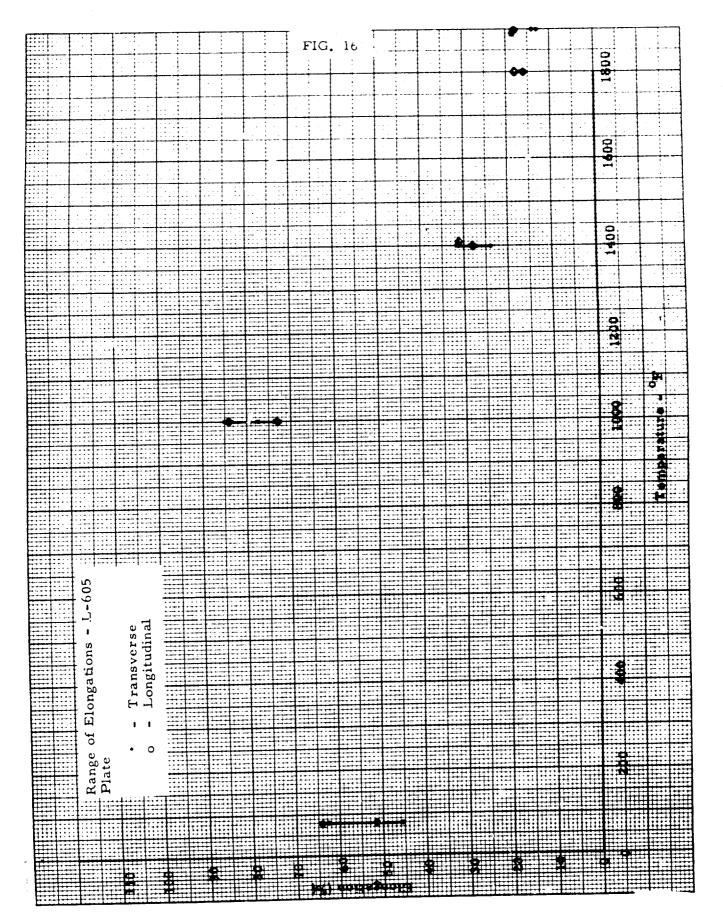


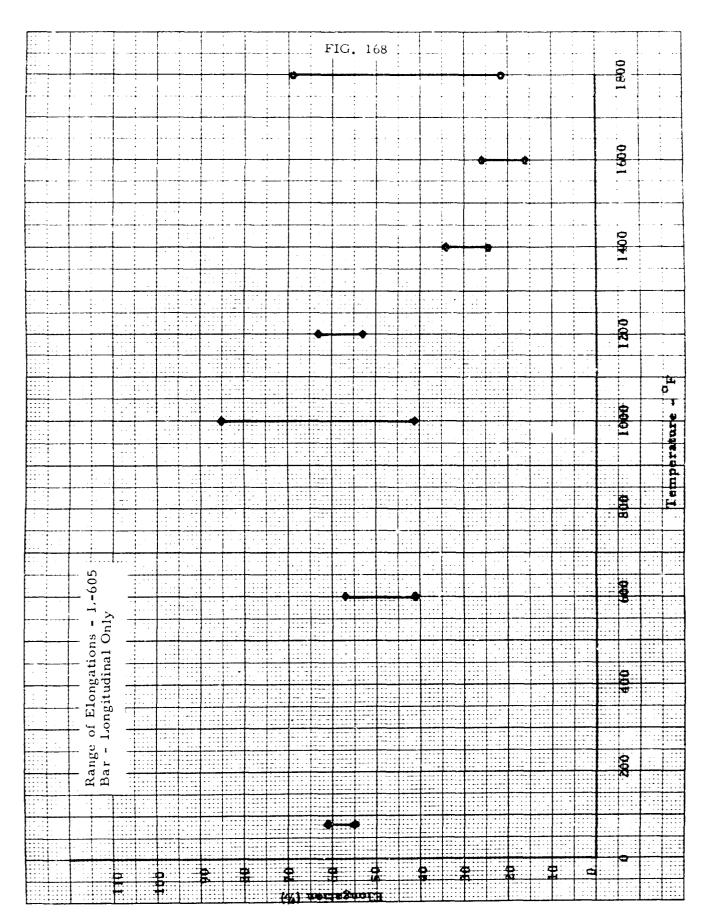












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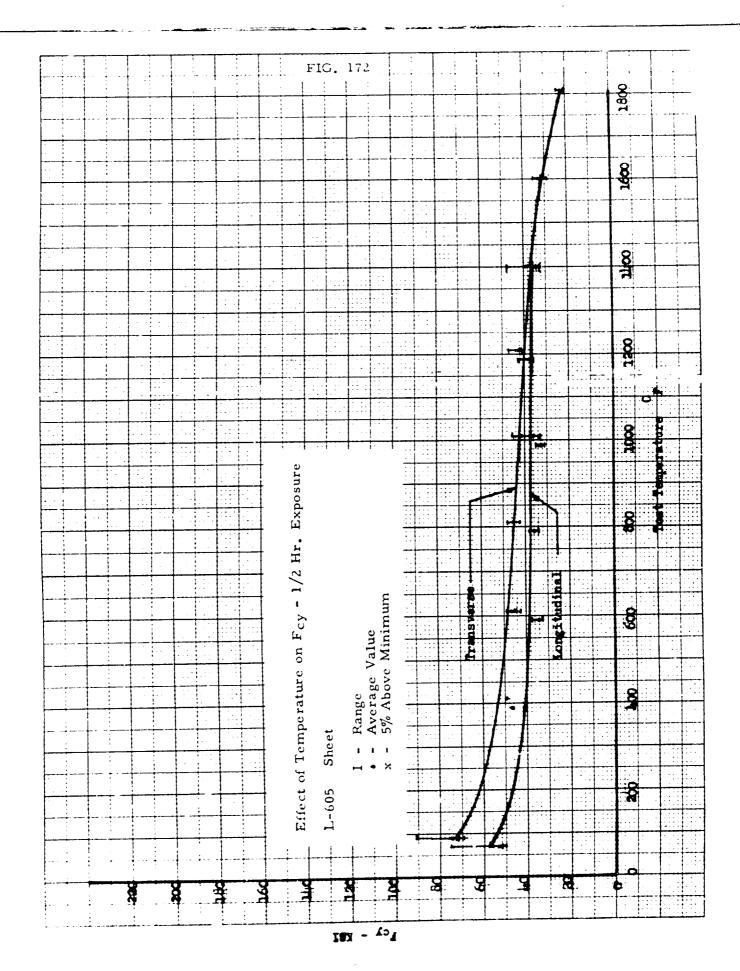
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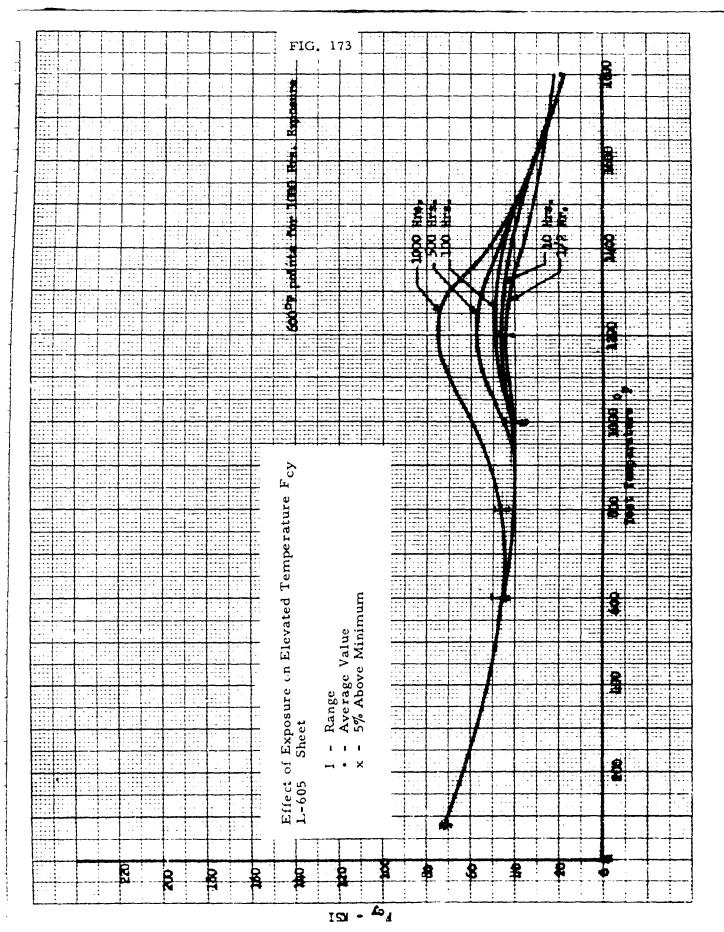
## SECTION 7.2.2 COMPRESSION

Figure 170 Effect of Temperature on Fcy
1/2 Hour Exposure - L605 Sheet (Longitudinal)

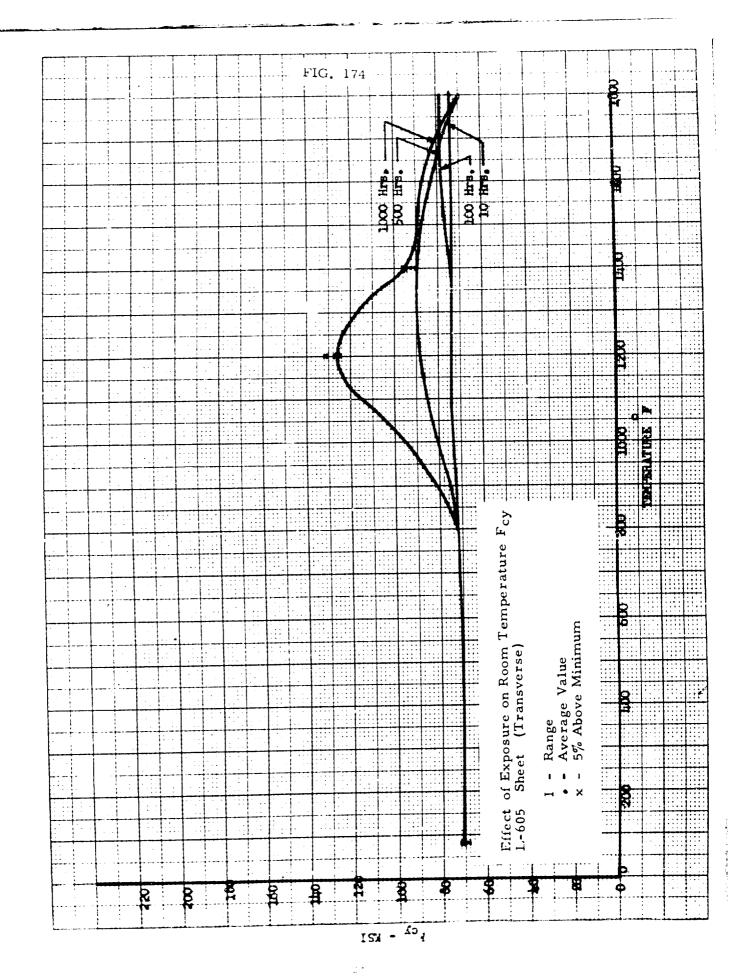
Figure 171 Effect of Temperature on Fcy
1/2 Hour Exposure - L605 Sheet (Transverse)

These two figures have been deleted, since they appear in Figure 172, Page 255.





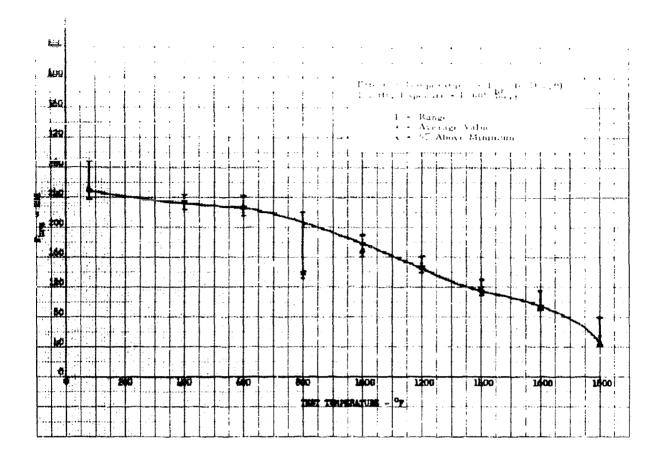
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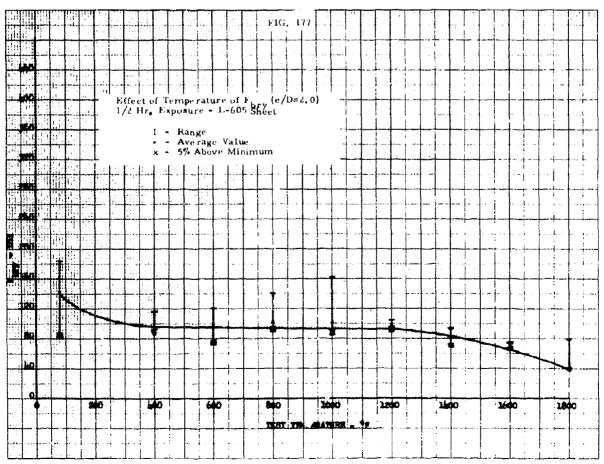


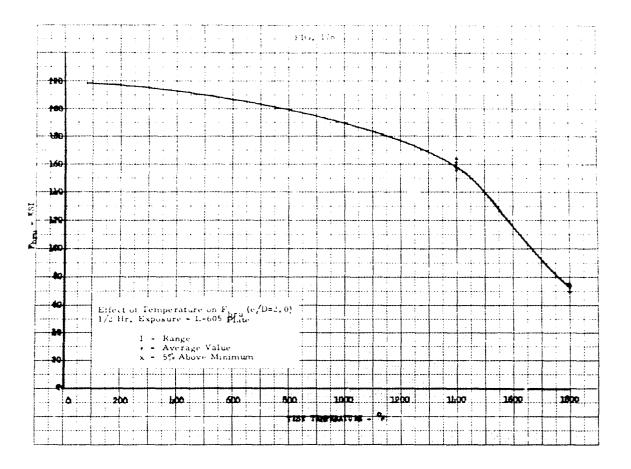
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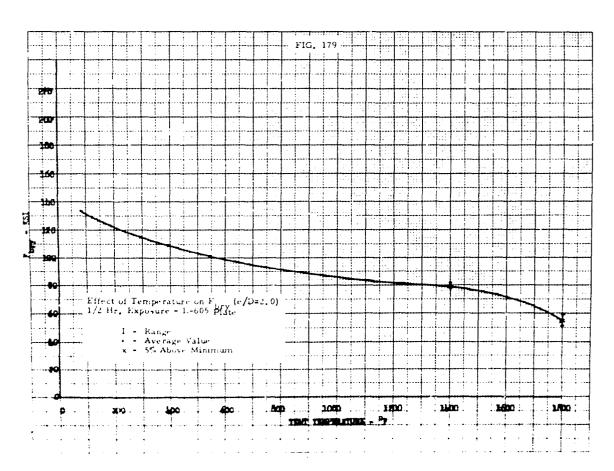
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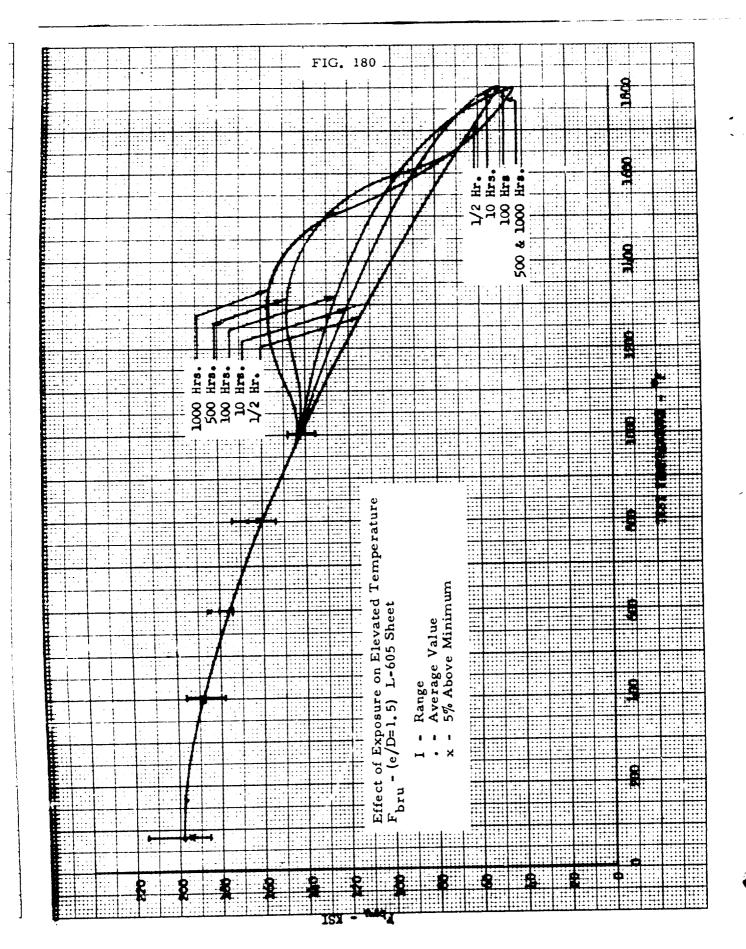
## SECTION 7.2.3 BRARING

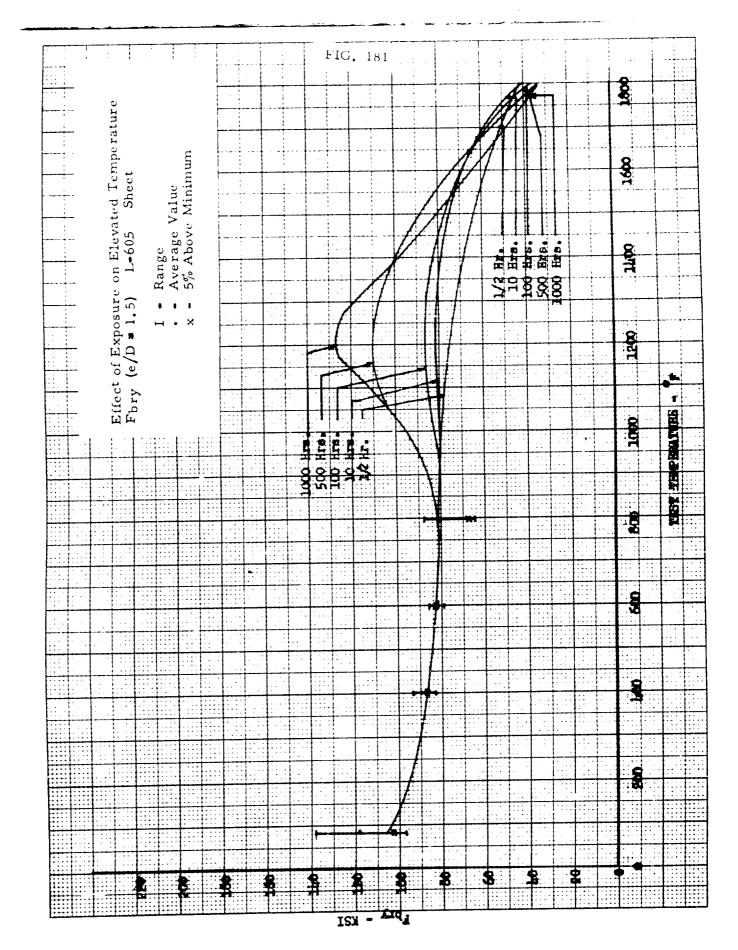


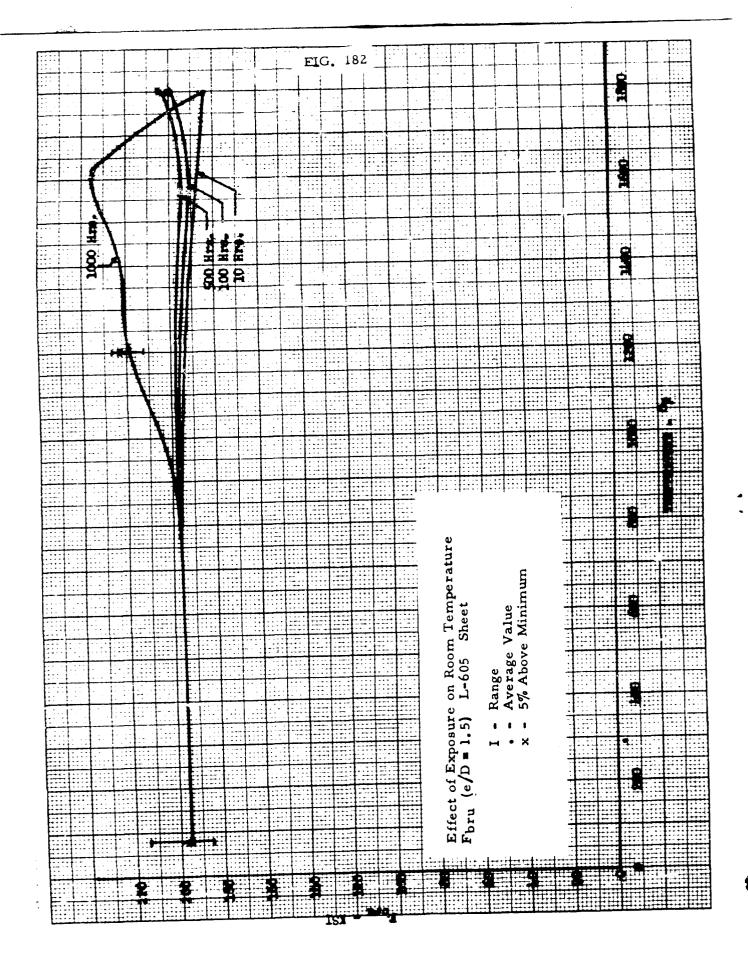


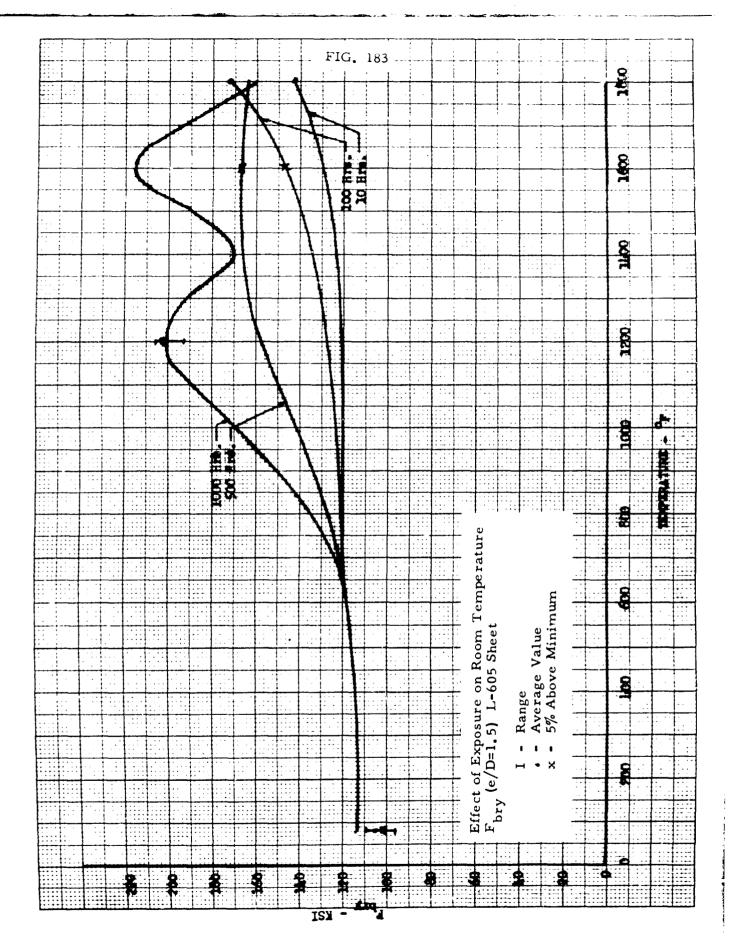


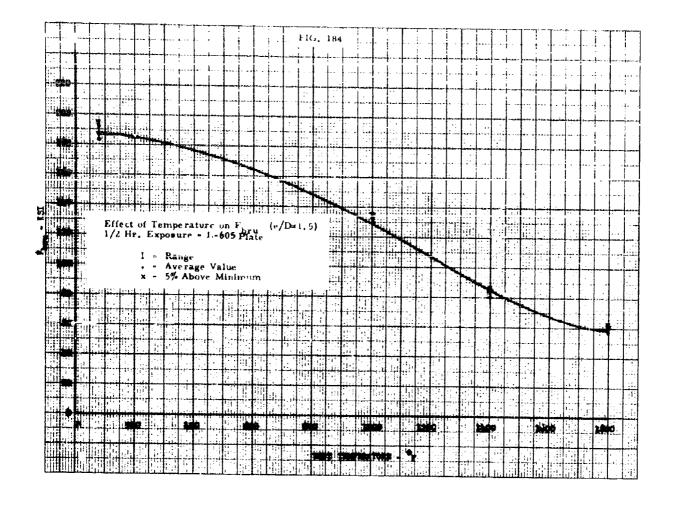


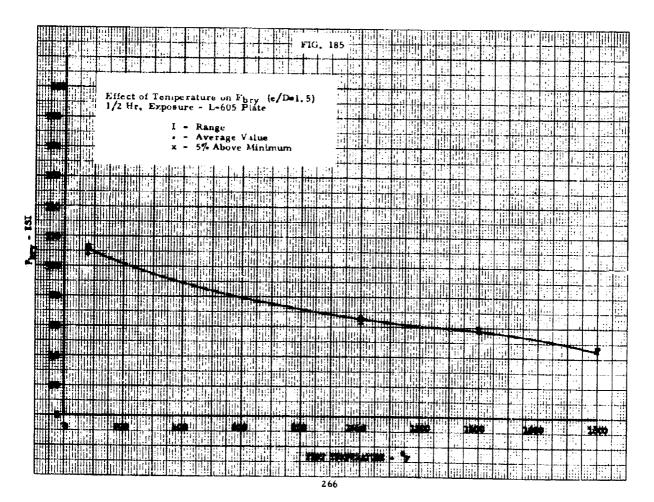




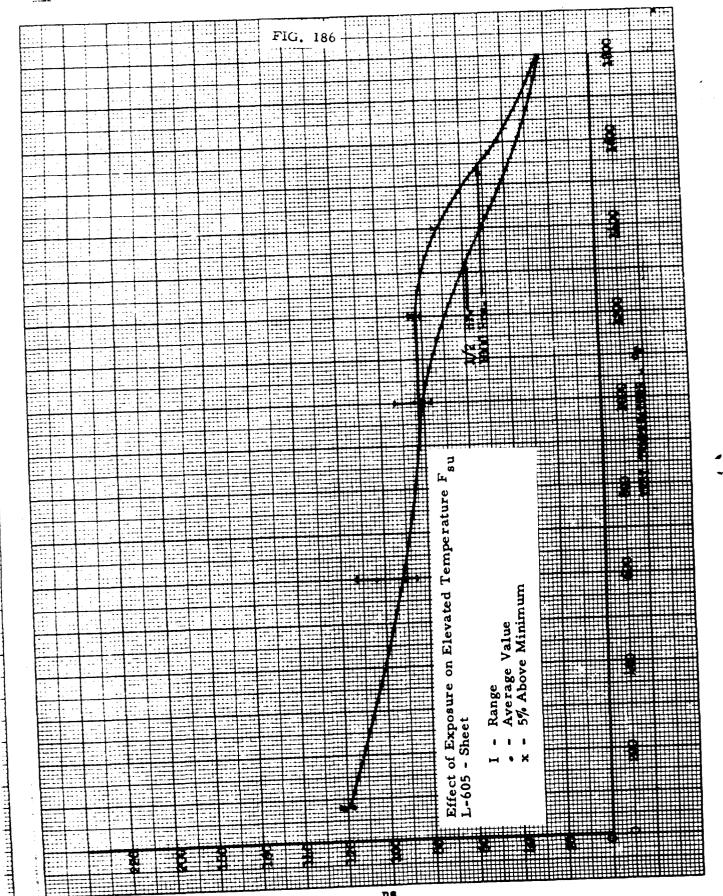




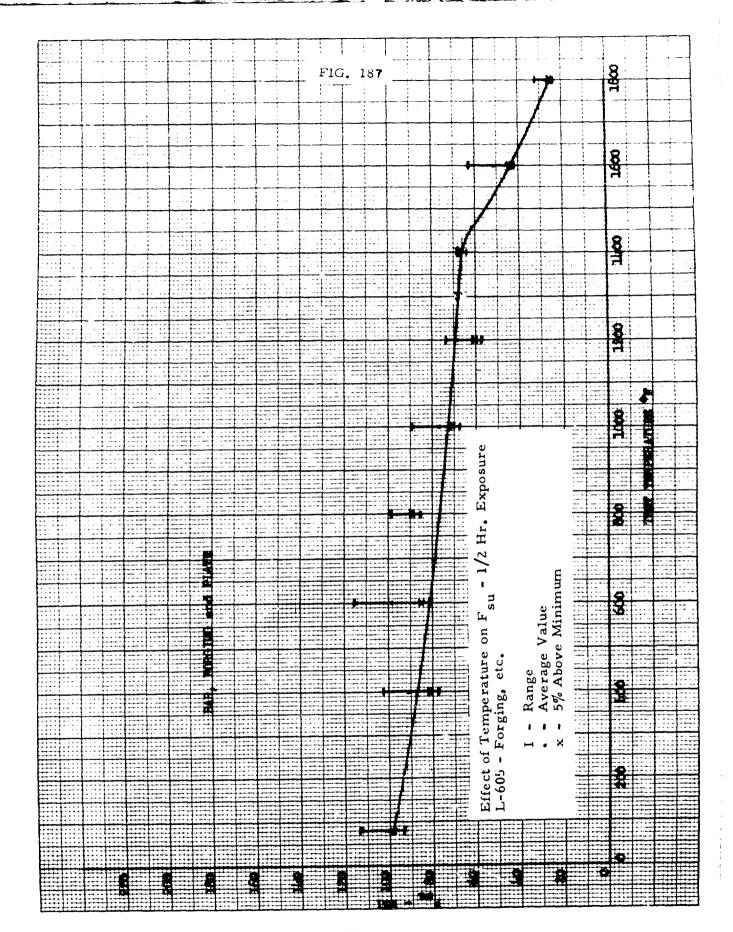




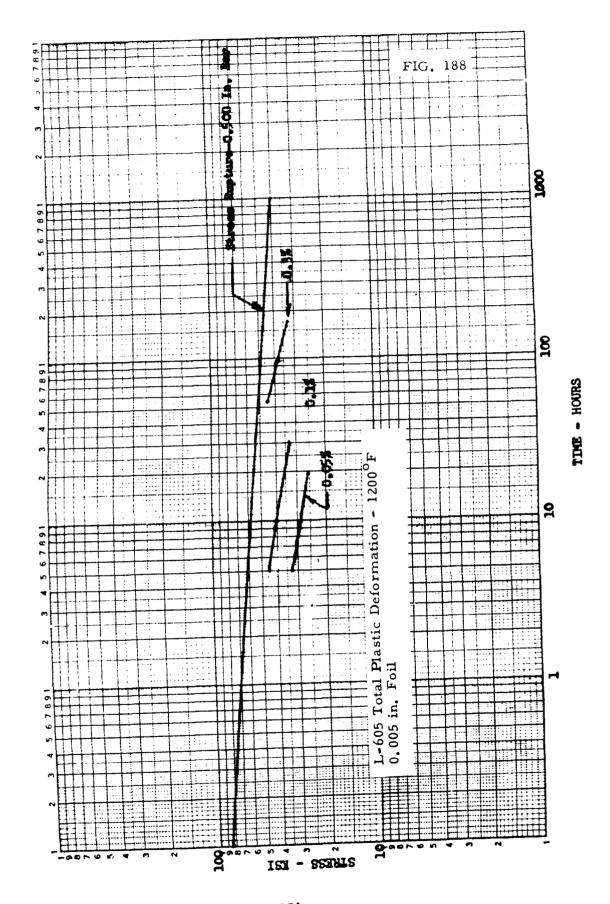
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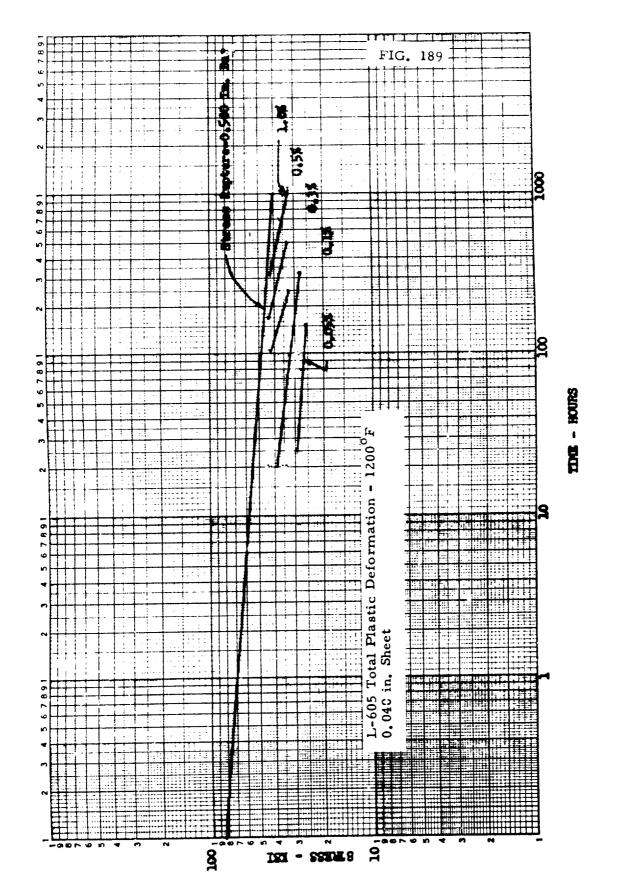


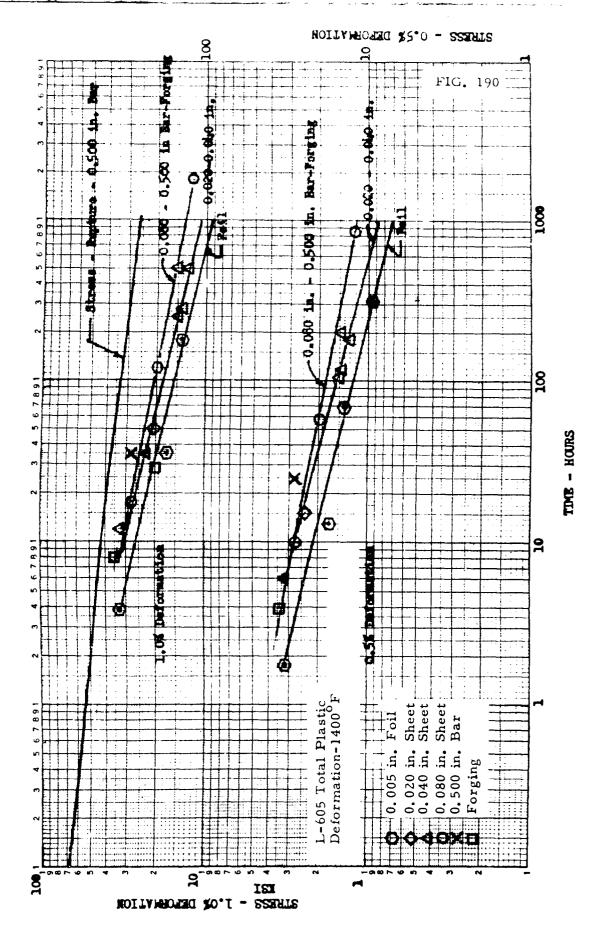
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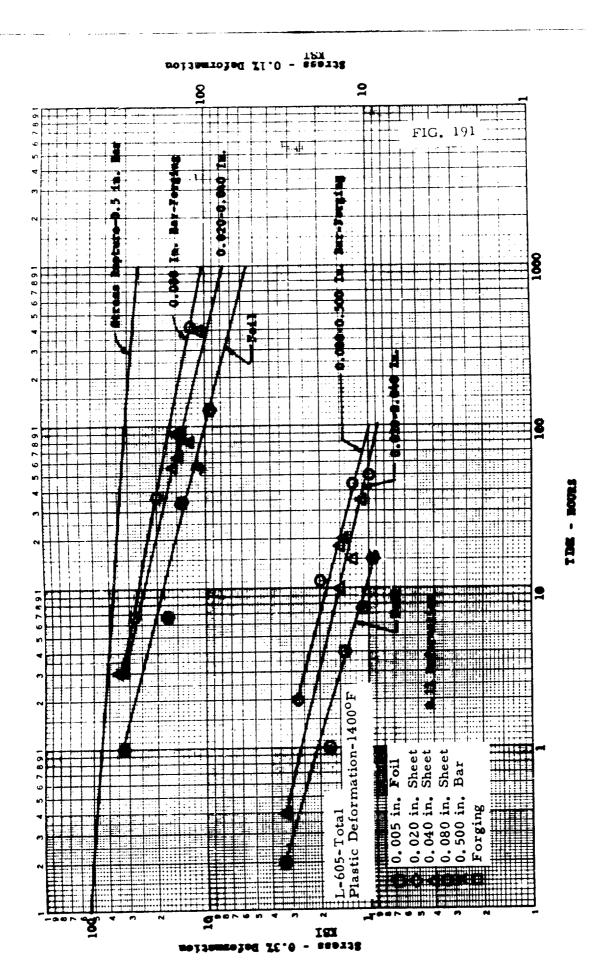


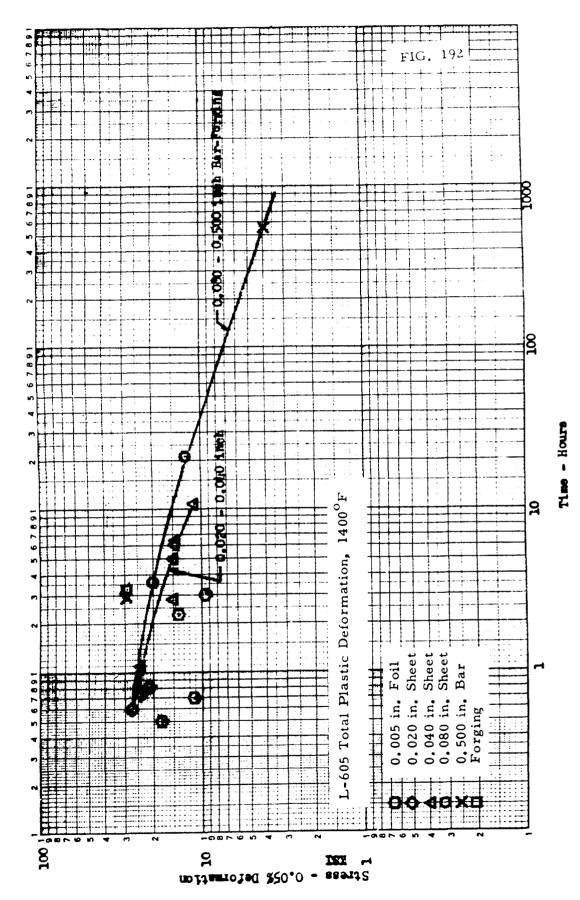
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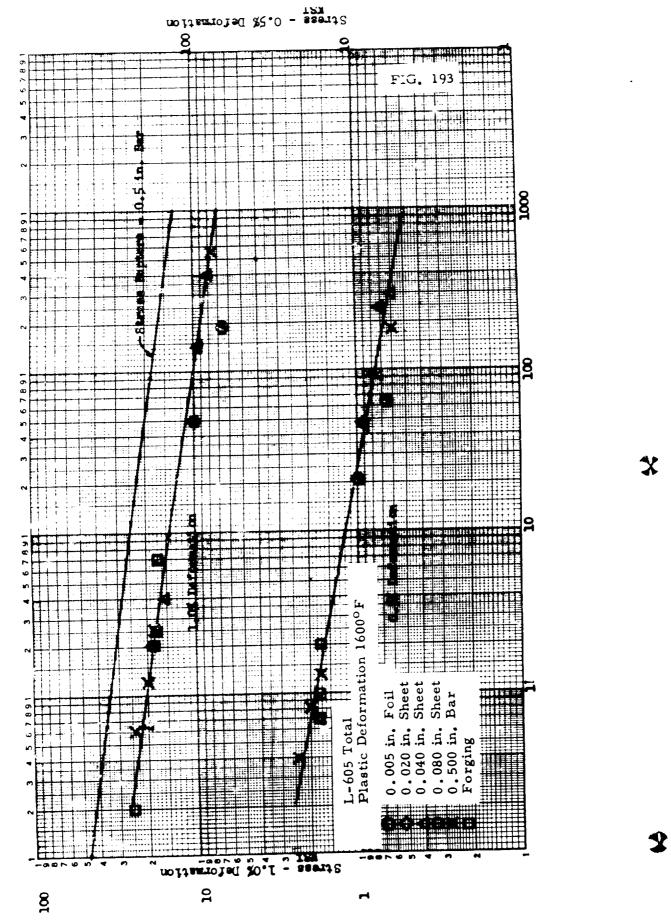


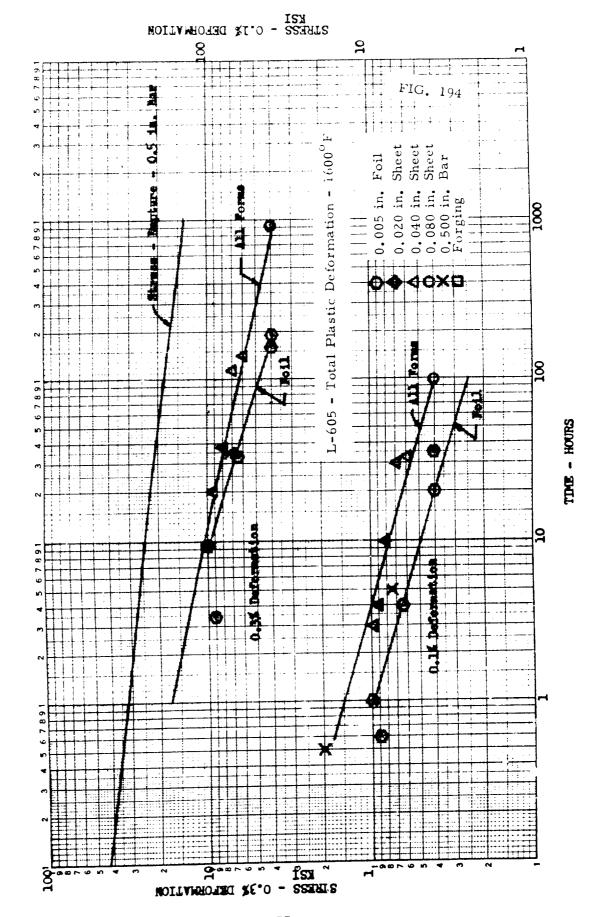


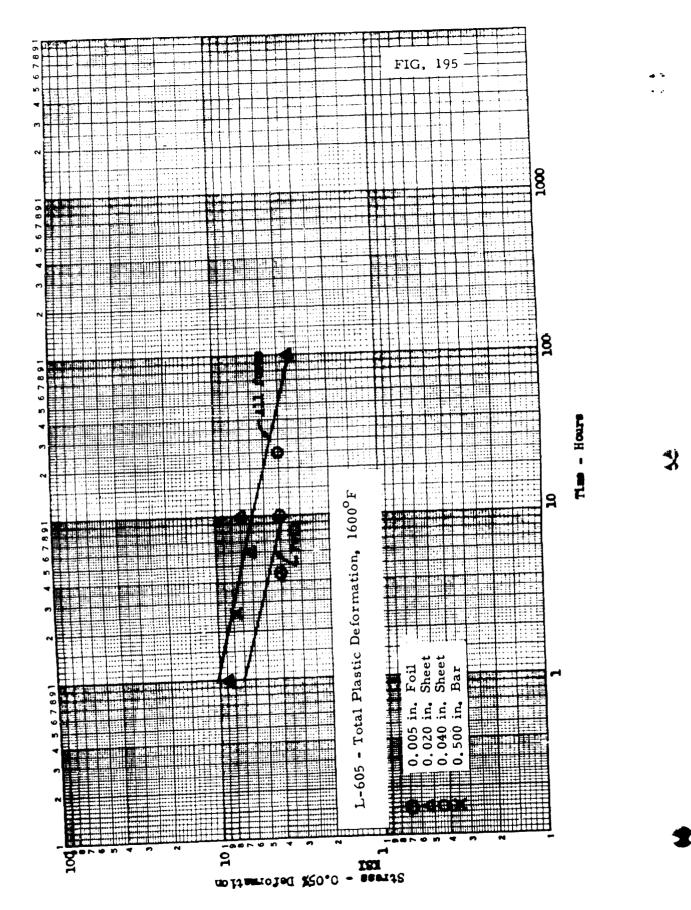


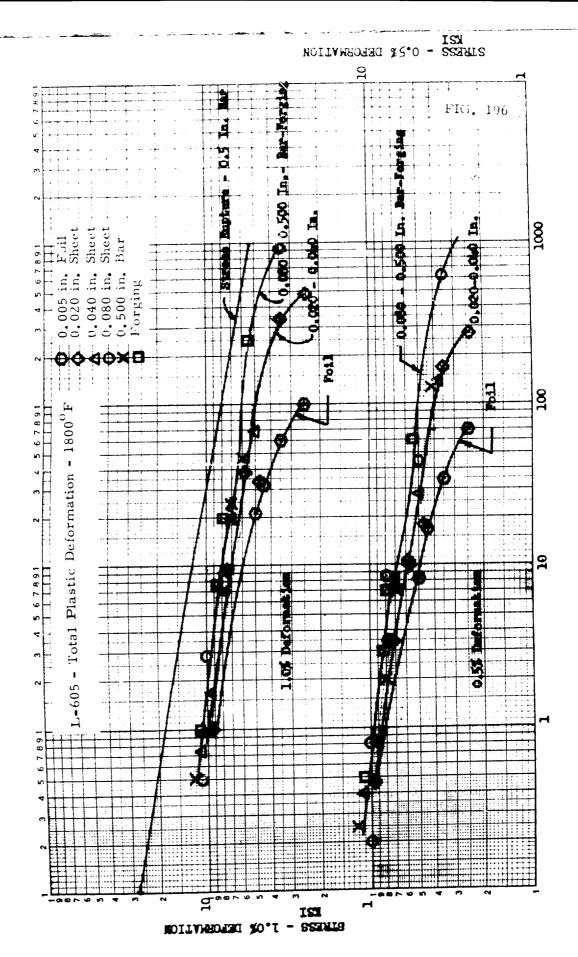


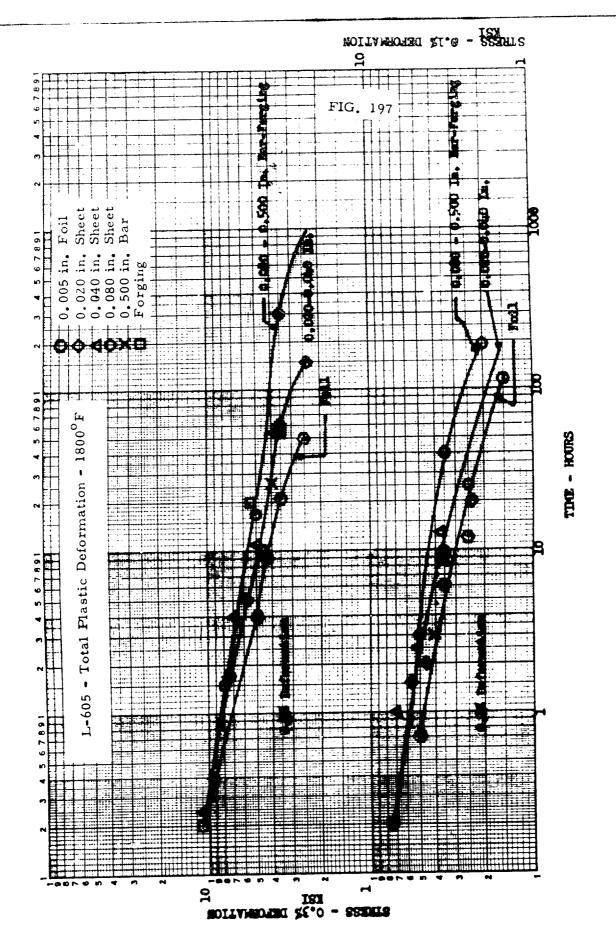


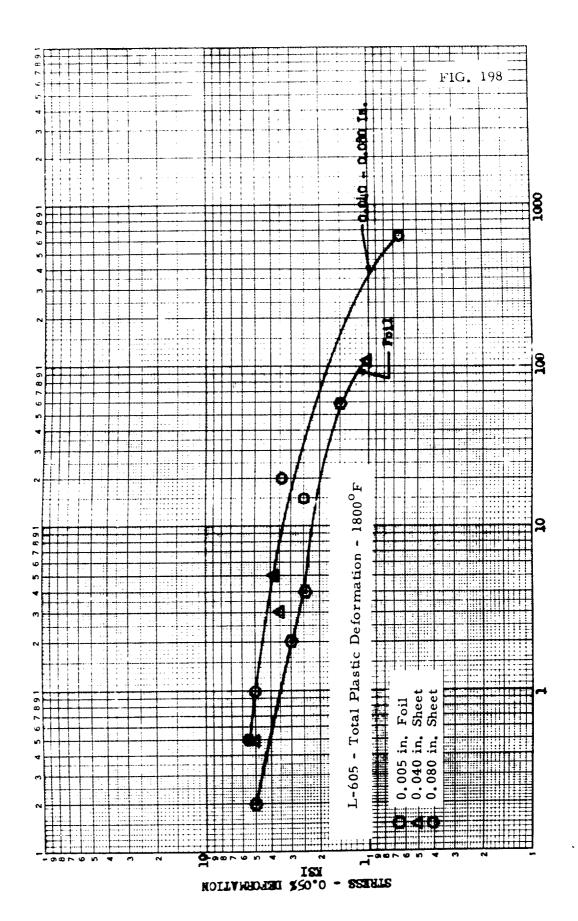












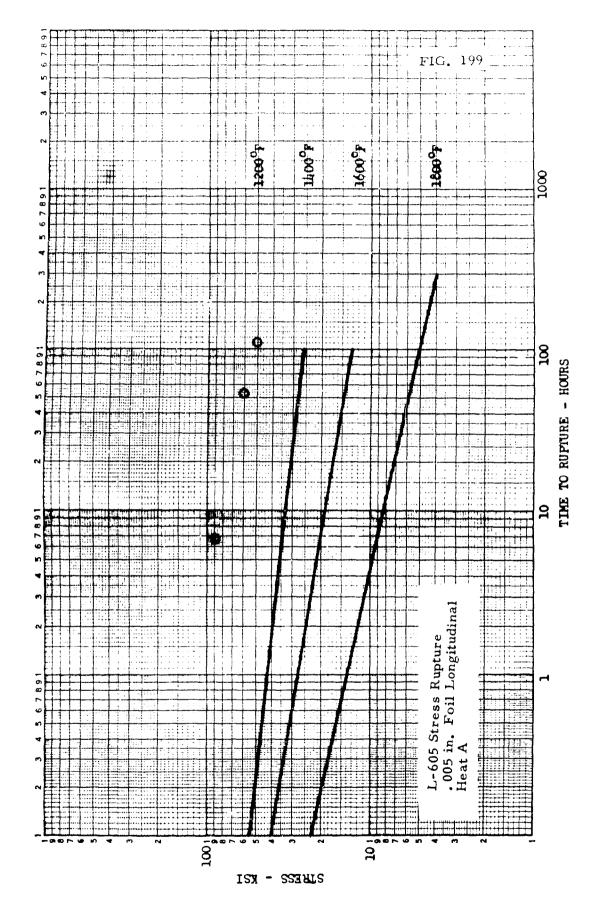
## SECTION VII

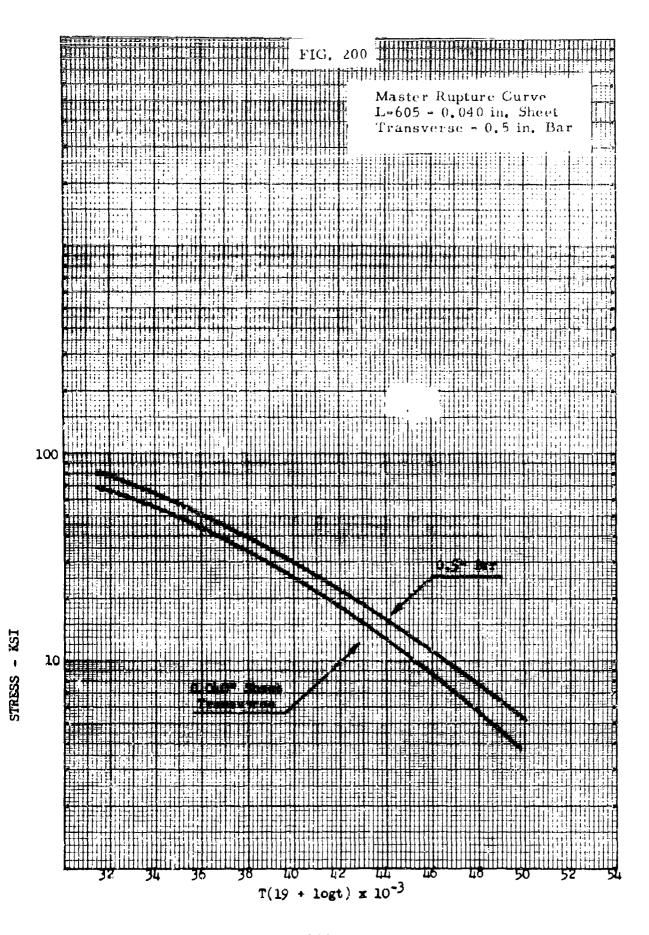
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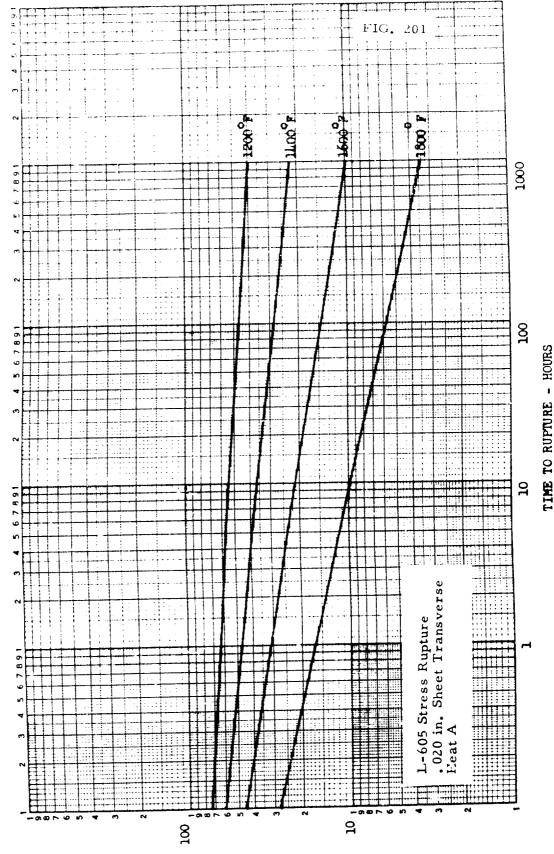
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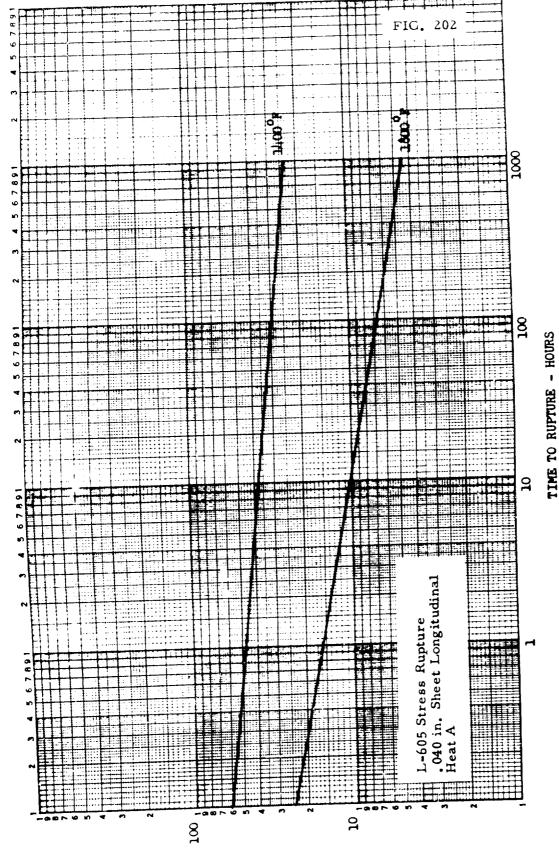




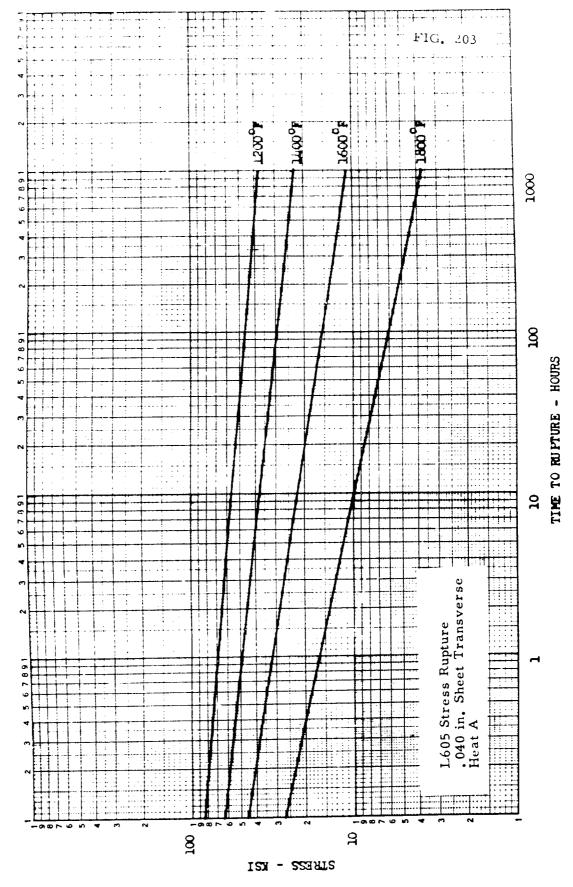
TIME TO RUPTURE - HOURS 

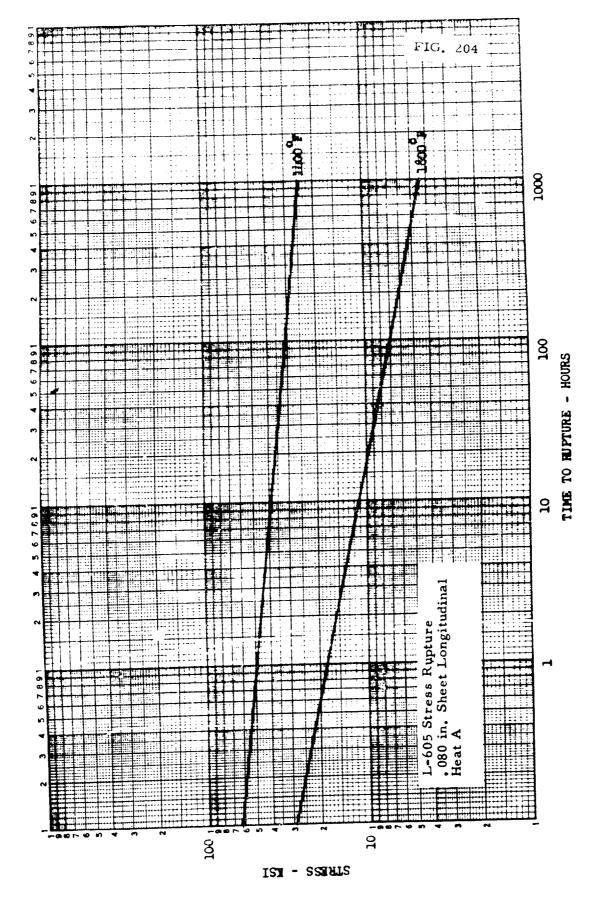


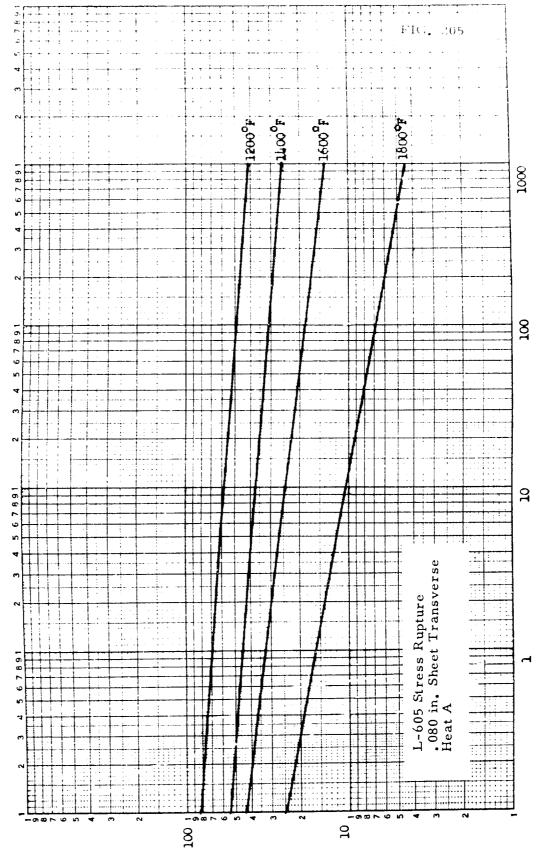
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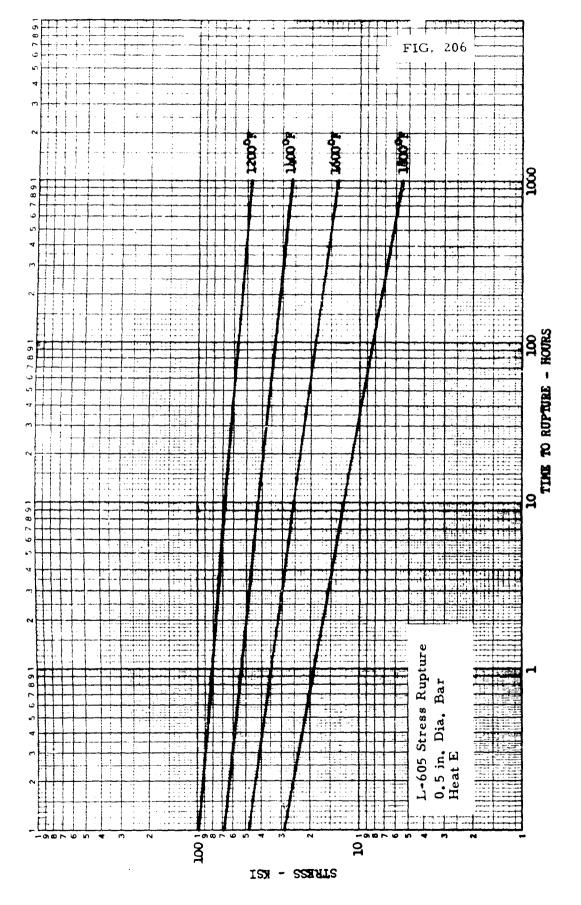




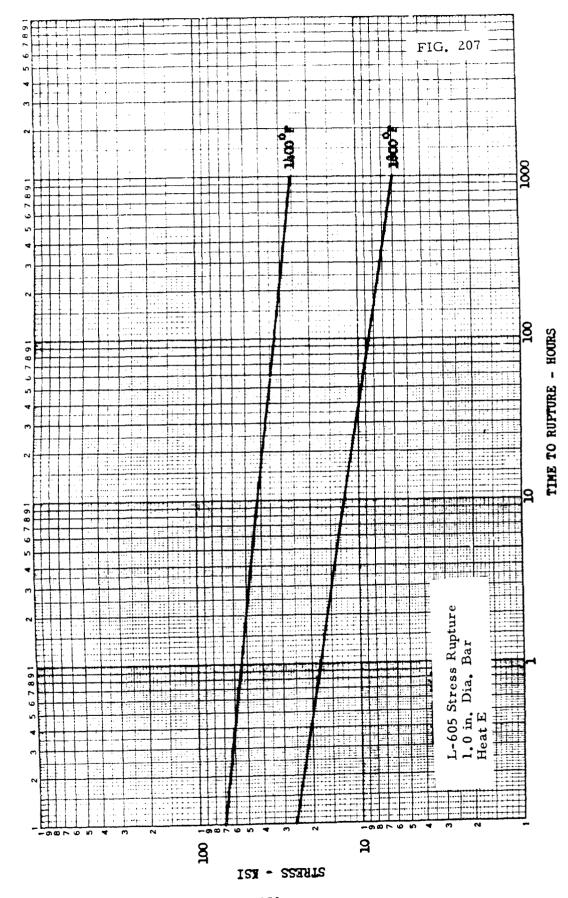


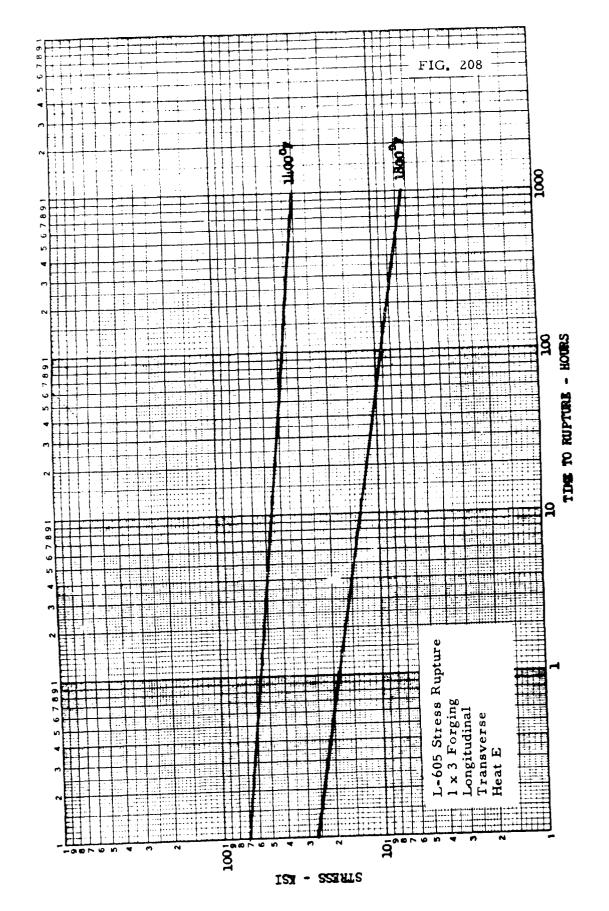
TIME TO RUPTURE - HOURS

STRESS - KSI



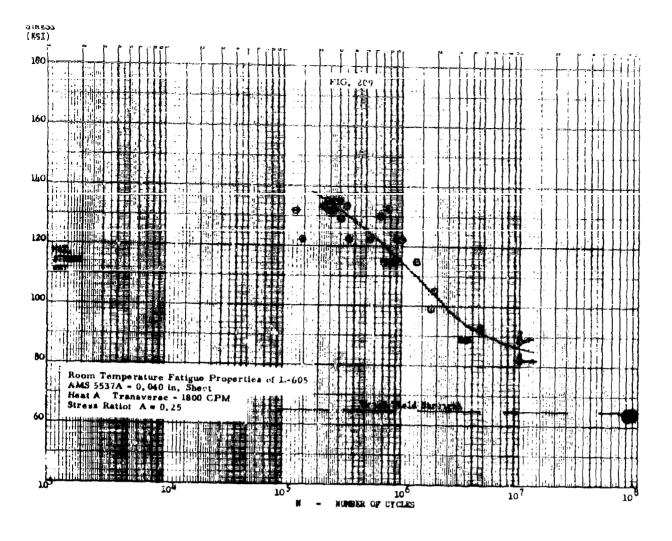
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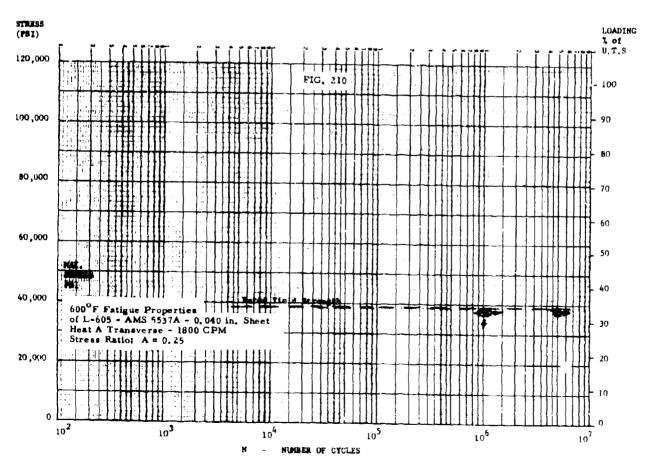


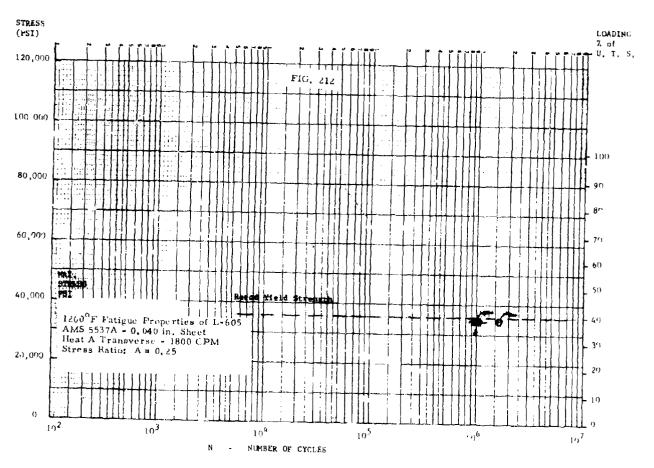


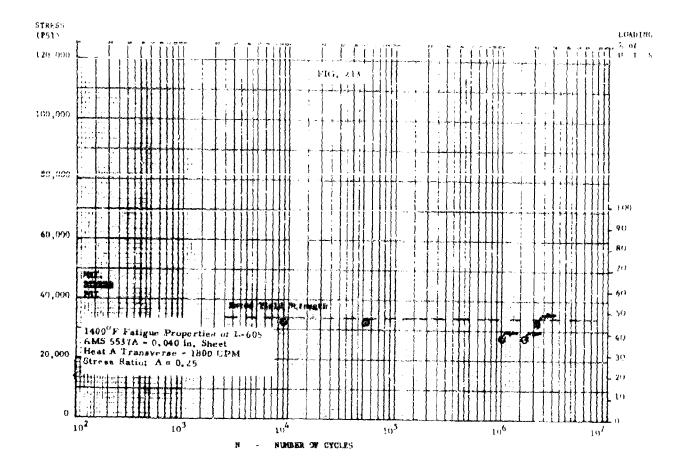
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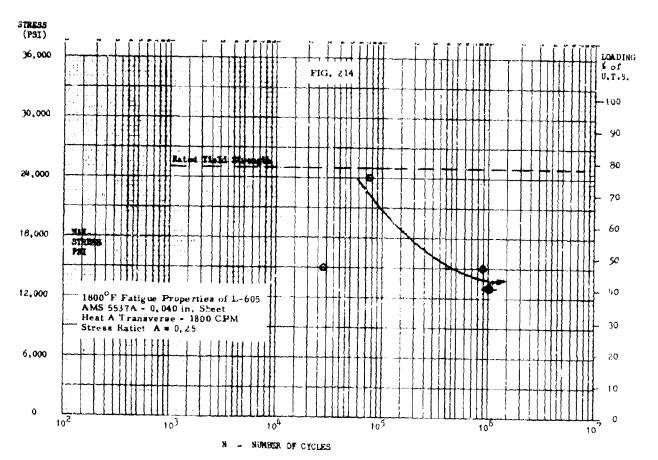
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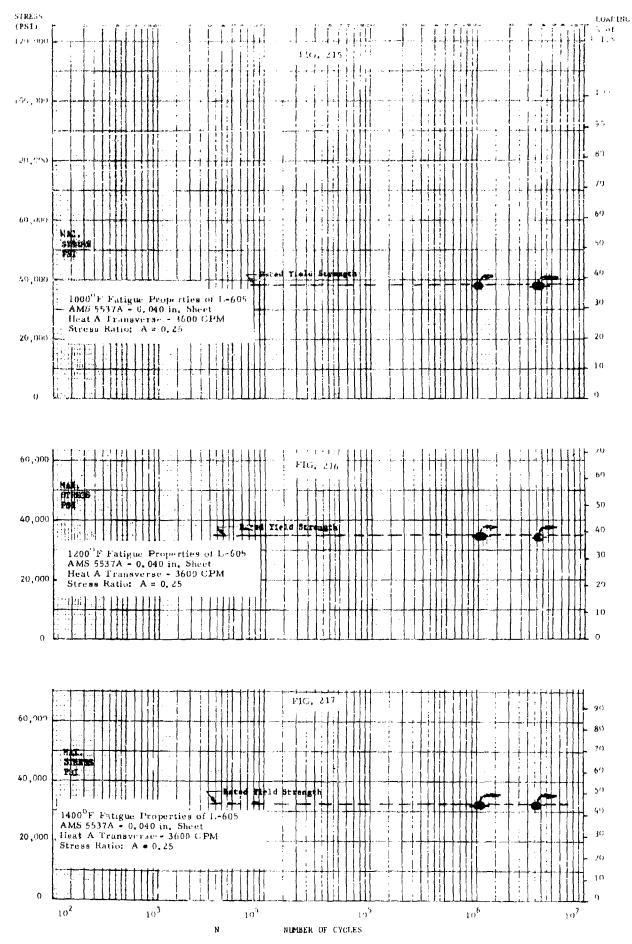


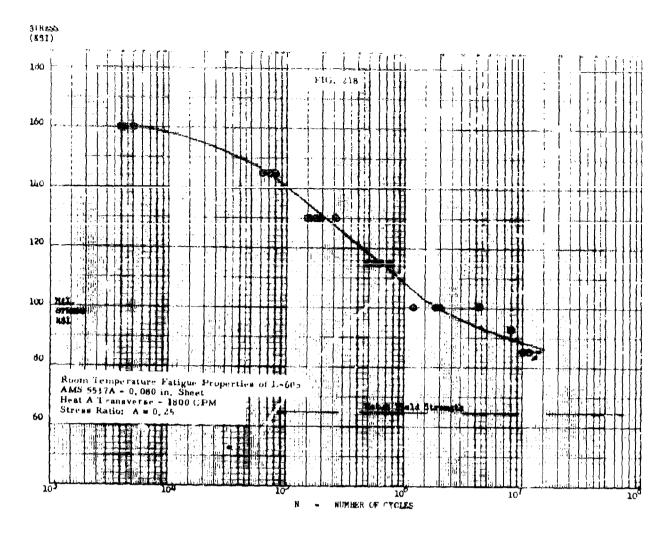


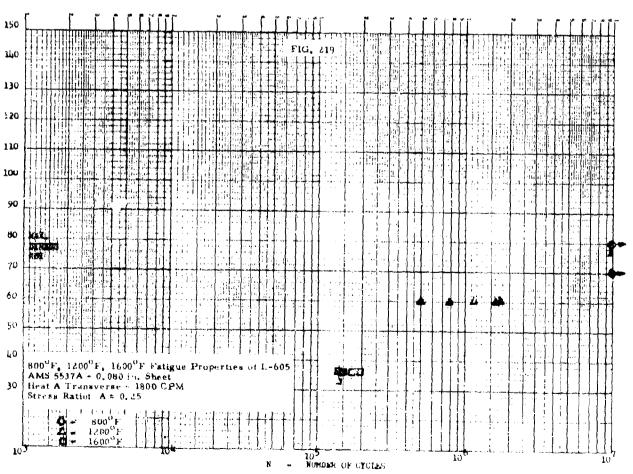


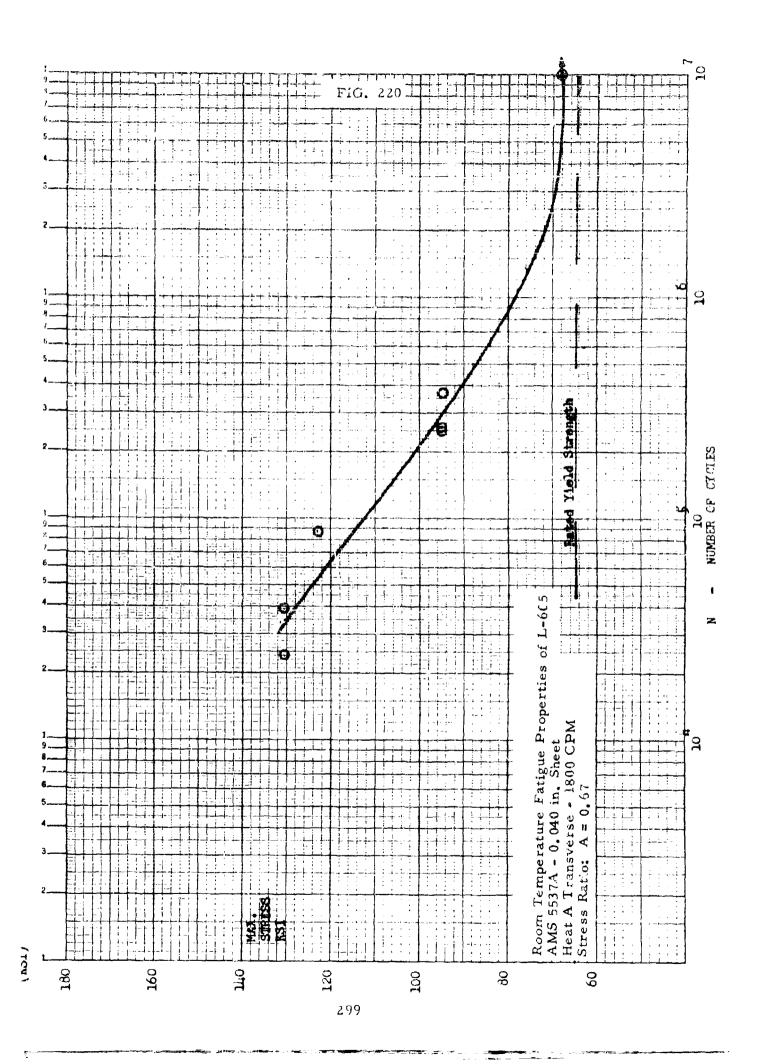


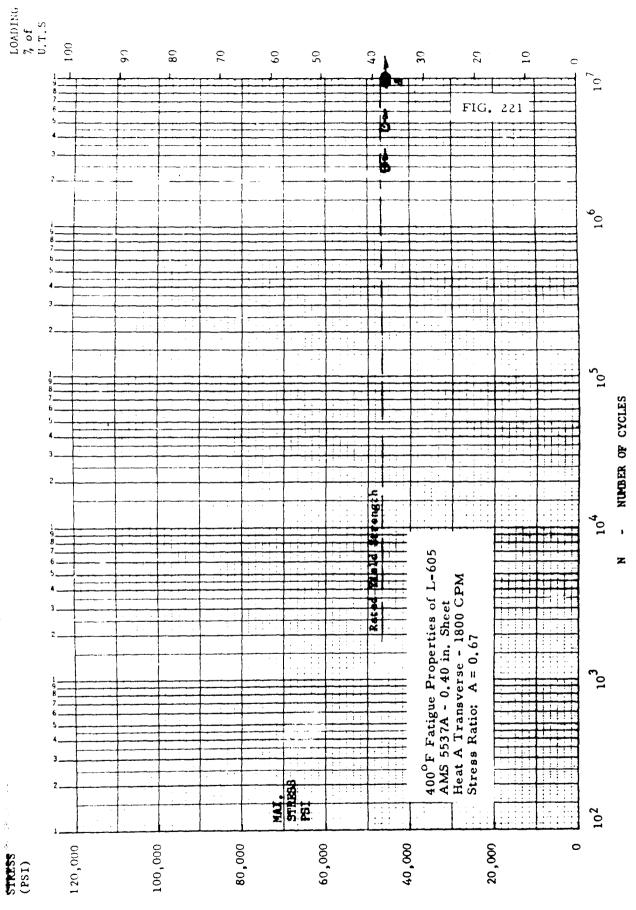


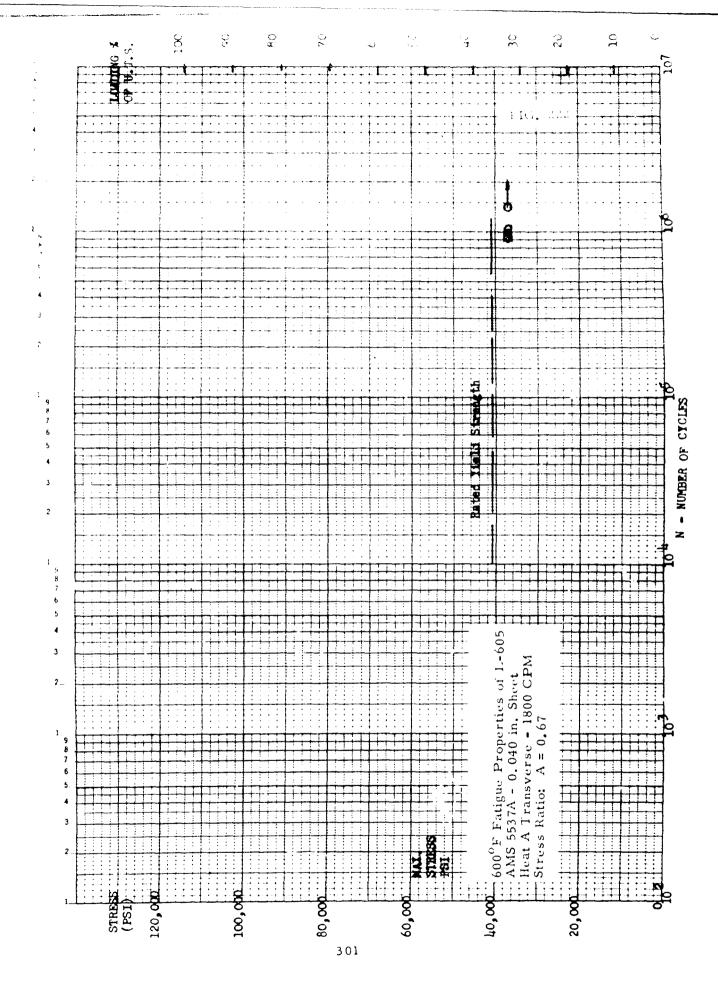


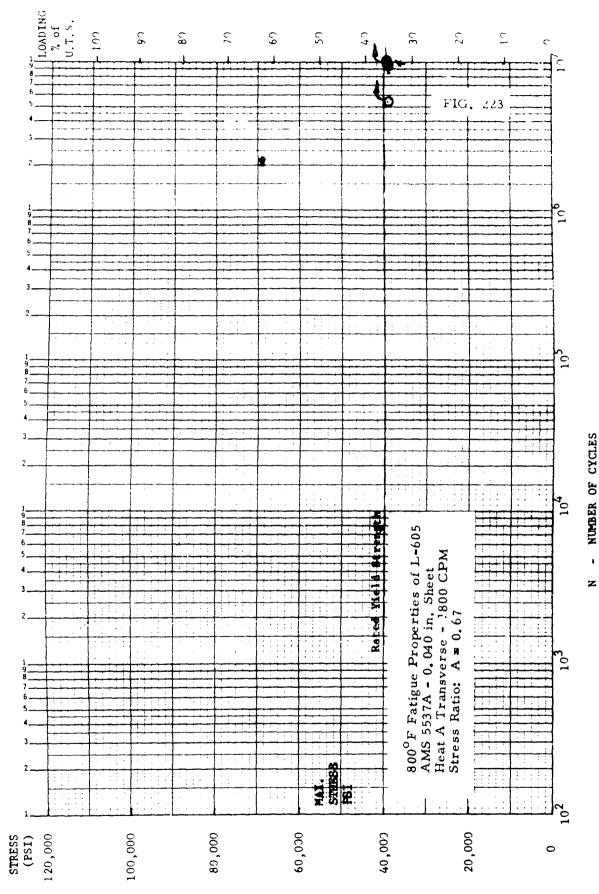




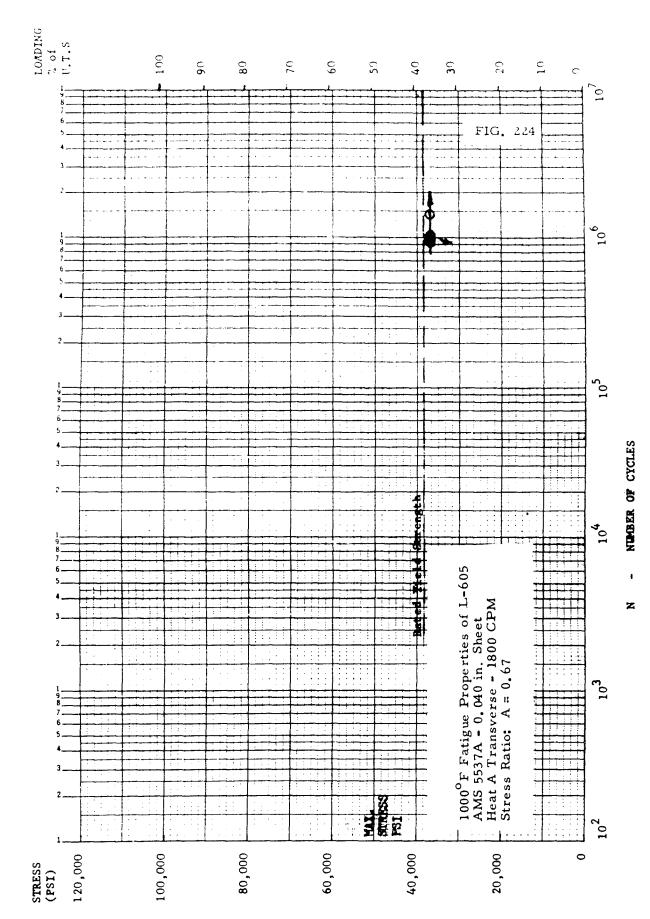


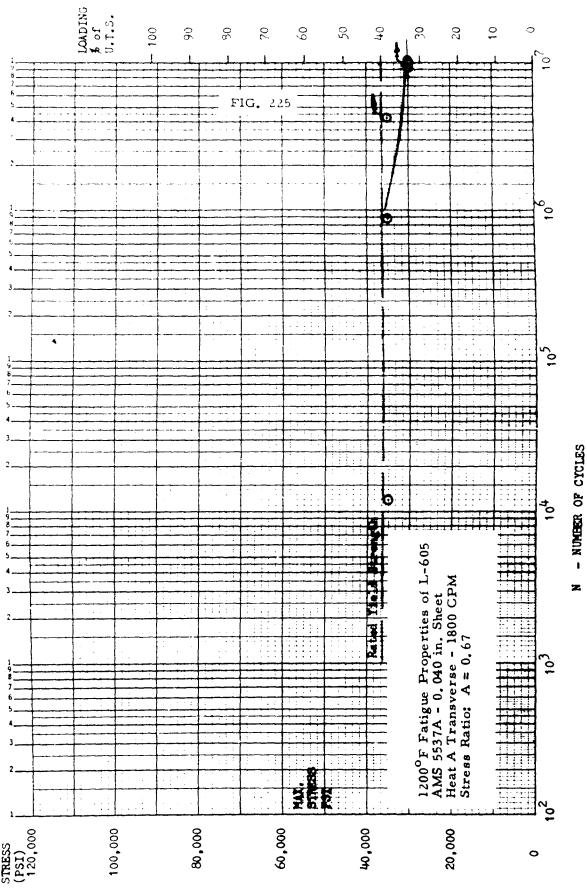


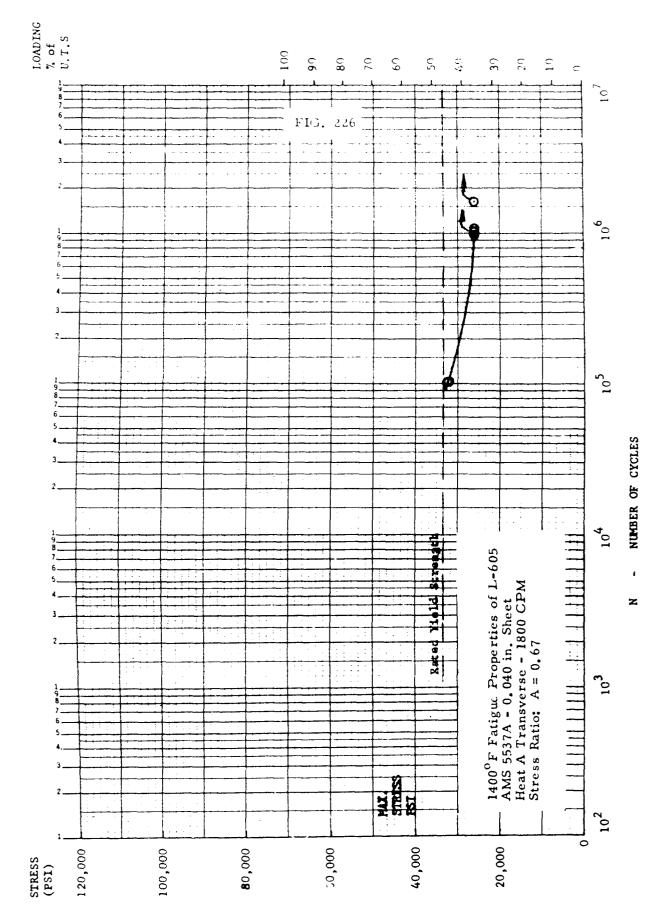


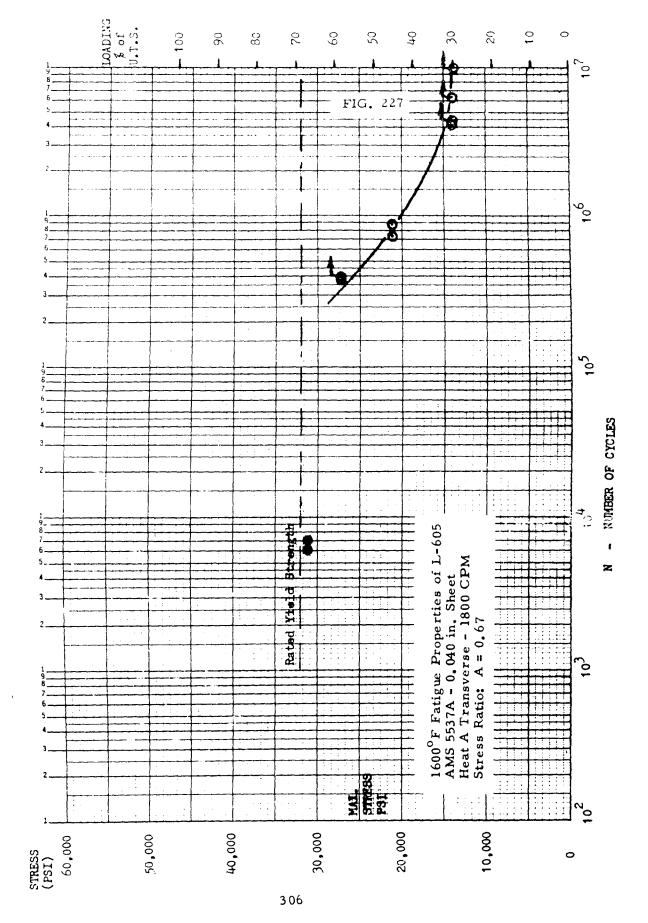


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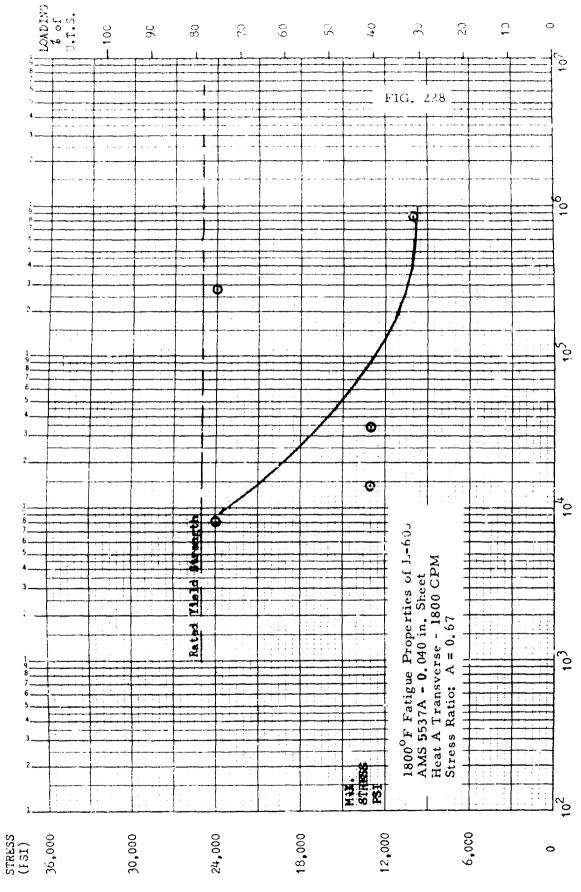








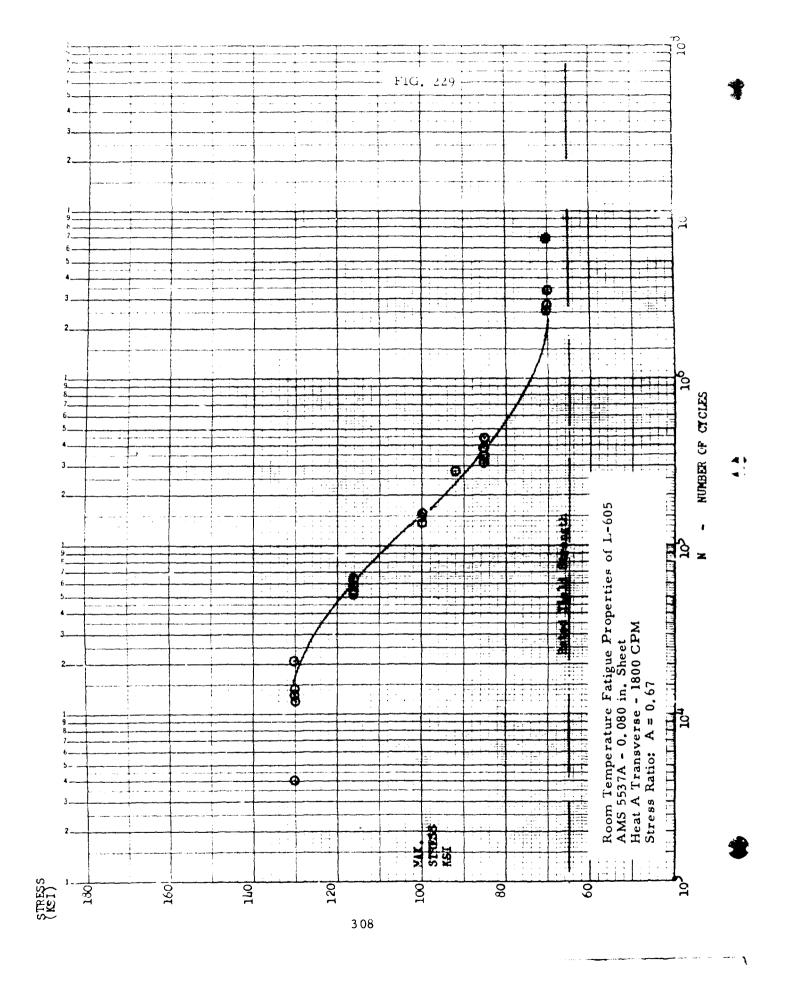
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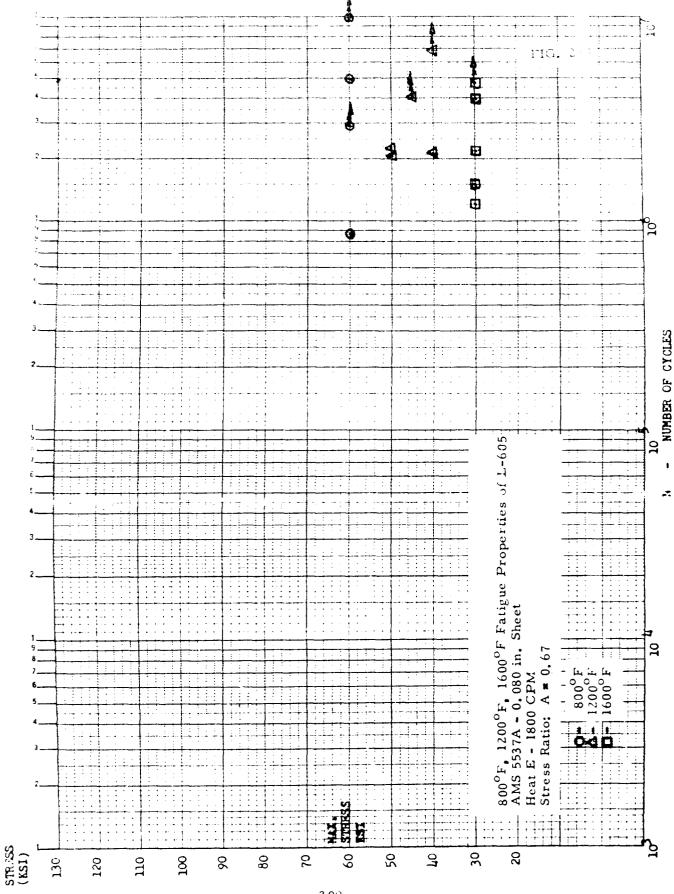


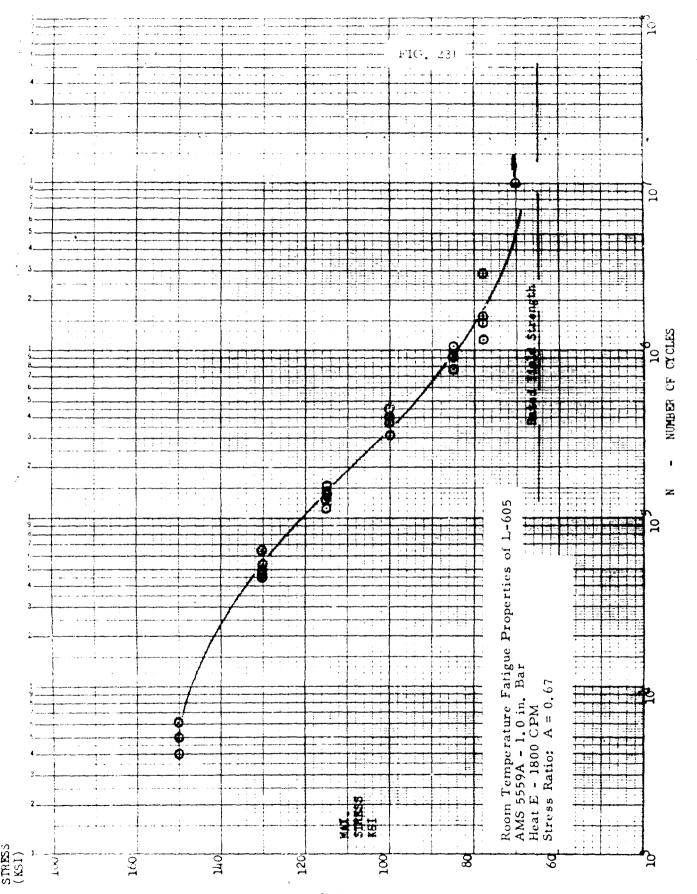
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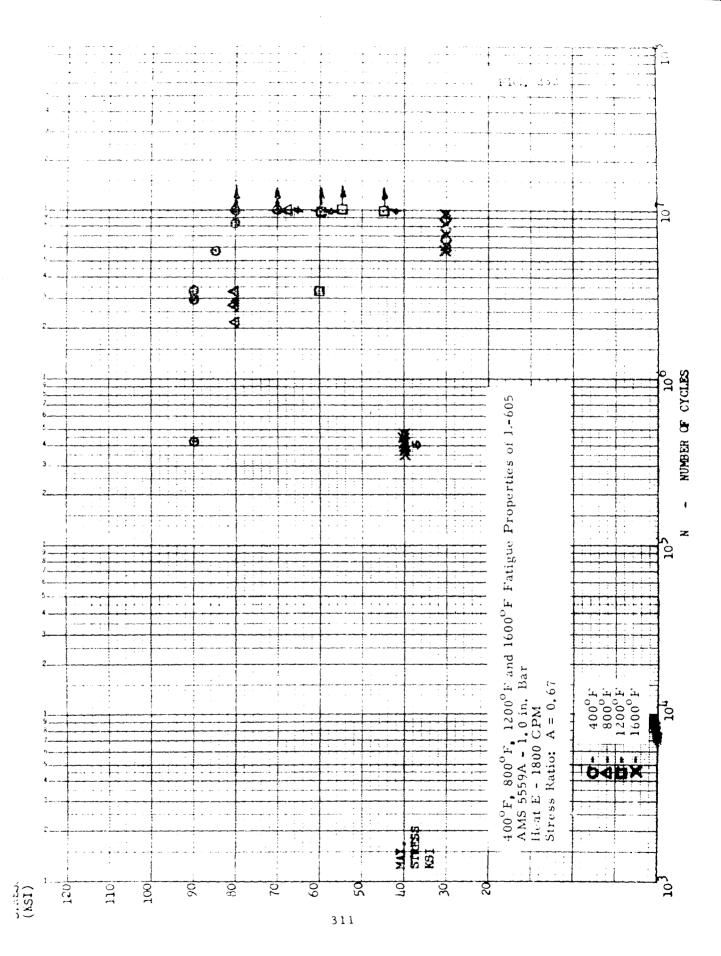
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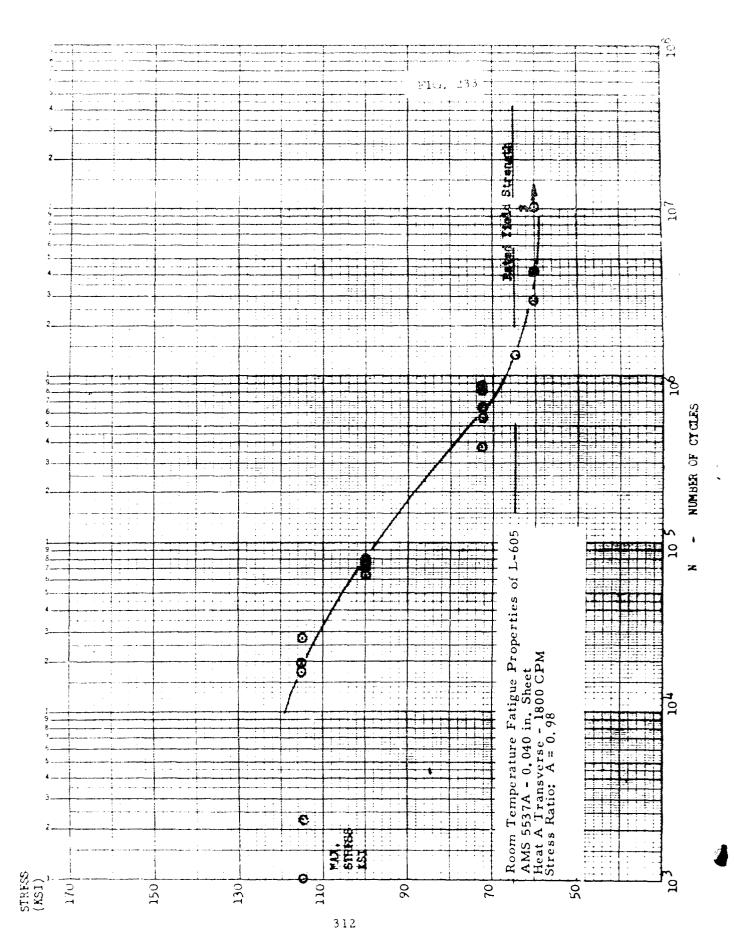


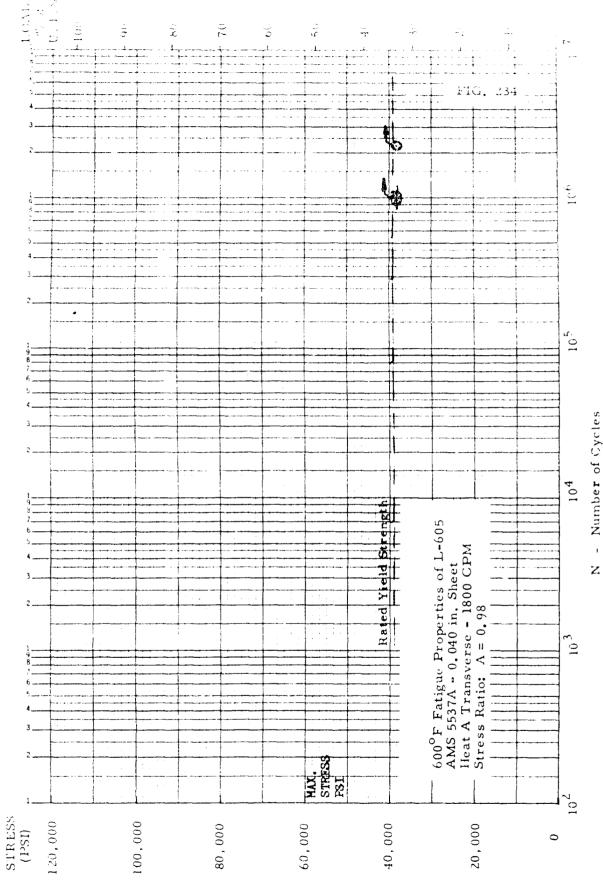


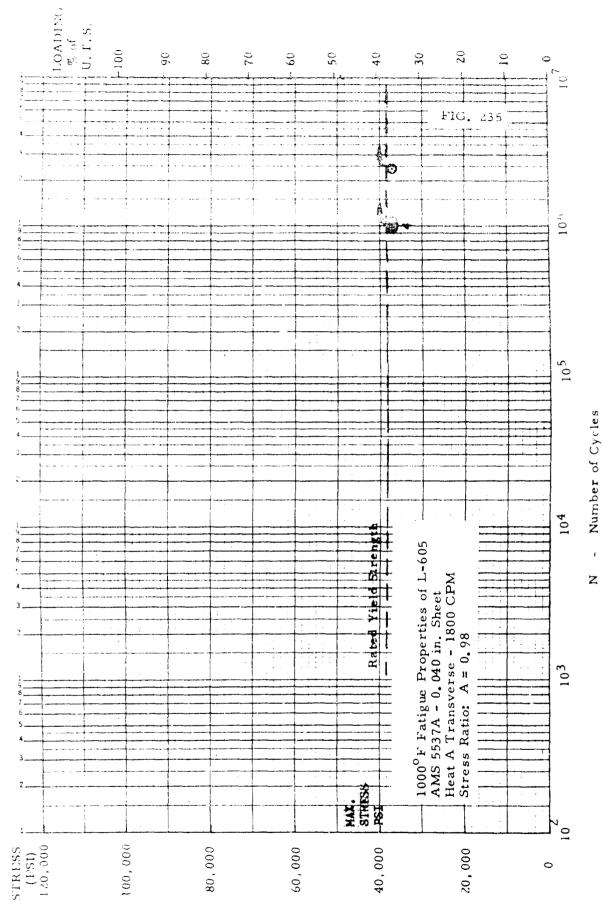


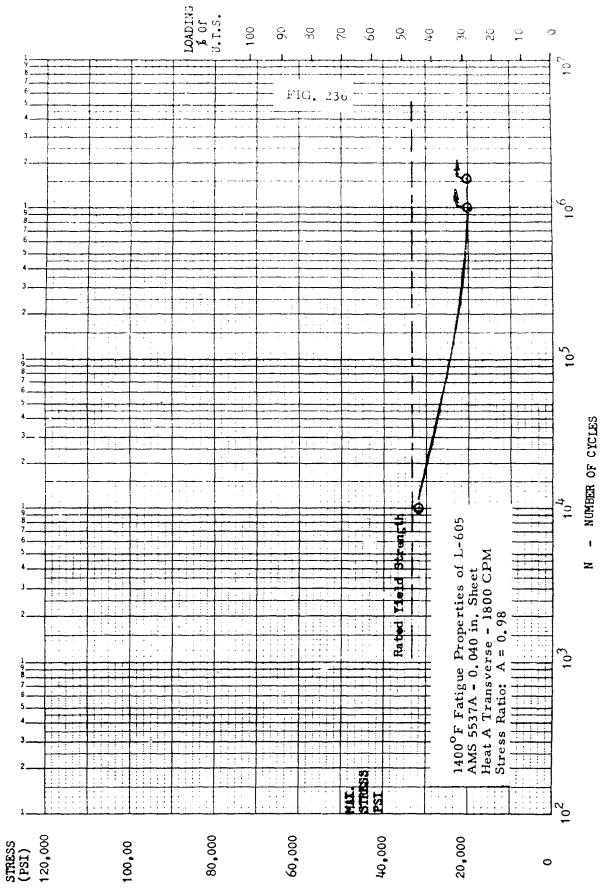


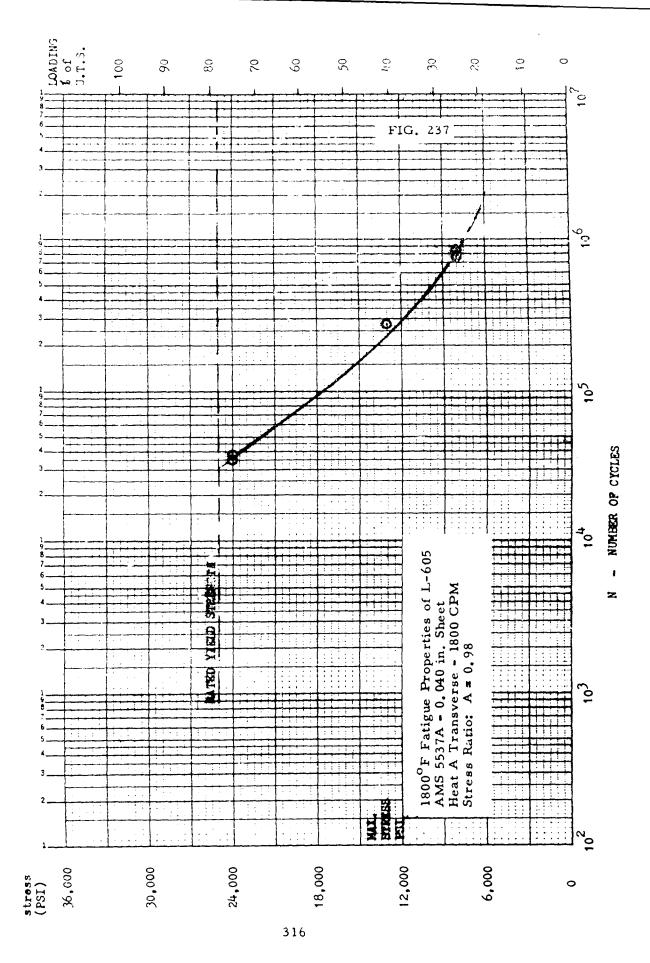
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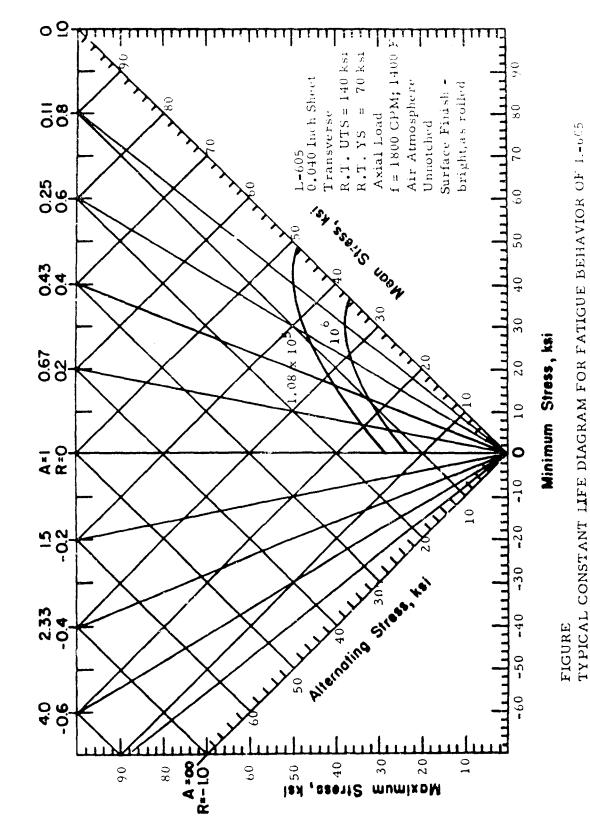












SHEET MATERIAL AT 1400 F.

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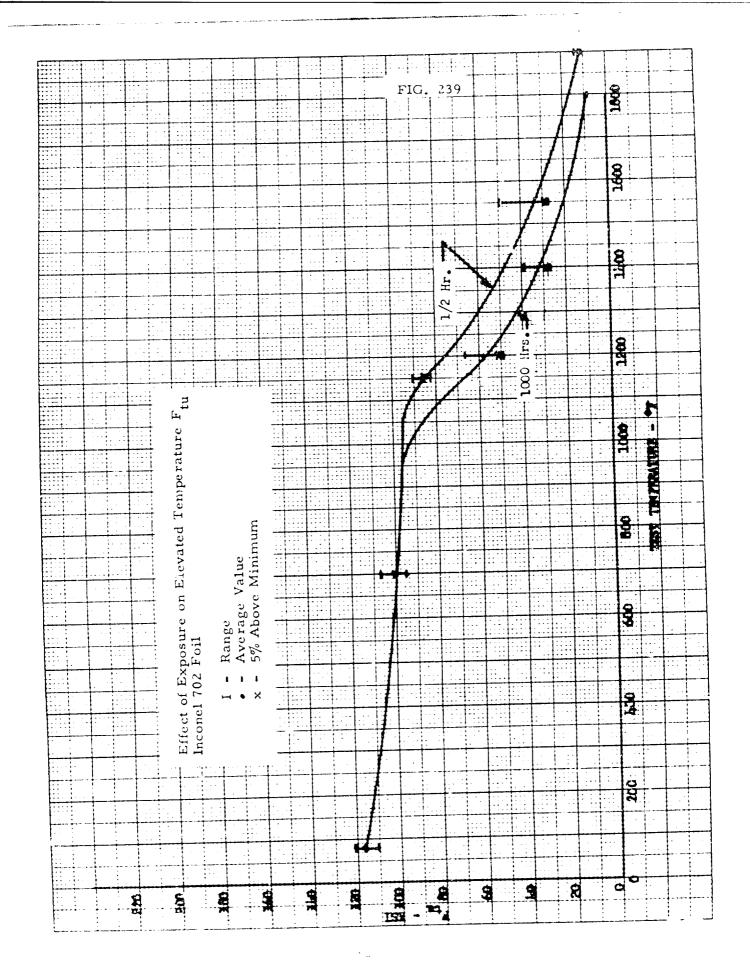
## SECTION VII - TEST RESULTS. TARLES AND GRAPHS

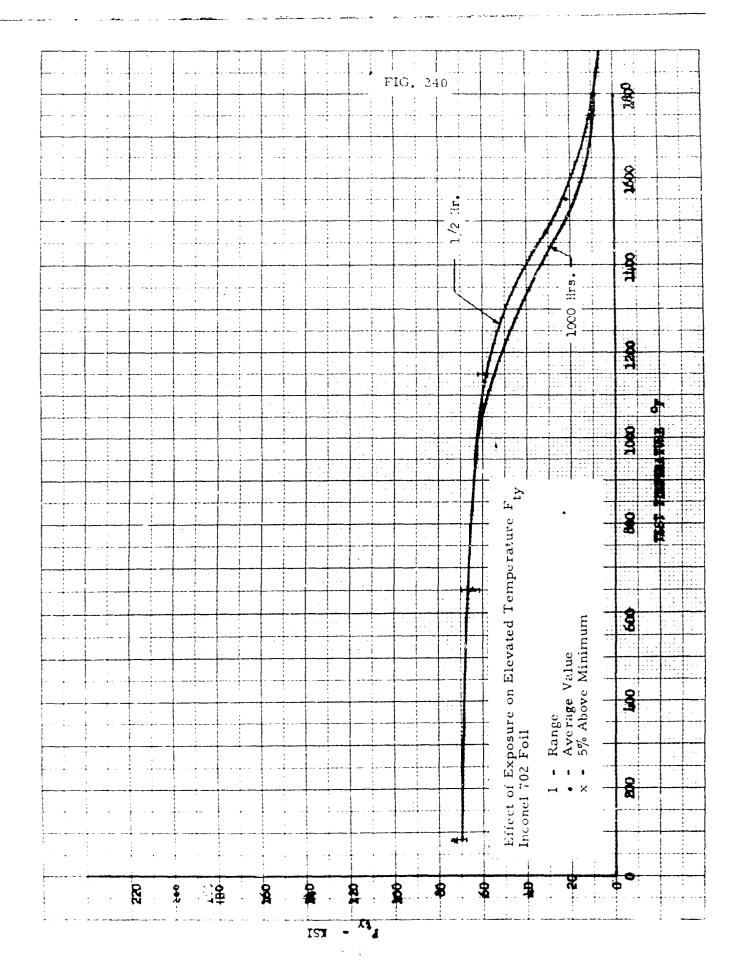
# SECTION 7.3 MATERIAL, INCONEL 702

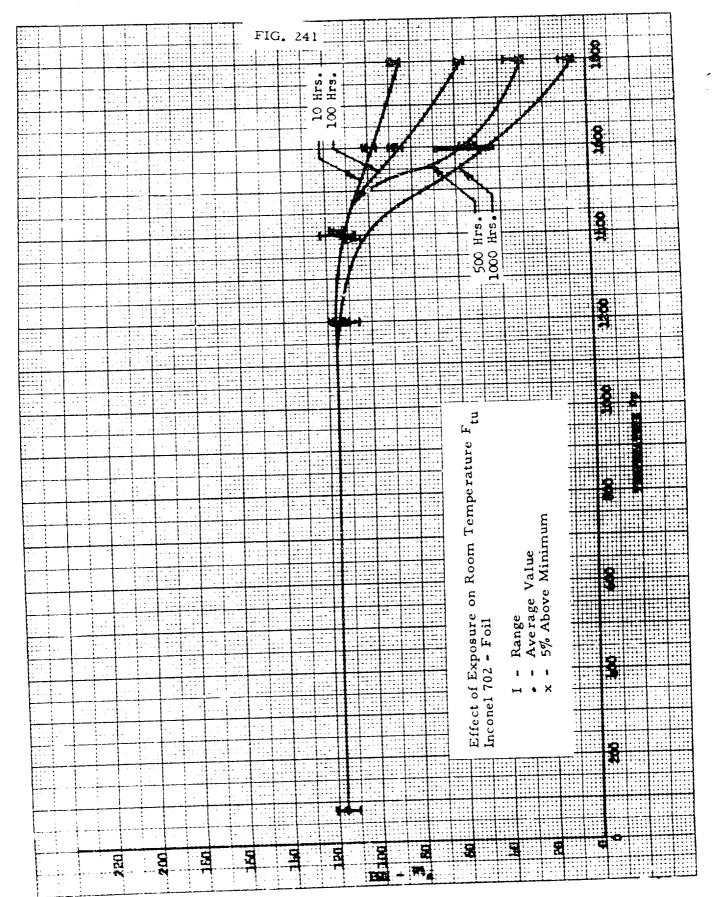
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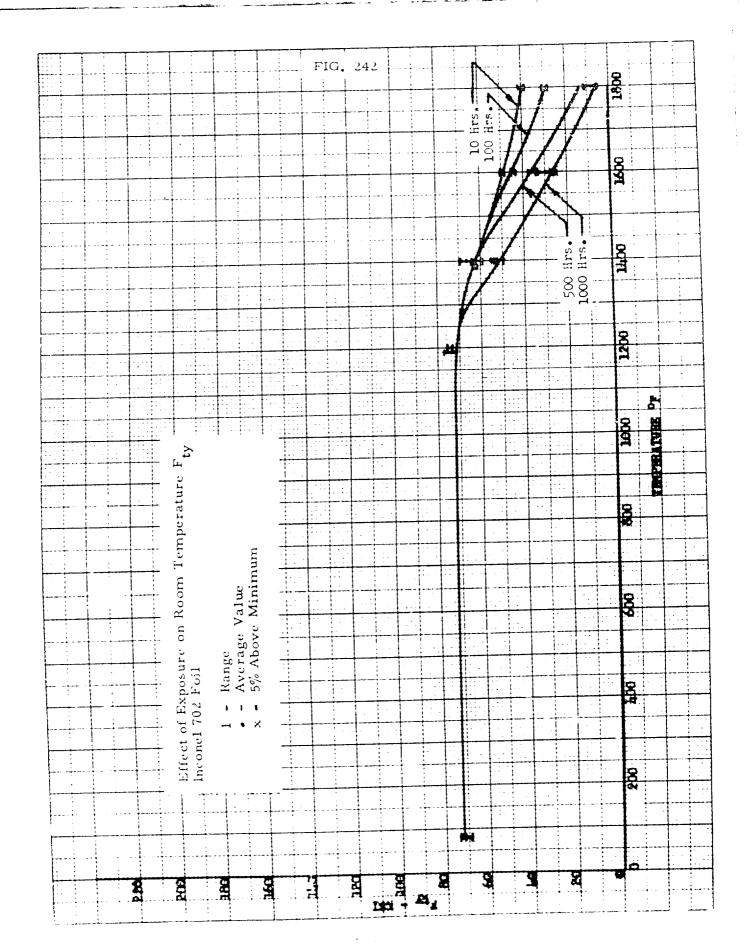
Section 7.3.1 Tension



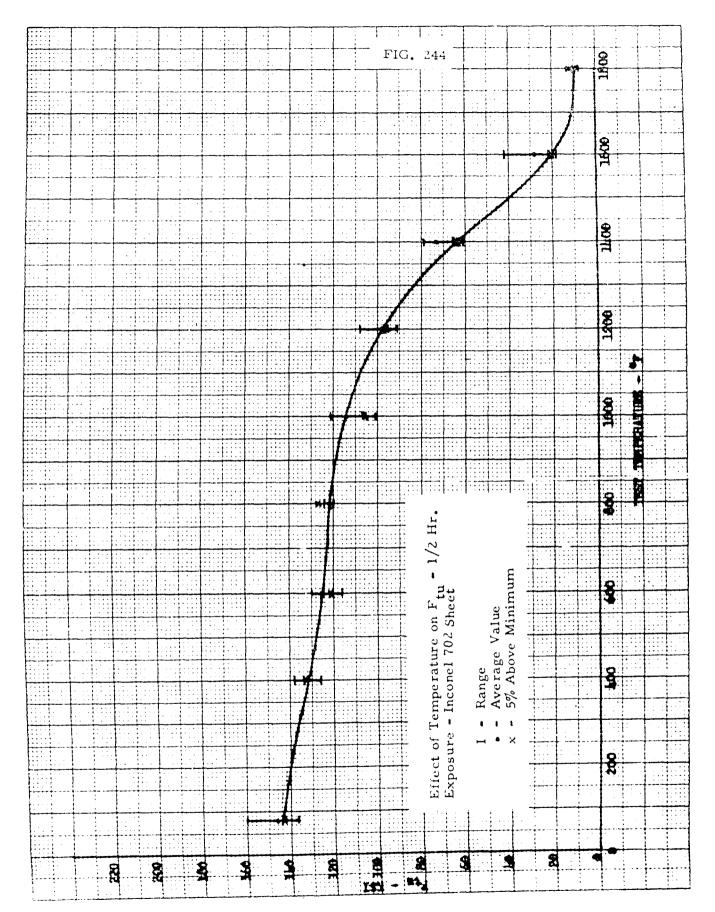


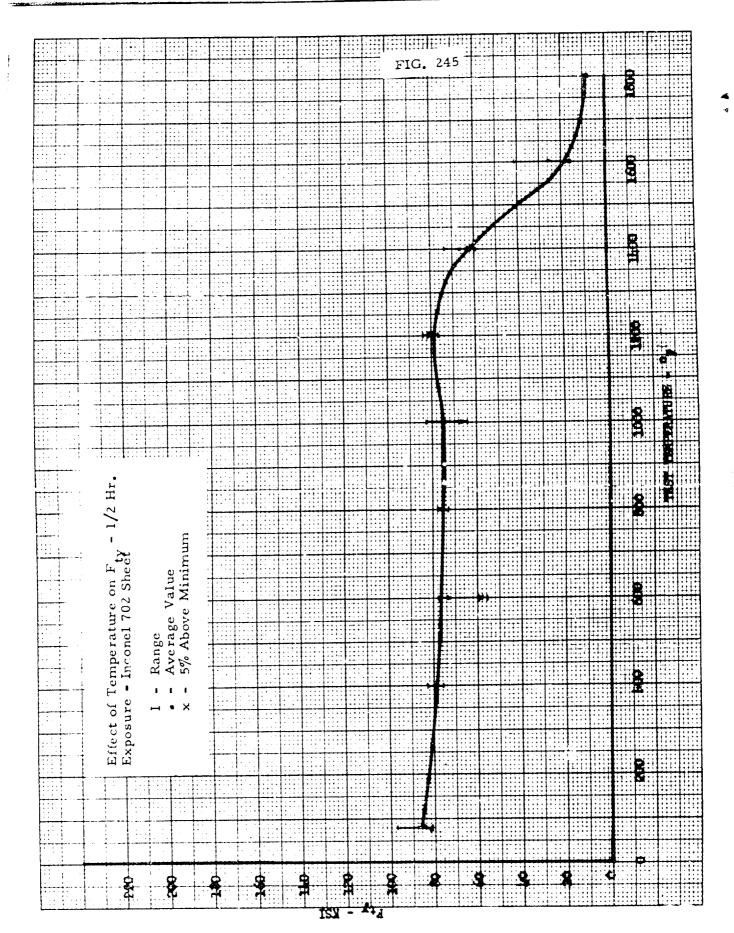


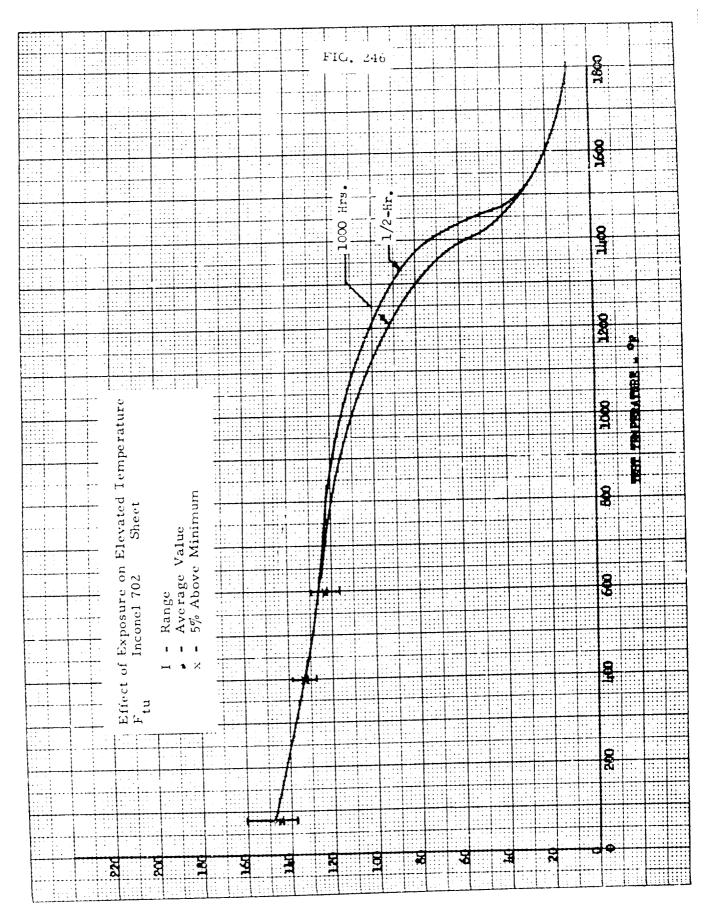
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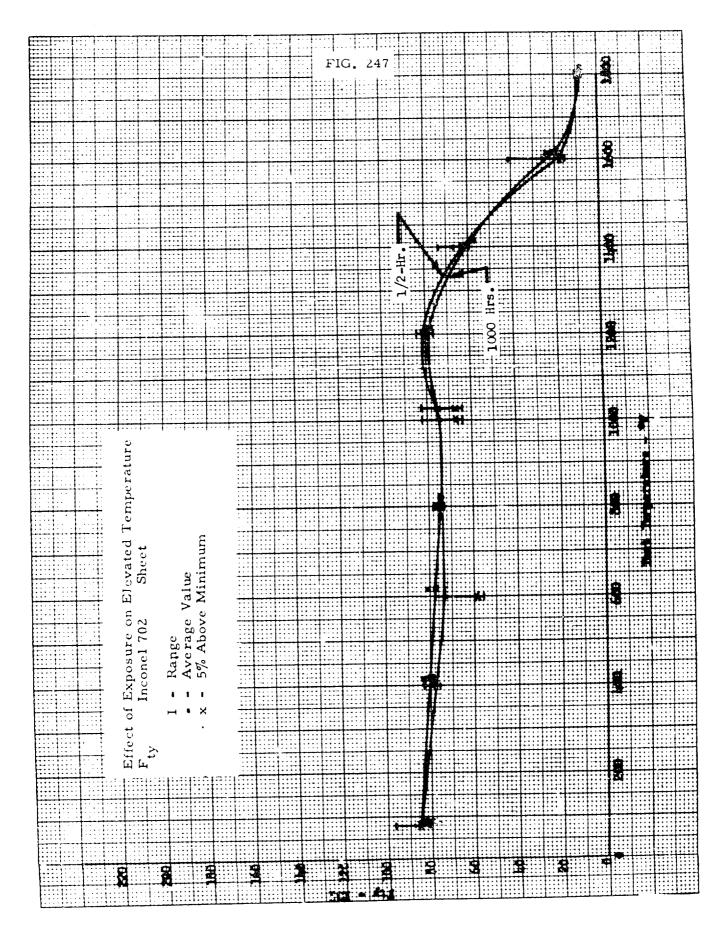


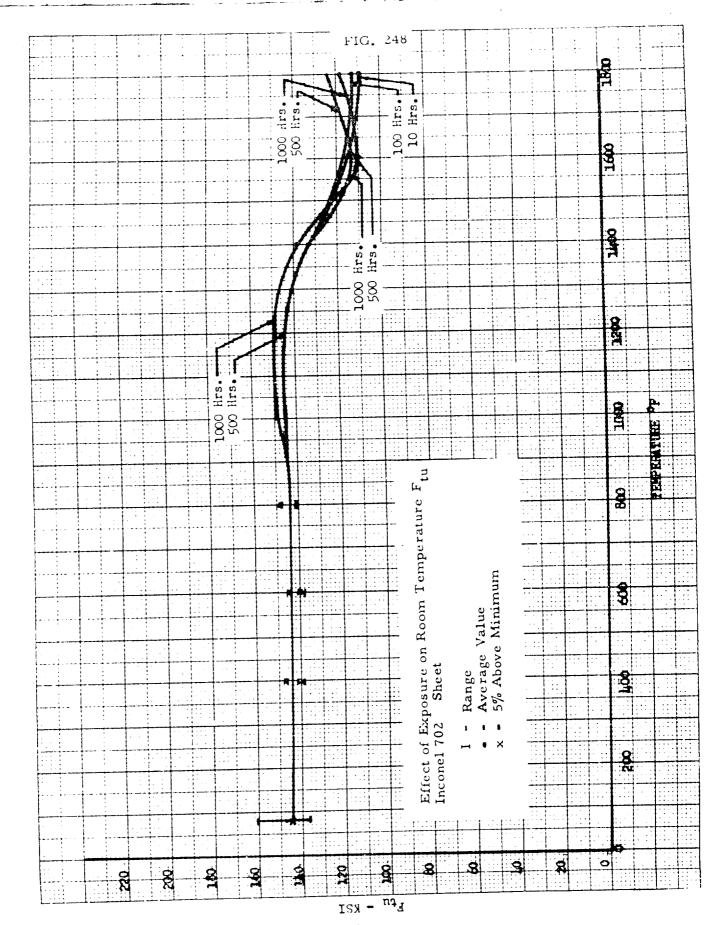
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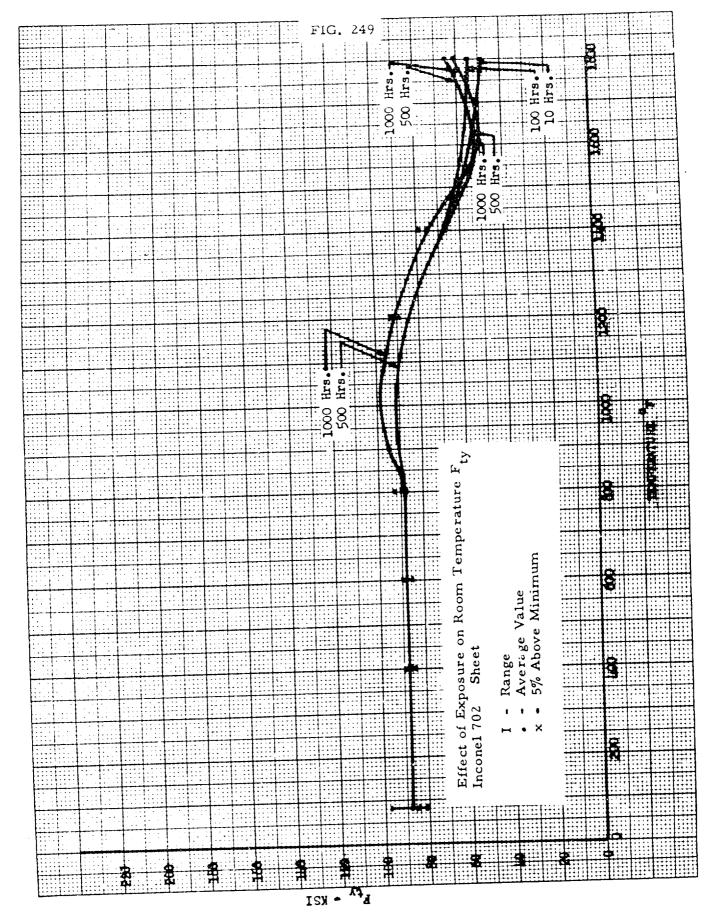


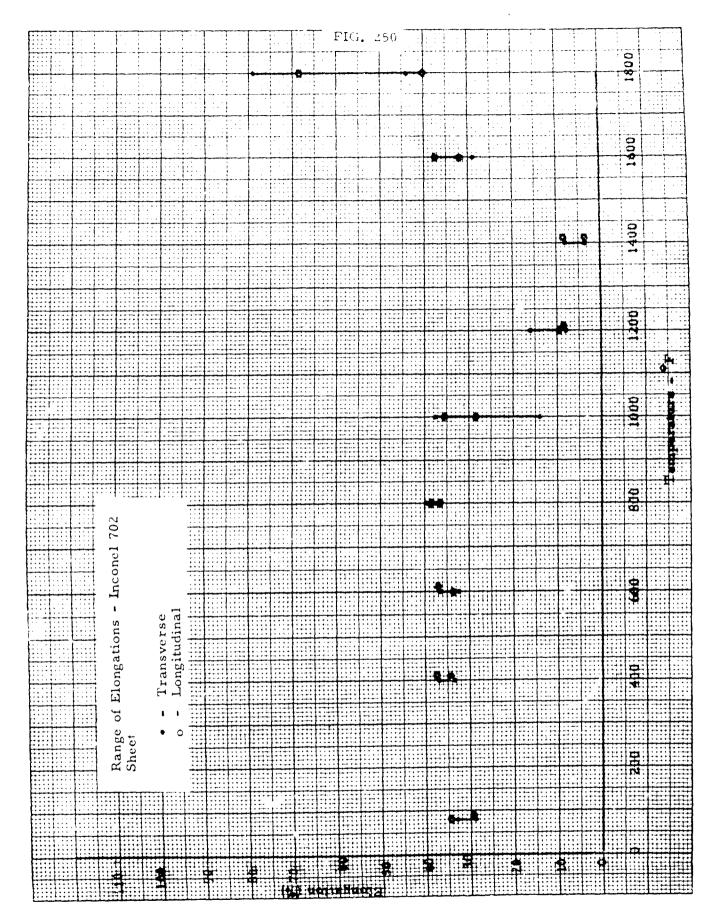




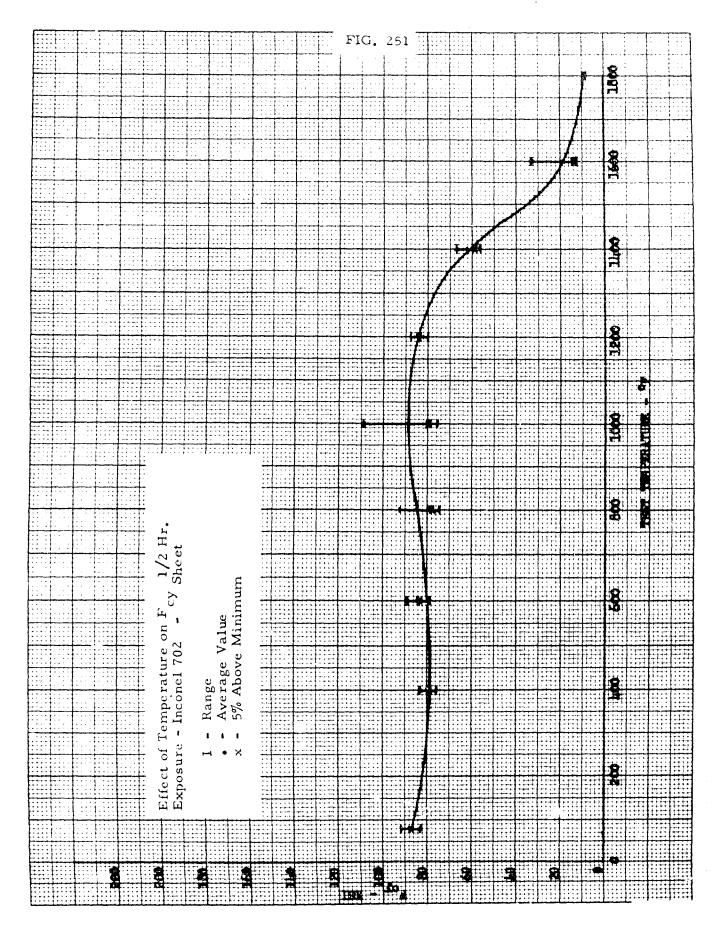


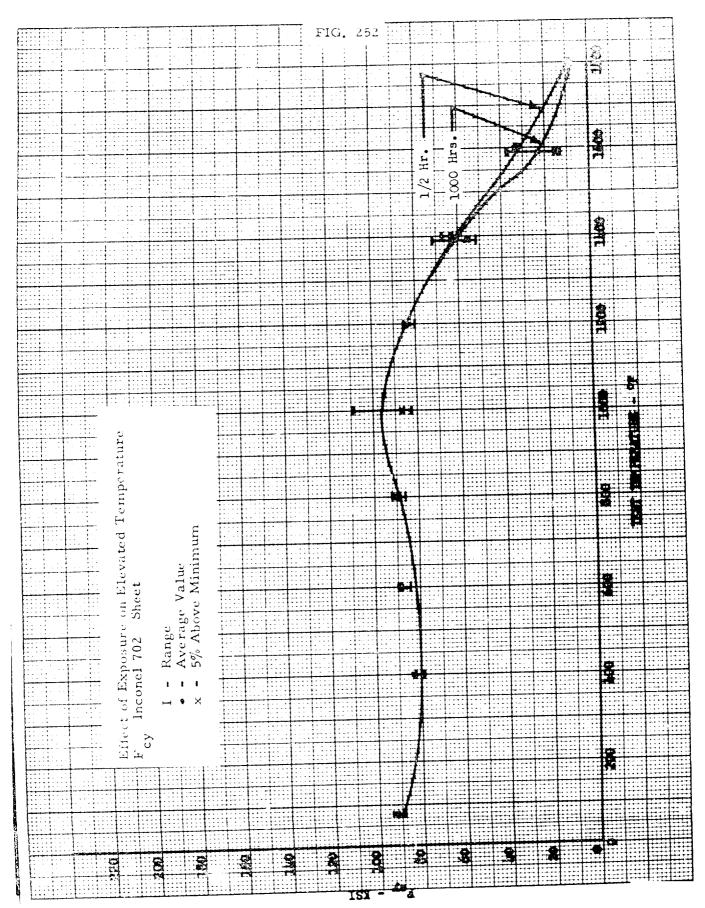
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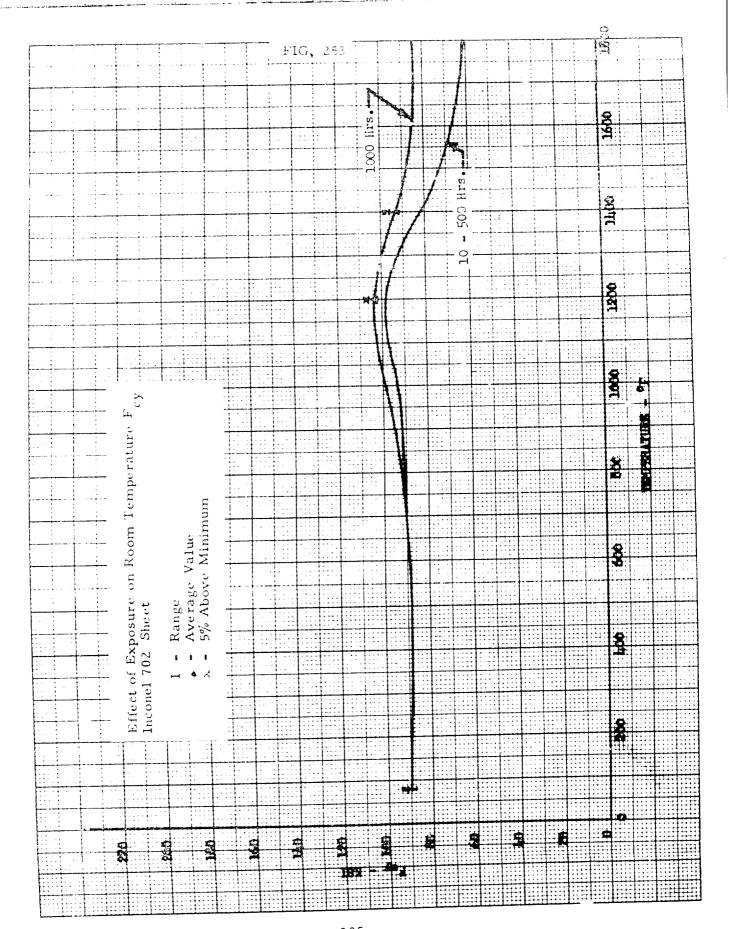




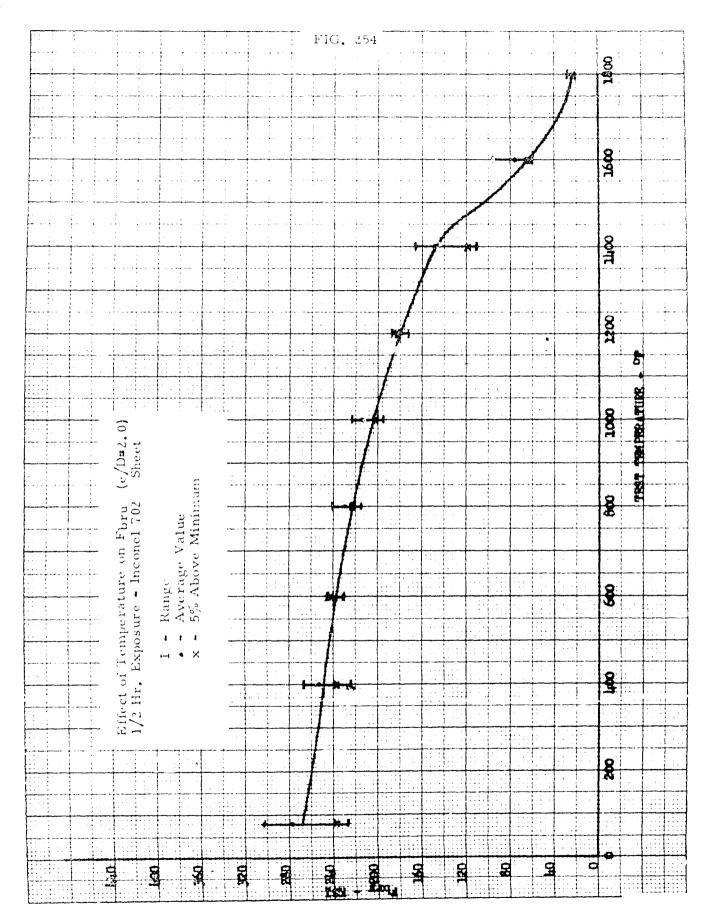
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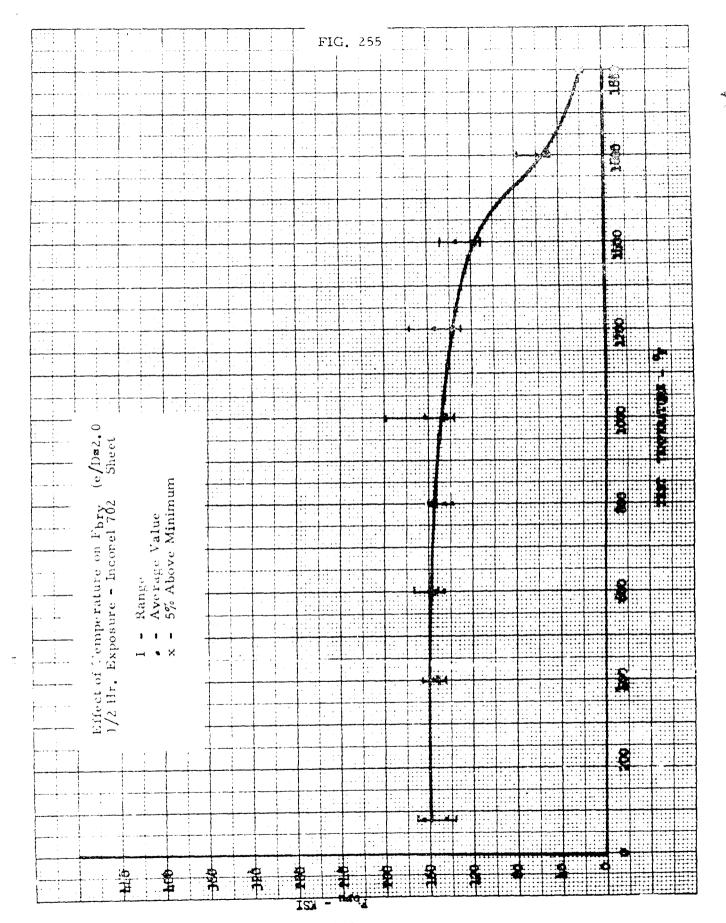


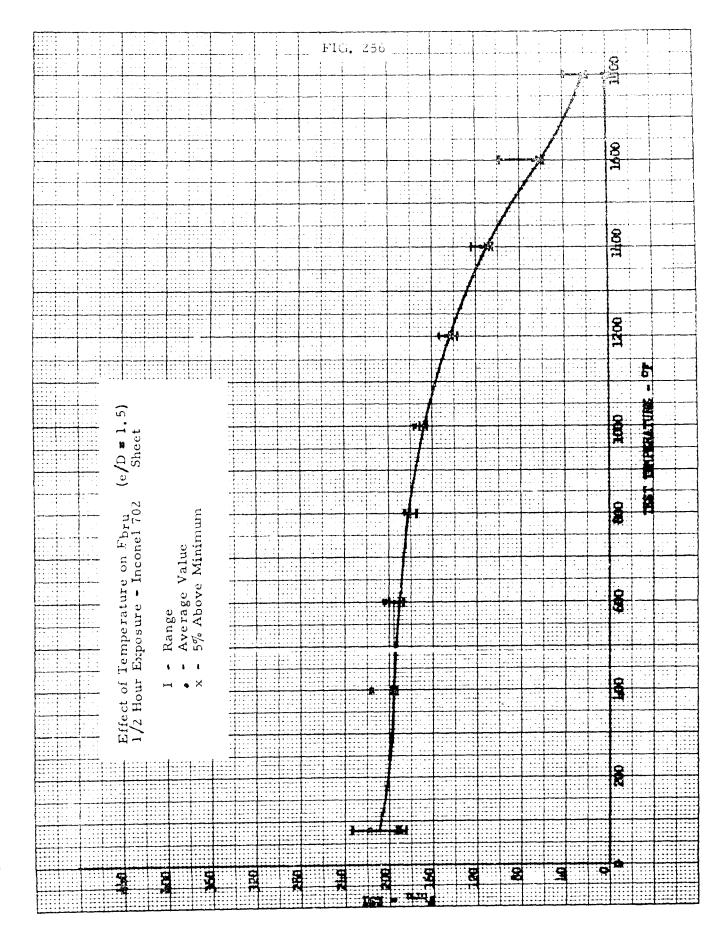


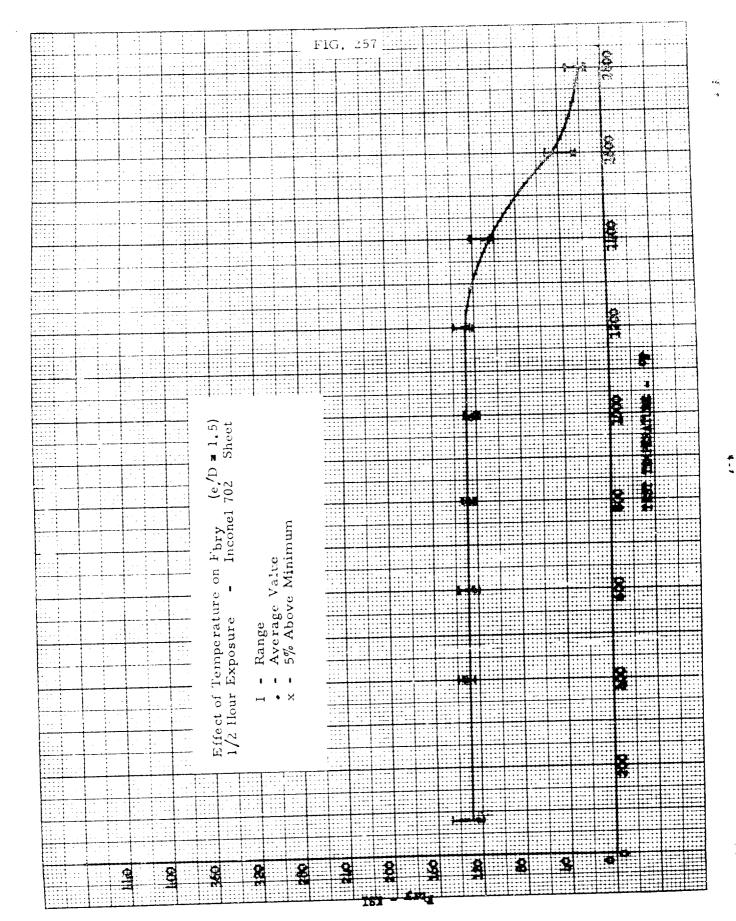


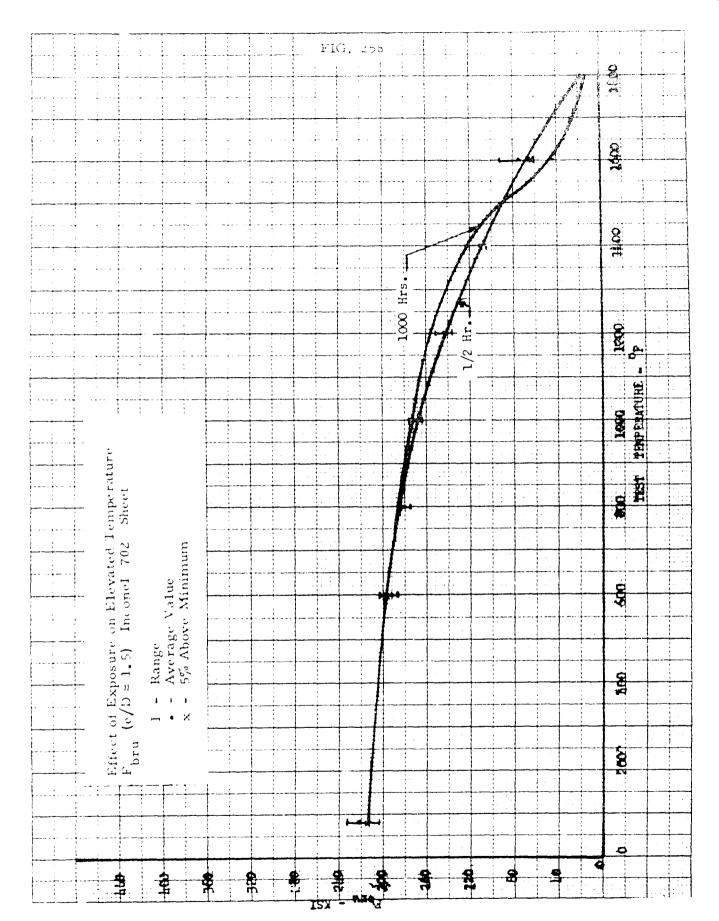
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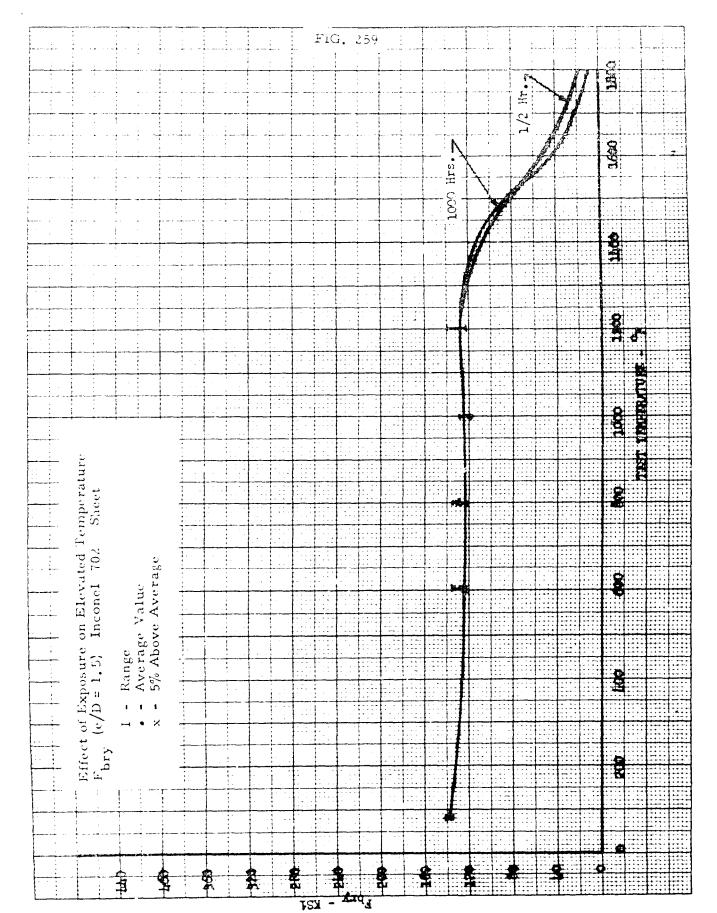


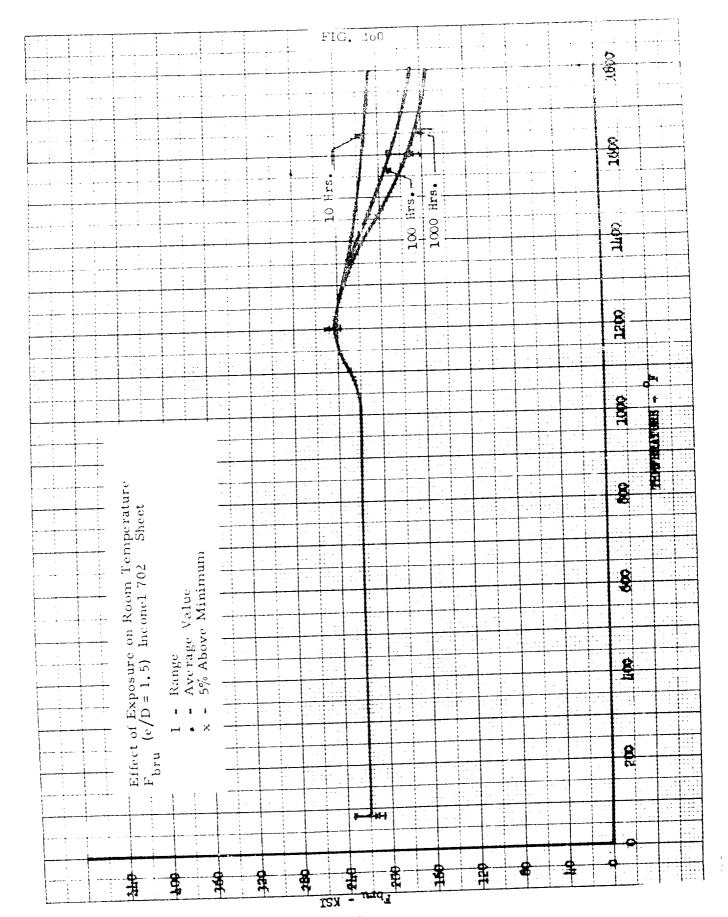


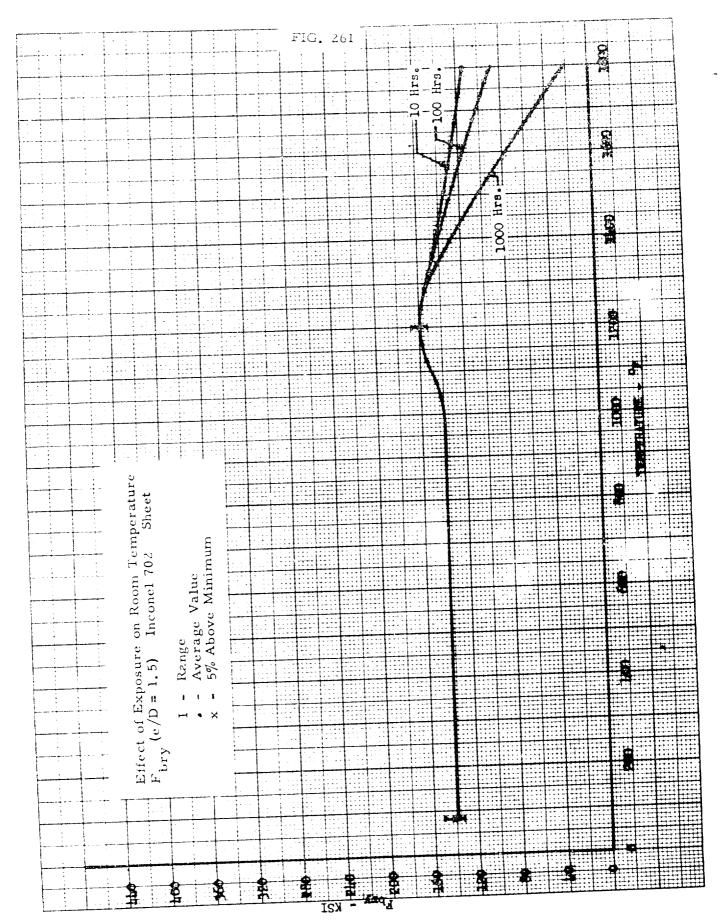




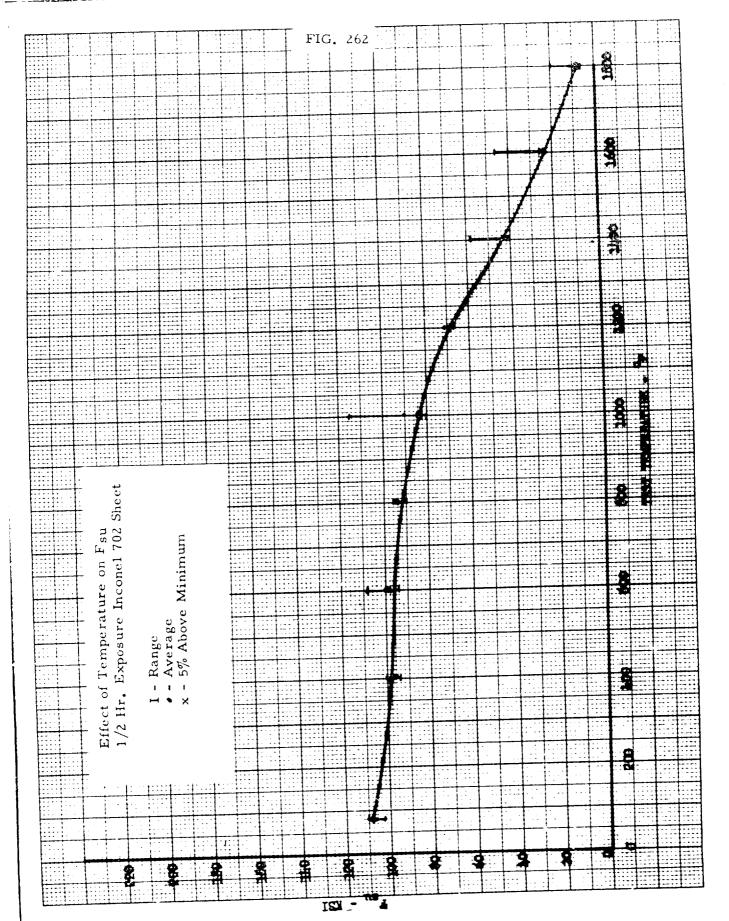


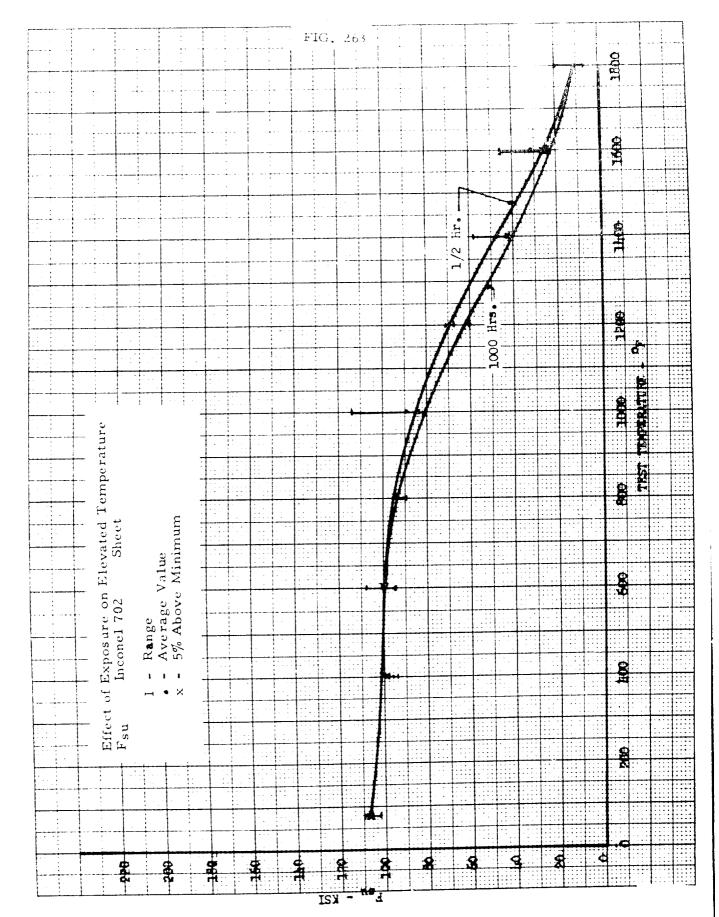


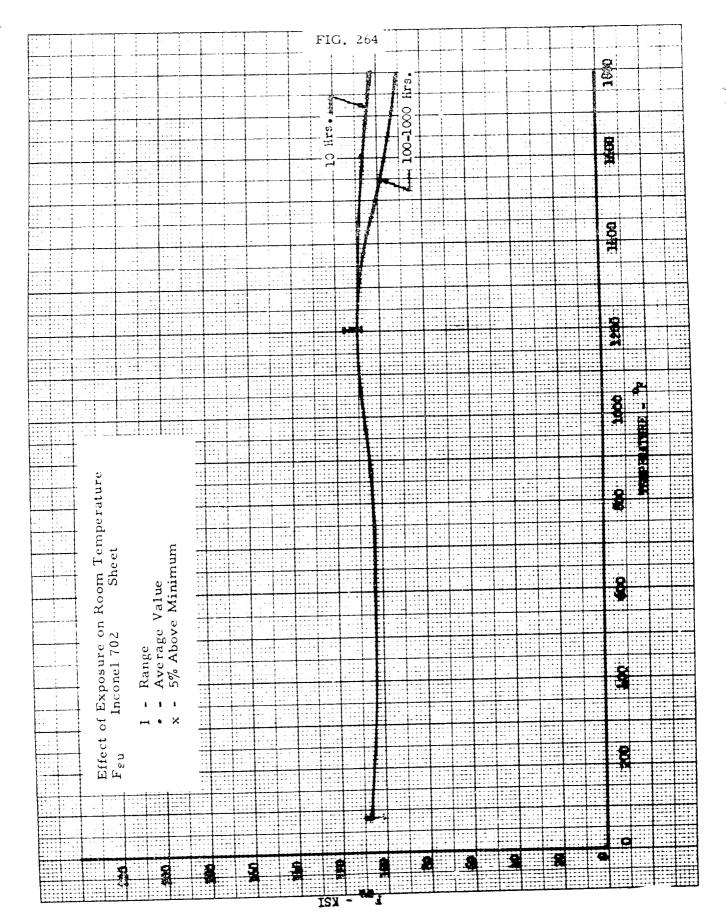




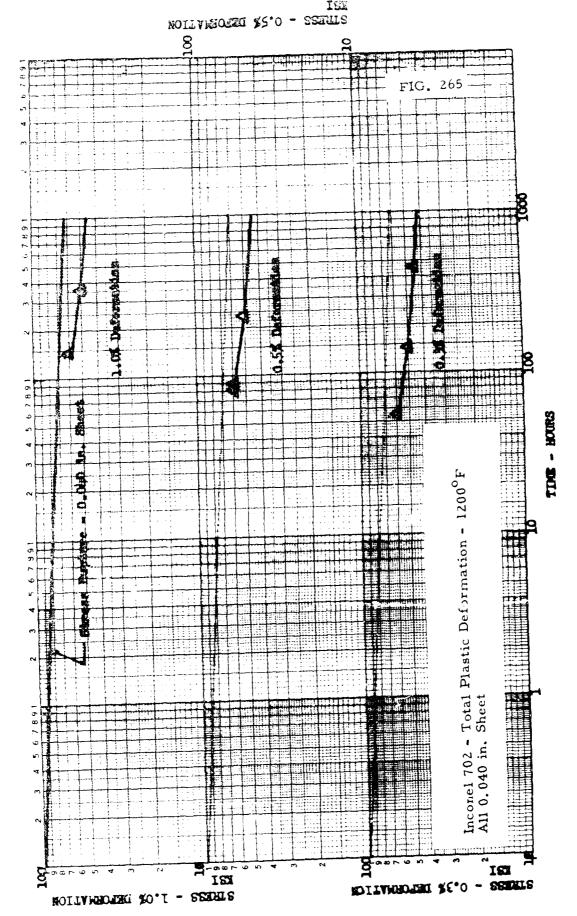
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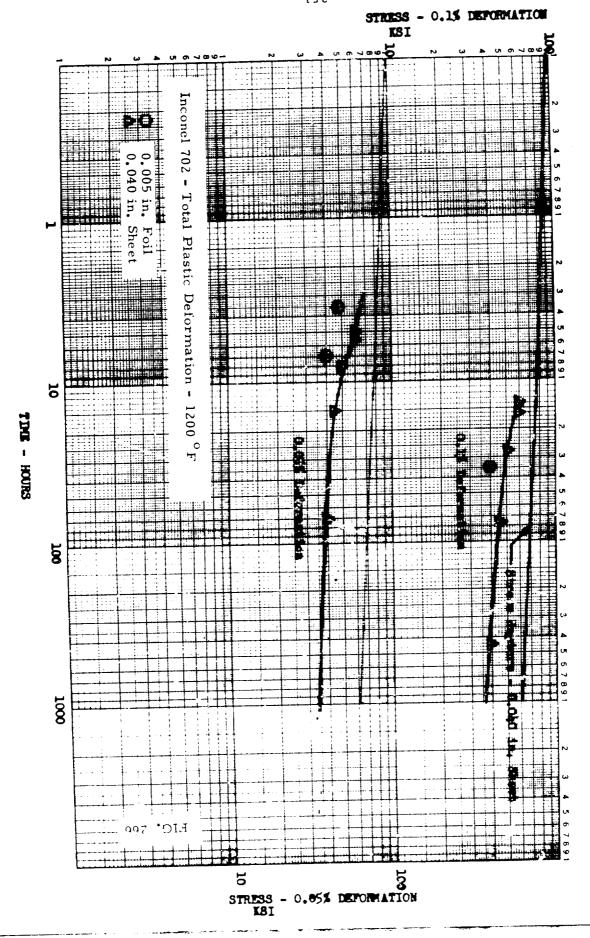


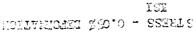


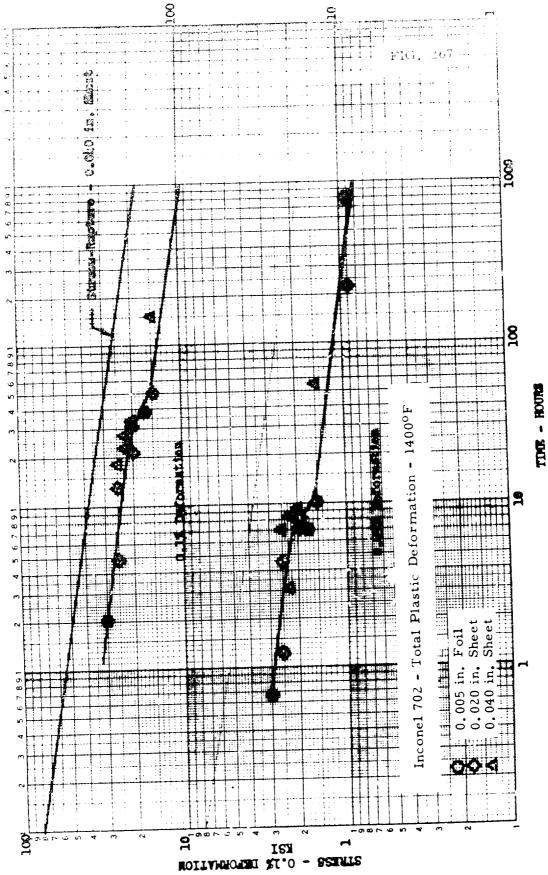


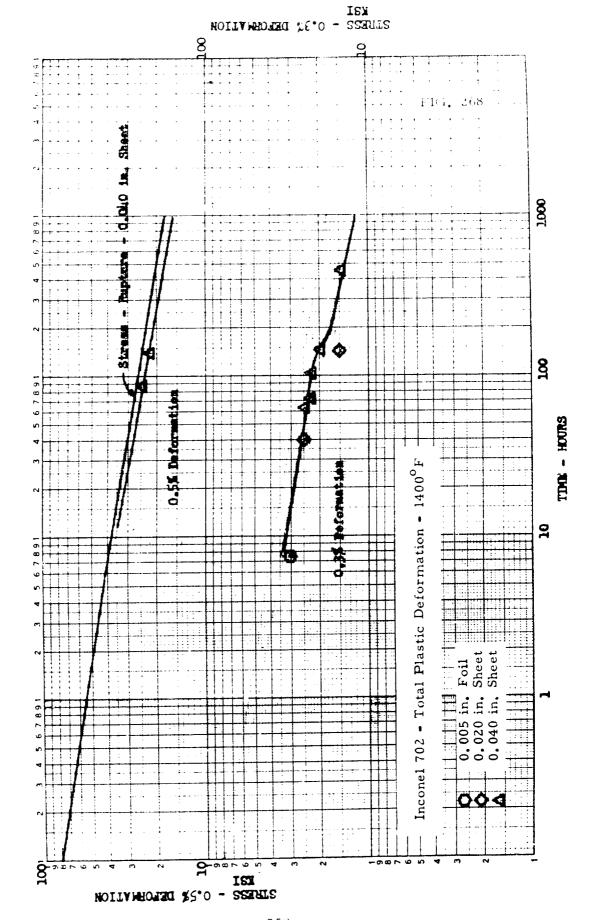
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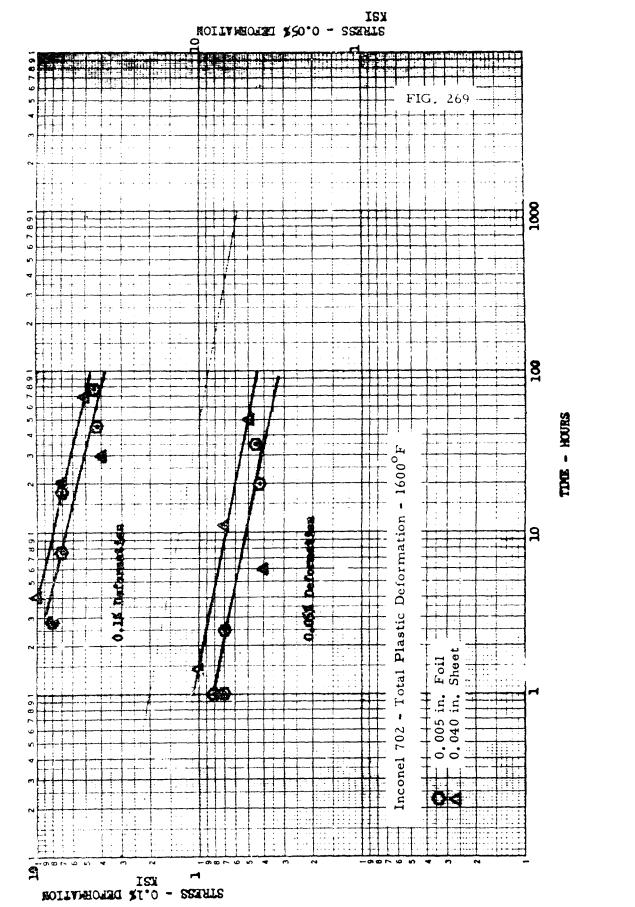


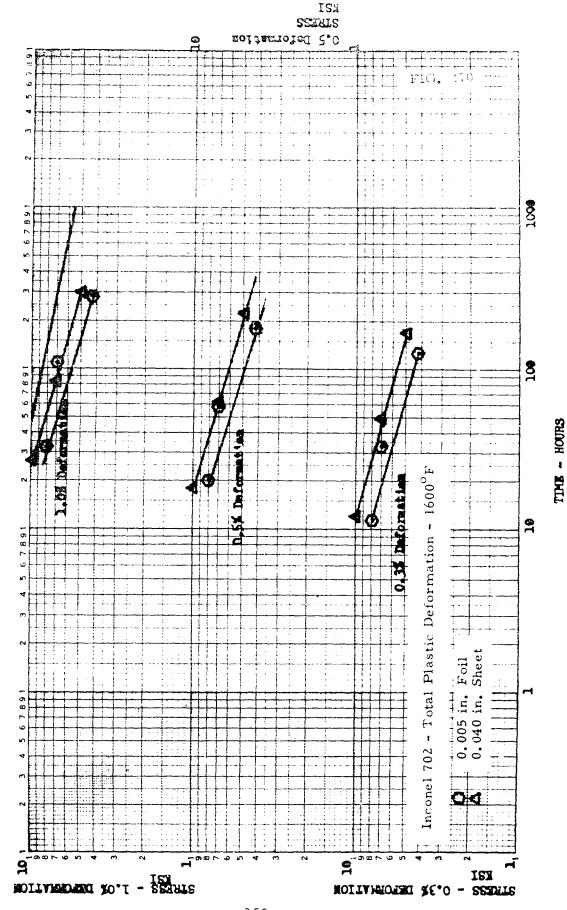


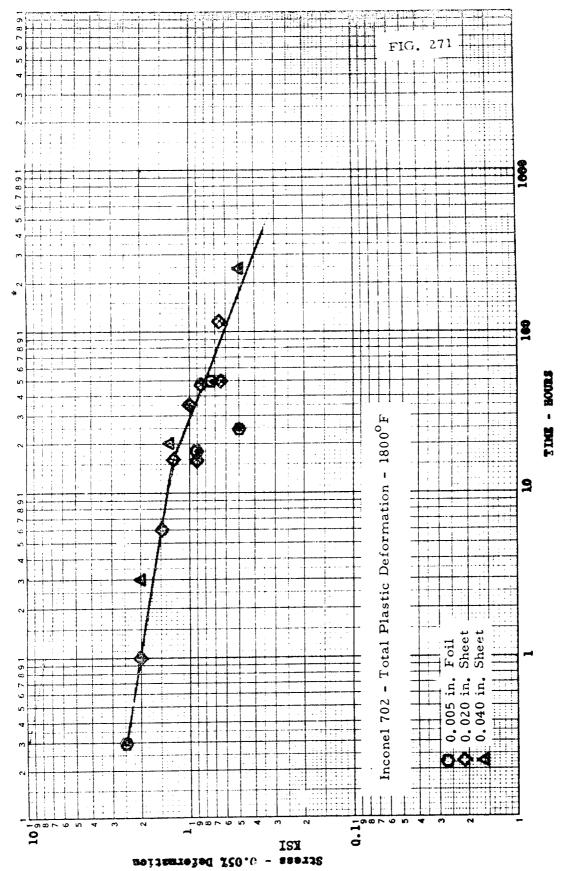


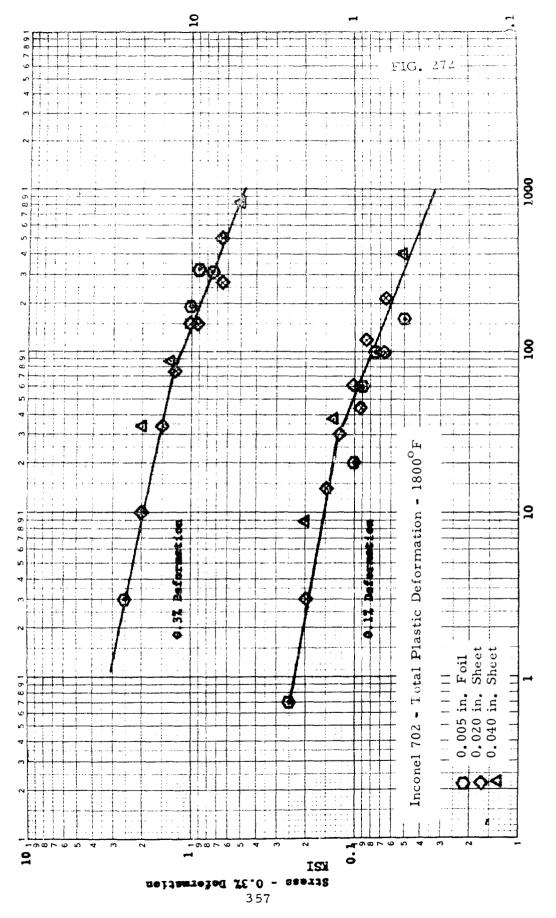






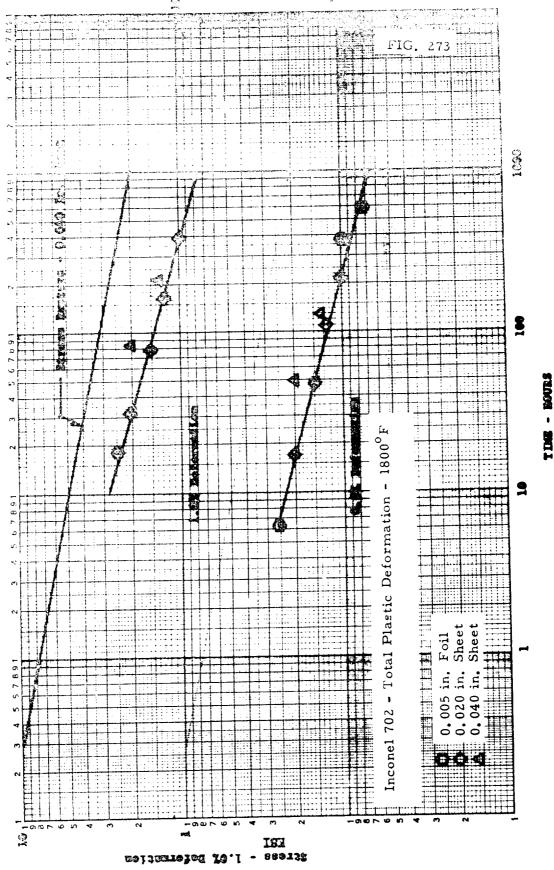






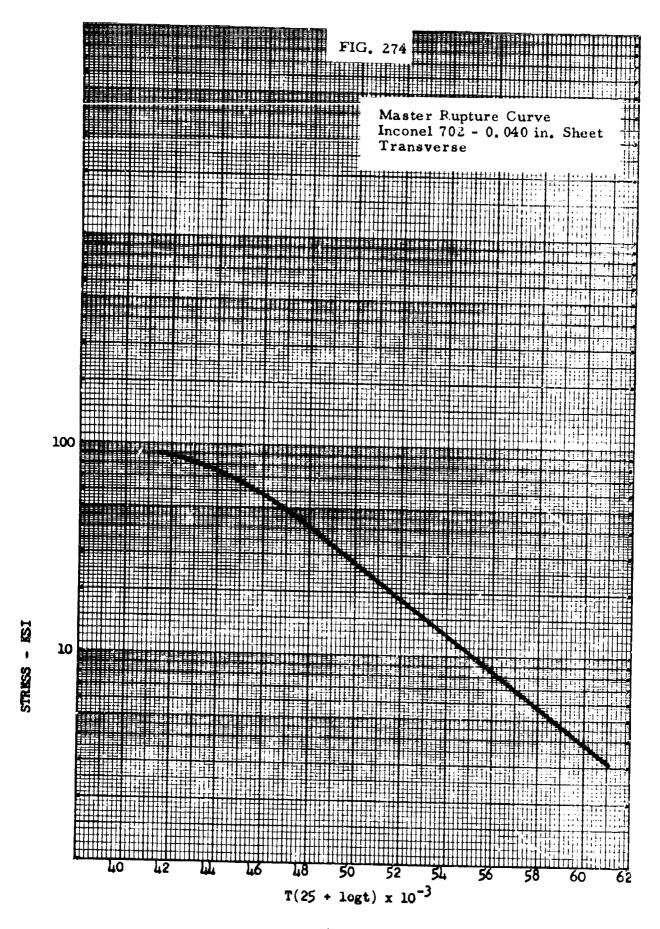
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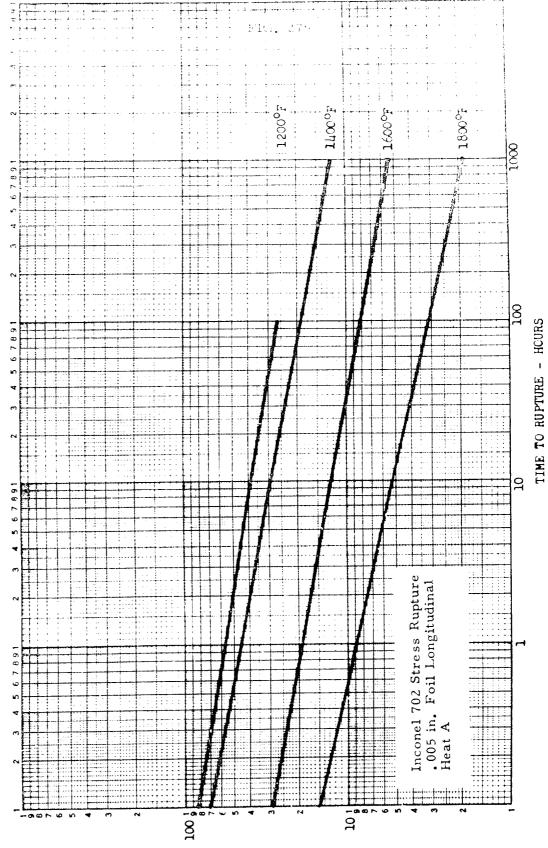




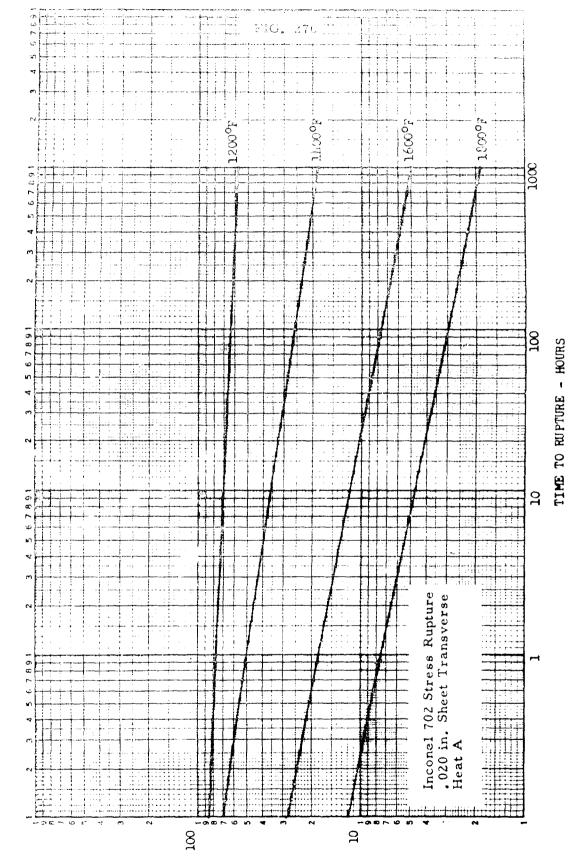
## SECTION VII

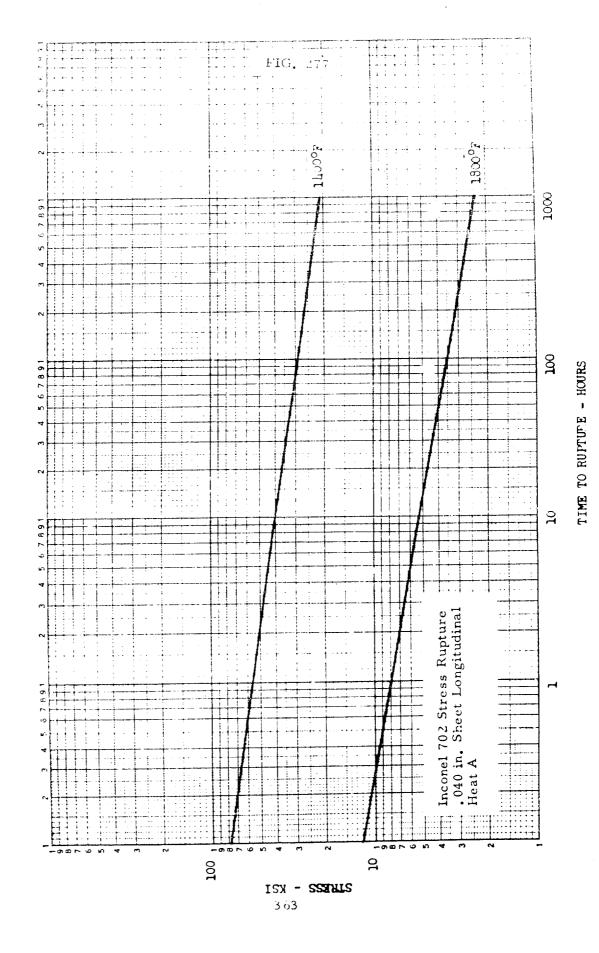
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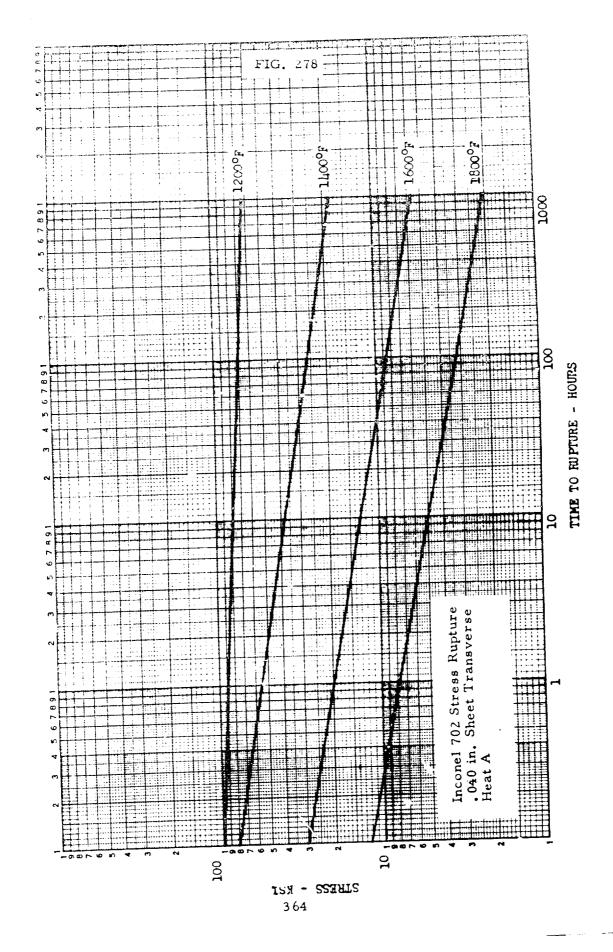




STRESS - KSI



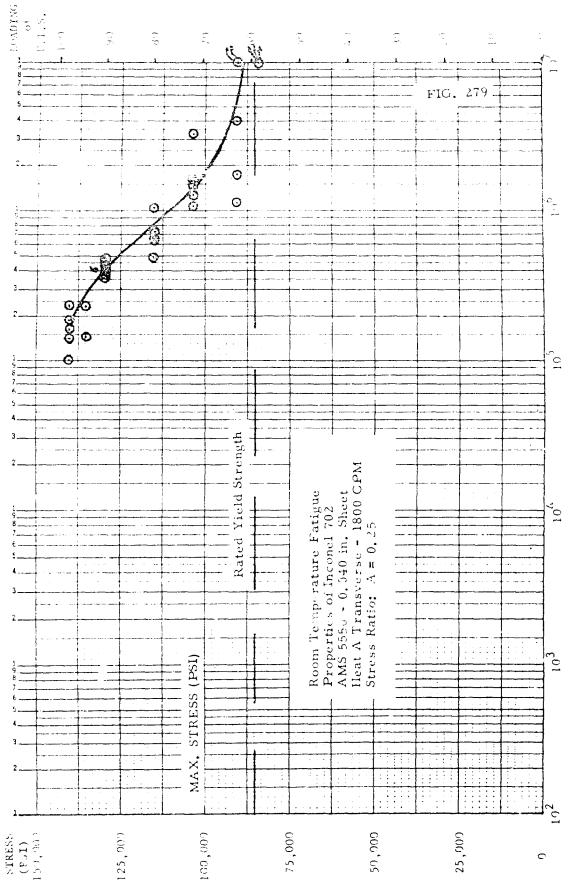




## SECTION VII

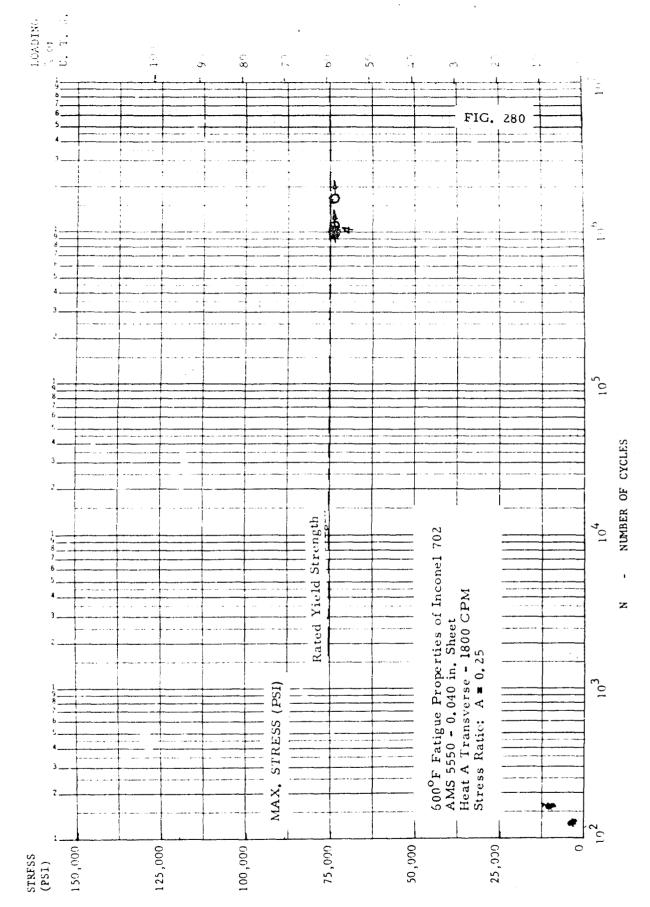
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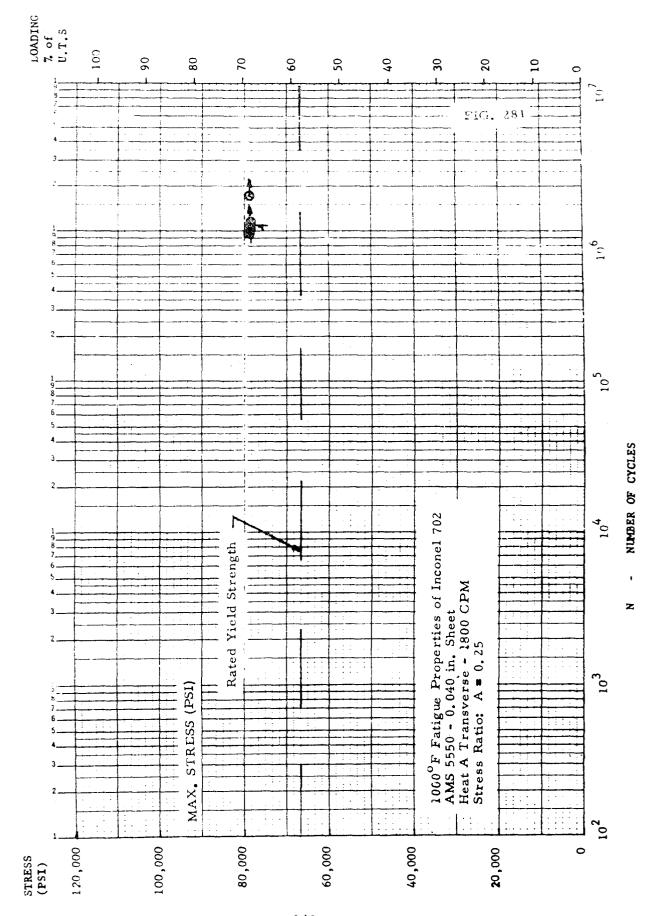
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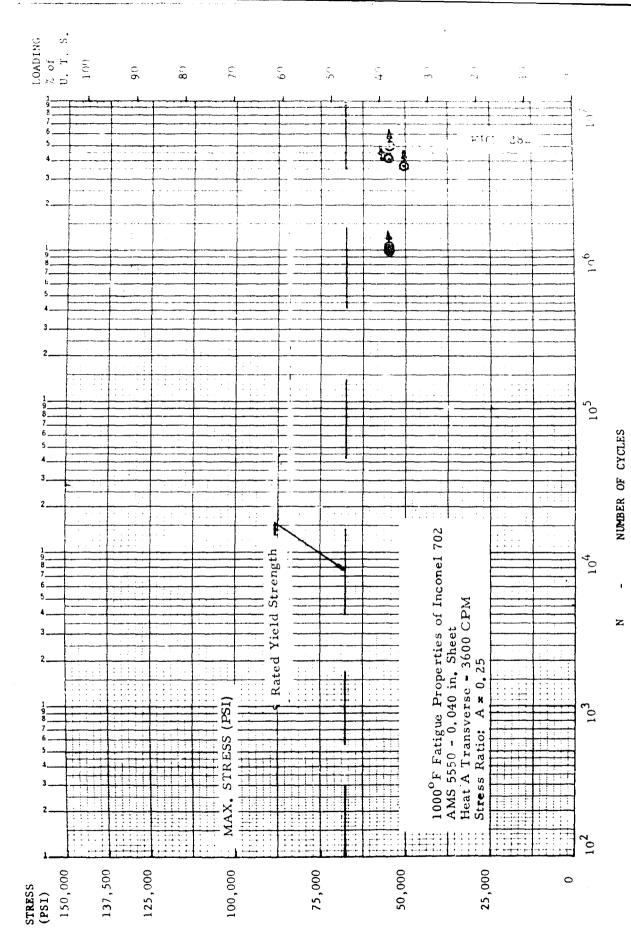


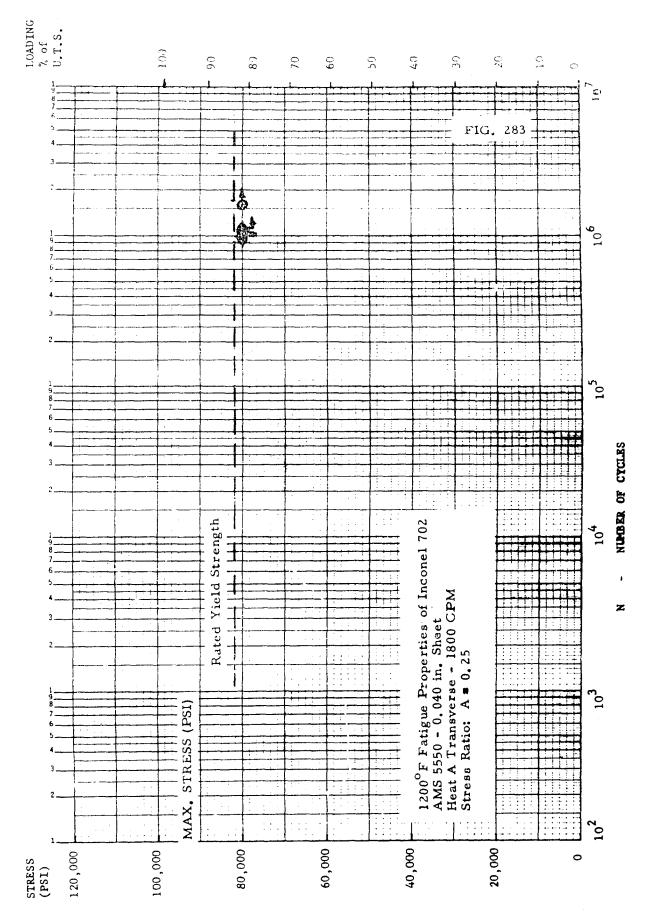
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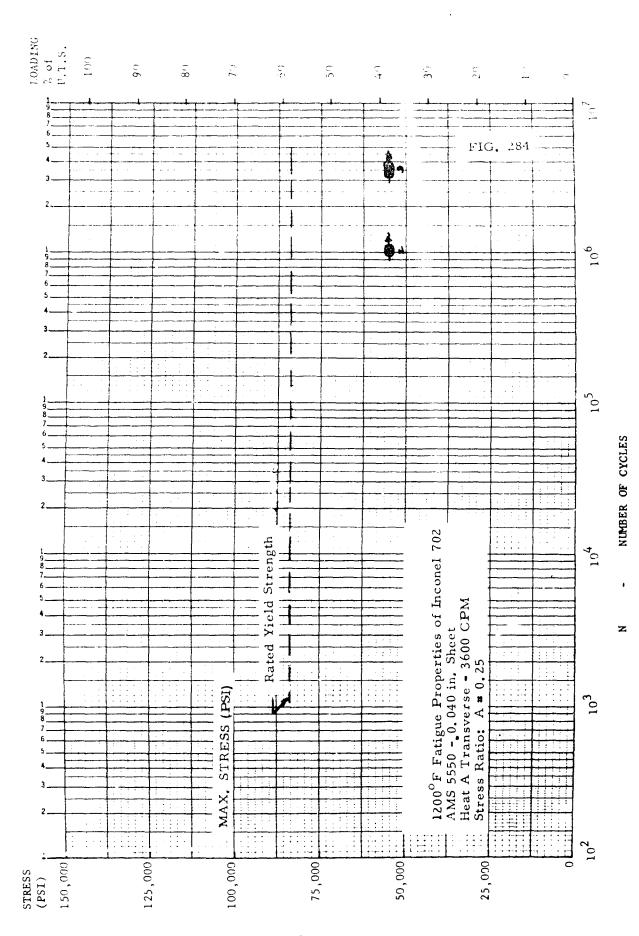
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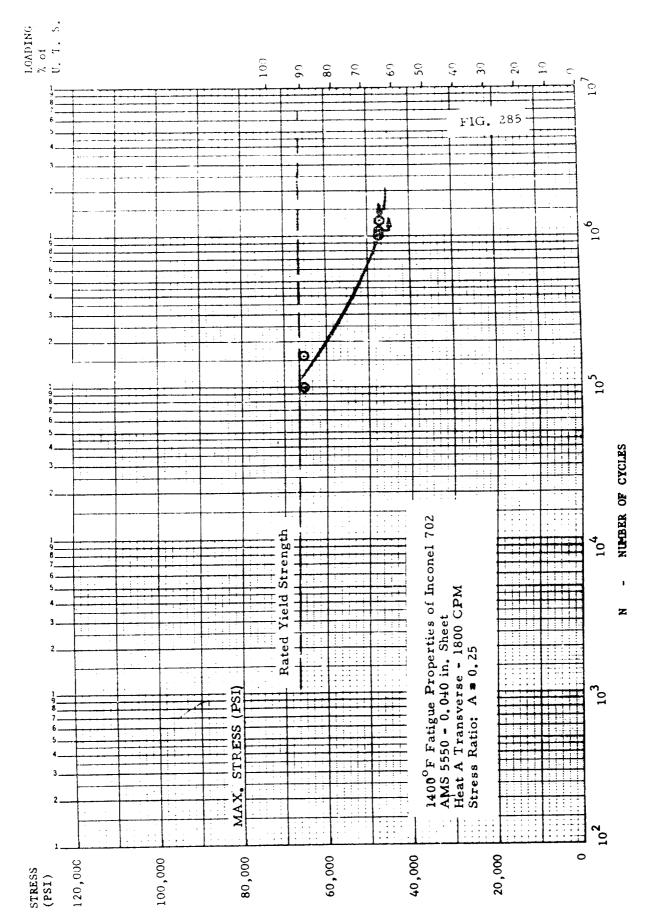


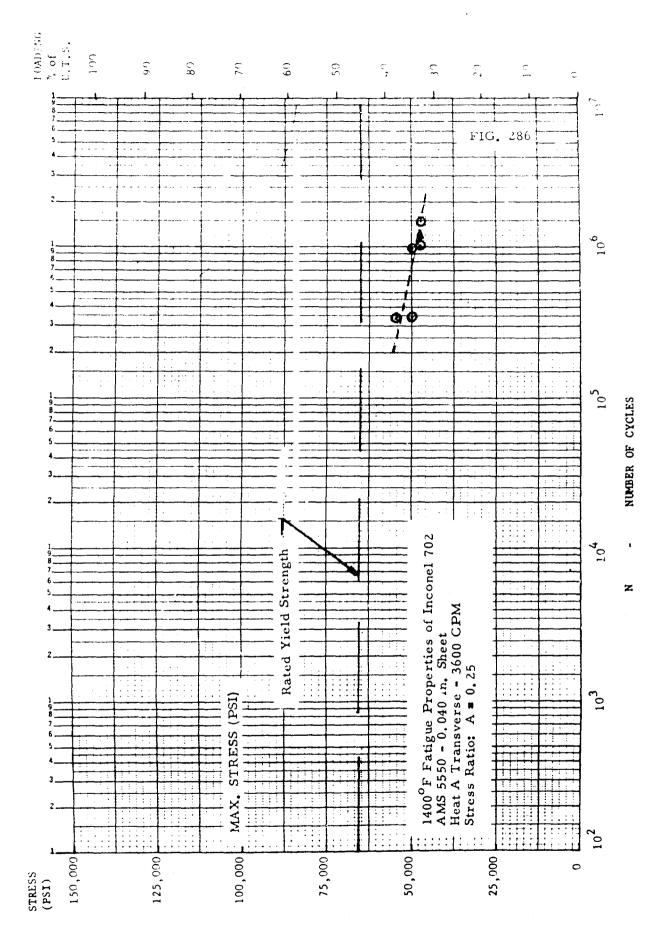


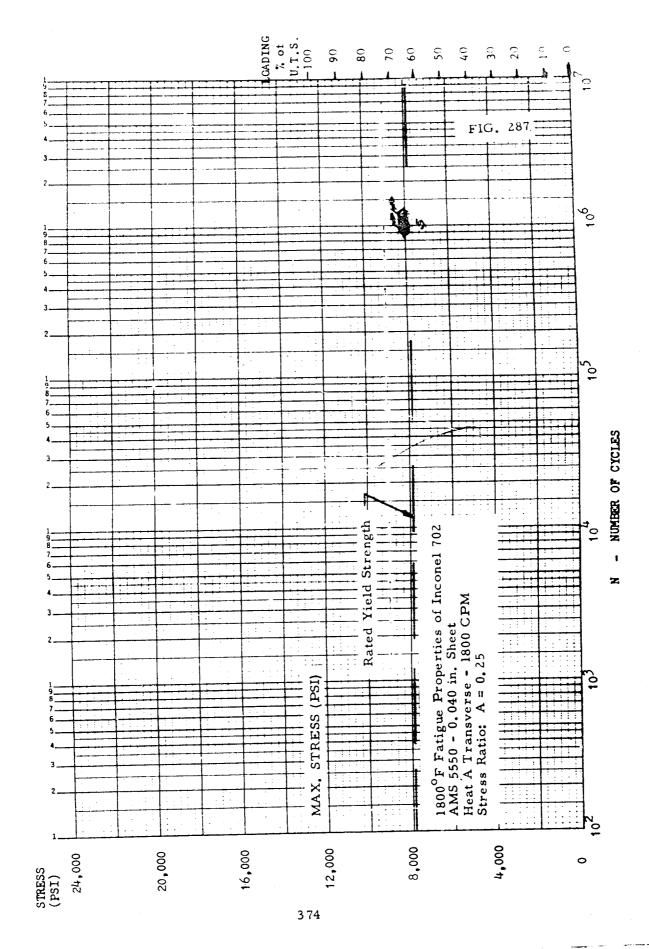


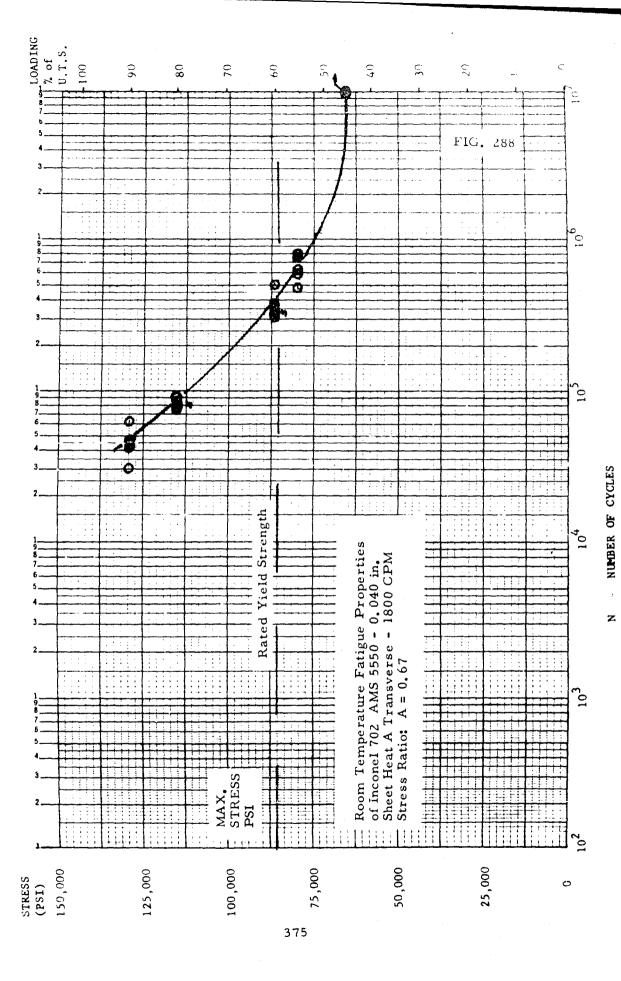


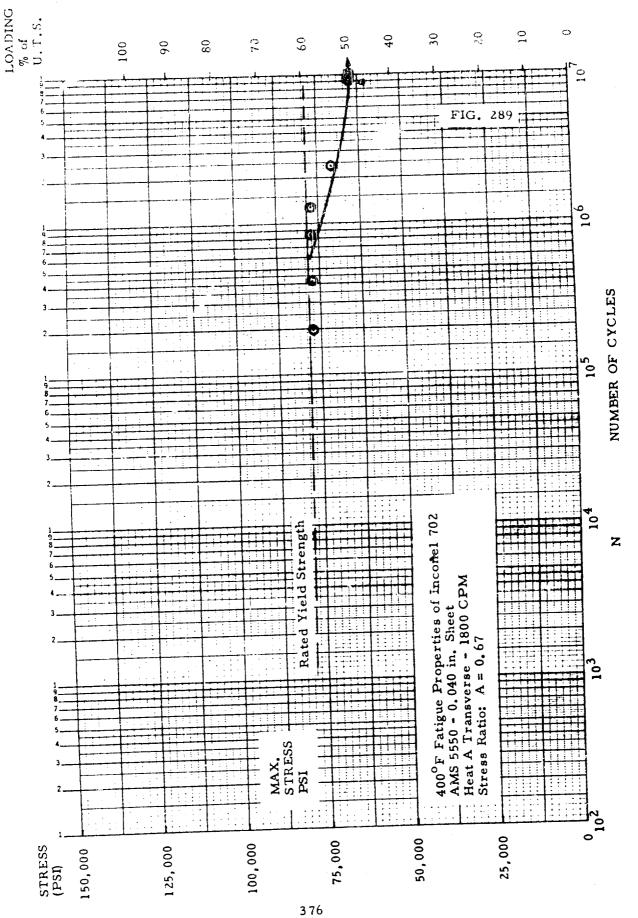


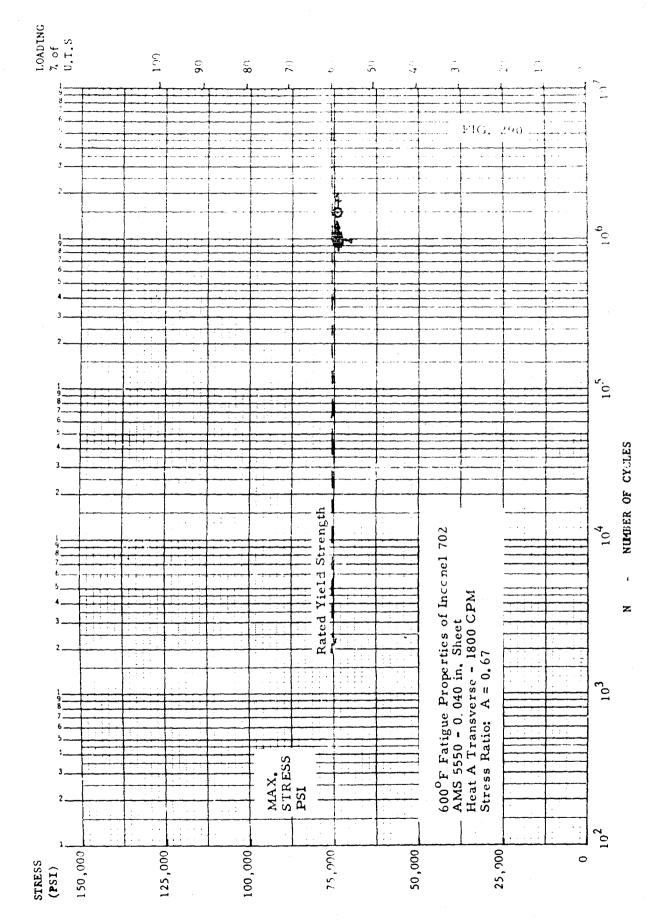


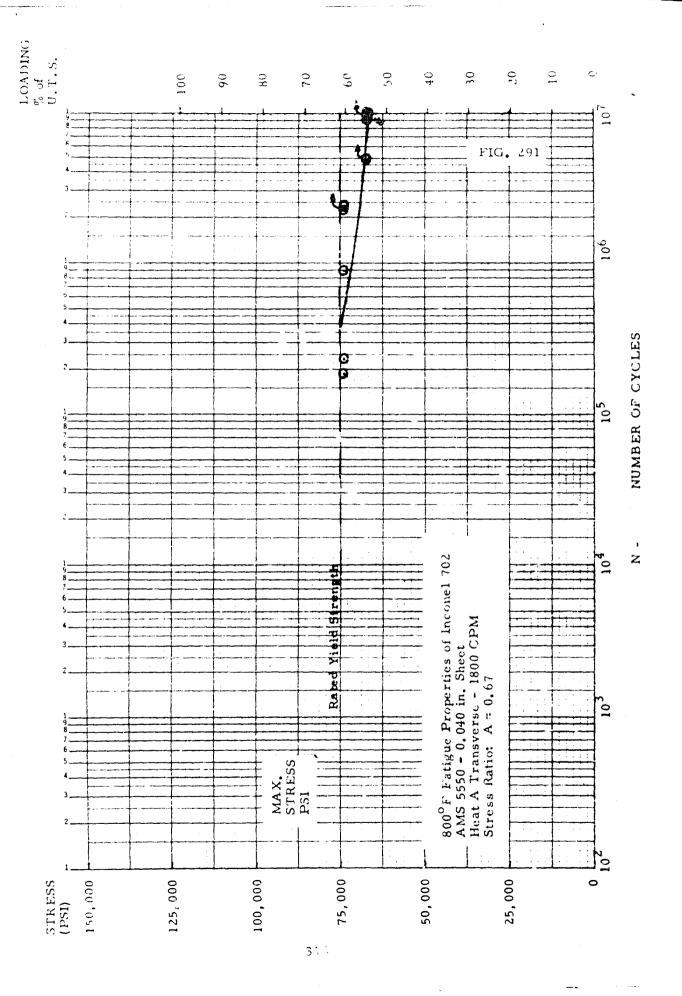


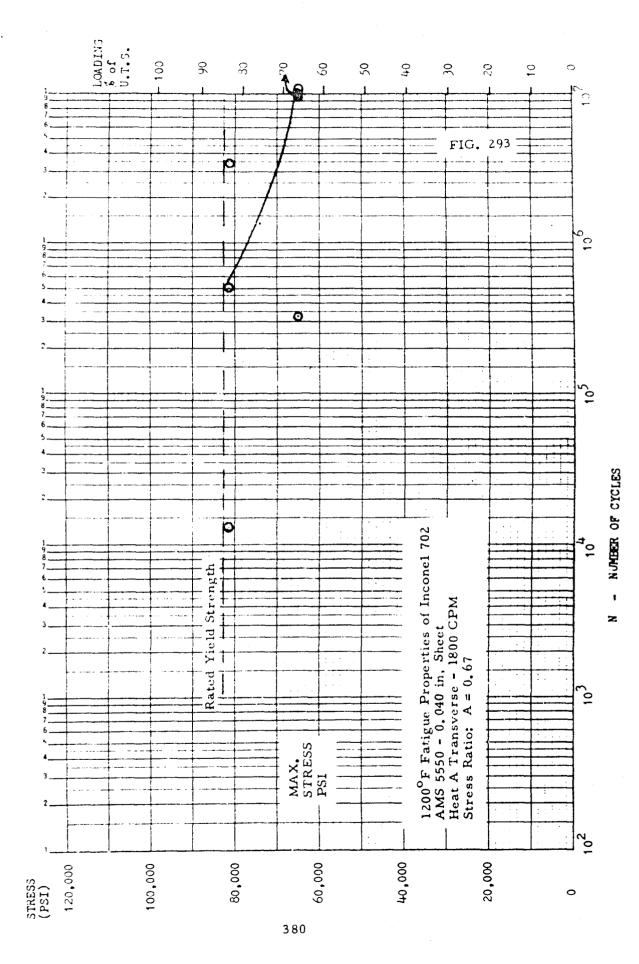


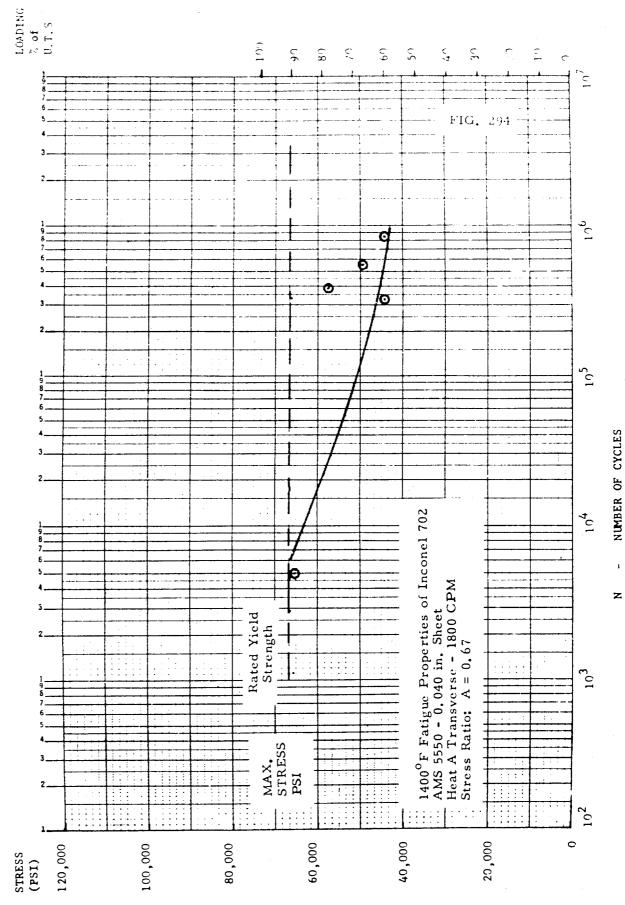




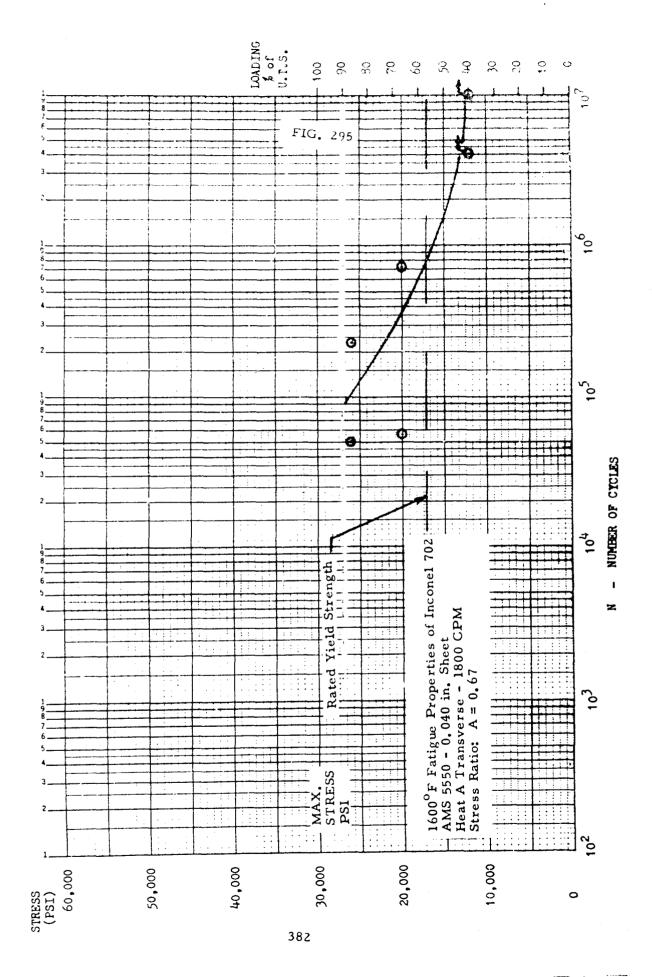


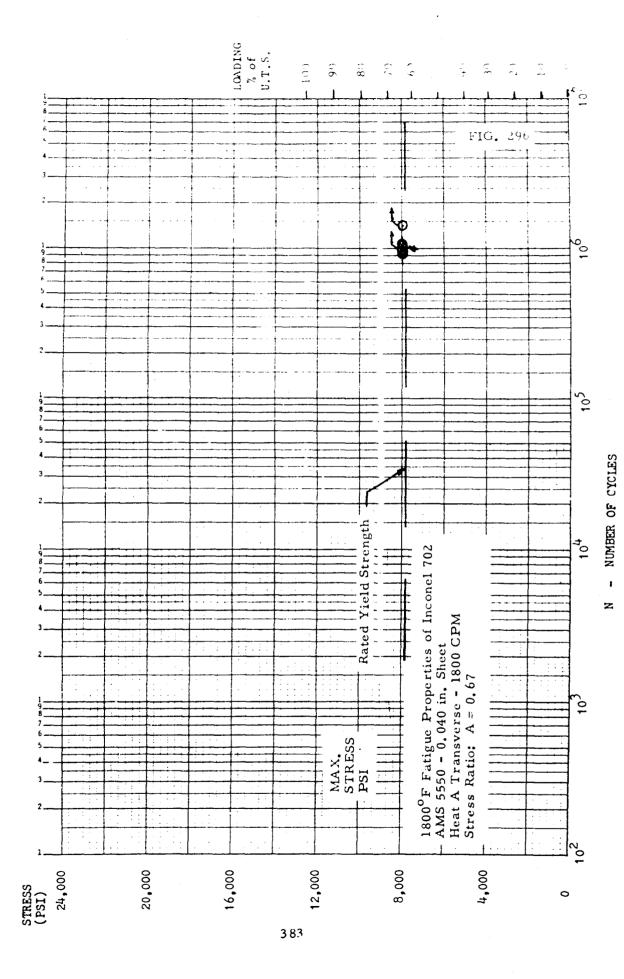


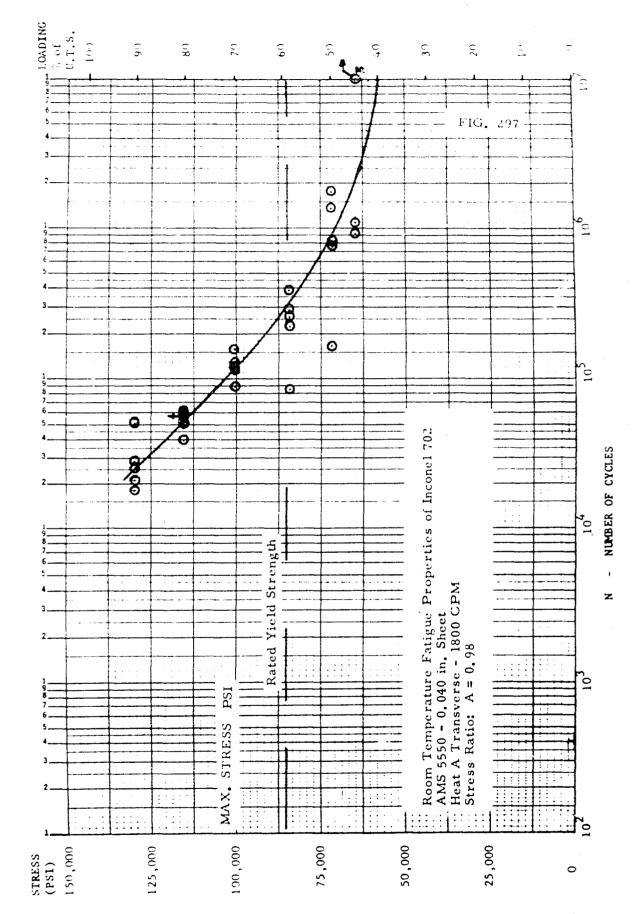


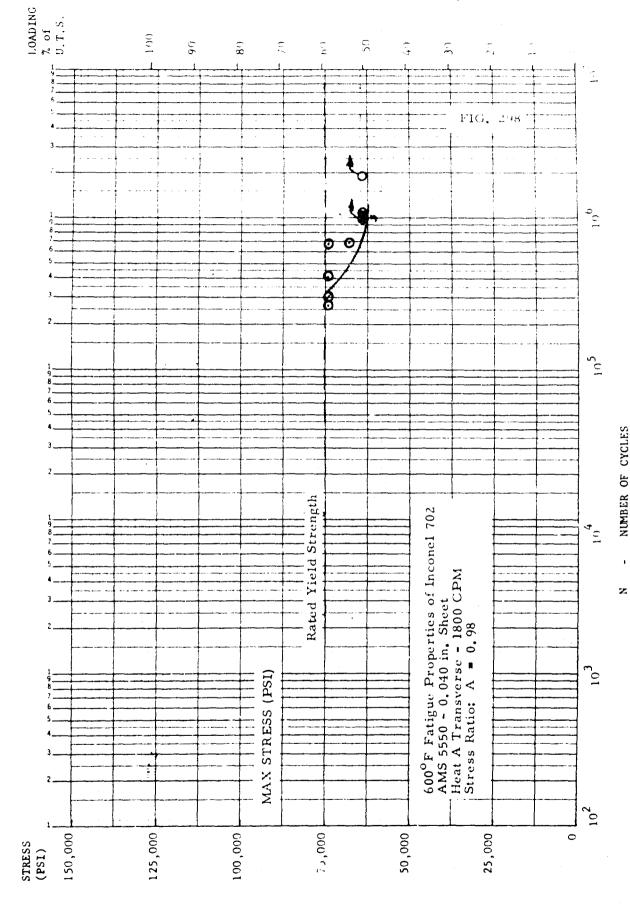


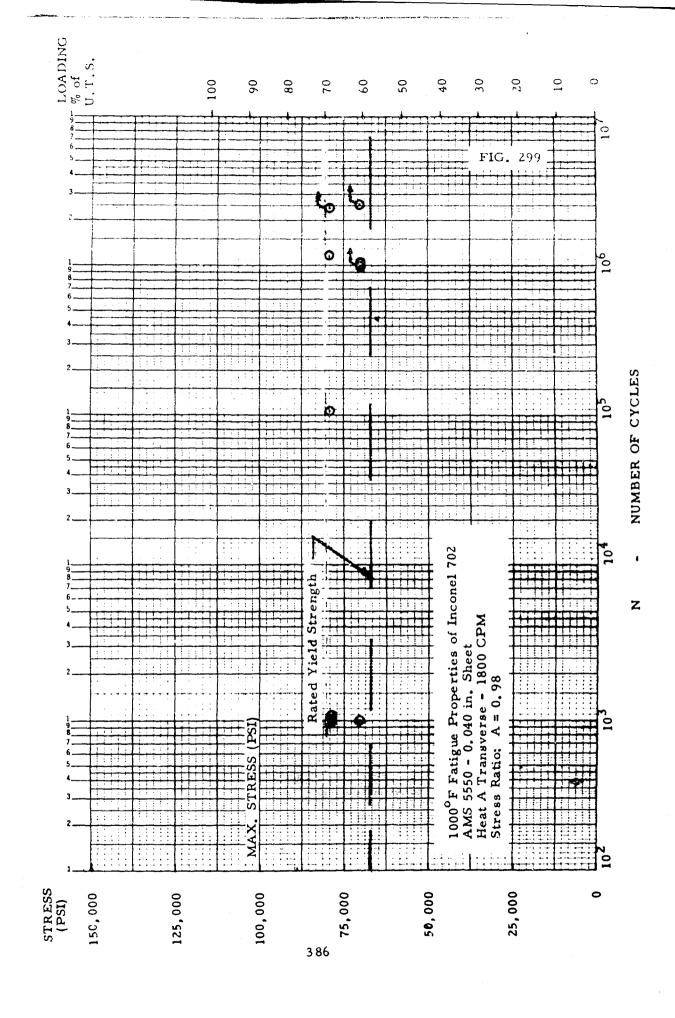
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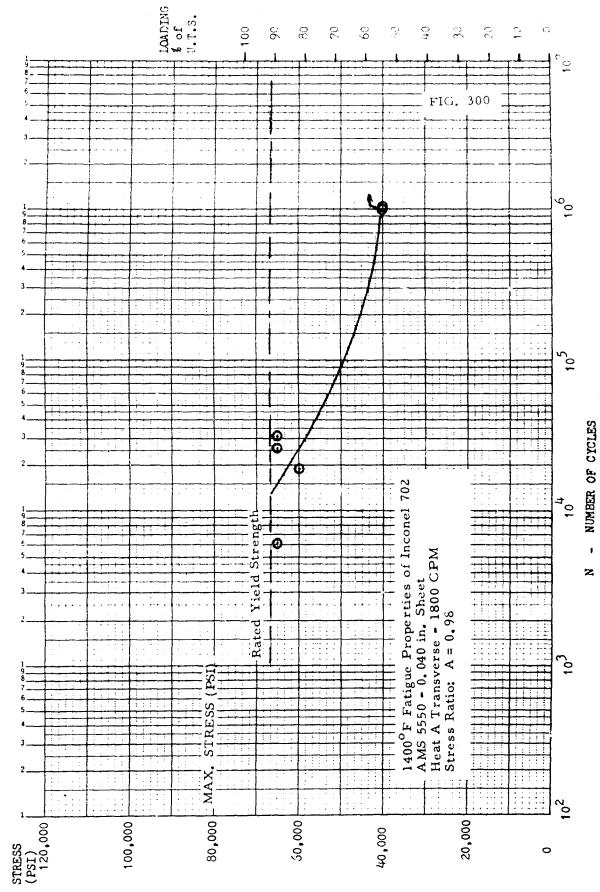


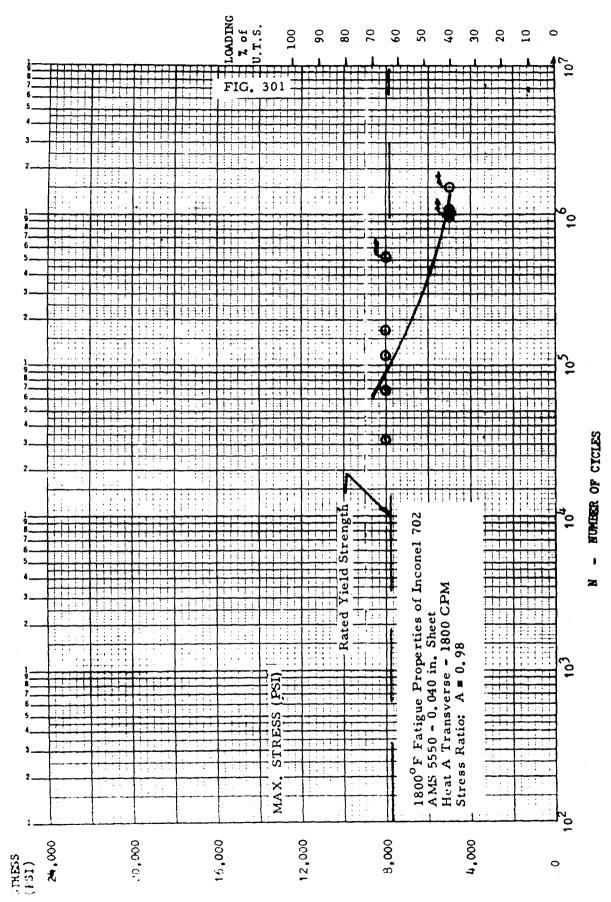


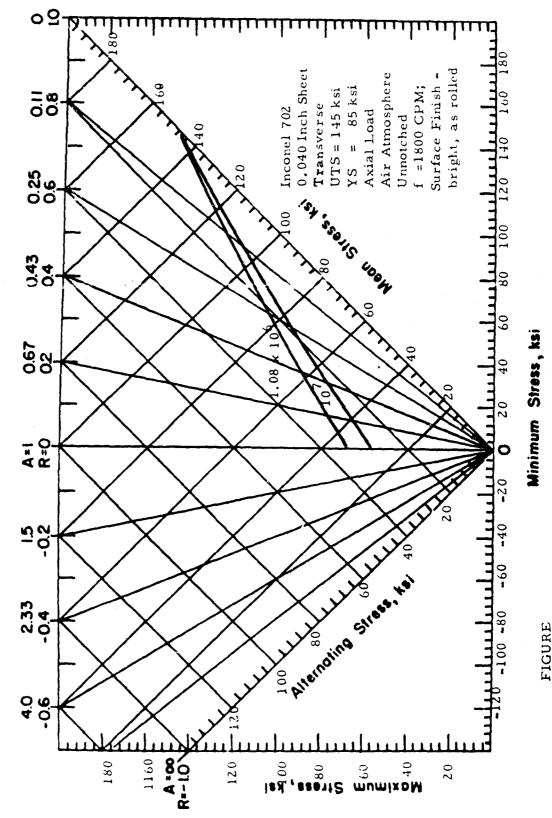








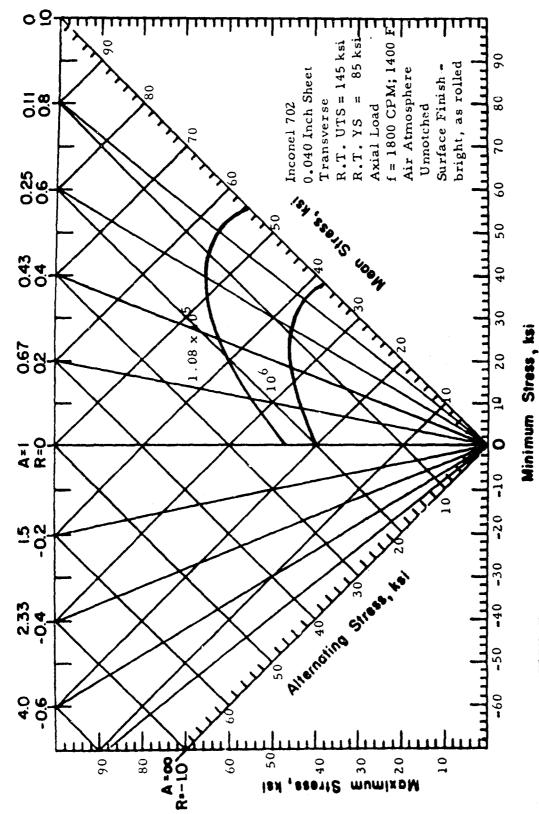




TYPICAL CONSTANT LIFE DIAGRAM FOR FATIGUE BEHAVIOR OF

INCONEL 702 SHEET MATERIAL AT ROOM TEMPERATURE.

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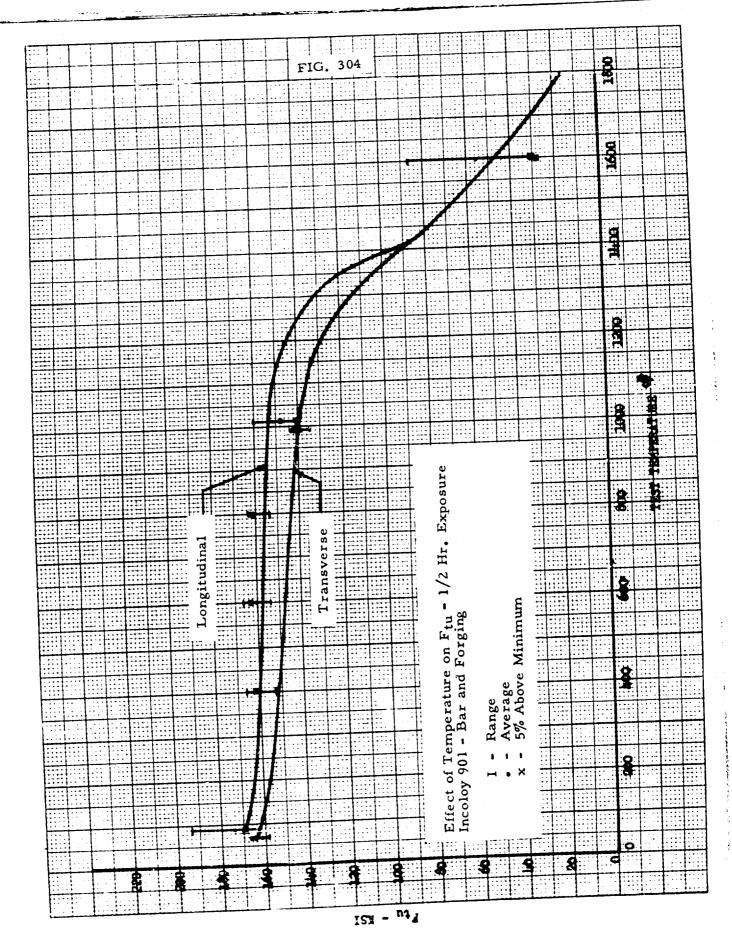
FIGURE
TYPICAL CONSTANT LIFE DIAGRAM OF FATIGUE BEHAVIOR OF
INCOLOY 702 AT 1400 F

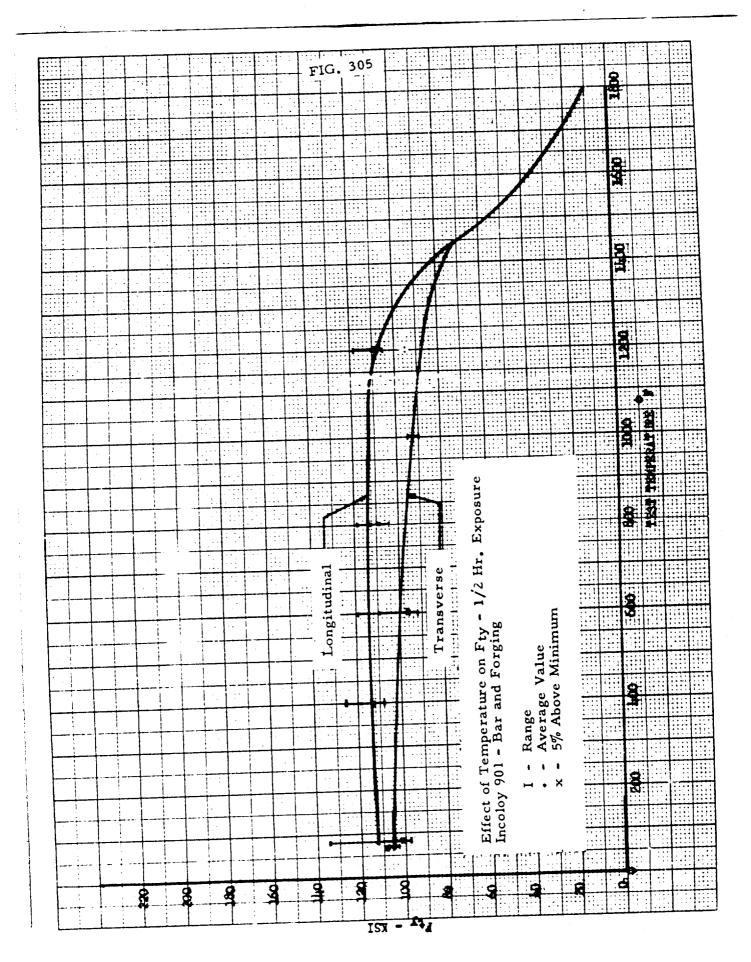
# SECTION 7.4 MATERIAL, INCOLOY 901

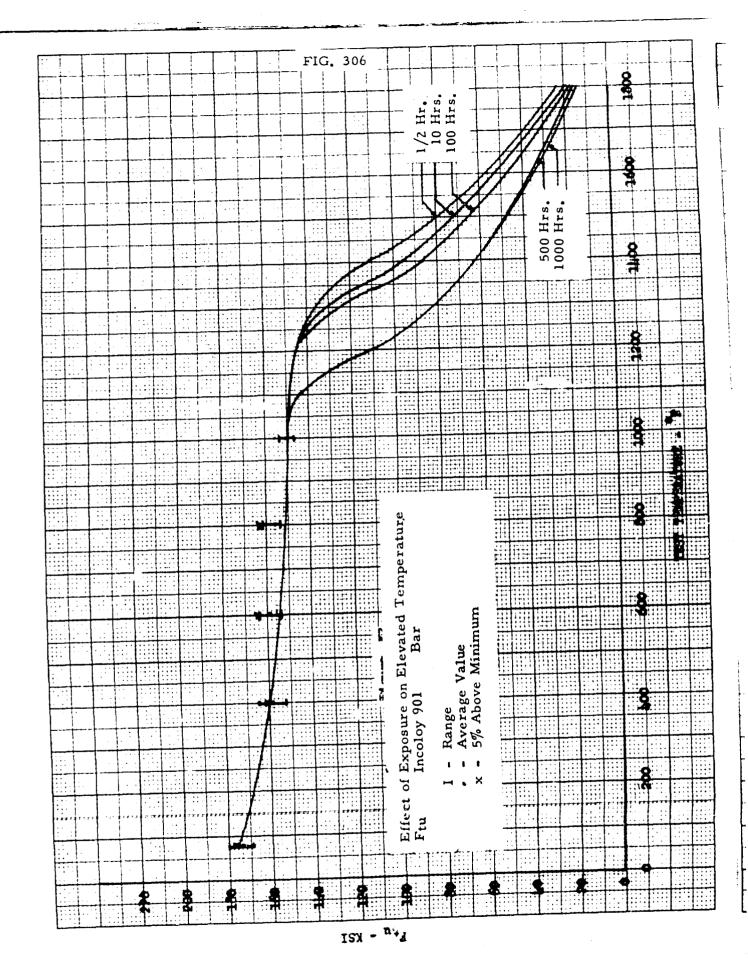
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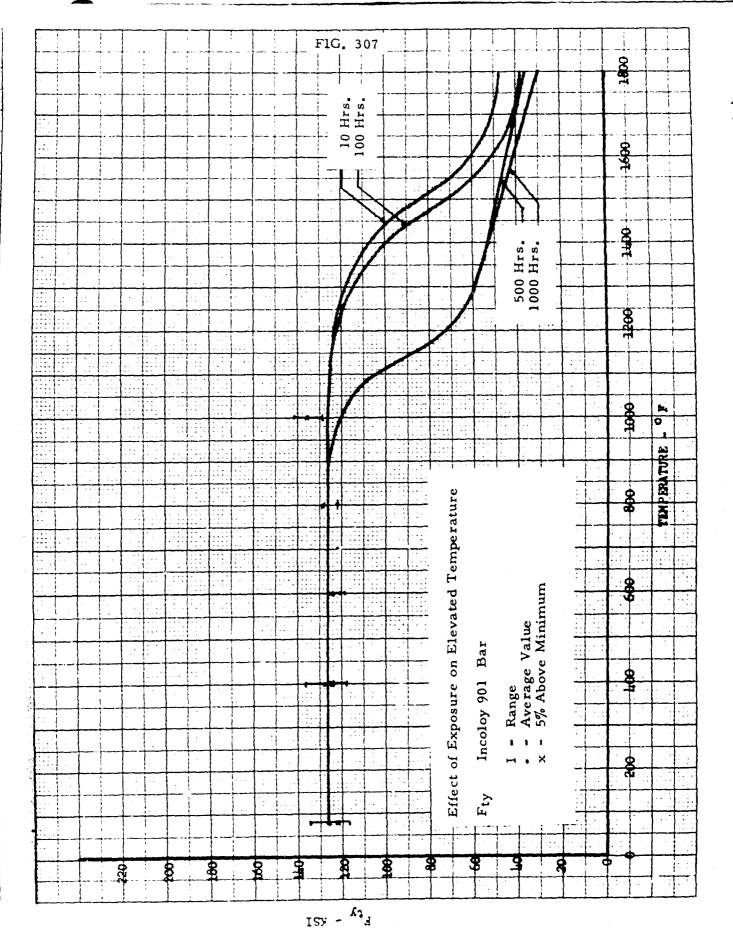
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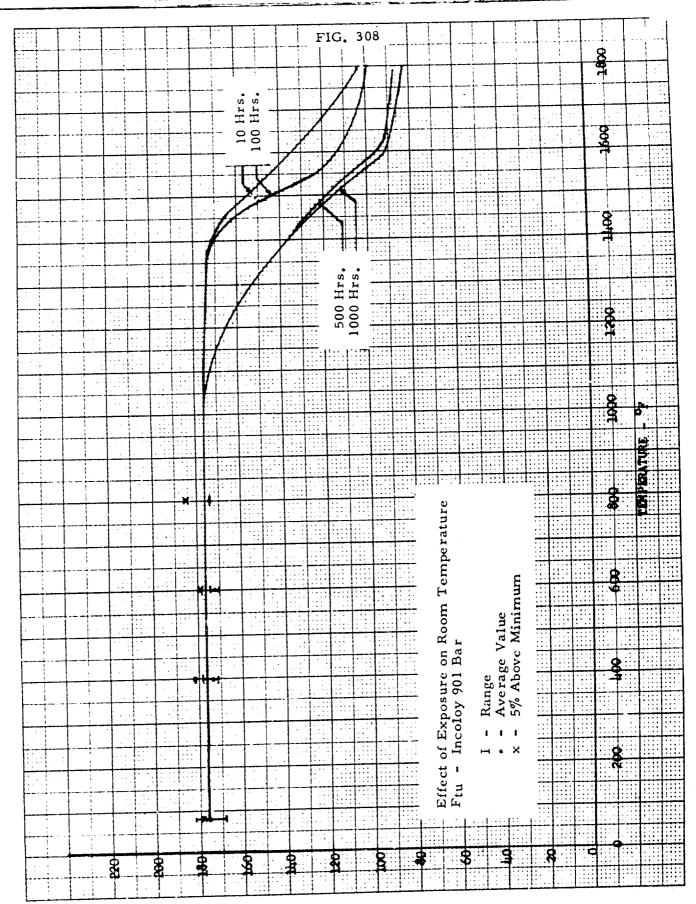




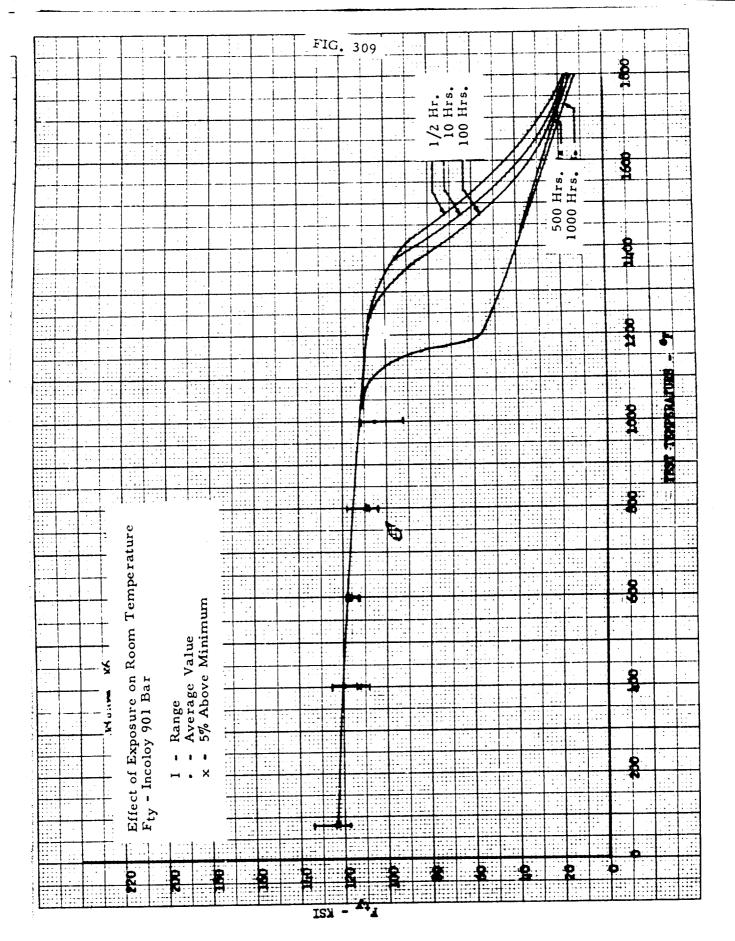


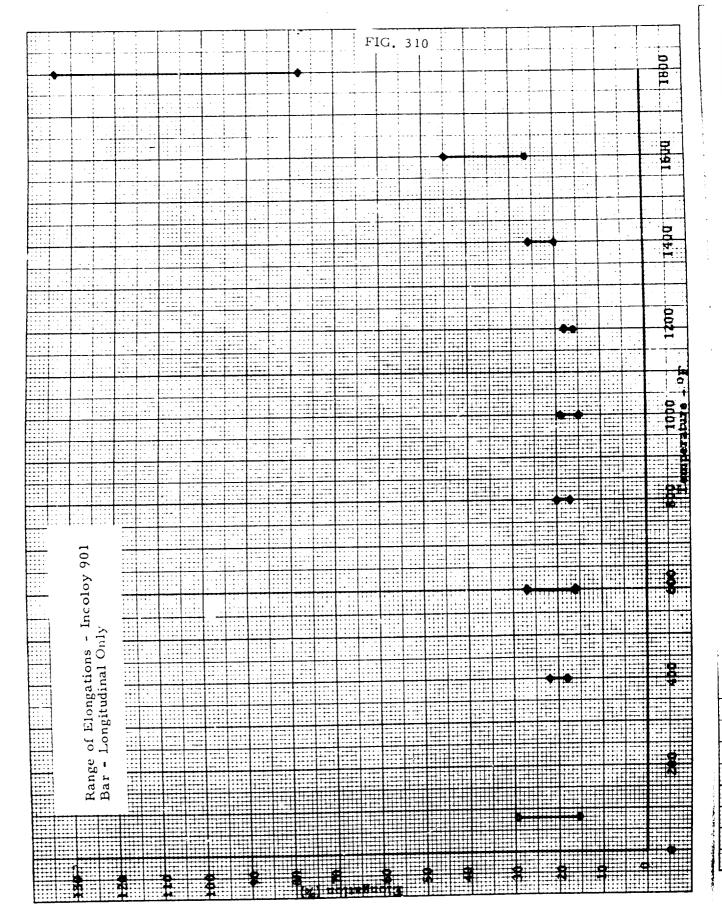


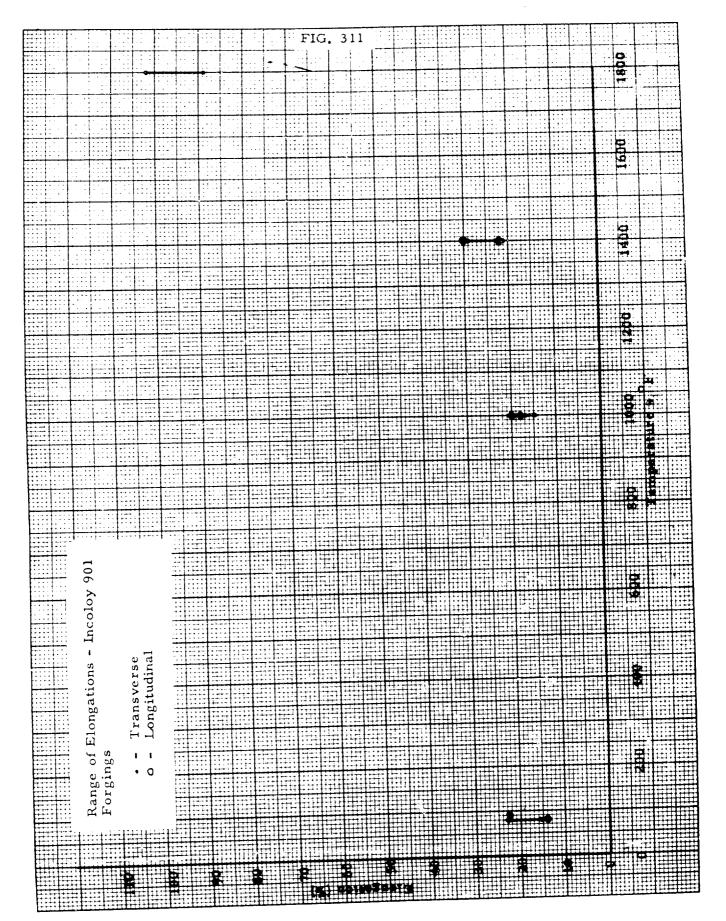




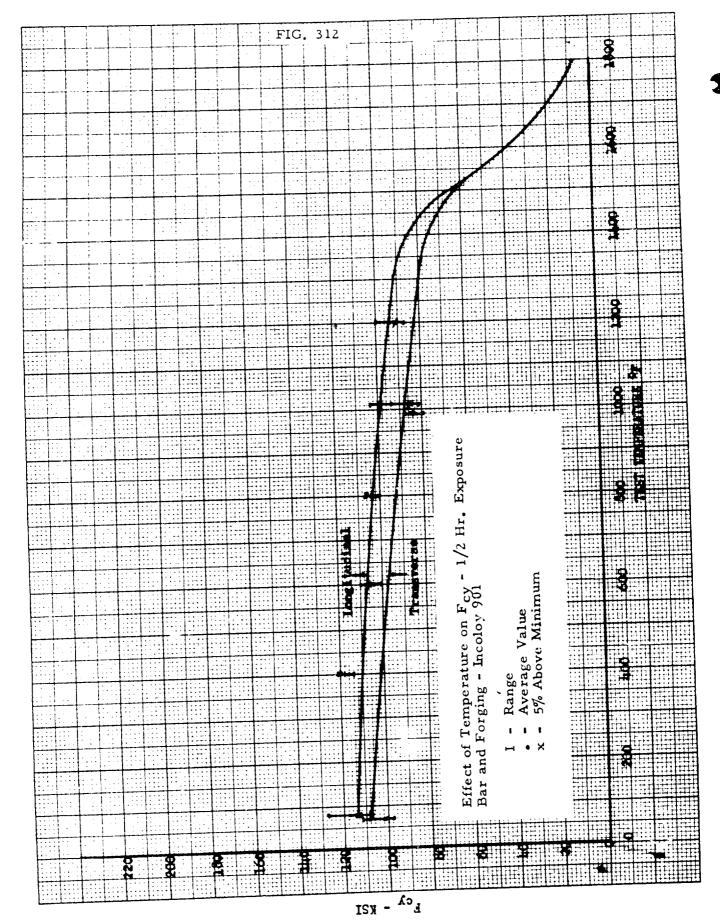
 $\mathbf{E}^{\mathfrak{L}n}$  - K2I





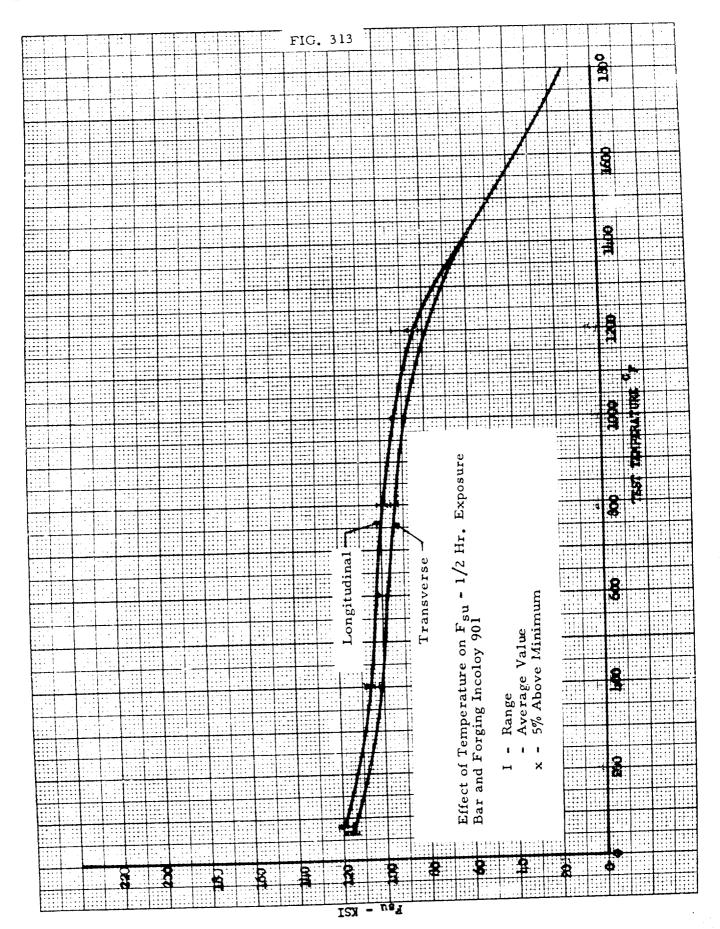


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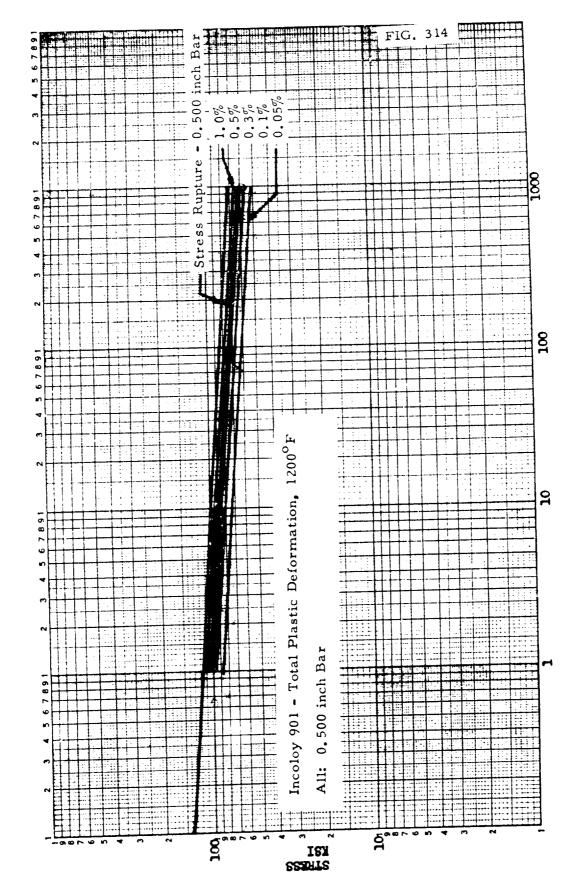


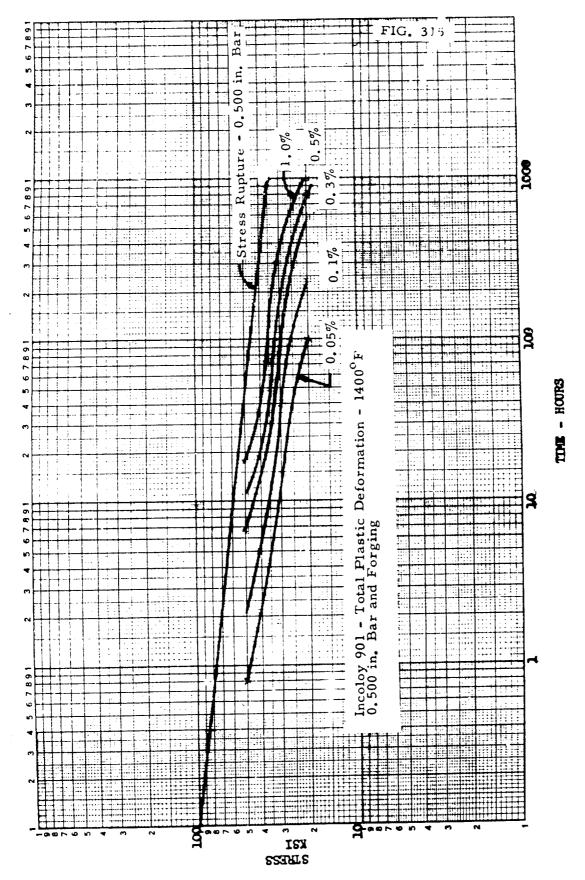
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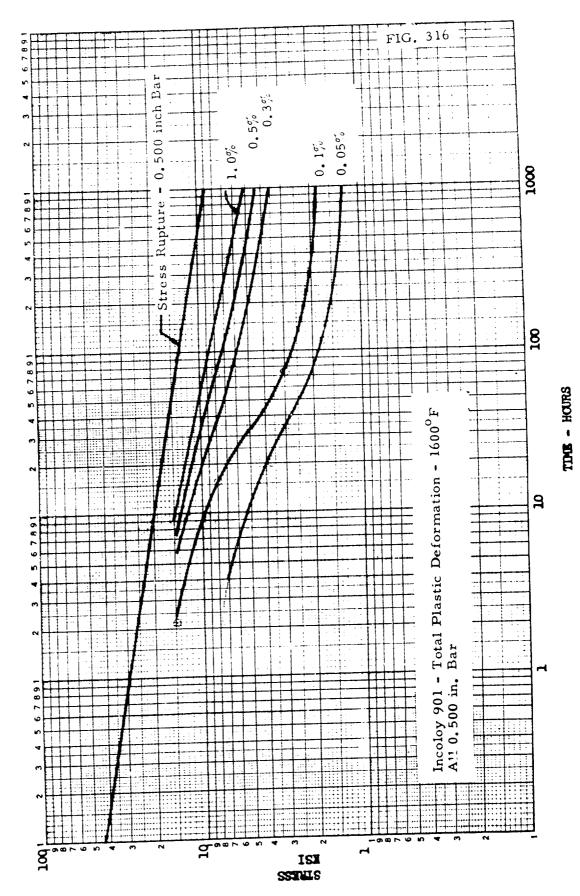
SECTION 7.4.4 SHEAR

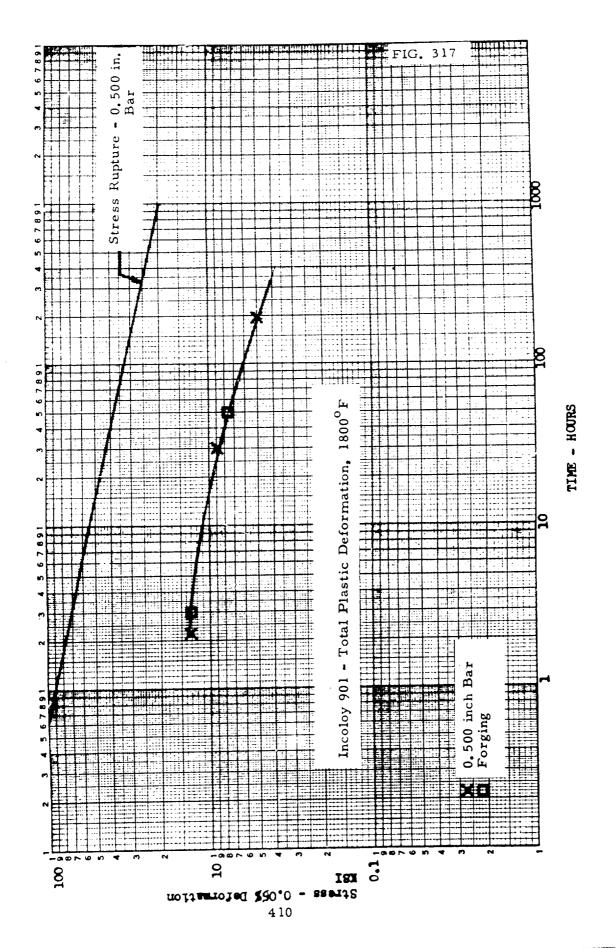


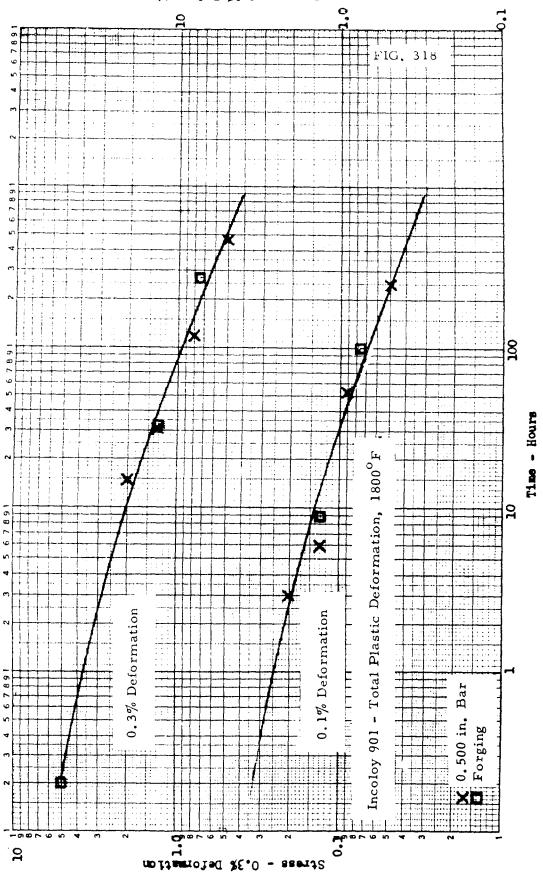
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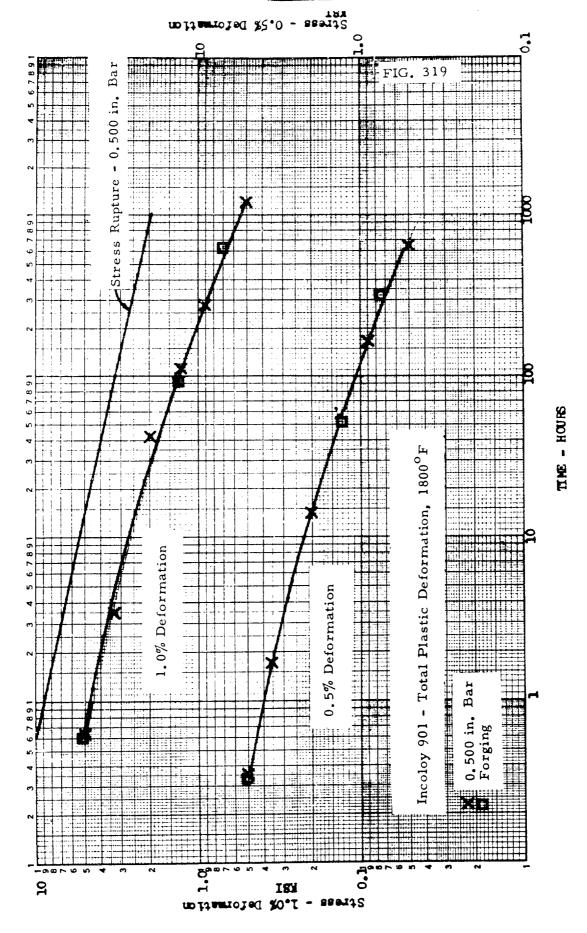




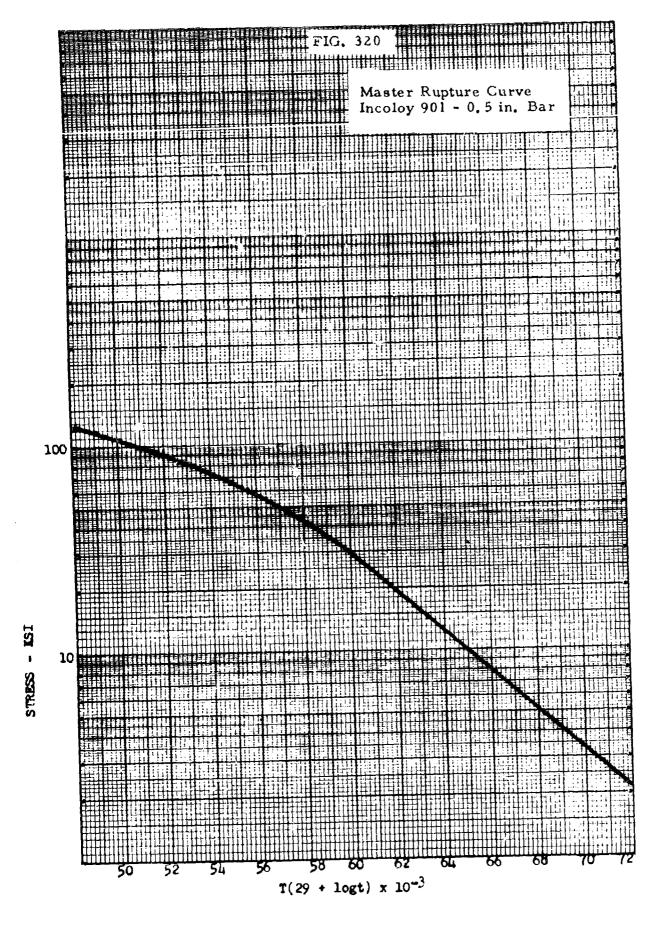


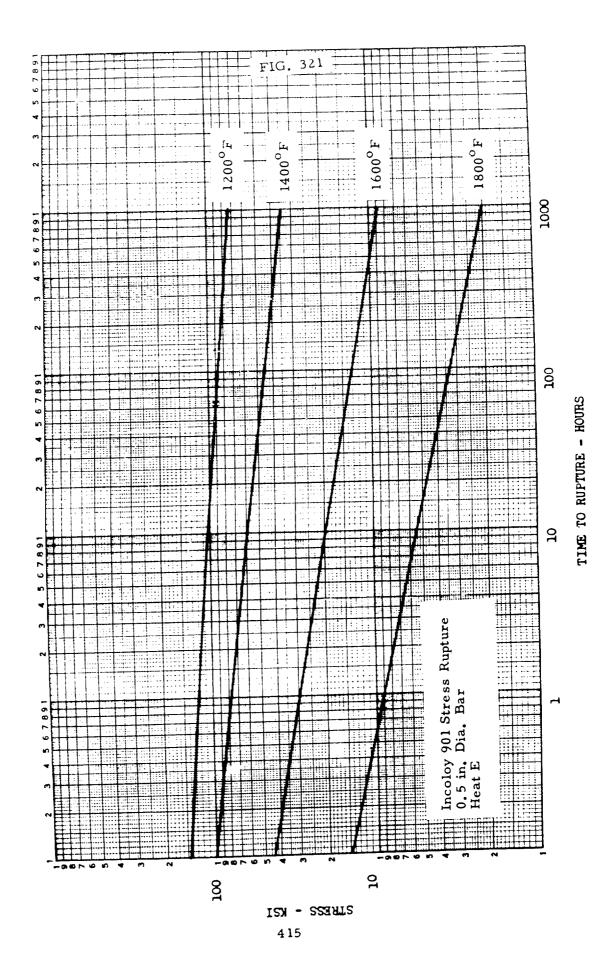


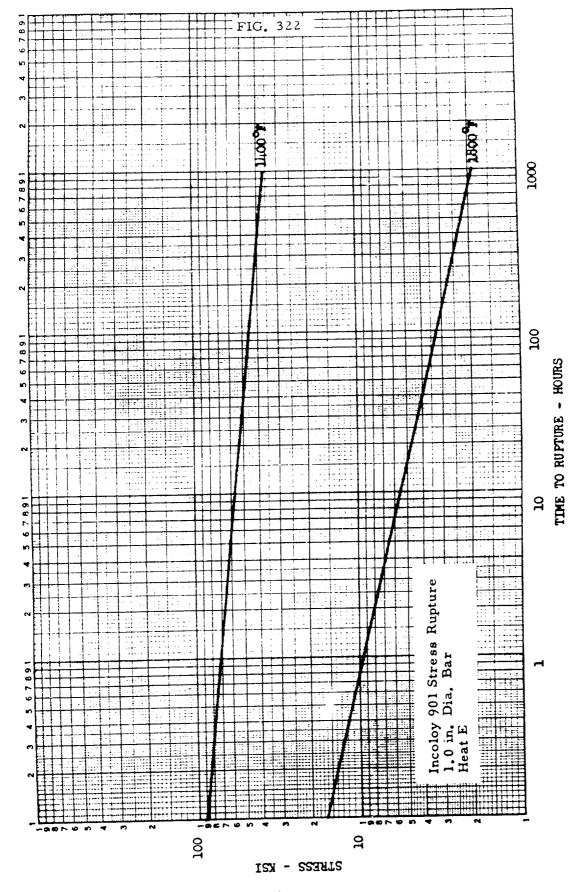


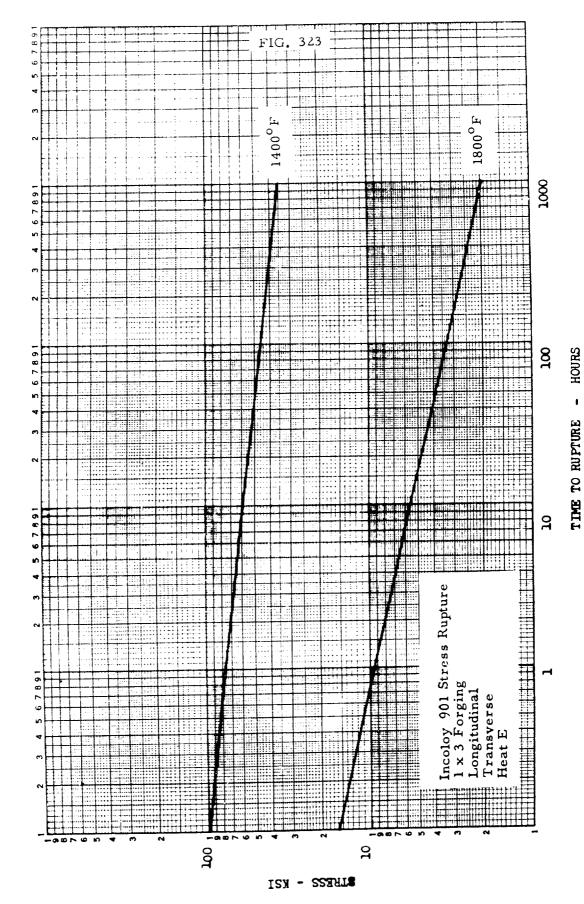


SECTION 7.4.6 STRESS RUPTURE

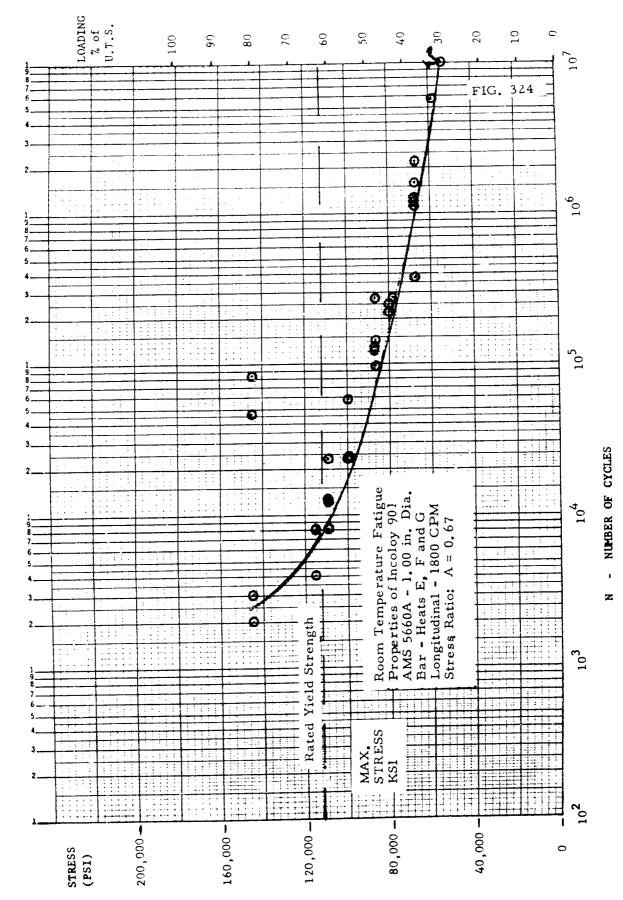


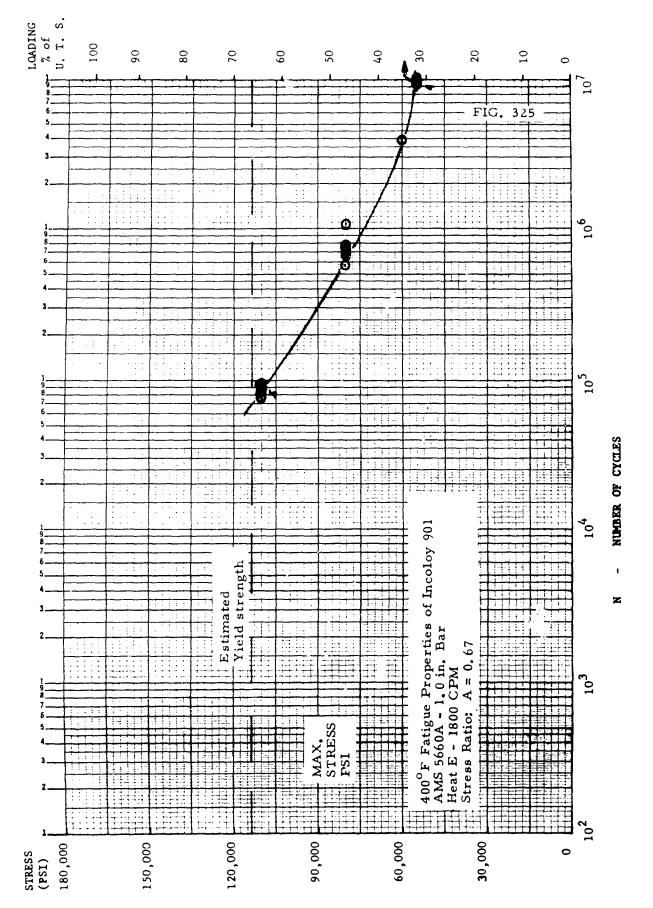


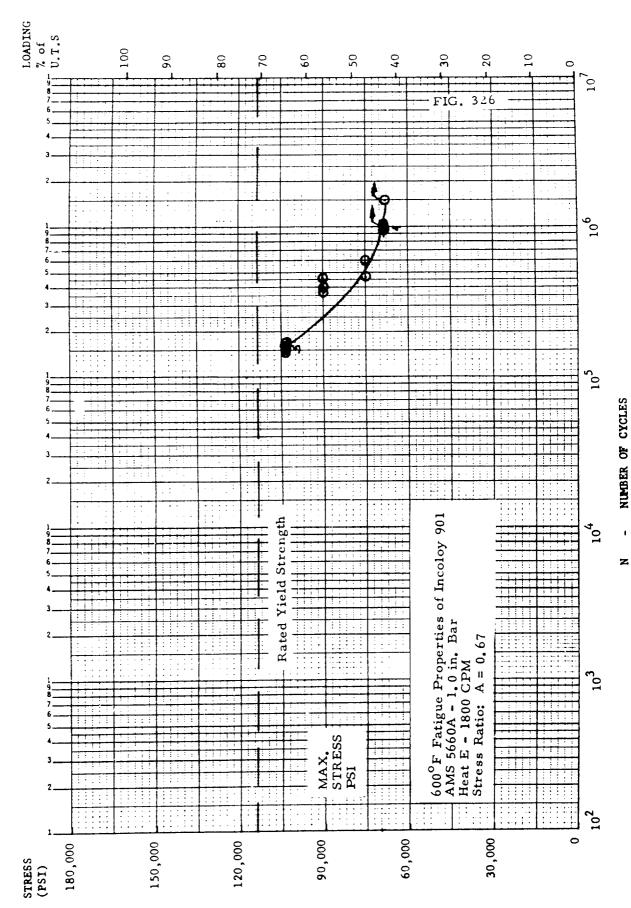


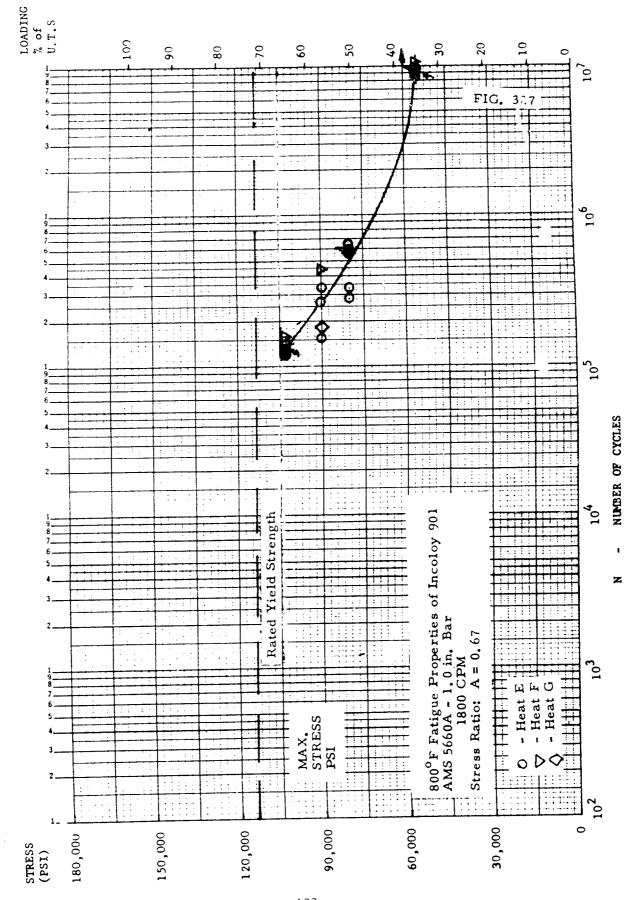


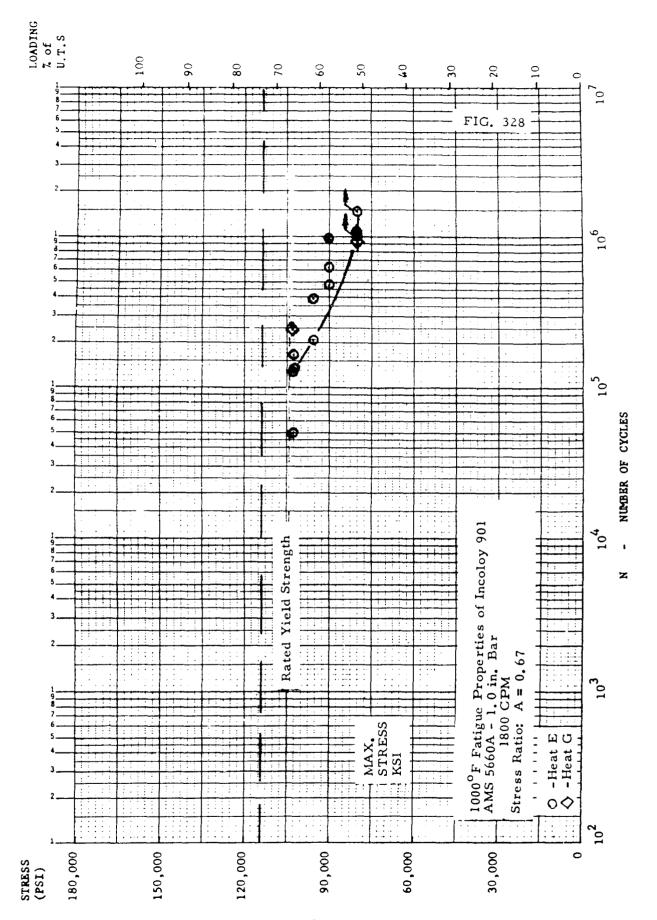
SECTION 7.4.7 PATICUE

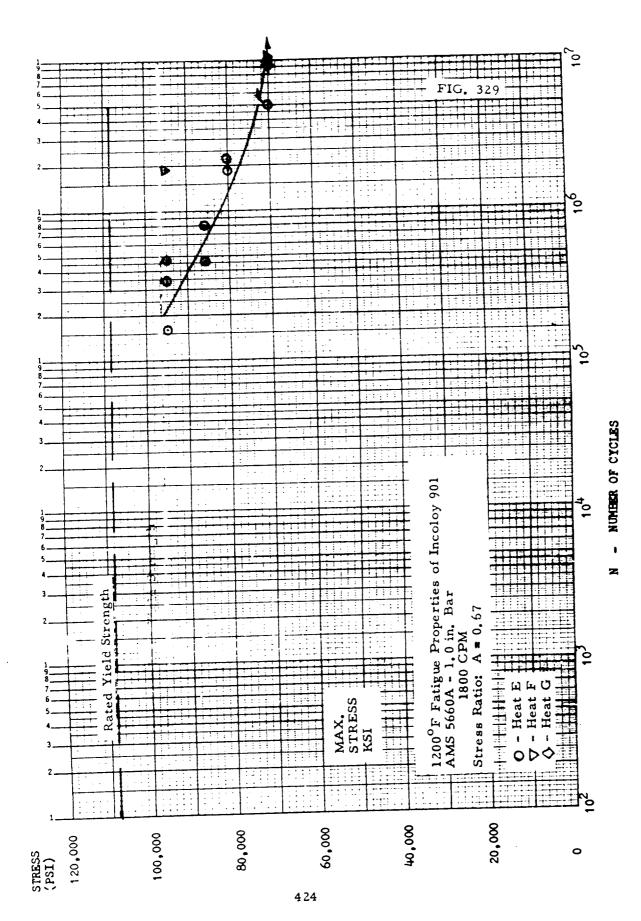


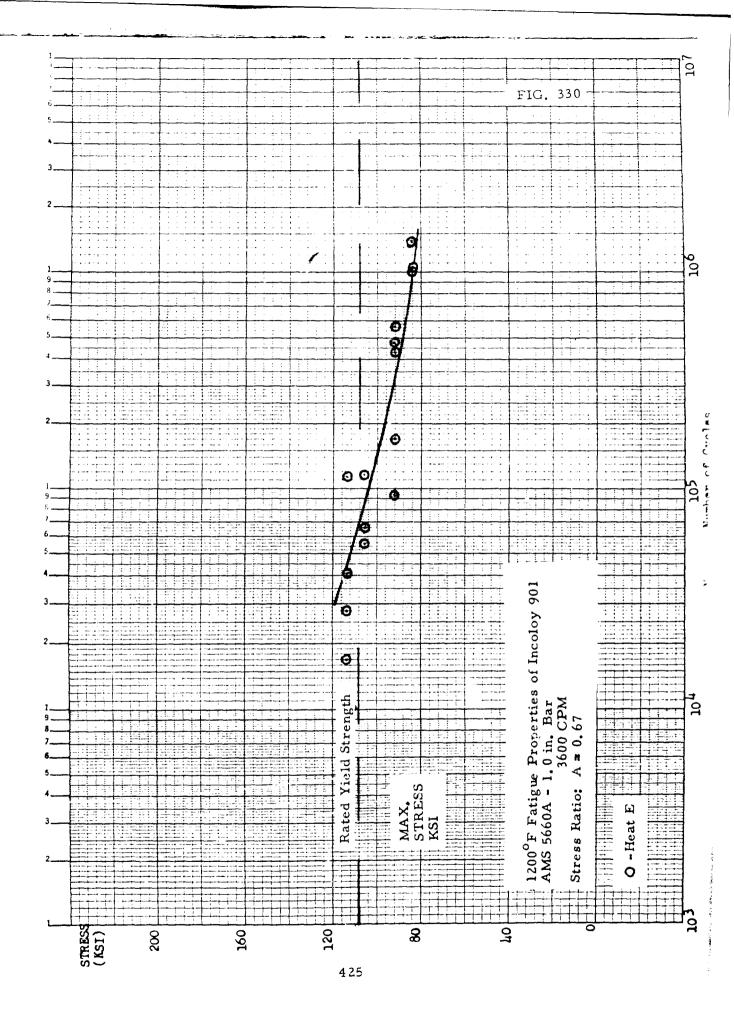


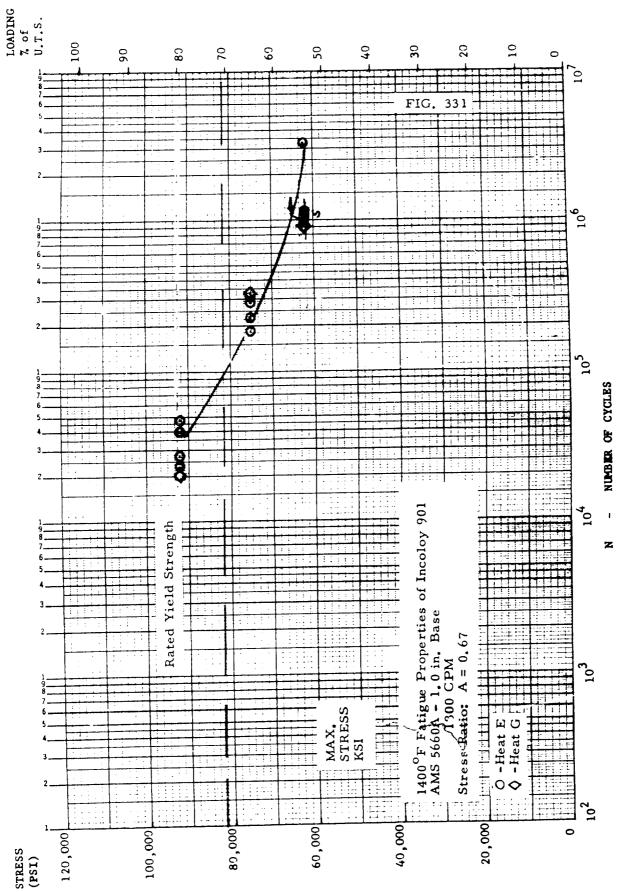


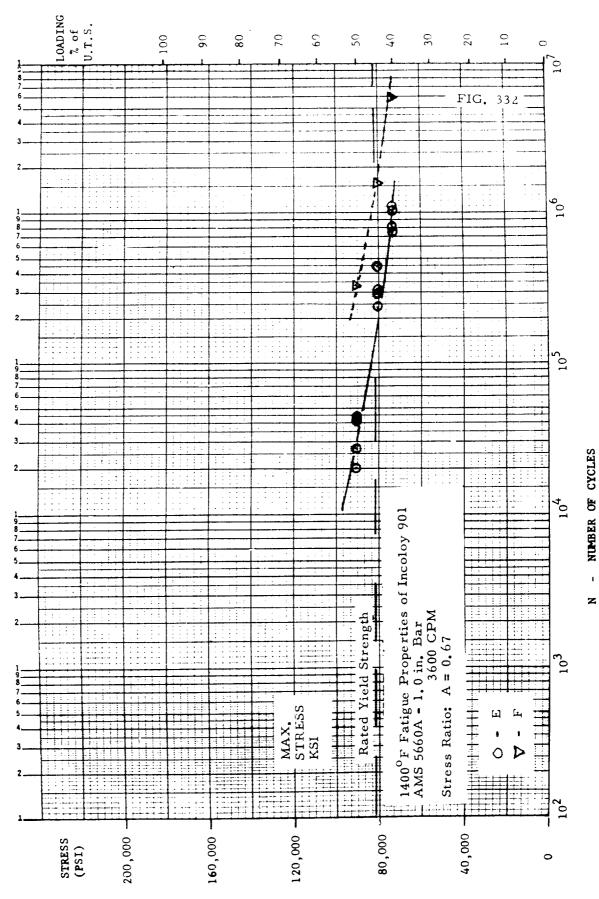


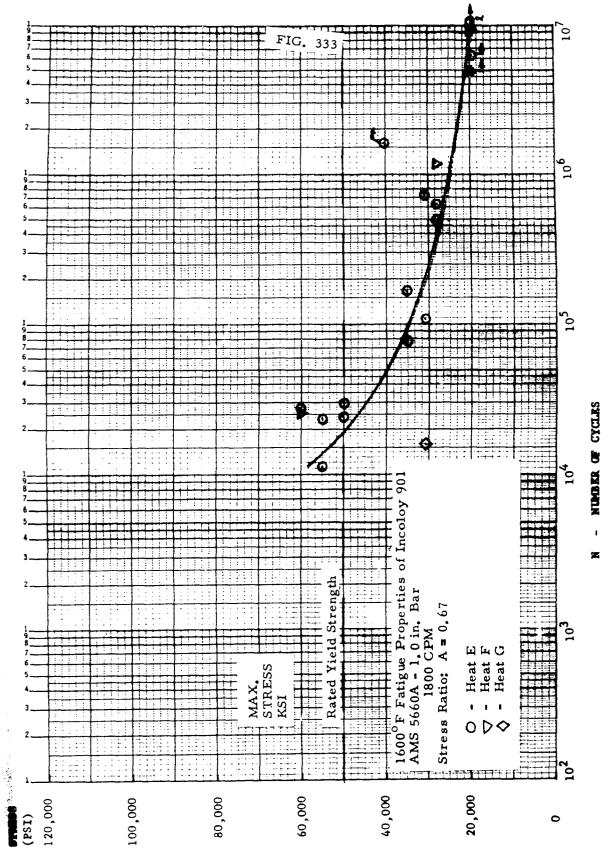


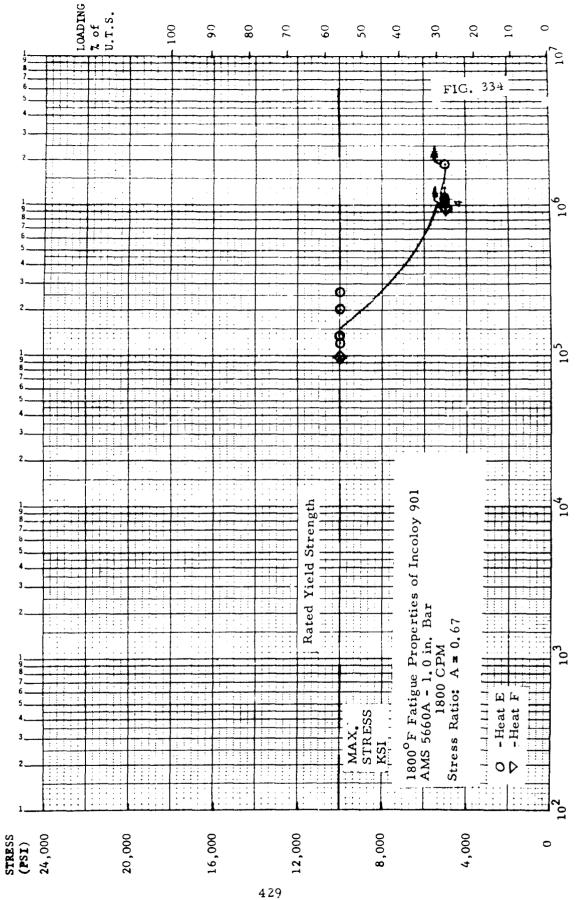






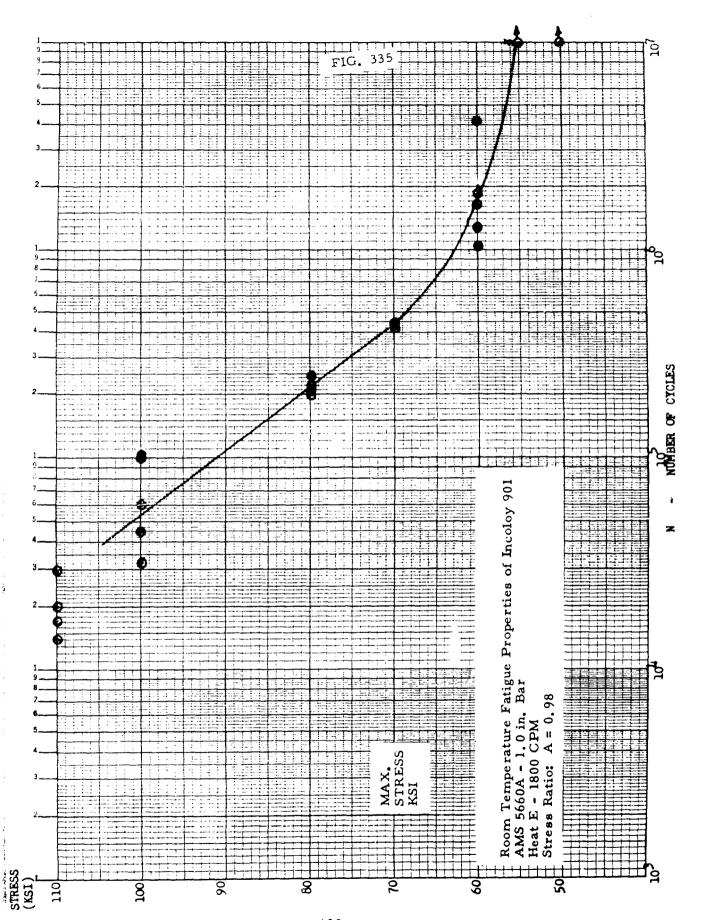


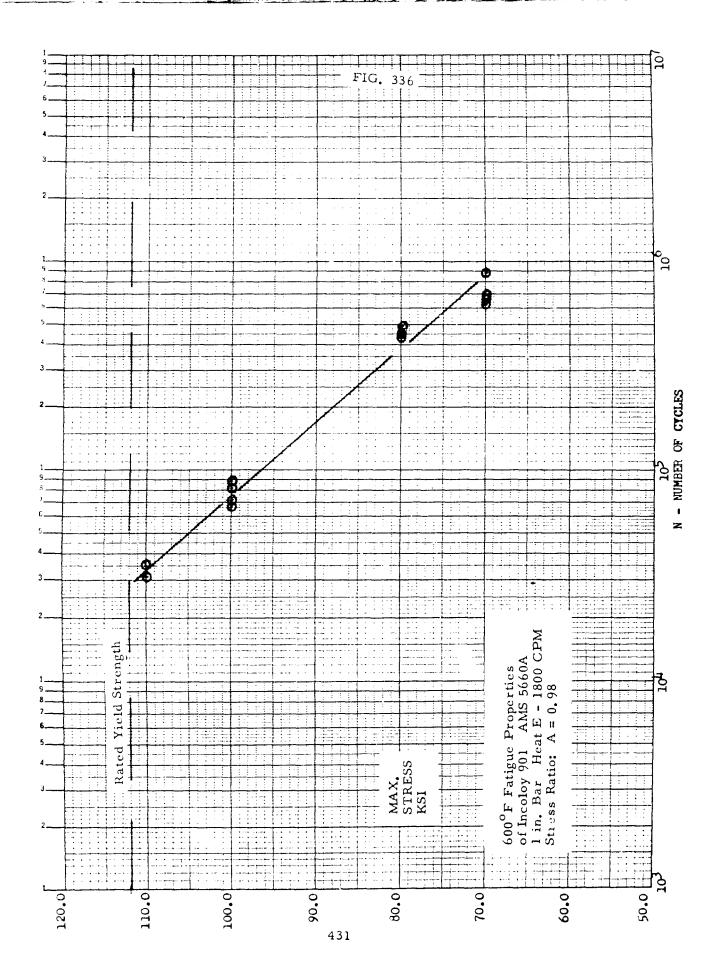


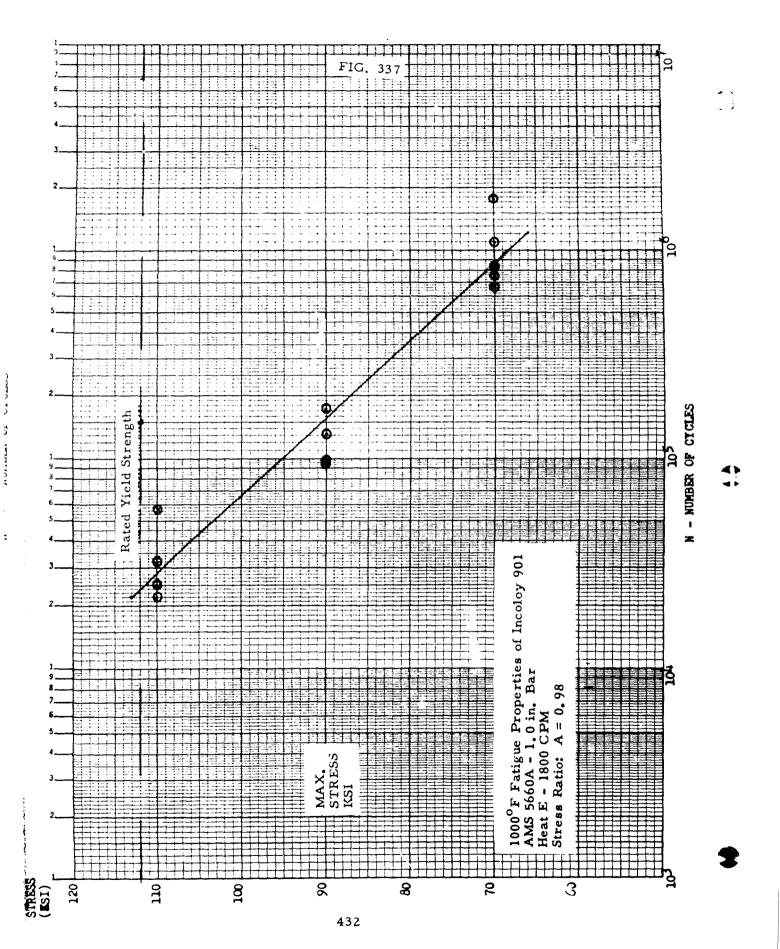


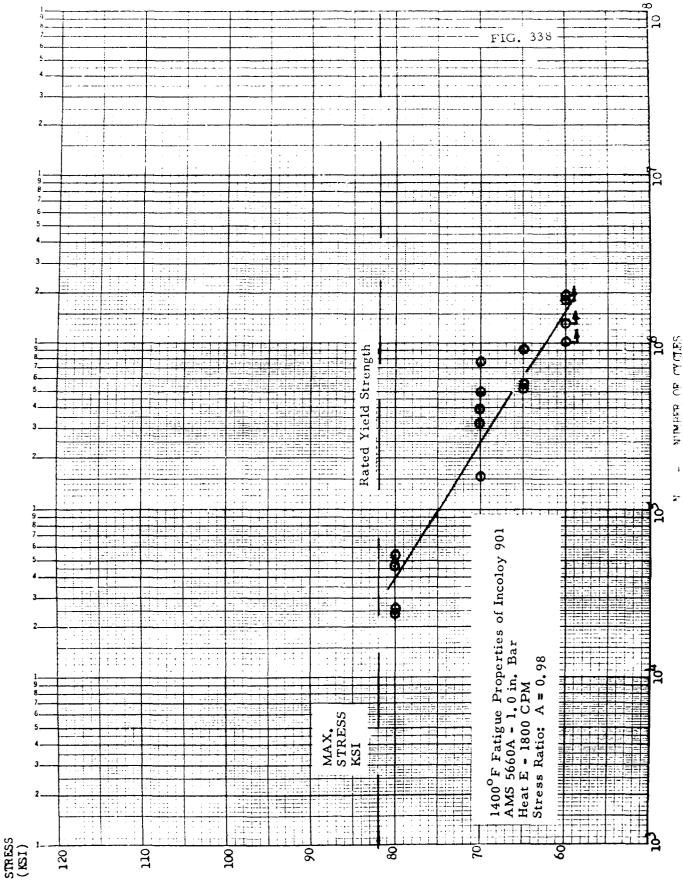
NUMBER OF CYCLES

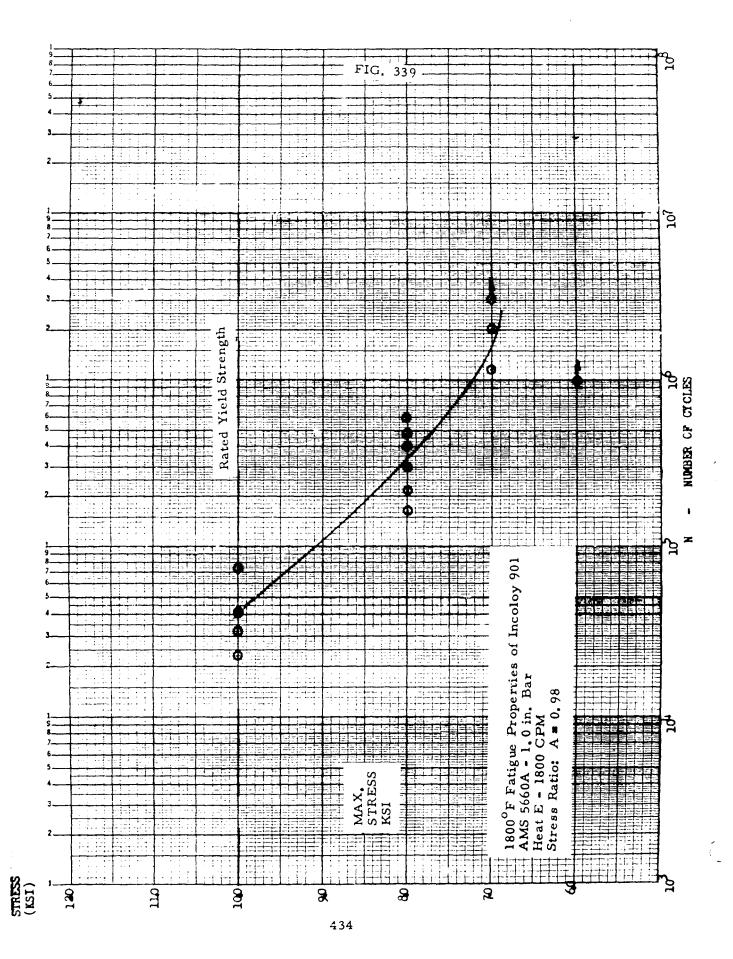
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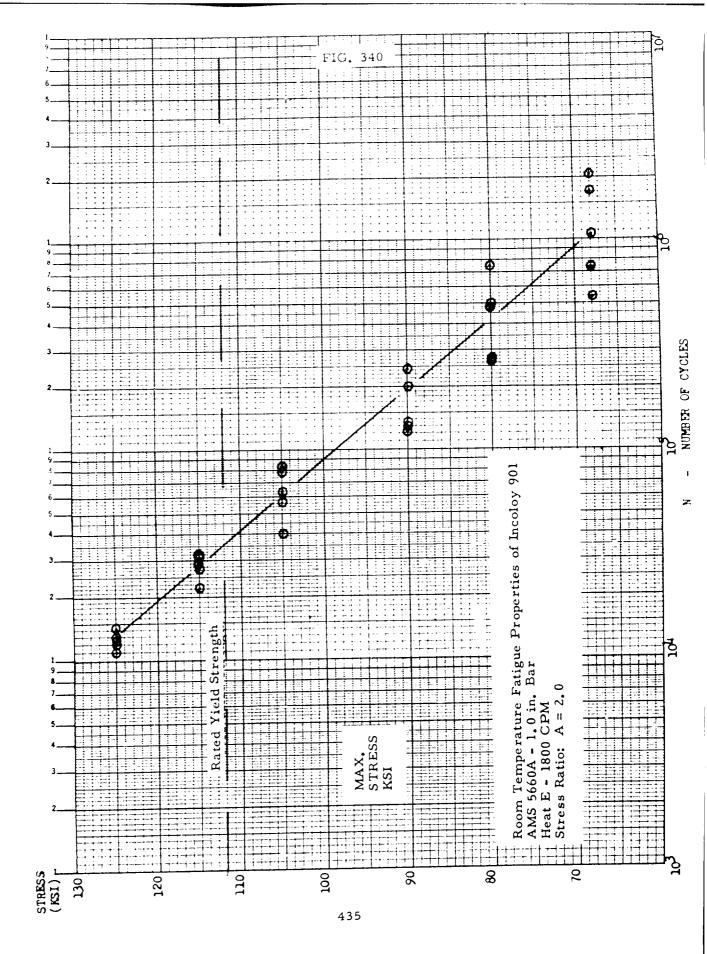


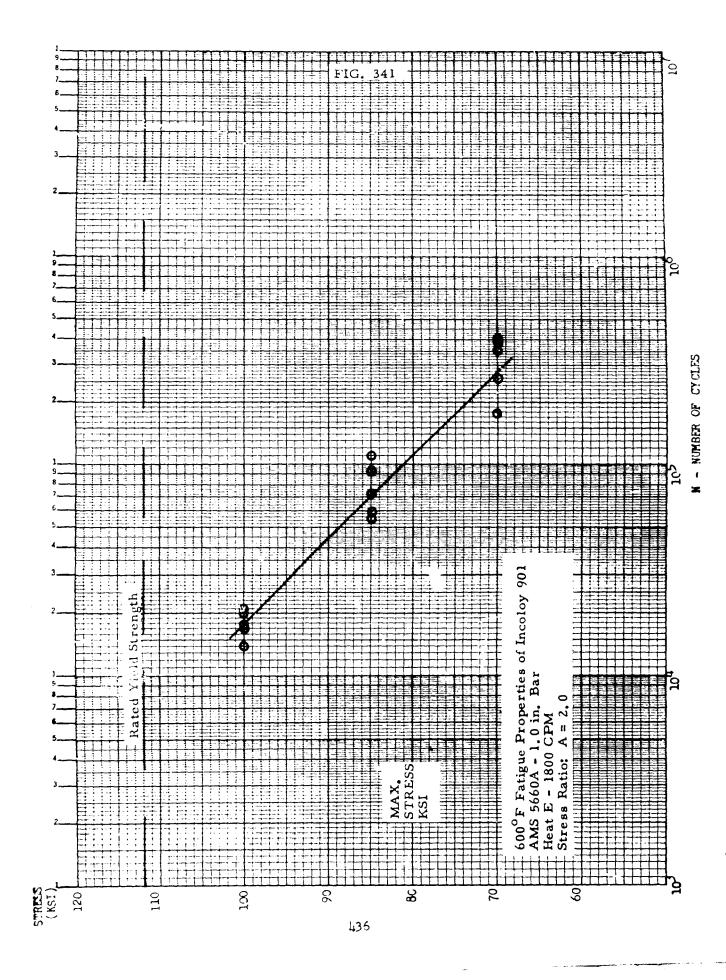


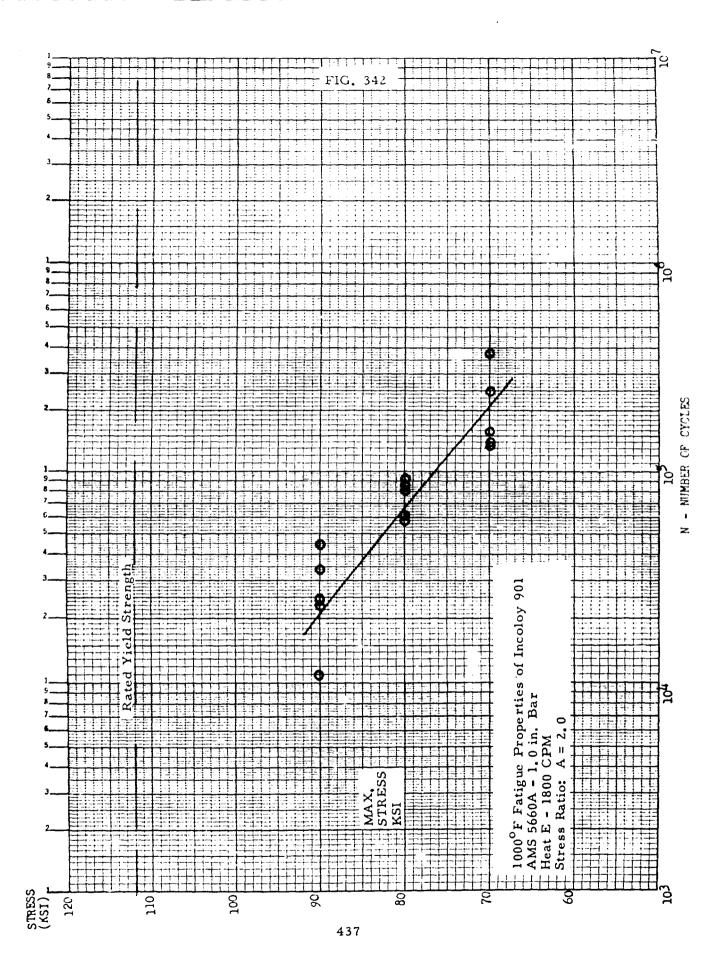


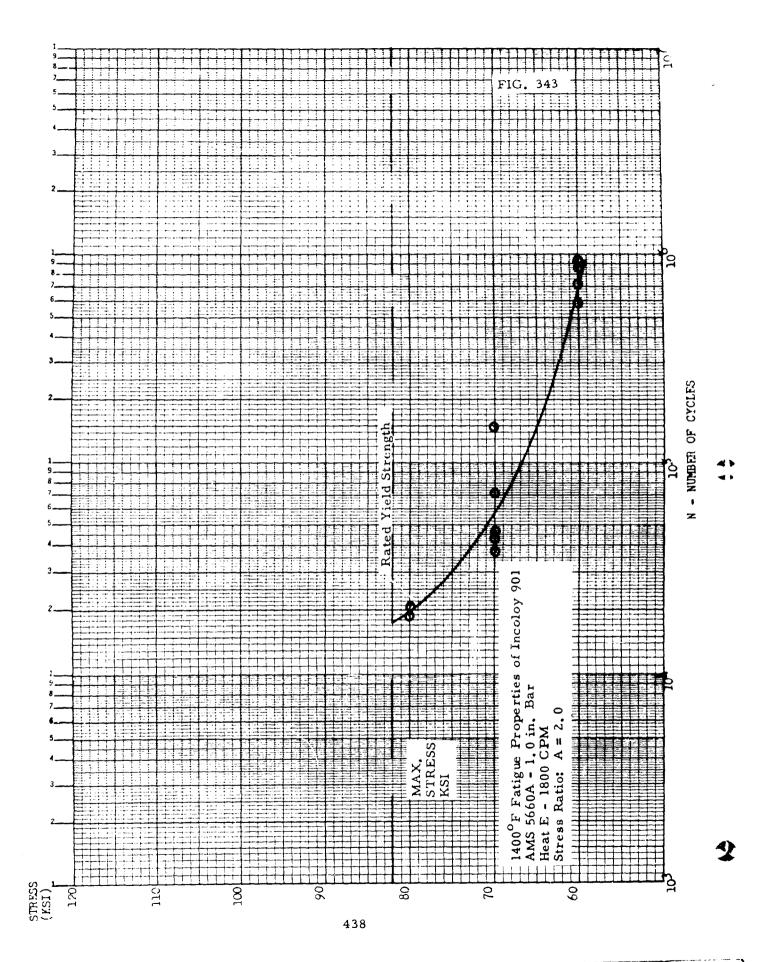


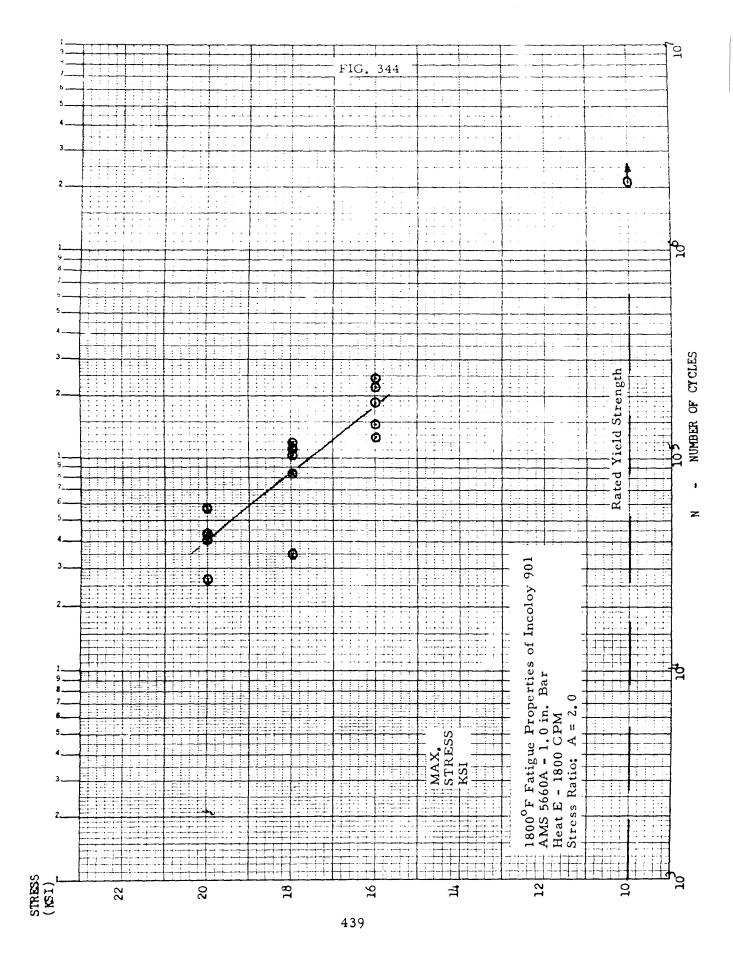


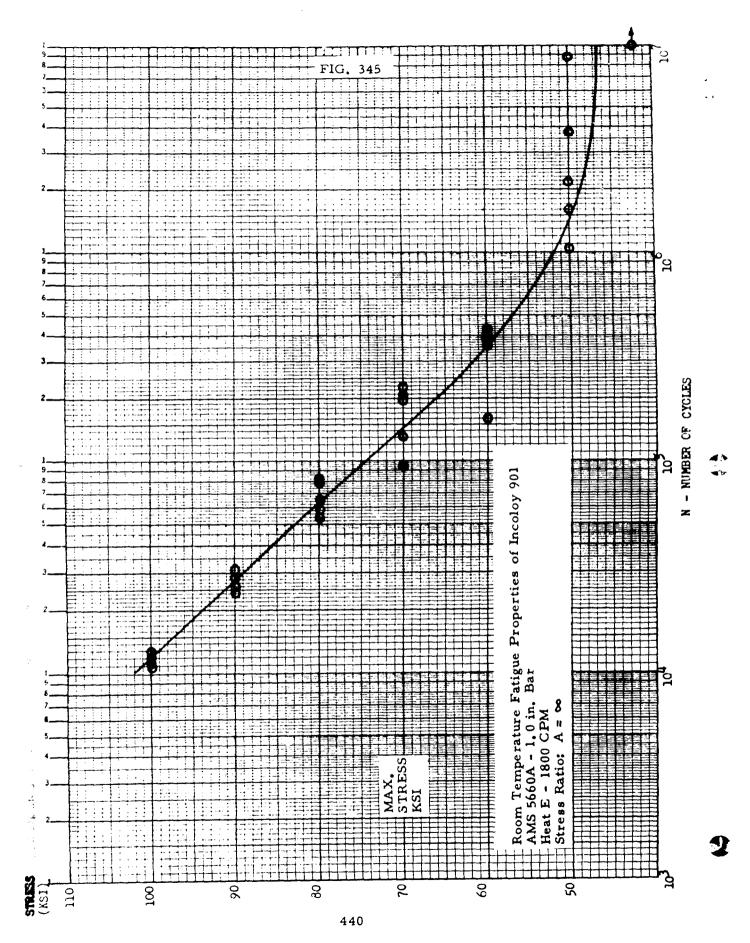


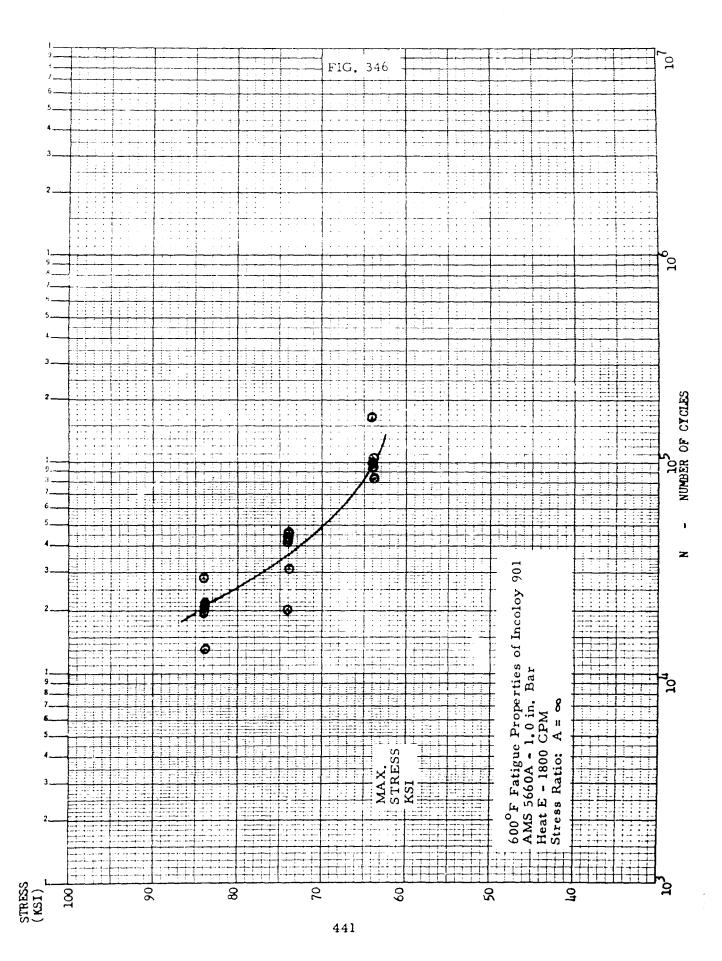


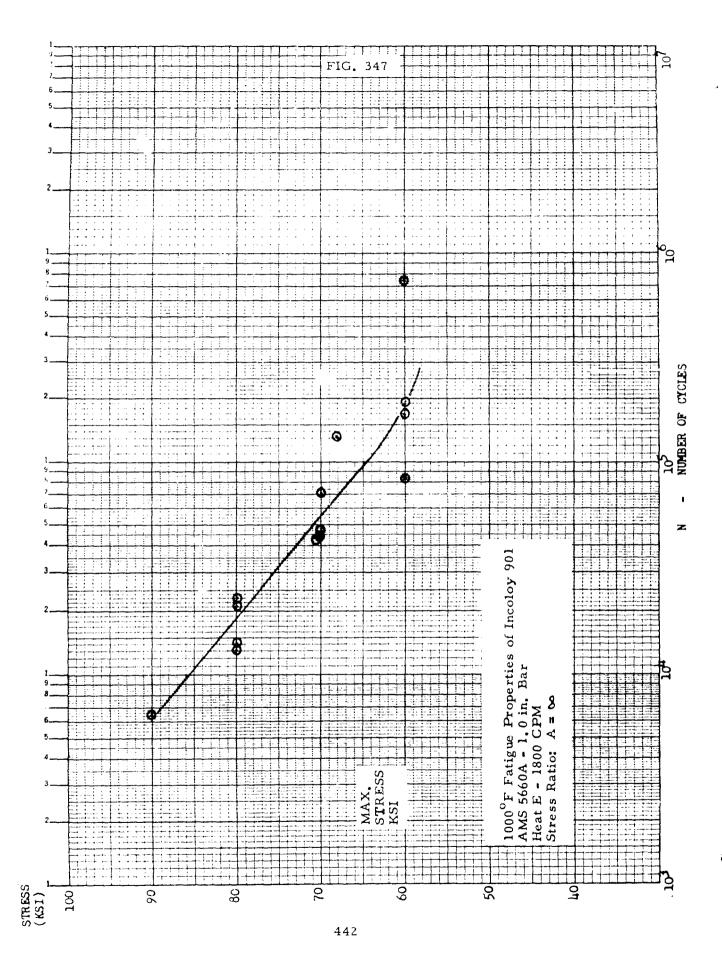


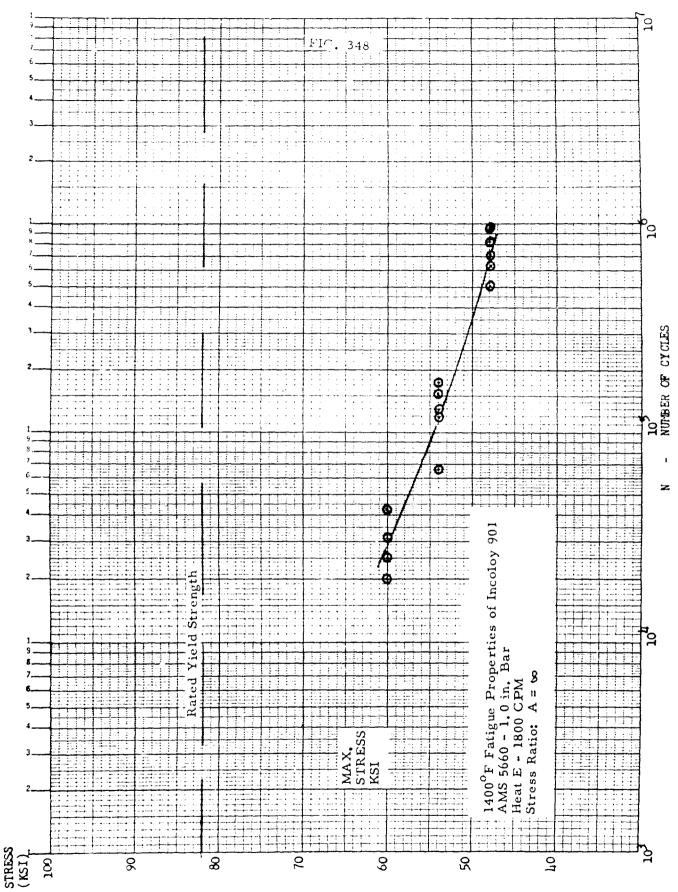


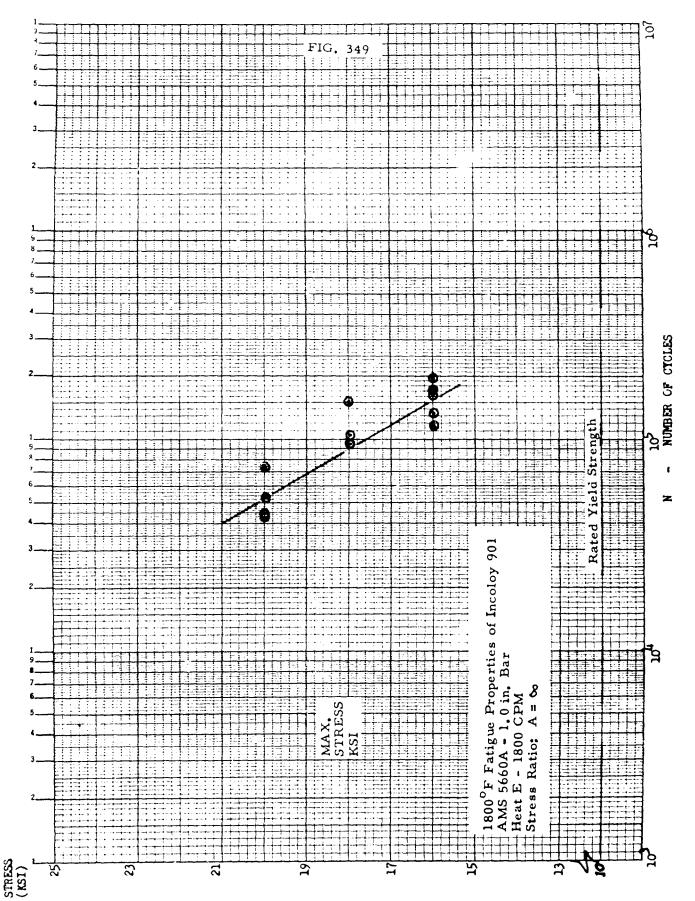


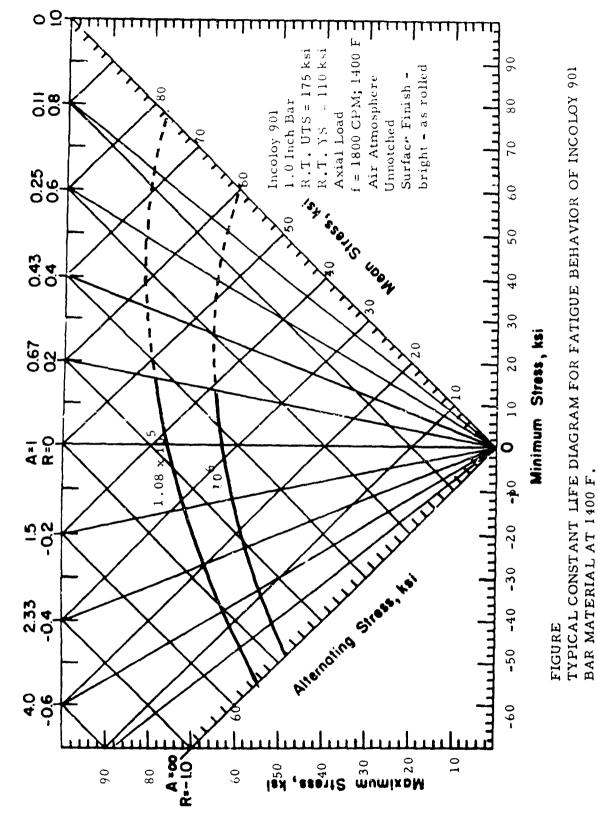












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## SECTION VIXI - MIL-HDB-5 DATA PRESENTATION

8.1 MATERIAL, RENE' 41

Alloy	Rene' 41						
Form	Foil (t ≤ .020")						
Condition	Solution Treated and Aged*						
DIRECTION	L			r			
Basis	À	B	A	В			
Mechanical Properties  Ftu, ksi	177.7	184.8	176.6	183.7			
F <sub>ty</sub> , ksi	120.3	125.1	122.8	127.7			
F <sub>cy</sub> , ksi							
F <sub>su</sub> , ksi F <sub>bru</sub> , ksi (e/D = 1.5) (e/D = 2.0) F <sub>bry</sub> , ksi (e/D = 1.5) (e/D = 2.0) e, per cent							
E, 10 <sup>6</sup> psi	31.6						
E <sub>c</sub> , 10 <sup>6</sup> pai							
G, 10 <sup>6</sup> psi							
Physical Properties		<del></del>	·				
ω, lb/in. <sup>3</sup>							
C, Btu/(lb)(F)							
K, Btu/[(hr)(ft <sup>2</sup> )(F)/ft]							
$\alpha$ , $10^{-6}$ in./in./F							

<sup>\*</sup> Solution Treat - 1975°F for 1/2-hour - Water Quench Age - 1400°F for 16-hours - Air Cool

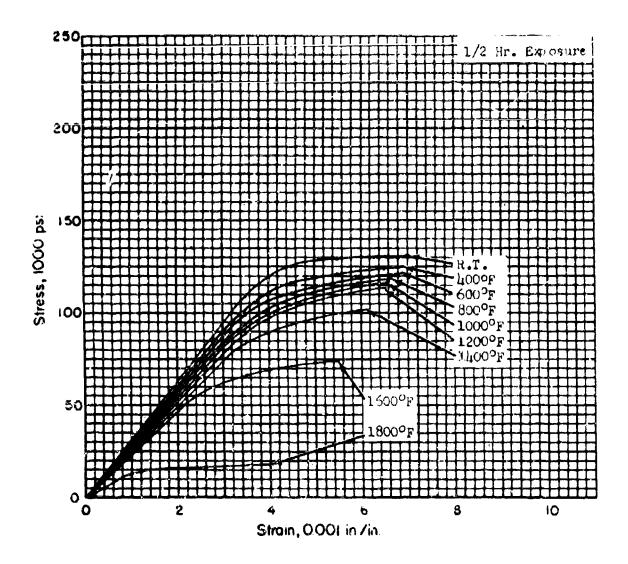
	, A							
Alloy	Rene' 41							
Form	Sheet (.020" $\leq t \leq .187$ )							
Condition	Solution treated and aged*							
Direction	L		Ţ					
Basis	A B		A	В				
Mechanical Properties  F <sub>tu</sub> , ksi	177.5	185.6	177.7	185.8				
F <sub>ty</sub> , kei	123.0	133.3	123.6	134.0				
F <sub>cy</sub> , kei	ութ.ր	154.4	137.1	148.6				
F <sub>su</sub> , ksi F <sub>bru</sub> , ksi	118.8	122.1	112.8	118.0				
(e/D = 1.5)	269.3	276.9	257.7	269.lı				
(e/D = 2.0)	331.2	340.6	323.7	338.5				
F <sub>brv</sub> , ksi								
$^{\circ}$ (e/D = 1.5)	200.6	232.6	196.2	213.0				
(e/D = 2.0) e, per cent	269.3	284.0	261.5	275.7				
E, 10 <sup>6</sup> psi	31.6							
E <sub>c</sub> , 10 <sup>6</sup> psi	31.6							
G, 10 <sup>6</sup> psi								
Physical Properties								
ω, lb/in. <sup>3</sup>								
C, Btu/(lb)(F)								
K, Btu/[(hr)(ft <sup>2</sup> )(F)/ft]	į							
$\alpha$ , $10^{-6}$ in./in./F								

Note: Solution trest 1975 F for 1/2 hour, water quench Age 1400 F for 16 hours. Air cool.

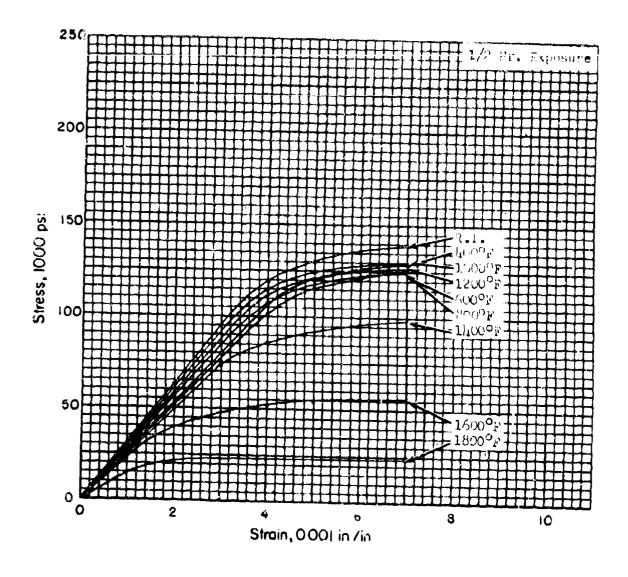
## DESIGN MECHANICAL AND PHYSICAL PROPERTIES OF

Alloy	Rene' lil								
Form	Plate & For	, Bar gings	Plate & Forging ‡				Bar		
Condition	Solution Treated and Aged*								
Direction		L		L		T		T	
Basis	Tent	Tentative A   B		Tentative A B		Tentative A   B		Tentative A B	
Mechanical Properties  F <sub>tu</sub> , ksi	189.9	191.7			183.4	185.2			
F <sub>ty</sub> , ksi	140.0	42.5			135.7	138.1			
F <sub>cy</sub> , kei	142.8	145.4			131.6	134.0			
F <sub>su</sub> , ksi F <sub>bru</sub> , ksi			132.6	133.8	128.9		126.9	128.1	
(e/D = 1.5) (e/D = 2.0) Fbry, ksi					175.5   353.4	356.8			
(e/D = 1.5) (e/D = 2.0) e, per cent					218.6	222.5	1		
E, 10 <sup>6</sup> psi			· · · · · · · · · · · · · · · · · · ·	31.6			1	J	
E <sub>c</sub> , 10 <sup>6</sup> psi	31.6								
G, 10 <sup>6</sup> psi									
Physical Properties									
ω, lb/in. <sup>3</sup>									
C, Btu/(lb)(F)									
K, Btu/[(hr)(ft <sup>2</sup> )(F)/ft]									
$\alpha$ , $10^{-6}$ in./in./F									

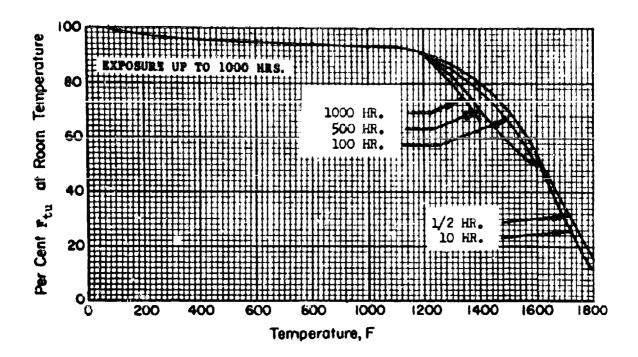
<sup>\*</sup> Solution Treat - 1975oF for 1/2-hour - Water Quench Age - 1400oF for 16-hours - Air Cool ‡ Bearing Results based on plate only



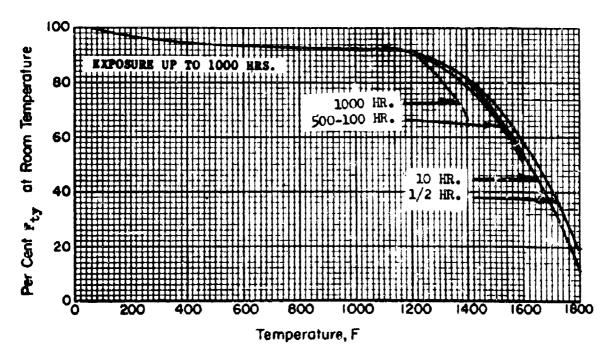
Typical tensile stress versus strain curves for Rene! 41 alloy sheet reduced to 'A' basis. Transverse direction only.



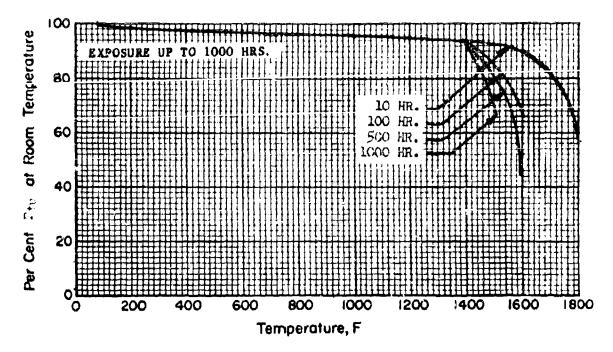
Typical compressive stress vs. strain curves for Rene! 41 alloy sheet, reduced to "A" basis. Transverse direction only.



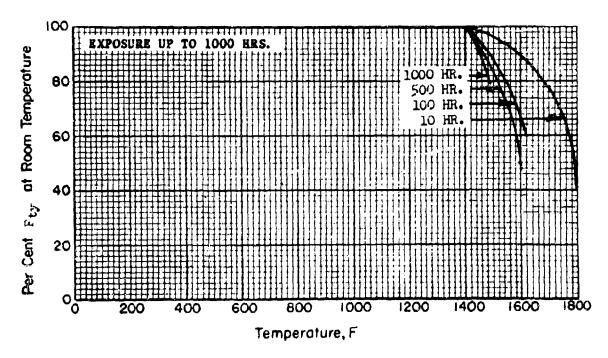
Effects of exposure temperature on elevated temperature ultimate tensile strength ( $F_{tu}$ ) of Rene' 41 alloy foil, transverse direction. Exposure up to 1000-hours.



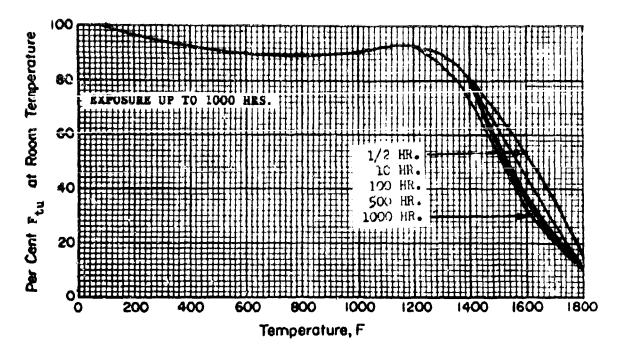
Effects of exposure temperature on elevated temperature tensile yield strength  $(F_{ty})$  of Rene' 41 alley foil, transverse direction. Exposure up to 1000-hours.



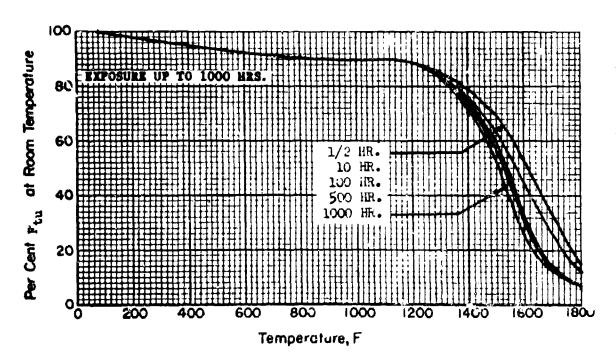
Effect of exposure temperature on room temperature ultimate tensile strength ( $F_{tu}$ ) of Rene' 41 alloy foil, transverse direction. Exposure up to 1000-hours.



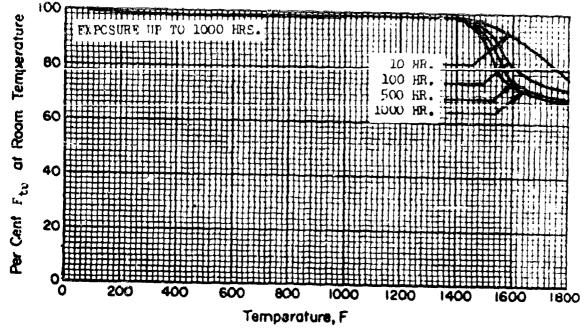
Effect of exposure temperature on room temperature tensile yield strength  $(F_{ty})$  of Rene' 41 alloy foil, transverse direction. Exposure up to 1000-hours.



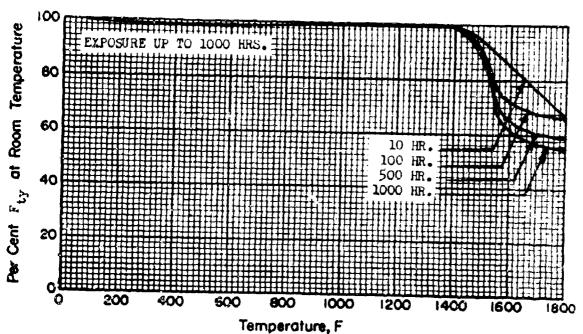
Effect of exposure temperature on elevated temperature ultimate tensile strength ( $F_{tu}$ ) of Rene' il alloy sheet, transverse direction. Exposure up to 1000-hours.



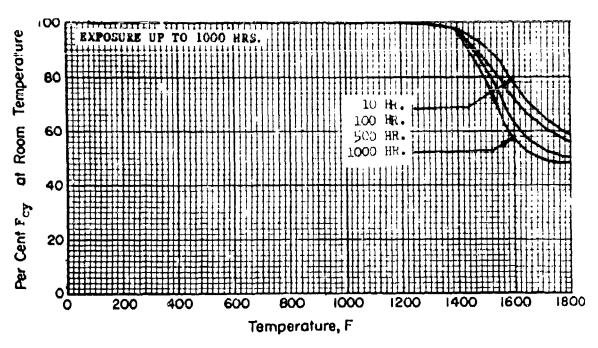
Effect of exposure temperature on elevated temperature tensile yield strength ( $F_{ty}$ ) of Rene' 41 alloy sheet, transverse direction. Exposure up to 1000-hours.



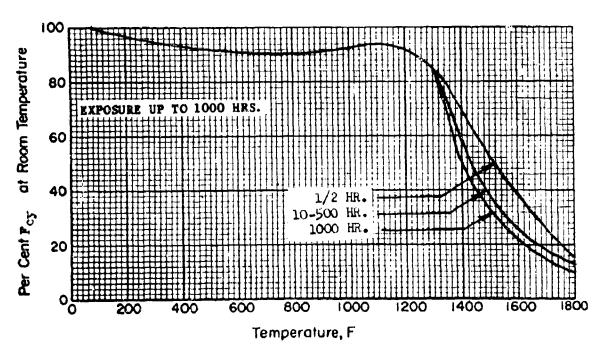
Effects of exposure to temperature on room temperature ultimate tensile strength  $(F_{\underline{t}\underline{u}})$  of Rene' 41 alloy sheet, transverse direction. Exposure up to 1000-hours.



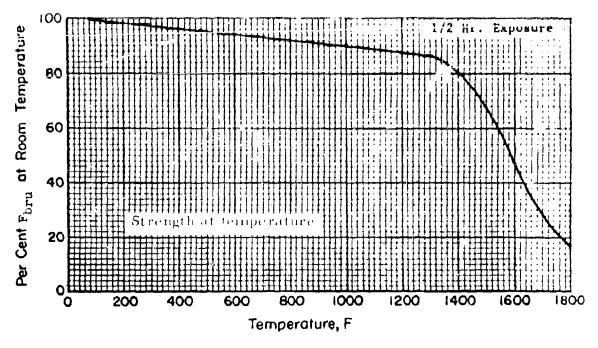
Effects of exposure temperature on room temperature tensile yield strength  $(F_{++})$  of Rene' 41 alloy sheet, transverse direction. Exposure up to 1000-hours.



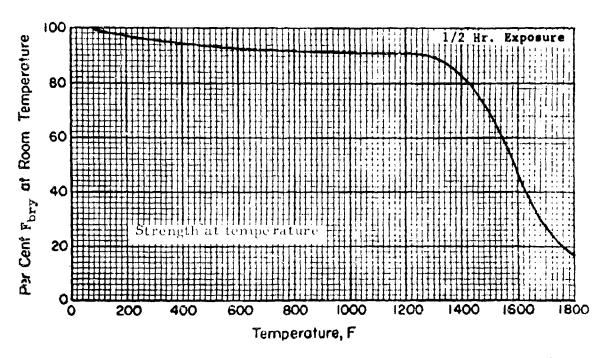
Effect of exposure temperature on room temperature compression yield strength  $(F_{Cy})$  of Rene' 41 alloy sheet, transverse direction. Exposure up to 1000-hours.



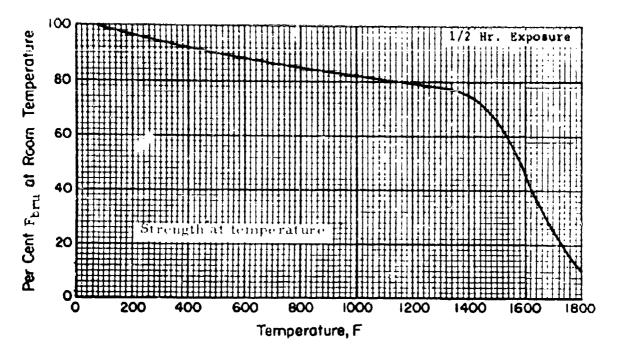
Effect of exposure temperature on elevated temperature compression yield strength (Fcy) of Rene! 41 alloy sheet, transverse direction. Exposure up to 1000-hours.



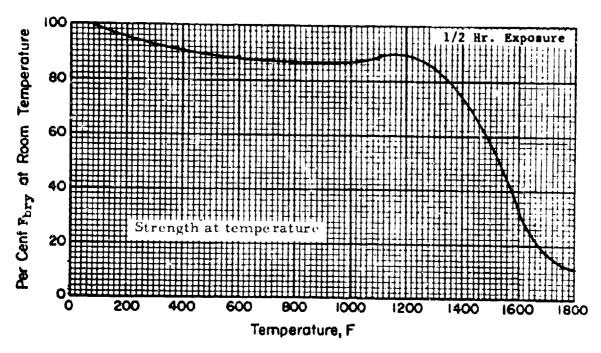
Effect of temperature on the ultimate bearing strength  $(F_{bru})$  of Rene' 41 alloy sheet. Exposure up to 1/2-hour. e/D = 2.0. Longitudinal direction only.



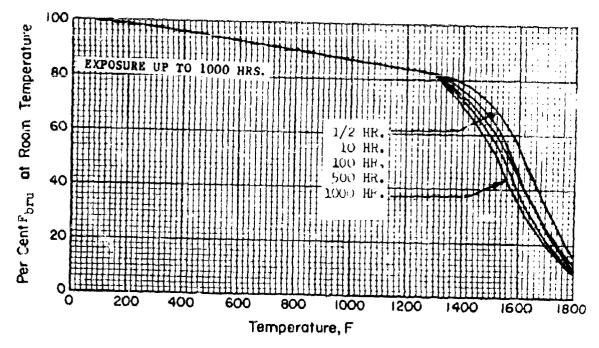
Effect of temperature on the bearing yield strength  $(F_{bry})$  of Rene' 41 alloy sheet. Exposure up to 1/2-hour. e/D = 2.0. Longitudinal direction only.



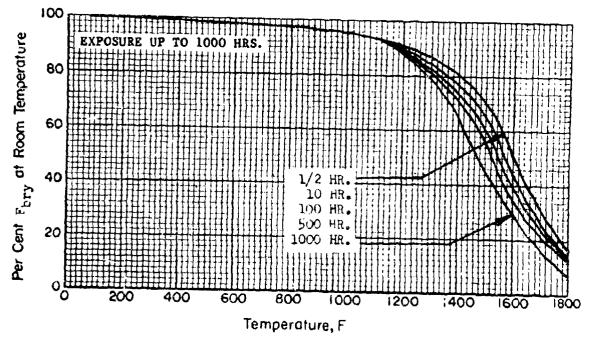
Effect of temperature on the ultimate bearing strength  $(F_{bru})$  of Rene' 41 alloy sheet. Exposure up to 1/2-hour. e/D = 2.0. Transverse direction only.



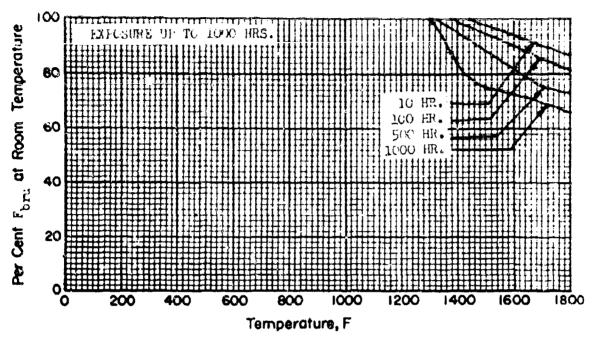
Effect of temperature on the bearing yield strength  $(F_{bry})$  of Rene' 41 alloy sheet. Exposure up to 1/2-hour. e/D = 2.0. Transverse direction only.



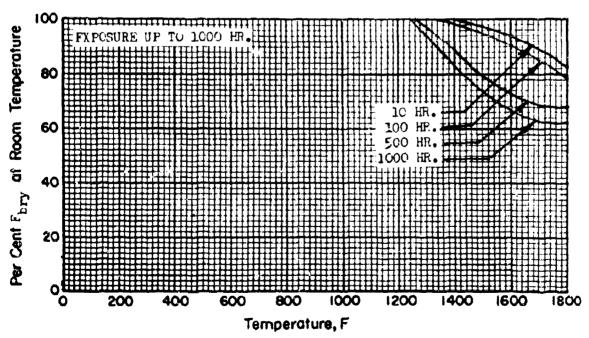
Effect of exposure temperature on elevated temperature bearing ultimate strength  $(F_{bru})$  of Rene' 41 alloy sheet, transverse direction, for e/D=1.5. Exposure up to 1000-hours.



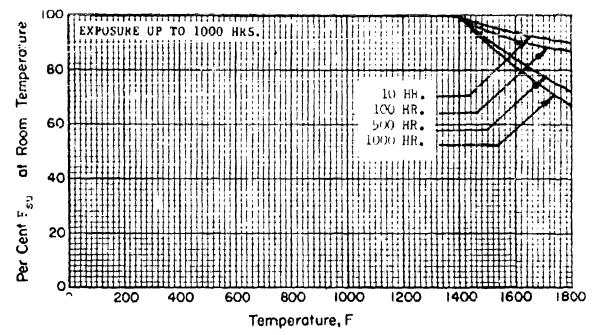
Effect of exposure temperature on elevated temperature bearing yield strength (Fbry) of Rene' 41 alloy sheet, transverse direction, e/D = 1.5. Exposure up to 1000-hours.



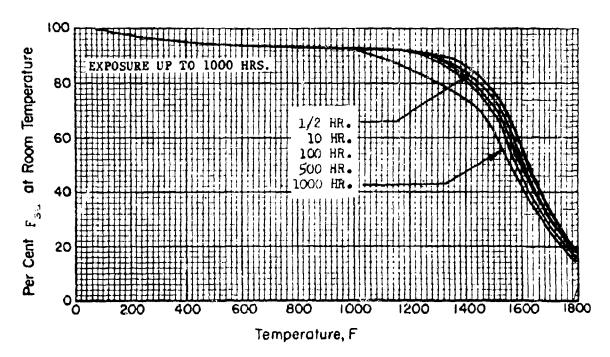
Effect of exposure temperature on room temperature bearing ultimate strength (F ) of Renet hl alloy sheet, transverse direction, for e/D = 1.5. Exposure up to 1000-hours.



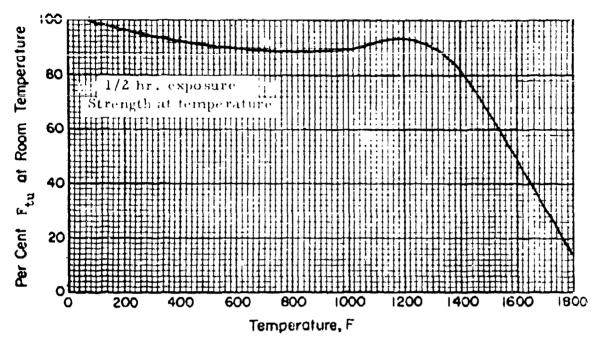
Effect of exposure temperature on room temperature bearing yield strength  $(r_{\rm bry})$  of Rene' 41 alloy sheet, transverse direction, for e/D 1.5. Exposure up to 1000-hours.



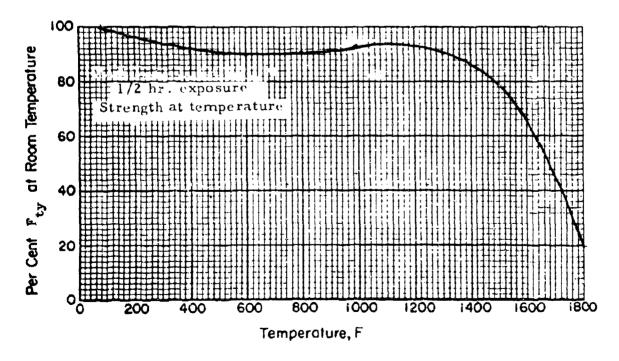
Effect of exposure temperature on room temperature shear ultimate strength  $(F_{su})$  of Rene!  $\frac{1}{2}$  alloy sheet transverse direction. Exposure up to 1000-hours.



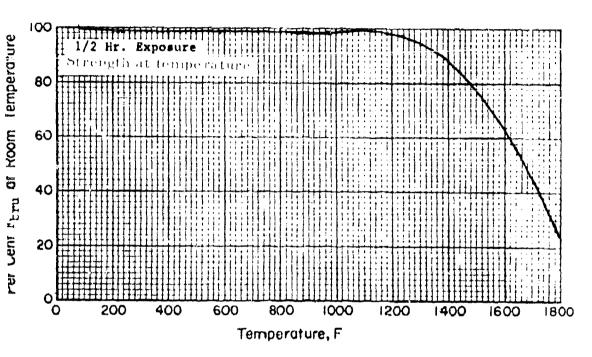
Effect of exposure temperature on elevated temperature shear ultimate strength  $(F_{\rm SU})$  of Rene' 41 alloy sheet, transverse direction. Exposure up to 1000-hours.



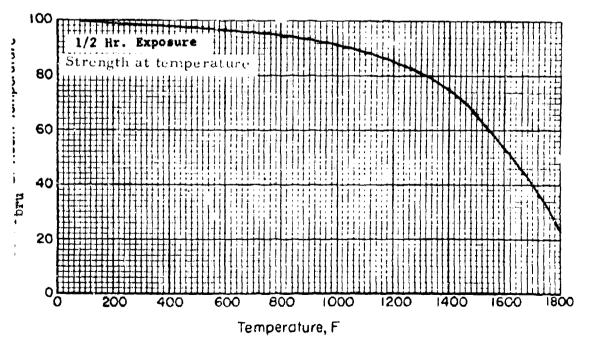
Effects of temperature on ultimate tensile strength ( $F_{tu}$ ) of Rene! In allow, bar, place and forging; transverse and long-itudinal directions.



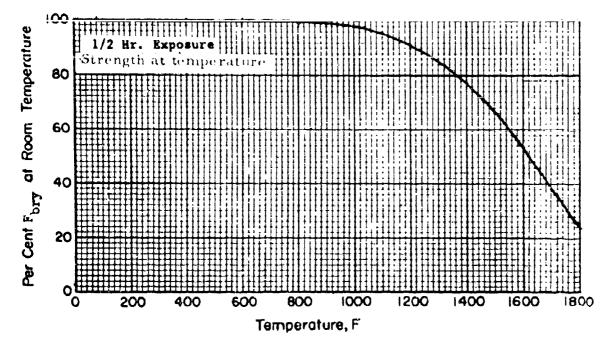
Effects of temperature on tensile yield strength  $(F_{\mbox{ty}})$  of Rene' 41 alloy bar, plate and forging; transverse and long-itudinal directions.



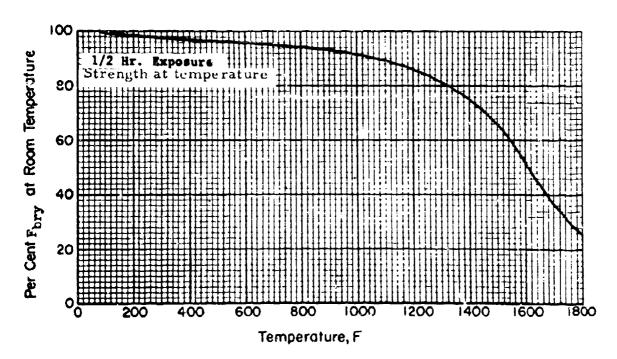
Effect of temperature on the bearing ultimate strength ( $F_{bru}$ ) of Rene' 41 alloy plate, transverse and longitudinal directions, e/D = 1.5. Exposure up to 1/2-hour.



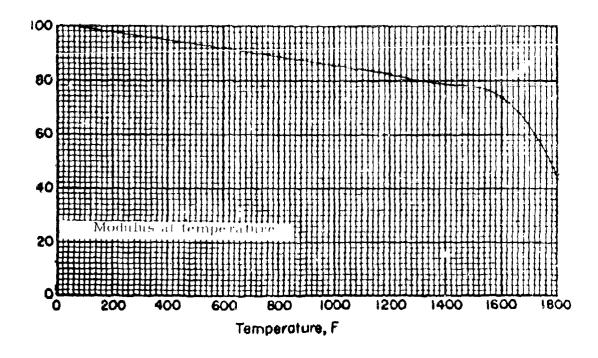
Effect of temperature on the bearing ultimate strength ( $F_{bru}$ ) of Rene' 41 alloy plate, transverse and longitudinal directions, e/D = 2.0. Exposure up to 1/2-hour.



Effect of temperature on the bearing yield strength  $(F_{bry})$  of Rene' 41 alloy plate, transverse and longitudinal directions, e/D = 1.5. Exposure up to 1/2-hour.



Effect of temperature on the bearing yield strength ( $F_{\rm bry}$ ) of Rene' 41 alloy plate, transverse and longitudinal directions, e/D = 2.0. Exposure up to 1/2-hour.



The effect of temperature on the elastic moduli, E and Ec, of Rene<sup>1</sup> 41 alloy.

## SECTION VIII - MIL-HDBK-5 DATA PRESENTATION

## 8.2 MATERIAL, L-605

## TABLE DESIGN MECHANICAL AND PHYSICAL PROPERTIES OF

Alloy	1 -005							
Form	Sheet ( $.020'' \le t \le .187''$ )							
Condition		Solutio	on Treate	ed.				
DIRECTION		L	r					
Basis	А	Б	Α	В				
Mechanical Properties  Fig. ksi	128.4	132.4	131.4	135,5				
F <sub>ty</sub> , ksi	62.8	66.3	60.9	64.3				
F <sub>cy</sub> , ksi	42.6	45.0	63.0	66.6				
F <sub>su</sub> , ksi F <sub>bru</sub> , ksi	108.9	112.3	112.0	115.4				
(e/D = 1.5) (e/D = 2.0)	184.9	190.7	187.9	193.8 233.3				
Fbry, kei (e/D = 1.5) (e/D = 2.0) e, per cent	93.6	98.8	102.9	108.7				
E, 10 <sup>6</sup> psi	32.9		32.9					
E <sub>c</sub> , 10 <sup>6</sup> pei	30.8		30,8					
G, 10 <sup>6</sup> psi								
Physical Properties		<del></del>	<del></del>					
ω, lb/in. <sup>3</sup>								
C, Btu/(lb)(F)								
K, Btu/[(hr)(ft <sup>2</sup> )(F)/ft]		•.						
α, 10 <sup>-3</sup> in./in./F								

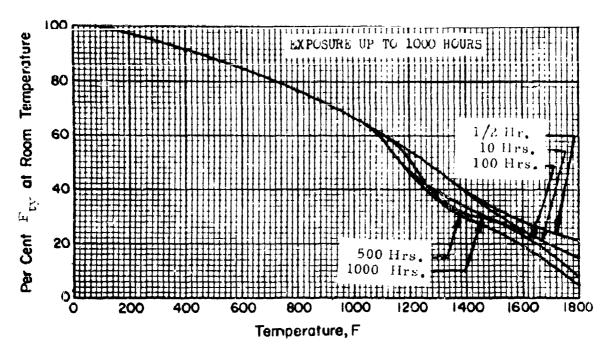
#### TABLE

### DESIGN MECHANICAL AND PHYSICAL PROPERTIES OF

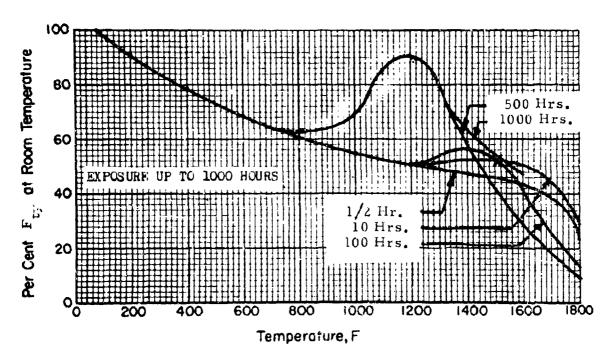
Alloy	L-605						
Form	Plate, Bar and Forging						
Condition	Solution Treated						
DIRECTION	L_		Т				
Basis	Α	В	A	В			
Mechanical Properties  Fiu, ksi	133.7	136.9	130,6	134.8			
F <sub>ty</sub> , ksi	60.2	63.6	56.2	62.1			
F <sub>cy</sub> , ksi	58.3	61.6	53.7	59.3			
Fau, kai	98.1	100.5	95.9*	98.9*			
F <sub>bru</sub> , kai (e/D = 1.5) (e/D = 2.0)		184.0	189.9				
F <sub>bry</sub> , ksi (e/D = 1.5) (e/D = 2.0)			96.9	107.1			
e, per cent			<u> </u>				
E, 10 <sup>6</sup> psi	32.9		32.9				
E <sub>c</sub> , 10 <sup>6</sup> psi	30.8		30.8				
G, 10 <sup>6</sup> psi							
Physical Properties							
ω, lb/in. <sup>3</sup>							
C, Btu/(lb)(F)							
K, Btu/[(hr)(ft <sup>2</sup> )(F)/ft]							
$\alpha$ , $10^{-6}$ in./in./F							

<sup>\*</sup> Tentative for Bar

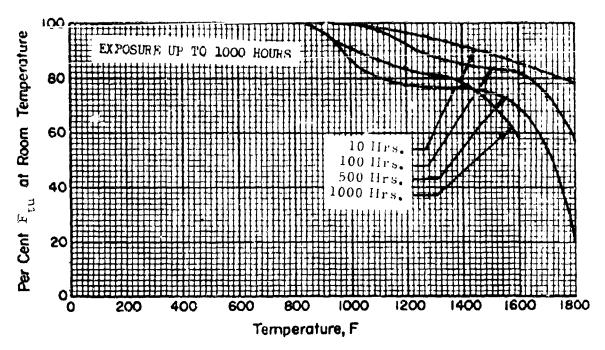
Page 469 has been eliminated data incorporated on Table, page 468.



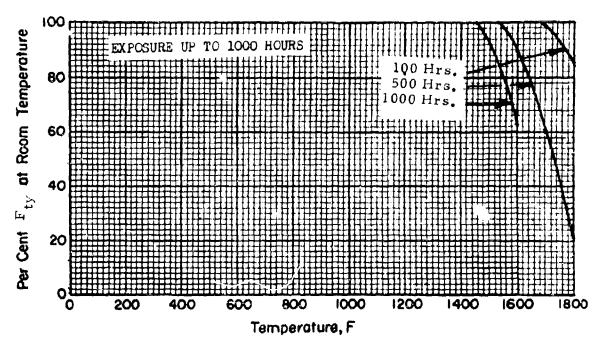
Effect of exposure time on the elevated temperature ultimate tensile strength ( $F_{tu}$ ) of L-605 alloy foil. Exposure up to 1000-hours. Transverse direction only.



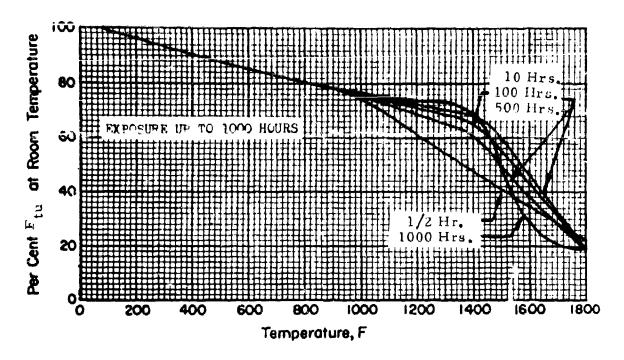
Effect of exposure time on the elevated temperature tensile yield strength (Fty) of L-605 alloy foil. Exposure up to 1000 hours. Transverse direction only.



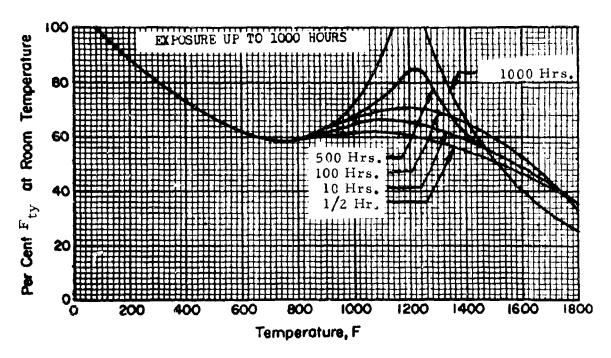
Effect of exposure at elevated temperature on the room temperature ultimate tensile strength ( $F_{tu}$ ) of L-605 alloy foil. Exposure up to 1000-hours. Transverse direction only.



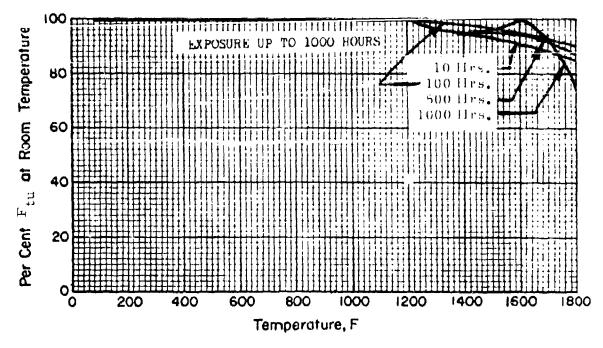
Effect of exposure at elevated temperature on the room temperature tensile yield strength  $(F_{ty})$  of I-605 alloy foil. Exposure up to 1000-hours. Transverse direction only.



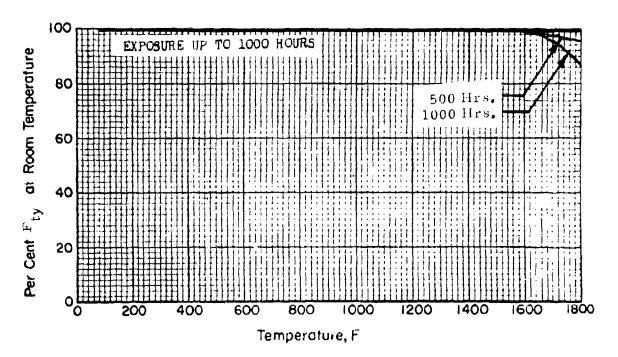
Effect of exposure time on the elevated temperature ultimate tensile strength ( $F_{tu}$ ) of L-605 alloy sheet. Exposure up to 1000-hours.



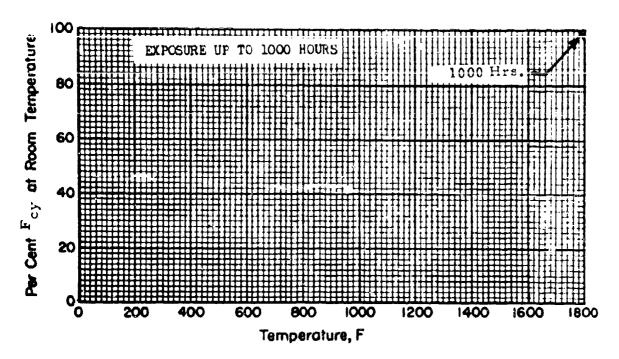
Effect of exposure time on the elevated temperature tensile yield strength ( $F_{ty}$ ) of L-605 alloy sheet. Exposure up to 1000-hours.



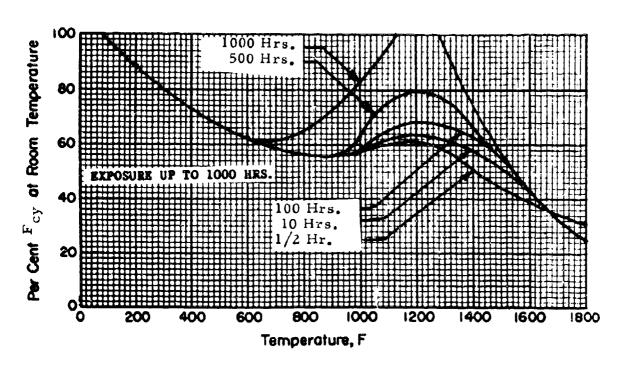
Effect of exposure at elevated temperature on the room temperature ultimate tensile strength ( $F_{tu}$ ) of L-605 alloy sheet. Exposure up to 1000-hours.Transverse direction only.



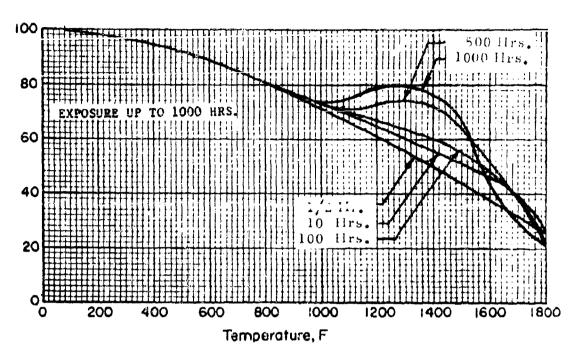
Effect of exposure at elevated temperature on the room temperature tensile yield strength ( $F_{ty}$ ) of L-605 alloy sheet. Exposure up to 1000-hours. Transverse direction only.



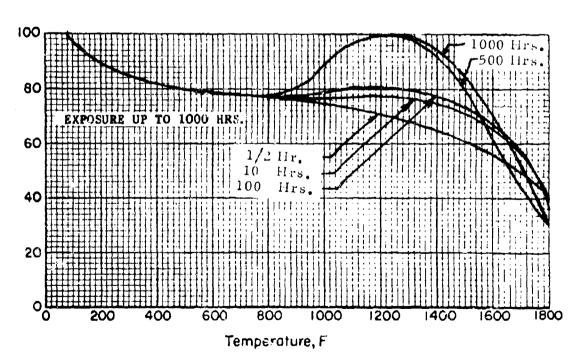
Effect of exposure at elevated temperature on the room temperature compressive yield strength ( $F_{\rm cy}$ ) of L-605 alloy sheet. Exposure up to 1000-hours. Transverse direction only.



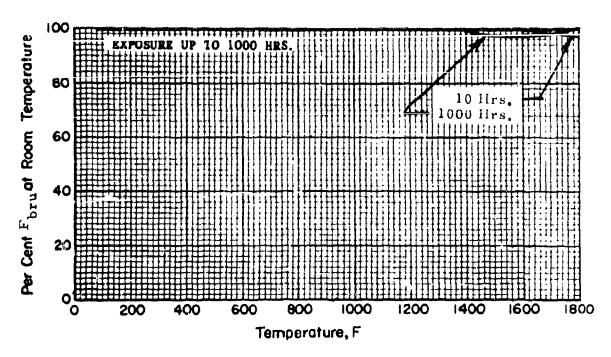
Effect of exposure time on elevated temps ature compressive yield strength ( $F_{\rm Cy}$ ) of L-605 alloy sheet. Exposure up to 1000-hours. Transverse direction only.



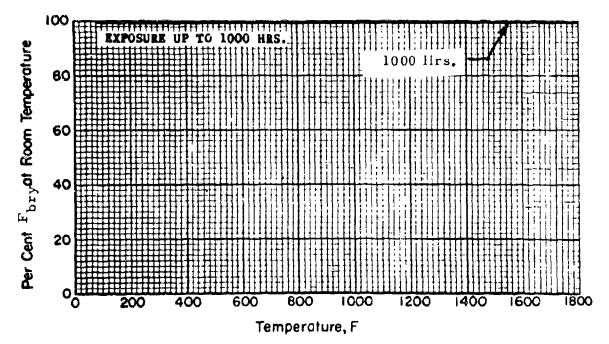
Effect of exposure time on the elevated temperature bearing ultimate strength  $(F_{bru})$  of 1-605 alloy sheet. Exposure up to 1000-hours. e/D = 1.5. Transverse direction only.



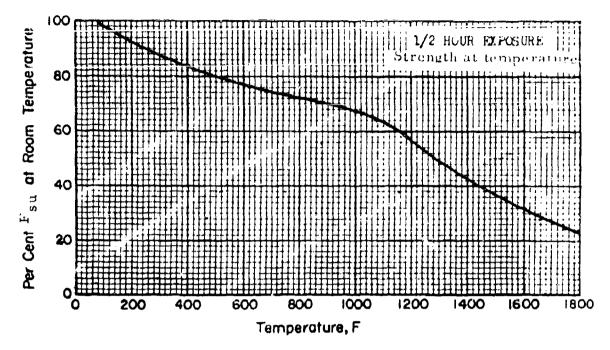
Effect of exposure time on the elevated temperature bearing yield strength ( $F_{\rm bry}$ ) of L-605 alloy sheet. Exposure up to 1000-hours. e/D = 1.5. Transverse direction only.



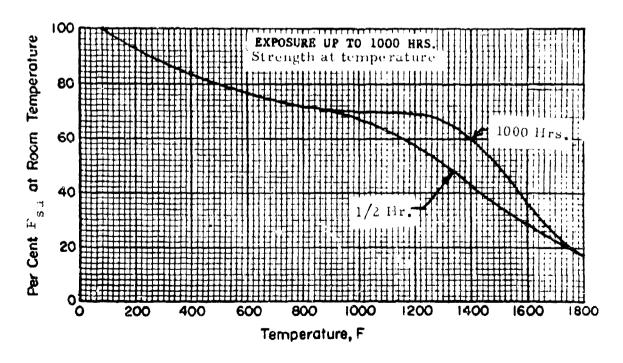
Effect of exposure at elevated temperature on the room temperature bearing ultimate strength ( $F_{\rm bru}$ ) of L-605 alloy sheet. Exposure up to 1000-hours. e/D = 1.5. Transverse direction only.



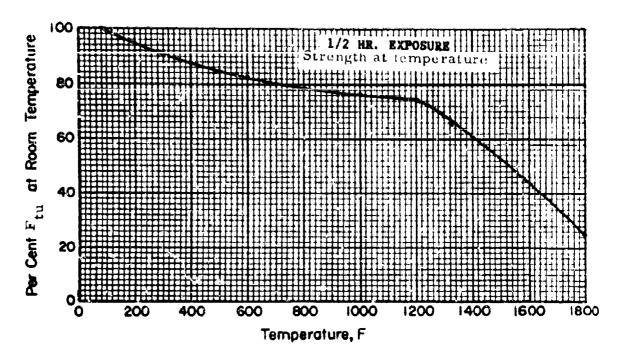
Effect of exposure at elevated temperature on the room temperature bearing yield strength  $(F_{\rm bry})$  of L-605 alloy sheet. Exposure up to 1000-hours. e/D = 1.5. Transverse direction only.



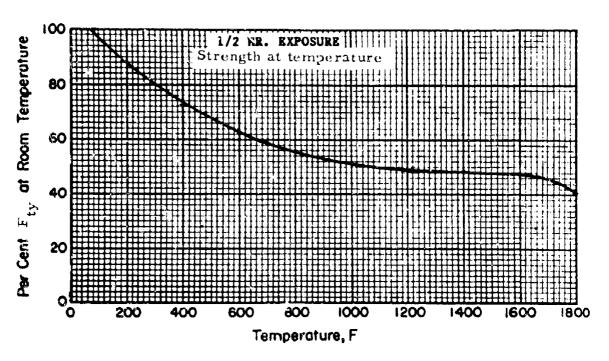
Effect of temperature on shear ultimate strength (Fsu) of L-605 alloy sheet. Exposure up to 1/2-hour.



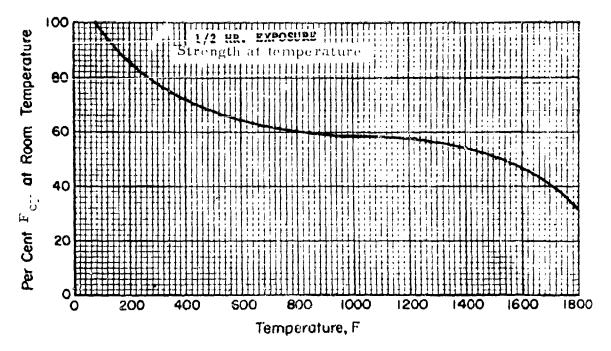
Effect of exposure time on the elevated temperature shear ultimate strength ( $F_{\rm SU}$ ) of L-605 alloy sheet. Exposure up to 1000-hours.



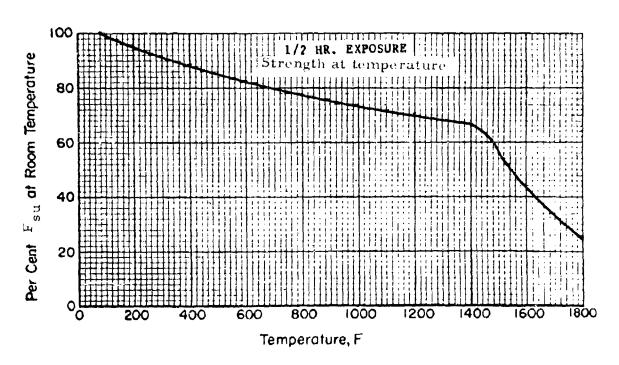
Effect of temperature on the ultimate tensile strength ( $F_{tu}$ ) of L-605 alloy plate, bar and forgings. Exposure up to 1/2-hour.



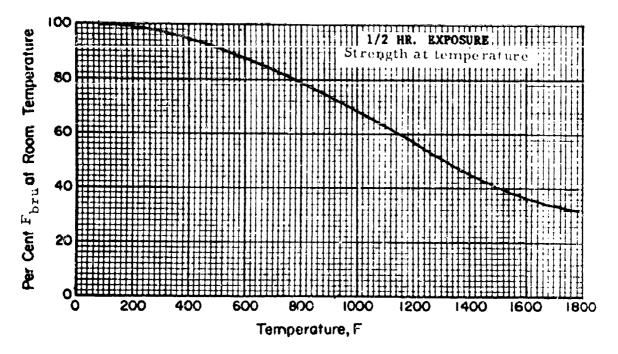
Effect of temperature on the tensile yield strength ( $F_{ty}$ ) of L-605 allow plate, bar and forgings. Exposure up to 1/2-hour.



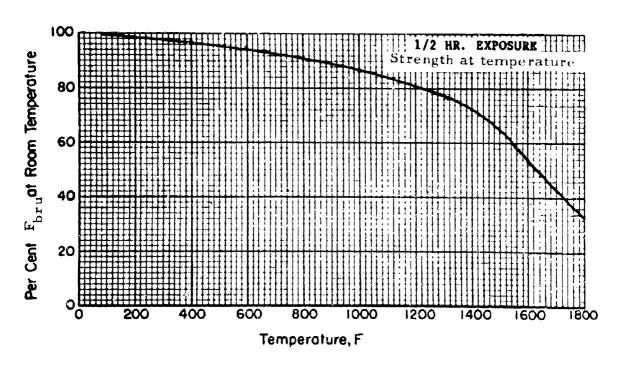
Effect of temperature on the compressive yield strength ( $F_{\rm CY}$ ) of 1-605 alloy plate, bar and forgings. Exposure up to 1/2-hour.



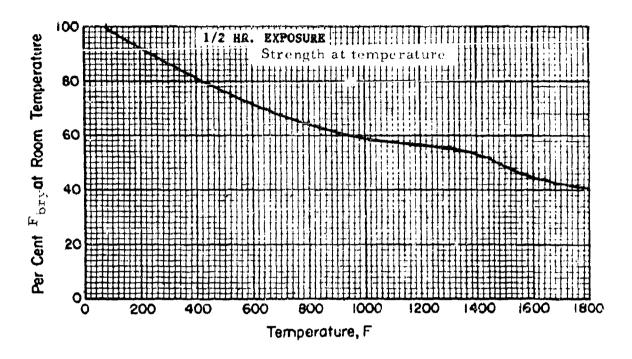
Effect of temperature on the shear ultimate strength ( $F_{su}$ ) of L-605 alloy plate, bar and forgings. Exposure up to 1/2-hour.



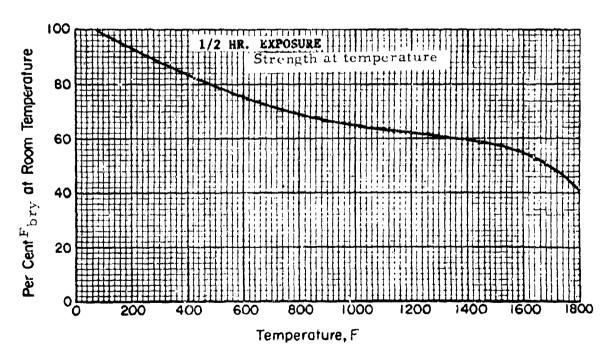
Effect of temperature on the bearing ultimate strength ( $F_{bru}$ ) of L-605 alloy plate. Exposure up to 1/2-hour. e/D = 1.5.



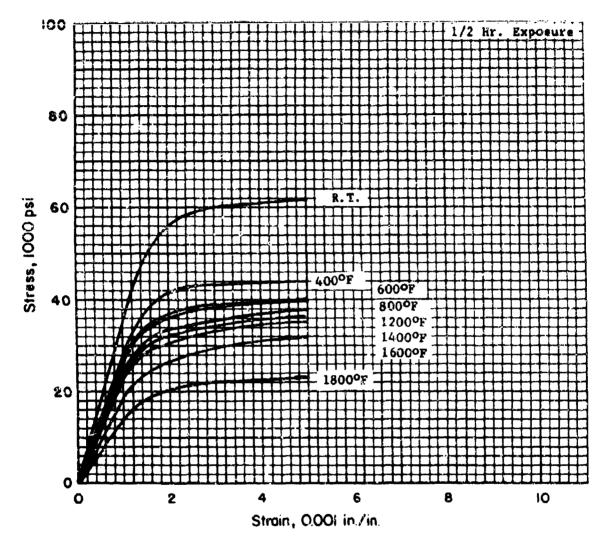
Effect of temperature on the bearing ultimate strength ( $F_{bru}$ ) of I-605 alloy plate. Exposure up to 1/2-hour. e/D = 2.0.



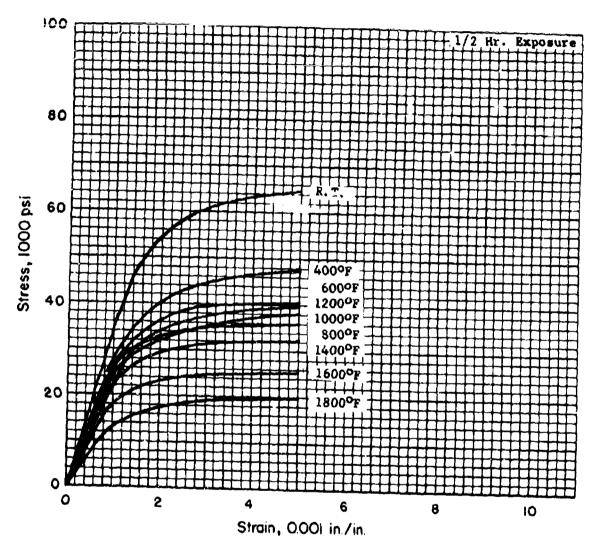
Effect of temperature on the bearing yield strength  $(F_{bry})$  of 1-605 alloy plate. Exposure up to 1/2-hour. e/D = 1.5.



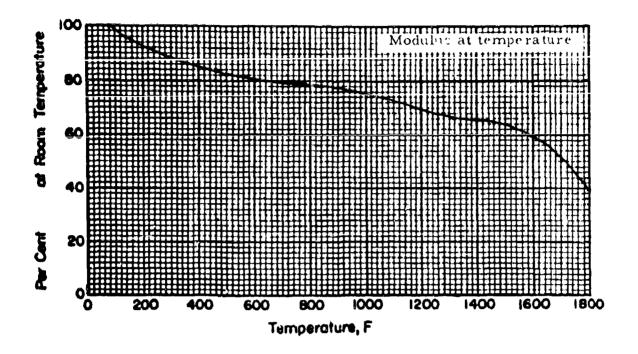
Effect of temperature on the bearing yield strength  $(F_{bry})$  of L-605 alloy plate. Exposure up to 1/2-hour. e/D = 2.0.



Typical tensile stress vs. strain curves for L-605 alloy sheet. Transverse direction only.



Typical compressive stress vs. strain curves for L-605 alloy sheet. Transverse direction only.

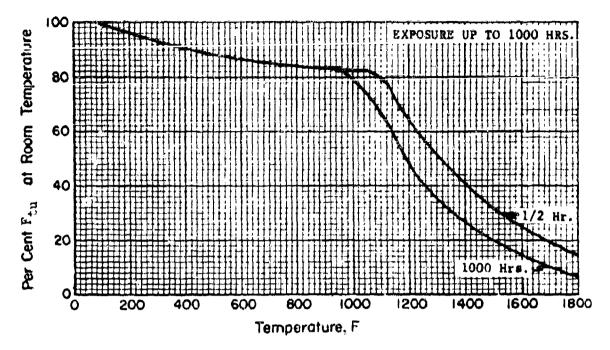


The effect of temperature on the elastic moduli, E and Ec, of L=605 alloy.

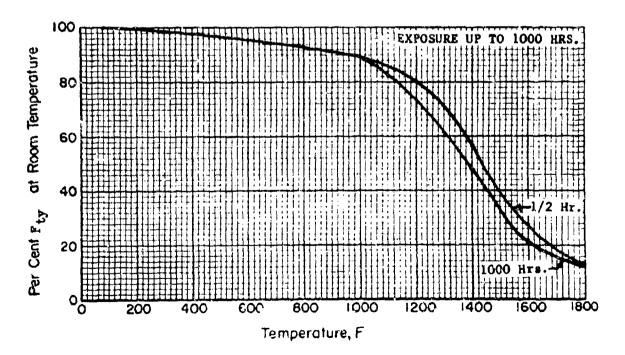
# SECTION VIII - MIL-HDBK-5 DATA PRESENTATION

8.3 MATERIAL, INCONEL 702

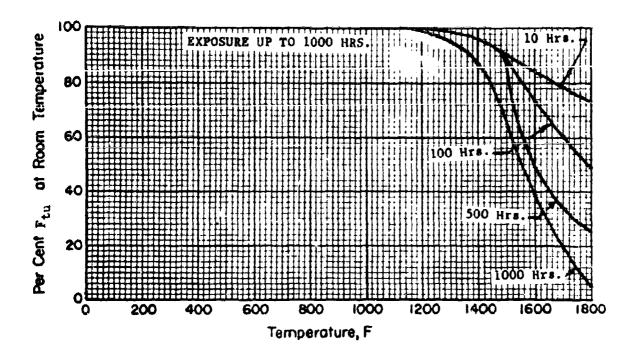
Alloy	Incomel 702						
Form	Sheet (.020" <= t <= .187")  Solution Treated and Aged						
Condition							
DIRECTION	L Tentative A   B		T Tentative				
Basis							
Mechanical Properties Ftu, kei	125.6	131.6	128.3	134.4			
F <sub>ty</sub> , ksi	67.4	73.4	69.5	75.7			
F <sub>cy</sub> , kai	67.6	73.6	71.7	78.1			
F <sub>su</sub> , ksi F <sub>bru</sub> , ksi	94.5	99,0	94.4	98.9			
(e/D = 1.5) (e/D = 2.0)	191.7 234.7	200.8	190.8	199.9 252.8			
F <sub>bry</sub> , ksi (a/D = 1.5) (e/D = 2.0) a, per cent	103.6 129.3	112.8	105.8 128.7	115.3			
E, 10 <sup>6</sup> psi		32	•6				
E <sub>c</sub> , 10 <sup>6</sup> psi G, 10 <sup>6</sup> psi	32.0						
Physical Properties							
ω, lb/in. <sup>3</sup>							
C, Btu/(lb)(F)							
K, Btu/[(hr)(ft <sup>2</sup> )(F)/ft]							
$\alpha$ , $10^{-6}$ in./in./F							



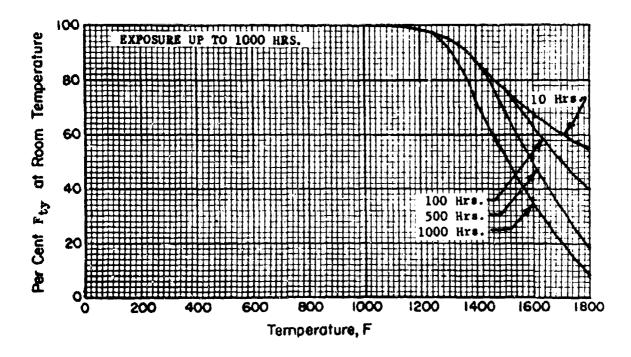
Effect of exposure temperature on the elevated temperature ultimate tensile strength ( $F_{tu}$ ) of Inconel 702 alloy foil, Transverse direction. Exposure up to 1000-hours.



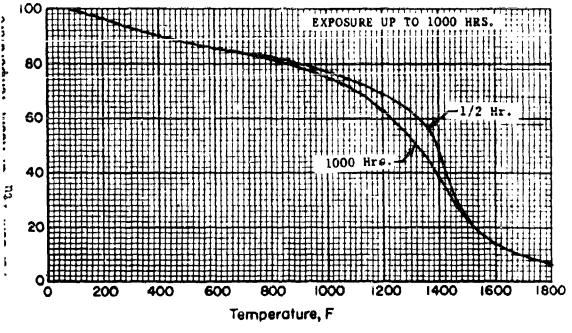
Effect of exposure temperature on the elevated temperature tensile yield strength (Fty) of Inconel 702 alloy foil, Transverse direction. Exposure up to 1000-hours.



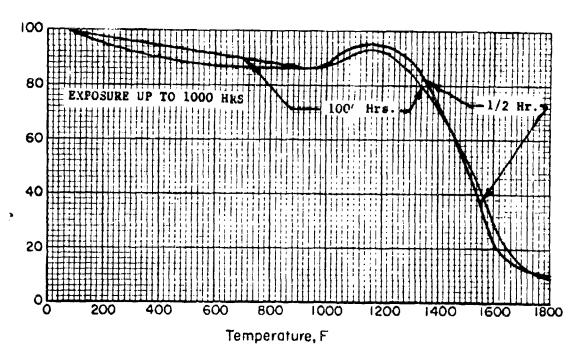
Effect of exposure temperature on room temperature ultimate tensile strength  $(F_{tu})$  of Inconel 702 alloy foil, transverse direction. Exposure up to 1000-hours.



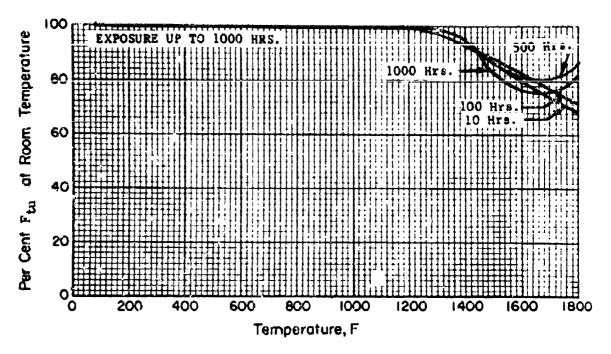
Effect of exposure temperature on room temperature tensile yield strength ( $F_{ty}$ ) of Inconel 702 alloy foil, transverse direction. Exposure up to 1000-hours.



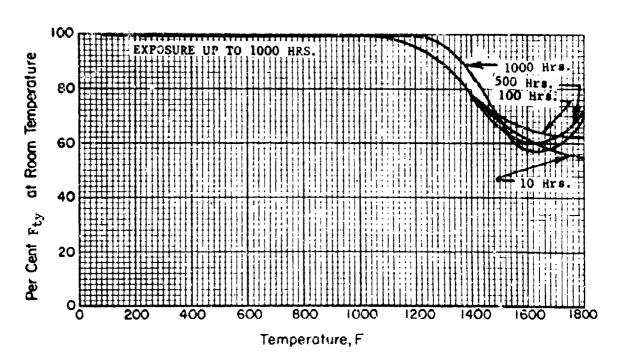
Effect of exposure temperature on the elevated temperature ultimate tensile strength  $(F_{tu})$  of Inconel 702 alloy sheet, transverse direction. Exposure up to 1000-hours.



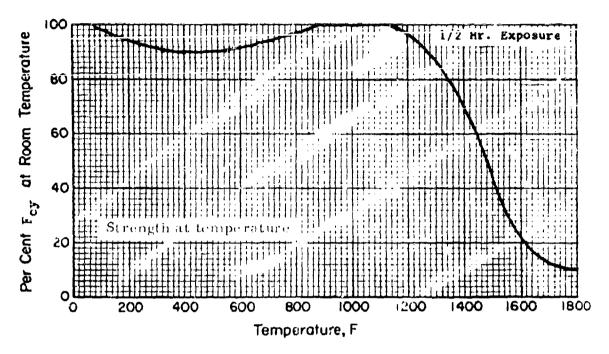
Effect of exposure temperature on the elevated temperature tensile yield strength (Fty) of Inconel 702 alloy sheet, transverse direction. Exposure up to 1000-Hours.



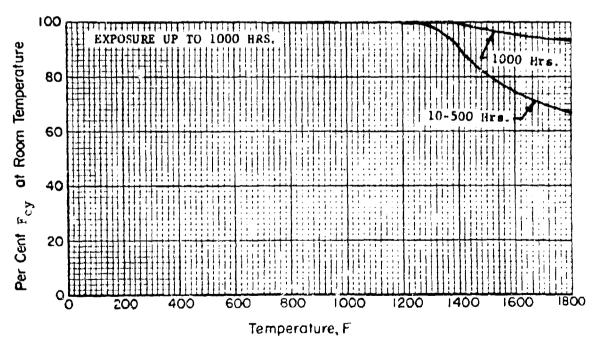
Effect of exposure temperature on the room temperature ultimate tensile strength ( $F_{tu}$ ) of Inconel 702 alloy sheet, transverse direction. Exposure up to 1000-hours.



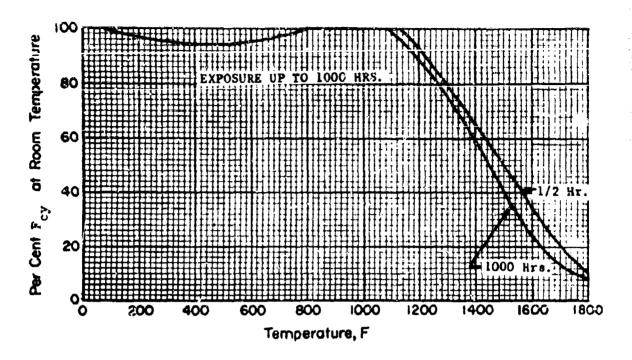
Effect of exposure temperature on the room temperature tensile yield strength (Fty of Inconel 702 alloy sheet, transverse direction. Exposure up to 1000-hours.



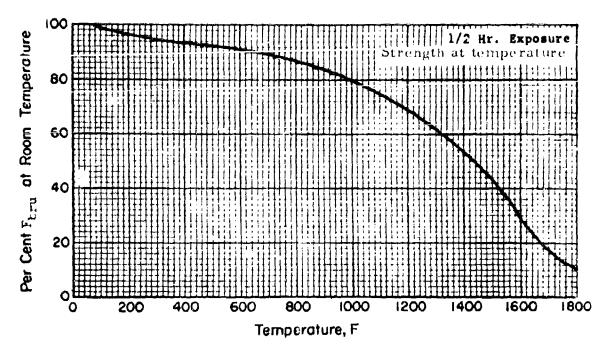
Effect of temperature on the compression yield strength ( $F_{cy}$ ) of Inconel 702 alloy sheet, transverse and longitudinal directions. Exposure up to 1/2-hour.



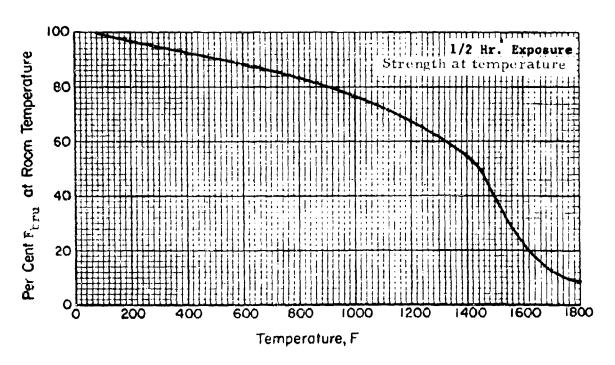
Effect of exposure temperature on room temperature compression yield strength ( $F_{\rm C}y$ ) of Inconel 702 alloy sheet, transverse direction. Exposure up to 1000-hours.



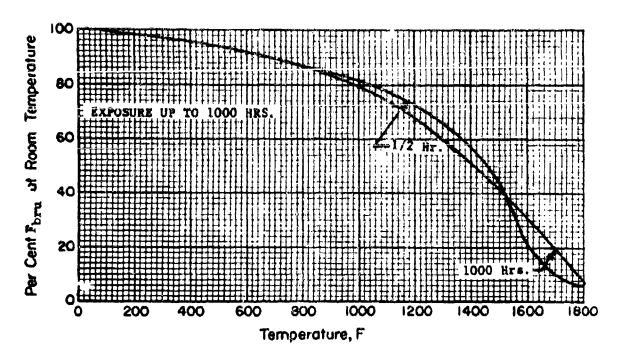
Effect of exposure temperature on elevated temperature on elevated temperature compression yield strength (Fcy) of Inconel 702 alloy sheet, transverse direction. Exposure up to 1000-hours.



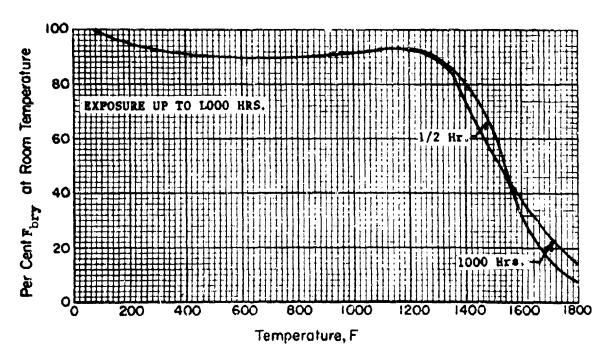
Eifect of temperature on the bearing ultimate strength ( $F_{bru}$ ) of Inconel 702 alloy sheet, transverse and longitudinal directions, e/D = 1.5. Exposure up to 1/2-hour.



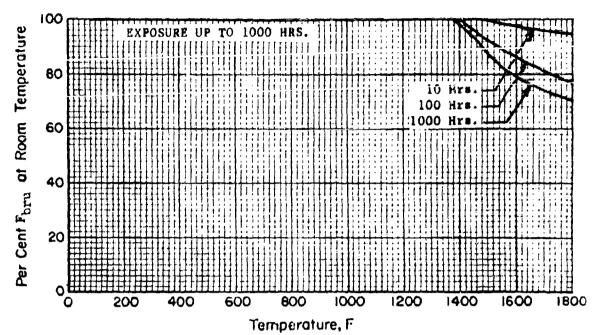
Effect of temperature on the bearing ultimate strength ( $F_{bru}$ ) of Inconel 702 alloy shect, transverse and longitudinal directions, e/D = 2.C. Exposure up to 1/2-hour.



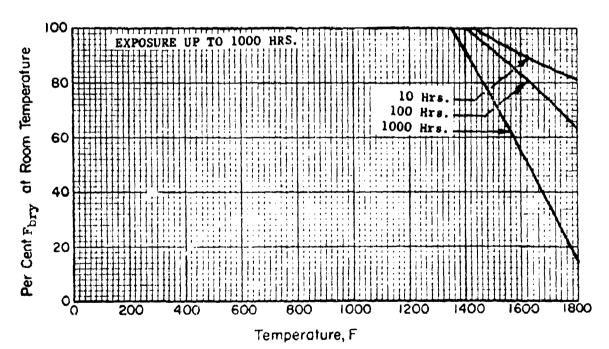
Effect of exposure temperature on the elevated temperature bearing ultimate strength  $(F_{\rm bru})$  of Inconel 702 alloy sheet, transverse direction, e/D = 1.5. Exposure up to 1000-hours.



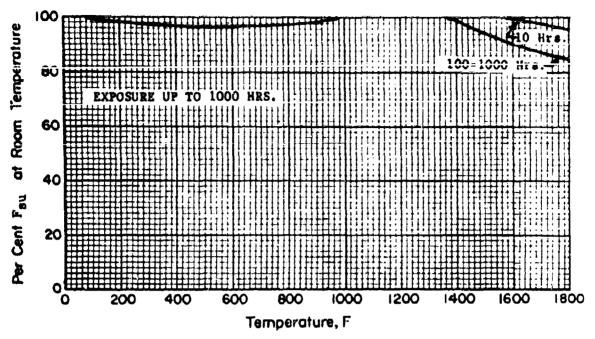
Effect of exposure temperature on elevated temperature bearing yield strength  $(F_{\rm bry})$  of Inconel 702 alloy sheet, transverse direction, e/D=1.5. Exposure up to 1000-hours.



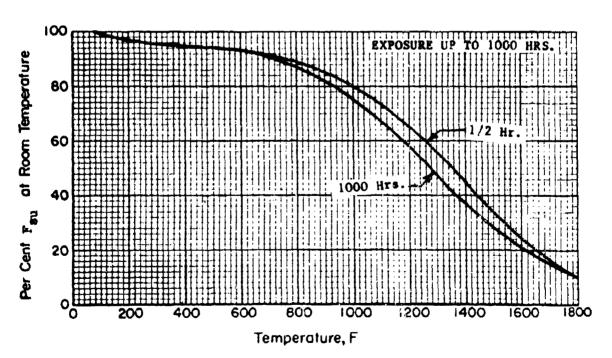
Effect of exposure temperature on the room temperature bearing ultimate strength ( $F_{bru}$ ) of Inconel 702 alloy sheet, transverse direction, e/D = 1.5. Exposure up to 1000-hours.



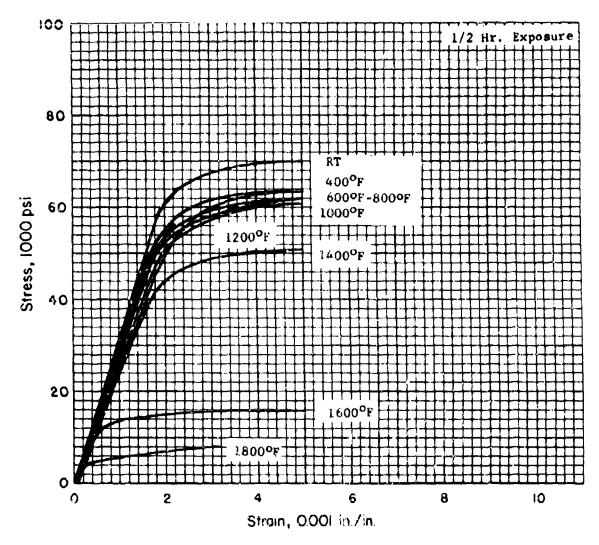
Effect of exposure temperature on the room temperature bearing yield strength (F<sub>bry</sub>) of Inconel 702 alloy sheet, trans-vorse direction, e/D = 1.5. Exposure up to 1000-hours.



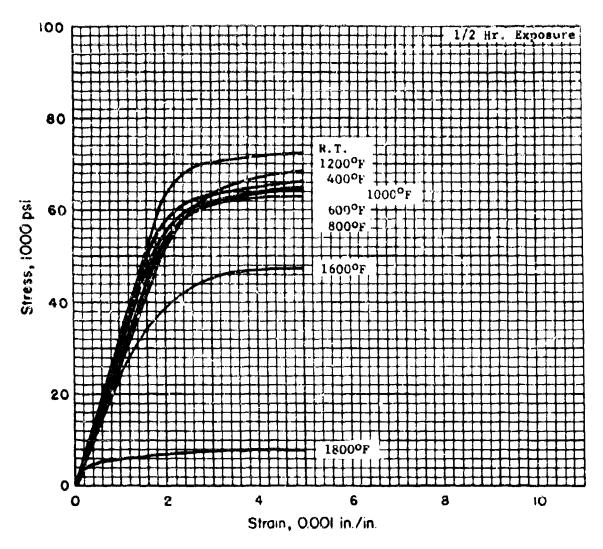
Effect of exposure temperature on the room temperature shear ultimate strength  $(F_{su})$  of Inconel 702 alloy sheet, transverse and longitudinal directions. Exposure up to 1000-hours.



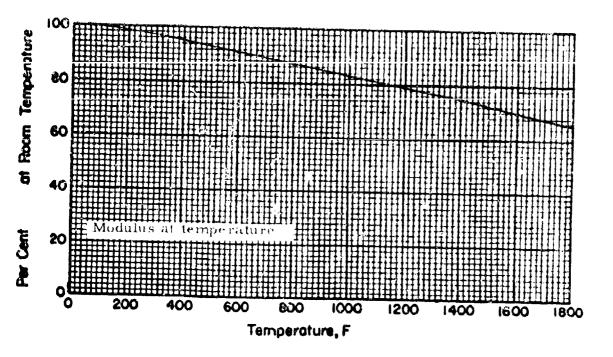
Effect of exposure temperature on the elevated temperature shear ultimate strength ( $F_{\rm su}$ ) of Inconel 702 alloy sheet, transverse and longitudinal directions. Exposure up to 1000-hours.



Typical tensile stress vs. strain curves for Incomel 702 alloy sheet. Transverse direction only.



Typical compressive stress vs. strain curves for Incomel 702 alloy sheet. Transverse direction only.



The effect of temperature on the elastic moduli, E and Ec, of Inconel 702 alloy

# SECTION VIII - MIL-HOBK-5 DATA PRESENTATION

8.4 MATERIAL, INCOLOY 901

### TABLE

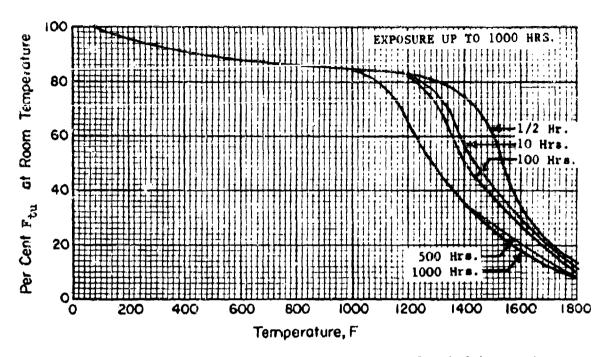
# DESIGN MECHANICAL AND PHYSICAL PROPERTIES OF

Alloy	Incoloy 90l						
Form	Forging				Bar		
Condition		e Aged					
DIRECTION	L T		·	T			
Basis	A	3	A	В	Tentative A B		
Mechanical Properties Ftu, ksi							
F <sub>ty</sub> , ksi							
F <sub>cy</sub> , kai			96.0	104.9			
F <sub>su</sub> , ksi F <sub>bru</sub> , ksi (e/D = 1.5) (e/D = 2.0)	108.8	112.7	104.2	107.9	99.1	102.7	
F <sub>bry</sub> , ksi (e/D = 1.5) (e/D = 2.0) e, per cent	1						
E, 10 <sup>6</sup> psi	29.9				29.9		
E <sub>c</sub> , 10 <sup>6</sup> psi	29.9			29.9			
G, 10 <sup>6</sup> psi							
Physical Properties				<u> </u>	<del></del>		
ω, lb/in, <sup>3</sup>							
C, Btu/(lb)(F)							
K, Btu/[(hr)(ft <sup>2</sup> )(F)/ft]							
α, 10 <sup>-6</sup> in./in./F							

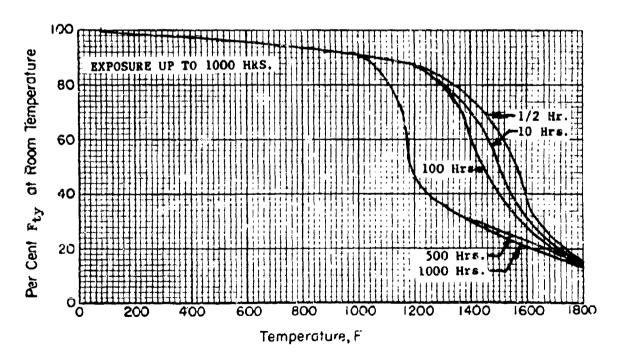
## TABLE DESIGN MECHANICAL AND PHYSICAL PROPERTIES OF

Alloy	Incoloy 901						
Form	Bar						
Condition	Solution Treated and Double Aged						
DI RECTION	L		Т				
Basis	А	В	А	В			
Mechanical Properties F <sub>tu</sub> , ksi	162.2	168.0	148.4	153.7			
F <sub>ty</sub> , ksi	101.7	111.1	95.8	104.7			
F <sub>cy</sub> , ksi	1 02 . 1	111.5					
F <sub>su</sub> , ksi F <sub>bru</sub> , ksi (e/D = 1.5) (e/D = 2.0) F <sub>bry</sub> , ksi (e/D = 1.5) (e/D = 2.0) e, per cent			99.1*	102.7*			
E, 10 <sup>6</sup> pei							
E <sub>c</sub> , 10 <sup>6</sup> psi	29.9						
G, 10 <sup>6</sup> psi							
Physical Properties					<u> </u>		
ω, lb/in. <sup>3</sup>							
C, Btu/(lb)(F)							
K, Btu/[(hx)(ft <sup>2</sup> )(F)/ft]							
α, 10 <sup>-6</sup> in./in./F							

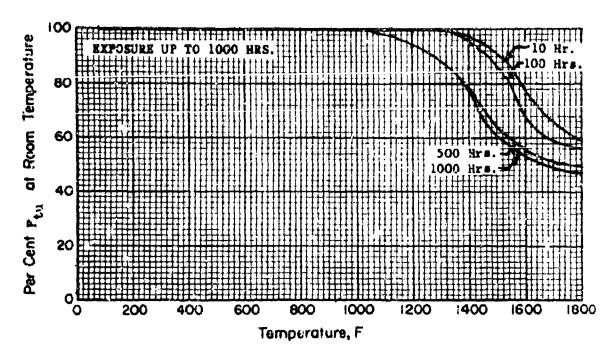
<sup>\*</sup> Tentative



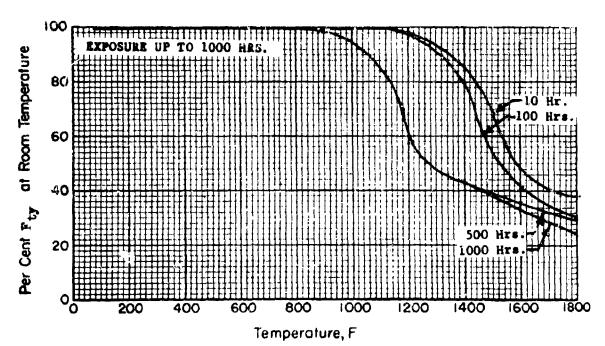
Effect of exposure temperature on the elevated temperature ultimate tensile strength ( $F_{tu}$ ) of Incoloy 901 alloy bar, longitudinal direction. Exposure up to 1000-hours.



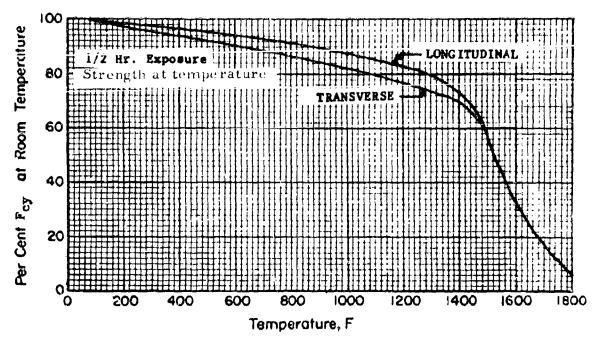
Effects of exposure temperature on the elevated temperature tensile yield strength ( $F_{ty}$ ) of Incoloy 901 alloy bar, longitudinal direction. Exposure up to 1000-hours.



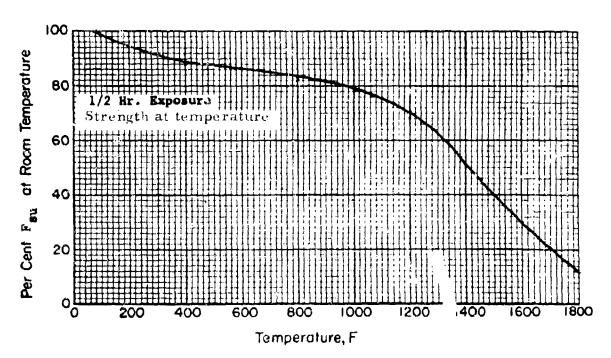
Effect of exposure temperature on the room temperature ultimate tensile strength  $(F_{tu})$  of Incoloy 901 alloy bar; longitudinal direction. Exposure up to 1000-hours.



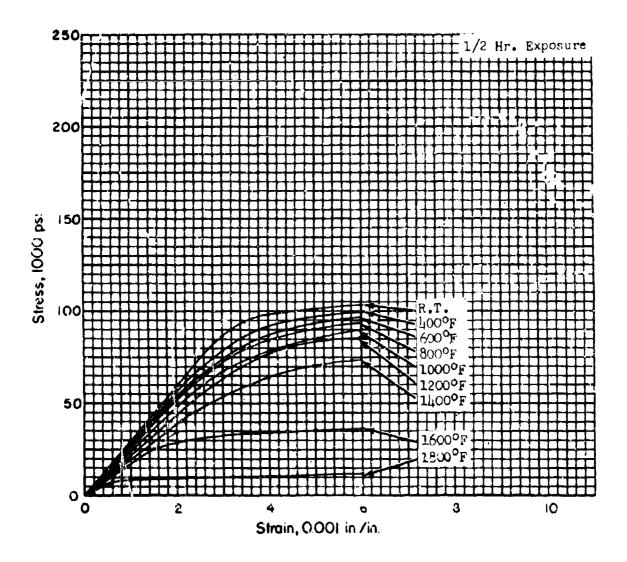
Effect of exposure temperature on the room temperature tensile yield strength  $(F_{\mbox{ty}})$  of Incoloy 901 alloy bar, longitudinal direction. Exposure up to 1000-hours.



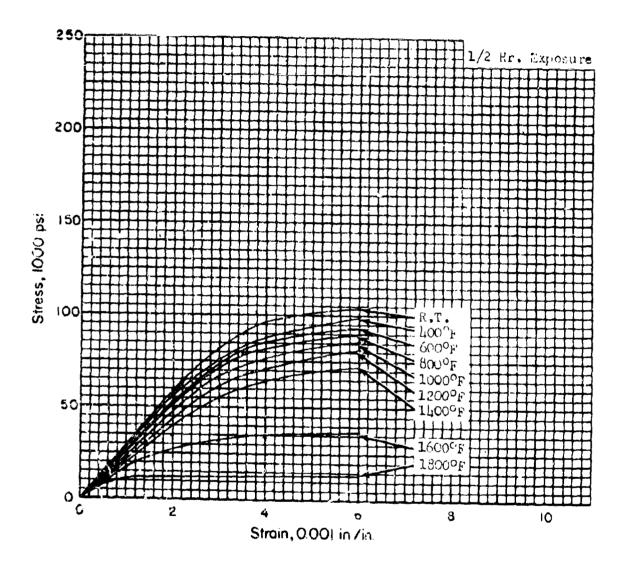
Effect of temperature on the compression yield strength  $(F_{\rm Cy})$  of Incoloy 901 alloy bar, longitudinal and transverse directions. Exposure to 1/2-hour.



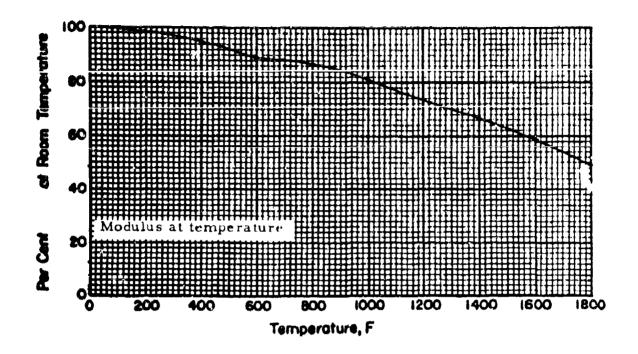
Effect of temperature on the shear ultimate strength ( $F_{su}$ ) of Incoloy 901 slloy bar, longitudinal and transverse directions. Exposure to 1/2-hour.



Typical tensile stress VS strain curves for Incoloy 901 alloy bar reduced to 'A' basis. Longitudinal direction only.



Typical compressive stress VS strain curves for Incoloy 901 alloy bar reduced to 'A' basis. Longitudinal direction only.



The effect of temperature on the elastic moduli, E and Ec, on Incoloy 901 alley.

## SECTION IX - REFERENCES

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Page 85 thru 89 - Table 20 Page 248, Appendix III Page 249 thru 252, Figures 149 thru 156

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Page 17 - Table III

3. "Effects of Severe Thermal and Stress Histories on Material Strength-Rate Process Theory Approach AlS1301 Extra Hard, Phl5-7 Mo RH, Rene 41, 7075-T6", ASD-TR-61-194 January 1962, Northrop Corp., AF33(616)-5769, by C.D. Brownfield & D.R. Apodaca

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4. "Effect of Creep-Emposure on Mechanical Properties of Rene'41," ASD-TR-61-73, August 1961, University of Michigan, AF33(616)-6462, by J.V. Gluck and J.W. Freeman

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- 5. "Structural Damage in Thermally Cycled Rene' 41 and Astroloy Sheet Materials", DMIC Report 126, February 1960, Battelle Memorial Institute, by D.P. Moon, J.A. VanEcho, W.F. Simmons & J.F. Baker
- 6. "Materials Property Data Compilation Part I Rene' 41 Sheet and Strip", AF33(657)-8017, General Electric Co., Cincinnati, Ohio, May 1962 by H.G. Popp

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10. "Investigation of Mechanical Properties of Rene 41", April 1962, ESRMR 137, Republic Aviation Corporation, by C. Shaver

#### Page 3 - Table 1

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- 11. "Creep Rupture Properties of Six Elevated Temperature Alloys", WADD-TR-199, August 1962, New England Materials Laboratory, by J. McBride, B. Mulhern, R. Widmer
- 12. "Manufacturing Methods for Hot Structures", Report #2-17059, Boeing Airplane Co., September 1959, by J. Claus, D. Meredith & F. Sutch

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- 17. "Corrosion Superalloys by Selected Fused Salts", WADD TR 60-115, Crucible Steel Co., March 1960, by A. Moskowitz and L. Redmuski
- 18. "Effect of Prior Creep on Mechanical Properties of Aircraft Structural Materials", AF33(616)-6462, University of Michigan, October 1960, by J. Gluck & J.W. Freeman
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- 20. "Gas Atmosphere Effects on Materials", Progress Report #1, AF33(616)-5667, August 1958, General Electric Co., FPLD Evendale, by R.A. Baughman
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  - 23. "Investigation of the Unnotched and Notched Fatigue Behavior of Several Heat Resistant Materials for Engine Bolts", WADD-TR-59-25, February 1960, Materials Laboratory, by D. Forney & D. Wang

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- 33. "Compression Testing at Elevated Temperatures", Metals Engineering Quarterly Vol. 1, No. 3, August 1961, by J.P. King
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# REPUBLIC AVIATION CORPORATION

FARMINGDALE, LONGISLAND, NEW YORK

January 26, 1965

Gentlemen:

SUBJECT: Contract AF 33(657)-8924

On December 1, 1964, ML-TDR-64-116 Research Investigation To Determine Mechanical Properties of Nickel and Cobalt Base Alloys for Inclusion in Military Handbook-5, Volume I, October 1964, was mailed to you in accordance with the distribution list received from the Air Force Systems Command.

Attached hereto is Errata sheet ML-TDR-64-116.

Very truly yours,

REPUBLIC AVIATION CORPORATION

J. A. Weglage, Contracts Administrator

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### ERRATA - JANUARY 1965

The following corrections are applicable to ML-TDR-64-116 Research Investigation To Determine Mechanical Properties of Nickel and Cobalt Base Alloys for Inclusion in Military Handbook-5, Volume I, October 1964.

### PAGE 449

Change - Plate, bar, and forging, B value for Fty should read 142.5 rather than 42.5

PAGE 501

Delete - tentative BAR properties

Add - additional forging properties

Direction	1	L	T		
Basis	A	В	A	В	
P <sub>tu</sub> , ksi	162.2	168.0	148.4	153.7	
Fty, ksi	101.7	111.1	95.8	104.7	
Foy, ksi	102.1	111.5			

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